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## SYNTHESIS OF CARBON NITRIDES AND COMPOSITE PHOTOCATALYST MATERIALS

by

Anthony Tristan Montoya

A thesis submitted in partial fulfillment of the requirements for the Doctor of Philosophy degree in Chemistry in the Graduate College of The University of Iowa

August 2018

Thesis Supervisor: Associate Professor Edward G. Gillan

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#### CERTIFICATE OF APPROVAL

#### PH.D. THESIS

This is to certify that the Ph.D. thesis of

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has been approved by the Examining Committee for the thesis requirement for the Doctor of Philosophy degree in Chemistry at the August 2018 graduation.

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To you, the reader

"Chemistry is the study of matter, but I prefer to see it as the study of change."

-Walter White

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iv

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#### ABSTRACT

This thesis describes the synthesis, characterization and photocatalytic applications of carbon nitride ( $C_3N_4$ ) and titanium dioxide ( $TiO_2$ ) materials.  $C_3N_4$  was prepared from the thermal decomposition of a trichloromelamine (TCM) precursor. Several different reactor designs and decomposition temperatures were used to produce chemically and thermally stable orange powders. These methods included a low temperature glass Schlenk reactor, a high mass scale stainless steel reactor, and decomposition at higher temperatures by the immersion of a Schlenk tube into a furnace. These products share many of the same structural and chemical properties when produced by these different methods compared to products from more common alternate precursors in the literature, determined by infrared spectroscopy (IR), nuclear magnetic resonance spectroscopy (NMR), powder X-ray diffraction (XRD), X-ray photoelectron spectroscopy (XPS), and elemental analysis.  $C_3N_4$  is capable of utilizing light for photocatalysis due to its moderate band gap ( $E_g$ ), measured to be between 2.2 and 2.5 eV. This enables  $C_3N_4$  to be used in the photocatalytic degradation of organic dyes and the production of hydrogen via the water-splitting reaction.  $C_3N_4$  degraded methylene blue dye to less than 10% of its initial concentration in less than an hour of UV light illumination and 60% under filtered visible light in 150 minutes. It also degraded methyl orange dye to below 20% in 70 minutes under UV light and below 60% in 150 minutes under visible light. Using precious metal co-catalysts (Pt, Pd, and Ag) photo-reduced onto the surface of  $C_3N_4$ , hydrogen was produced from a 10% aqueous solution of triethanolamine at rates as high as 260  $\mu$ mol h<sup>-1</sup> g<sup>-1</sup>.

vi

 $C_3N_4$  was also modified by mixing the precursor with different salts (NaCl, KBr, KI, KSCN, and  $NH_4SCN$ ) as hard templates. Many of these salts reacted with TCM by exchanging the anion with the chlorine in TCM. The products were mostly prepared using the high temperature Schlenk tube reactor, and resulted in yellow, orange, or tanbrown products with  $E_g$  values between 2.2 and 2.7 eV. Each of these products had subtle differences in the IR spectra and elemental composition. The morphology of these C<sub>3</sub>N<sub>4</sub> products appeared to be more porous than unmodified C<sub>3</sub>N<sub>4</sub>, and the surface area for some increased by a factor of 4. These products demonstrated increased activity for photocatalytic hydrogen evolution, with the product from TCM-KI reaching a peak rate as high as 1,300  $\mu$ mol h<sup>-1</sup> g<sup>-1</sup>. C<sub>3</sub>N<sub>4</sub> was coated onto metal oxide supports (SiO<sub>2</sub>, Al<sub>2</sub>O<sub>3</sub>, TiO<sub>2</sub>, and WO<sub>3</sub>) with the goal of utilizing enhanced surface area of the support or synergy between two different semiconductors. These products typically required higher temperature synthesis conditions in order to fully form. The compositions of the  $SiO_2$  and  $Al_2O_3$  products were richer in nitrogen and hydrogen compared to unmodified  $C_3N_4$ . The higher temperature reactions with  $C_3N_4$  and  $WO_3$  resulted in the formation of the  $H_xWO_3$ phase, and an alternate approach of coating  $WO_3$  on  $C_3N_4$  was used. The degradation of methyl orange showed a significant increase in adsorption of dye for the composites with SiO<sub>2</sub> and Al<sub>2</sub>O<sub>3</sub>, which was not seen with any of the individual components. The composite between  $C_3N_4$  and TiO<sub>2</sub> showed improved activity for hydrogen evolution compared to unmodified  $C_3N_4$ .

The surface of  $TiO_2$  was modified by the reductive photodeposition of several first row transition metals (Mn, Fe, Co, Ni, and Cu). This process resulted in the slight color change of the white powder to shades of light yellow, blue or grey. Bulk elemental analysis showed that these products contained between 0.04-0.6 at% of the added metal, which was lower than the targeted deposit amount. The Cu modified  $TiO_2$  had the largest enhancement of photocatalytic hydrogen evolution activity with a rate of 8,500 µmol h<sup>-1</sup> g<sup>-1</sup>, a factor of 17 higher than unmodified  $TiO_2$ .

#### PUBLIC ABSTRACT

Carbon nitride and titanium dioxide are semiconductors capable of harnessing energy from light and using it to facilitate different chemical reactions in a process called photocatalysis. This thesis describes the preparation and modification of these materials with the intent of testing and improving their photocatalytic activity. Carbon nitride is formed via the decomposition of trichloromelamine at temperatures above 180 °C. This produces an orange powder that is largely inert to common solvents and thermally stable. Carbon nitrides have been modified by incorporating different salts or metal oxide supports to increase available surface area, an important property when studying catalytic applications. The surface of titanium dioxide was modified using different earth abundant metals as an alternative to a more expensive platinum co-catalyst.

The photocatalytic reactions studied were the degradation of dyes in water and the production of hydrogen gas from water. The degradation of dyes serves as an analog to the removal of pollutants from environmental water systems. Using the described catalysts, up to 90% of the dye can be degraded in the presence of UV light or up to 60% in the presence of visible light. Hydrogen has applications as a clean renewable fuel, though challenges lie in its storage and transportation. By using these catalysts and light energy, hydrogen can be produced from water.

ix

#### TABLE OF CONTENTS

LIST OF TABLES	xiv
LIST OF FIGURES	xvi
CHAPTER 1 INTRODUCTION AND BACKGROUND	1
1.1 Introduction	1
1.2 Carbon Nitrides	1
1.3 Titanium Dioxide	4
1.4 Principles of Photocatalysis	6
1.5 Thesis Overview	9
CHAPTER 2 SYNTHESIS AND ACTIVATION OF CARBON NITRIDE FROM TRICHLOROMELAMINE DECOMPOSITION	. 11
2.1 Introduction	. 11
2.2 Experimental	. 13
Reagents	. 13
Thermal decomposition of TCM	. 14
Characterization	. 16
Photocatalytic reactions	. 17
2.3 Results and Discussion	. 19
Formation of carbon nitride from the decomposition of TCM	. 19
Solid state characterization of carbon nitride from TCM decomposition	. 22
Compositional analysis of carbon nitride	. 26
Microstructure and morphology	. 31
Optical properties	. 33
Photocatalytic studies	. 36
2.4 Conclusions	. 51
CHAPTER 3 POROUS ENGINEERING BY SALT MODIFICATIONS TO	
CARBON NITRIDE SYNTHESIS	. 53
3.1 Introduction	. 53
3.2 Experimental	. 55
Reagents	. 55
$C_3N_4$ synthesis from alkali halide mixtures with TCM	. 55

$C_3N_4$ synthesis from thiocyanate mixtures with TCM	57
Solid state mixtures with salts	58
Water wash of C <sub>3</sub> N <sub>4</sub> -salt composites	59
Characterization	59
Photocatalytic reactions	60
3.3 Results and Discussion	62
Initial reactions between TCM and salt additives	62
Carbon nitride formation from decomposition of TCM-salt mixtures	67
Chemical and structural characterization of carbon nitride salt composites	72
Morphology and surface area	77
Optical properties	80
Photocatalytic studies	83
3.4 Conclusions	90
CHAPTER 4 DEPOSITION OF CARBON NITRIDE ONTO METAL OXIDE SUPPORTS	91
4.1 Introduction	
4.2 Experimental	
Reagents	
Coating of $C_3N_4$ onto SiO <sub>2</sub> , TiO <sub>2</sub> and Al <sub>2</sub> O <sub>3</sub> oxide supports	
Preparation of $C_3N_4$ -WO <sub>3</sub> composite materials	
Annealing of $C_3N_4$ -metal oxide composites	
Reaction of a physical mixture of $C_3N_4$ and metal oxide support	
Characterization	
Photocatalytic reactions	
4.3 Results and Discussion	
Decomposition reaction and thermal stability of C <sub>3</sub> N <sub>4</sub> in oxide composites	
Chemical and structural characterization of carbon nitride and oxide composites	106
Morphology and surface area	115
Optical properties	119
Photocatalytic studies	122

4.4 Conclusions	131
CHAPTER 5 ENHANCED PHOTOCATALYSIS OF TRANSITION-METAL	
SURFACE-MODIFIED TITANIUM DIOXIDE	132
5.1 Introduction	132
5.2 Experimental	134
Reagents	134
Modification of TiO <sub>2</sub> surface with photodeposited transition metals	134
Characterization	136
Photocatalytic reactions	137
5.3 Results and Discussion	139
Photochemical M:TiO <sub>2</sub> synthesis	139
M:TiO <sub>2</sub> structure, composition, and surface analysis	139
Optical properties	146
Photocatalytic reactions	149
5.4 Conclusions	164
CHADTER 6 DREI IMINARY STUDIES ON METAL NITRIDES SUICA	
CHAI TER UT RELIMINART STUDIES ON METAL NITRIDES, SILICA	
ETCHING, SELECTIVE ORGANIC PHOTO-OXIDATIONS, CONCLUSIONS,	1.55
ETCHING, SELECTIVE ORGANIC PHOTO-OXIDATIONS, CONCLUSIONS, AND FUTURE OUTLOOK	166
ETCHING, SELECTIVE ORGANIC PHOTO-OXIDATIONS, CONCLUSIONS, AND FUTURE OUTLOOK	166 166
ETCHING, SELECTIVE ORGANIC PHOTO-OXIDATIONS, CONCLUSIONS, AND FUTURE OUTLOOK	166 166 167
ETCHING, SELECTIVE ORGANIC PHOTO-OXIDATIONS, CONCLUSIONS, AND FUTURE OUTLOOK	166 166 167 167
ETCHING, SELECTIVE ORGANIC PHOTO-OXIDATIONS, CONCLUSIONS, AND FUTURE OUTLOOK	166 166 167 167 167
ETCHING, SELECTIVE ORGANIC PHOTO-OXIDATIONS, CONCLUSIONS, AND FUTURE OUTLOOK	166 166 167 167 167 168
<ul> <li>ETCHING, SELECTIVE ORGANIC PHOTO-OXIDATIONS, CONCLUSIONS,</li> <li>AND FUTURE OUTLOOK</li> <li>6.1 Introduction</li> <li>6.2 Experimental</li> <li>Reagents</li> <li>Metal nitride formation on C<sub>3</sub>N<sub>4</sub></li> <li>Etching of silica from C<sub>3</sub>N<sub>4</sub>@SiO<sub>2</sub></li> <li>Photocatalytic oxidation of small organic molecules</li> </ul>	166 166 167 167 167 168 168
<ul> <li>ETCHING, SELECTIVE ORGANIC PHOTO-OXIDATIONS, CONCLUSIONS,</li> <li>AND FUTURE OUTLOOK</li> <li>6.1 Introduction</li> <li>6.2 Experimental</li> <li>Reagents</li> <li>Metal nitride formation on C<sub>3</sub>N<sub>4</sub></li> <li>Etching of silica from C<sub>3</sub>N<sub>4</sub>@SiO<sub>2</sub></li> <li>Photocatalytic oxidation of small organic molecules</li> <li>Characterization</li> </ul>	166 166 167 167 167 168 168 169
<ul> <li>ETCHIAI TEK OT KELIMINART STODIES ON METAL MITKIDES, SILICA</li> <li>ETCHING, SELECTIVE ORGANIC PHOTO-OXIDATIONS, CONCLUSIONS, AND FUTURE OUTLOOK.</li> <li>6.1 Introduction.</li> <li>6.2 Experimental.</li> <li>Reagents.</li> <li>Metal nitride formation on C<sub>3</sub>N<sub>4</sub>.</li> <li>Etching of silica from C<sub>3</sub>N<sub>4</sub>@SiO<sub>2</sub>.</li> <li>Photocatalytic oxidation of small organic molecules .</li> <li>Characterization.</li> <li>6.3 Results and Discussion</li> </ul>	166 166 167 167 167 168 168 169 170
<ul> <li>ETCHING, SELECTIVE ORGANIC PHOTO-OXIDATIONS, CONCLUSIONS, AND FUTURE OUTLOOK.</li> <li>6.1 Introduction</li></ul>	166 166 167 167 167 168 168 169 170 170
<ul> <li>ETCHING, SELECTIVE ORGANIC PHOTO-OXIDATIONS, CONCLUSIONS,</li> <li>AND FUTURE OUTLOOK</li></ul>	166 167 167 167 167 168 168 169 170 170 172
ETCHING, SELECTIVE ORGANIC PHOTO-OXIDATIONS, CONCLUSIONS, AND FUTURE OUTLOOK	166 167 167 167 167 168 168 169 170 170 172 174
ETCHING, SELECTIVE ORGANIC PHOTO-OXIDATIONS, CONCLUSIONS, AND FUTURE OUTLOOK6.1 Introduction6.2 ExperimentalReagentsMetal nitride formation on $C_3N_4$ .Etching of silica from $C_3N_4$ @SiO2Photocatalytic oxidation of small organic moleculesCharacterization6.3 Results and DiscussionMetal nitrides on $C_3N_4$ @SiO2Photocatalytic oxidation in the vapor phase6.4 Overall Thesis Summary and Conclusions	166 167 167 167 167 168 168 168 169 170 170 172 174 176
ETCHING, SELECTIVE ORGANIC PHOTO-OXIDATIONS, CONCLUSIONS,AND FUTURE OUTLOOK6.1 Introduction6.2 ExperimentalReagentsMetal nitride formation on $C_3N_4$ .Etching of silica from $C_3N_4$ .Etching of silica from $C_3N_4$ .Characterization6.3 Results and DiscussionMetal nitrides on $C_3N_4$ .Etching of SiO <sub>2</sub> from C <sub>3</sub> N <sub>4</sub> .Etching of SiO <sub>2</sub> from C <sub>3</sub> N <sub>4</sub> .Etching of SiO <sub>2</sub> from C <sub>3</sub> N <sub>4</sub> .Etching of SiO <sub>2</sub> from C <sub>3</sub> N <sub>4</sub> .Etching of SiO <sub>2</sub> from C <sub>3</sub> N <sub>4</sub> .Etching of SiO <sub>2</sub> from C <sub>3</sub> N <sub>4</sub> .Etching of SiO <sub>2</sub> from C <sub>3</sub> N <sub>4</sub> .Etching of SiO <sub>2</sub> from C <sub>3</sub> N <sub>4</sub> .Etching of SiO <sub>2</sub> from C <sub>3</sub> N <sub>4</sub> .Etching of SiO <sub>2</sub> from C <sub>3</sub> N <sub>4</sub> .Etching of SiO <sub>2</sub> from C <sub>3</sub> N <sub>4</sub> .SiDia Summary and Conclusions.6.5 Future Outlook.	166 167 167 167 167 168 168 168 169 170 170 172 174 176 177

A.1 Photocatalytic Hydrogen Evolution	
Signal calibration	
Photocatalysis set-up	
A.2 Quantum Yield Calculations	
A.3 Protocol for CHN Elemental Analysis	
Start-up	
Calibration and sampling (Manual 4-58)	
Shutdown	
Changing the combustion and reduction tubes	
REFERENCES	

#### LIST OF TABLES

Table 1.1.	Band gaps and corresponding absorption edges for $C_3N_4$ , TiO <sub>2</sub> , and WO <sub>3</sub> .	. 6
Table 2.1.	NMR assignments and shifts for TCM-C <sub>3</sub> N <sub>4</sub> (S)	24
Table 2.2.	CHN elemental analysis of C <sub>3</sub> N <sub>4</sub> products.	27
Table 2.3.	XPS signal positions and surface composition for C <sub>3</sub> N <sub>4</sub>	30
Table 2.4.	Rate constants $(min^{-1})$ for UV light degradation of MB dye using TCM- C <sub>3</sub> N <sub>4</sub> products.	36
Table 2.5.	Rate constants $(min^{-1})$ for visible light degradation of MB dye using TCM, MA, and DCDA-C <sub>3</sub> N <sub>4</sub> products	38
Table 2.6.	Rate constants $(min^{-1})$ for UV and visible light degradation of methyl orange using TCM-C <sub>3</sub> N <sub>4</sub> and DCDA-C <sub>3</sub> N <sub>4</sub> products.	42
Table 2.7.	XPS surface analysis data for metals deposited on TCM-C <sub>3</sub> N <sub>4</sub> (P/A)	45
Table 2.8.	Average hydrogen production rates (with standard deviation) for Pt, Pd, and Ag deposited $C_3N_4$ products with ICP-OES quantification of metal loading.	48
Table 3.1.	Heats of formation for tested additive salts. <sup>100</sup>	64
Table 3.2.	CHN elemental analysis for selected washed TCM-salt C <sub>3</sub> N <sub>4</sub> products from 500 °C tube reactions.	73
Table 3.3.	Summary of XRD phases observed in different salt mixtures before and after decomposition reaction (major phase in bold).	77
Table 3.4.	BET surface areas of $C_3N_4$ products from TCM and TCM-salt mixtures	80
Table 3.5.	Rate constants $(min^{-1})$ for UV light degradation of methyl orange dye using washed $C_3N_4$ products from TCM-NaCl mixtures.	83
Table 3.6.	Rate constants $(min^{-1})$ for UV light degradation of methyl orange dye using washed $C_3N_4$ produced from different TCM-salt mixtures	85
Table 3.7.	Average hydrogen production rates (with standard deviation) for Pt deposited $C_3N_4$ and ICP-OES quantification of metal loading	87
Table 4.1.	CHN elemental analysis for $C_3N_4$ metal oxide composite products	)8

Table 4.3.	Rate constants (min <sup>-1</sup> ) for UV and visible light degradation of methyl orange dye using $C_3N_4$ -metal oxide composites	124
Table 4.4.	Rate constants (min <sup>-1</sup> ) for UV and visible light degradation of methyl orange dye using $C_3N_4$ and milled $C_3N_4$ -metal oxide composites	126
Table 4.5.	Average hydrogen evolution rates for 1 wt% Pt deposited $TiO_2$ , $C_3N_4$ , and $C_3N_4$ -metal oxide composites over 4 1-hour cycles (standard deviation reported in parentheses).	129
Table 5.1.	ICP-OES and XPS compositions and peak positions of M(x%)TiO <sub>2</sub> samples.	142
Table 5.2.	XPS analysis of surface chlorine relative to deposited metals on $M(x\%)$ :TiO <sub>2</sub> .	146
Table 5.3.	First-order rate constants $(min^{-1})$ for the degradation of methylene blue dye using M(2%):TiO <sub>2</sub> products.	149
Table 5.4.	First-order rate constants $(min^{-1})$ for the degradation of methylene blue dye using M(1%)M'(1%):TiO <sub>2</sub> products	151
Table 5.5.	Hydrogen evolution rates (µmol/h) and quantum yields for M:TiO <sub>2</sub> materials.	156
Table 5.6.	Comparison of several Ni and Cu modified titania for hydrogen evolution photocatalysis.	157
Table 5.7.	Hydrogen evolution rates (µmol/h) and quantum yields for select recycled M:TiO <sub>2</sub> materials	158
Table 5.8.	XPS data on select M:TiO <sub>2</sub> products after use in photocatalytic hydrogen evolution.	159
Table 6.1.	Summary of XRD phases observed for annealing reactions between $C_3N_4$ and $MCl_2$	171
Table 6.2.	CHN elemental analysis of C <sub>3</sub> N <sub>4</sub> @SiO <sub>2</sub> and etched products	174
Table A.1.	Volume additions of hydrogen and corresponding pressure signals for hydrogen and argon used for calibration	181

#### LIST OF FIGURES

Figure 1.1 – F	Franklin's proposed pathway of deammonation products, which lead to the melon polymer	2
Figure 1.2 – 7	The generally accepted ideal structure of $g-C_3N_4$ .	4
Figure 1.3 – U	Unit cell structures of $TiO_2$ (A) anatase and (B) rutile. Ti atoms are in cyan and O atoms are in red.	5
Figure 1.4 – C	Generalized mechanism for photocatalysis at a semiconductor particle surface.	7
Figure 1.5 – C	Chemical structures of (A) methyl orange and (B) methylene blue dyes	8
Figure 2.1 – 7	TGA and DTA plots of TCM decomposition under argon flow and a 10 °C/min ramp rate	.0
Figure 2.2 – F	T-IR spectrum of gas phase products from TCM decomposition2	1
Figure 2.3 – I	Decomposition of TCM to an ideal $C_3N_4$ structure of condensed hepatazines	1
Figure 2.4 – F	FT-IR spectra of precursors used to prepare $C_3N_4$ (a) TCM, (b) MA, and (c) DCDA	.2
Figure 2.5 – F	FT-IR spectra of (a) TCM-C <sub>3</sub> N <sub>4</sub> (S), (b) TCM-C <sub>3</sub> N <sub>4</sub> (P), (c) TCM-C <sub>3</sub> N <sub>4</sub> (T), (d) TCM-C <sub>3</sub> N <sub>4</sub> (P/A), (e) DCDA-C <sub>3</sub> N <sub>4</sub> (T), and (f) MA-C <sub>3</sub> N <sub>4</sub> (T)	3
Figure 2.6 – C	CP-MAS-NMR spectra for TCM- $C_3N_4(S)$ , (a) <sup>13</sup> C, (b) <sup>15</sup> N 2	5
Figure 2.7 – F	Powder X-ray diffraction of $C_3N_4$ products (a) TCM- $C_3N_4(S)$ , (b) TCM- $C_3N_4(P)$ , (c) TCM- $C_3N_4(T)$ , (d) TCM- $C_3N_4(P/A)$ , (e) MA- $C_3N_4(T)$ , and (f) DCDA- $C_3N_4(T)$	6
Figure 2.8 – X	XPS C 1s, N 1s, O 1s, and Cl 2p spectra for TCM-C <sub>3</sub> N <sub>4</sub> (P/A) 2	9
Figure 2.9 – S	SEM of (A,B) TCM-C <sub>3</sub> N <sub>4</sub> (S), (C,D) TCM-C <sub>3</sub> N <sub>4</sub> (P), (E,F) TCM- C <sub>3</sub> N <sub>4</sub> (T), (G,H) TCM-C <sub>3</sub> N <sub>4</sub> (P/A)	2
Figure 2.10 –	Optical microscope images of (a) TCM- $C_3N_4(S)$ , (b) TCM- $C_3N_4(P)$ , (c) TCM- $C_3N_4(T)$ , (d) TCM- $C_3N_4(P/A)$ , (e) MA- $C_3N_4(T)$ , and (f) DCDA- $C_3N_4(T)$ .	4
Figure 2.11 –	DRS UV-vis of C <sub>3</sub> N <sub>4</sub> products (a) raw reflectance spectra and (b) Tauc plots	5

<ul> <li>Figure 2.13 – Visible light degradation of methylene blue with C<sub>3</sub>N<sub>4</sub> products, (A) degradation plots and (B) linear rate fits.</li> <li>Figure 2.14 – UV light degradation of methyl orange with C<sub>3</sub>N<sub>4</sub> products, (A) degradation plots and (B) linear rate fits.</li> <li>Figure 2.15 – Visible light degradation of methyl orange with C<sub>3</sub>N<sub>4</sub> products, (A) degradation plots and (B) linear rate fits.</li> <li>Figure 2.16 – XRD TCM-C<sub>3</sub>N<sub>4</sub>(P/A) with photodeposited (a) Pt, (b), Pd, and (c) Ag.</li> <li>Ag.</li> <li>Figure 2.17 – Photocatalytic hydrogen evolution using TCM-C<sub>3</sub>N<sub>4</sub>(P) with and without Pt co-catalyst.</li> <li>Figure 2.18 – Valence band XPS spectra of (A) TCM-C<sub>3</sub>N<sub>4</sub>(S), (B) TCM-C<sub>3</sub>N<sub>4</sub>(P), (C) TCM-C<sub>3</sub>N<sub>4</sub>(T), and (D) TCM-C<sub>3</sub>N<sub>4</sub>(P/A).</li> <li>Figure 2.19 – Band structure diagram for TCM-C<sub>3</sub>N<sub>4</sub> products based on XPS valence band spectra and calculated band gaps.</li> <li>Figure 3.1 – Method for introducing porosity into C<sub>3</sub>N<sub>4</sub> by hard templating with salts.</li> <li>Figure 3.2 – XRD patterns of TCM-KBr mixtures prepared by (a) mortar and pestle, (b) solution mixture, and (c) ball milling.</li> <li>Figure 3.4 – XRD patterns of TCM-KSCN mixtures prepared by (a) mortar and pestle, (b) solution mixture, and (c) ball milling.</li> <li>Figure 3.5 – IR spectra of C<sub>3</sub>N<sub>4</sub> precursors (a) TCM, (b) TCM-NaCl, (c) TCM-KBr, (d) TCM-KI, (e) TCM-KSCN, and (f) TCM-NH<sub>4</sub>SCN.</li> <li>Figure 3.6 – C<sub>3</sub>N<sub>4</sub> products prepared from (a) 3:1, (b) 2:1, (c) 1:1, and (d) 1:2 mixtures of TCM-KI com the for the large form 1:1 TCM:NaCl mixtures under air flow and a 10 °C/min ramp rate.</li> <li>Figure 3.7 – TGA plots of C<sub>3</sub>N<sub>4</sub> products from 1:1 TCM:NaCl mixtures under air flow and a 10 °C/min ramp rate.</li> </ul>	Figure 2.12 – UV lig (A) de	ght degradation of methylene blue with TCM-C <sub>3</sub> N <sub>4</sub> products, gradation plots and (B) linear rate fits.	37
<ul> <li>Figure 2.14 – UV light degradation of methyl orange with C<sub>3</sub>N<sub>4</sub> products, (A) degradation plots and (B) linear rate fits.</li> <li>41</li> <li>Figure 2.15 – Visible light degradation of methyl orange with C<sub>3</sub>N<sub>4</sub> products, (A) degradation plots and (B) linear rate fits.</li> <li>43</li> <li>Figure 2.16 – XRD TCM-C<sub>3</sub>N<sub>4</sub>(P/A) with photodeposited (a) Pt, (b), Pd, and (c) Ag.</li> <li>44</li> <li>Figure 2.17 – Photocatalytic hydrogen evolution using TCM-C<sub>3</sub>N<sub>4</sub>(P) with and without Pt co-catalyst.</li> <li>47</li> <li>Figure 2.18 – Valence band XPS spectra of (A) TCM-C<sub>3</sub>N<sub>4</sub>(S), (B) TCM-C<sub>3</sub>N<sub>4</sub>(P), (C) TCM-C<sub>3</sub>N<sub>4</sub>(T), and (D) TCM-C<sub>3</sub>N<sub>4</sub>(P/A).</li> <li>50</li> <li>Figure 2.19 – Band structure diagram for TCM-C<sub>3</sub>N<sub>4</sub> products based on XPS valence band spectra and calculated band gaps.</li> <li>51</li> <li>Figure 3.1 – Method for introducing porosity into C<sub>3</sub>N<sub>4</sub> by hard templating with salts.</li> <li>54</li> <li>Figure 3.2 – XRD patterns of TCM-KBr mixtures prepared by (a) mortar and pestle, (b) solution mixture, and (c) ball milling.</li> <li>64</li> <li>Figure 3.4 – XRD patterns of TCM-KSCN mixtures prepared by (a) mortar and pestle, (b) solution mixture, and (c) ball milling.</li> <li>65</li> <li>Figure 3.4 – XRD patterns of TCM-KSCN mixtures prepared by (a) mortar and pestle, (b) solution mixture, and (c) ball milling.</li> <li>65</li> <li>Figure 3.5 – IR spectra of C<sub>3</sub>N<sub>4</sub> precursors (a) TCM, (b) TCM-NaCI, (c) TCM-KBr, (d) TCM-KI, (e) TCM-KSCN, and (f) TCM-NH<sub>4</sub>SCN.</li> <li>67</li> <li>Figure 3.6 – C<sub>3</sub>N<sub>4</sub> products prepared from (a) 3:1, (b) 2:1, (c) 1:1, and (d) 1:2 mixtures of TCM:NaCI.</li> <li>69</li> <li>Figure 3.7 – TGA plots of C<sub>3</sub>N<sub>4</sub> products from 1:1 TCM:NaCI mixtures under air flow and a 10 °C/min ramp rate.</li> <li>71</li> </ul>	Figure 2.13 – Visibl degrad	le light degradation of methylene blue with C <sub>3</sub> N <sub>4</sub> products, (A) lation plots and (B) linear rate fits	39
Figure 2.15 - Visible light degradation of methyl orange with $C_3N_4$ products, (A) degradation plots and (B) linear rate fits.43Figure 2.16 - XRD TCM- $C_3N_4(P/A)$ with photodeposited (a) Pt, (b), Pd, and (c) Ag.44Figure 2.17 - Photocatalytic hydrogen evolution using TCM- $C_3N_4(P)$ with and without Pt co-catalyst.47Figure 2.18 - Valence band XPS spectra of (A) TCM- $C_3N_4(P)$ , (C) TCM- $C_3N_4(T)$ , and (D) TCM- $C_3N_4(P/A)$ .50Figure 2.19 - Band structure diagram for TCM- $C_3N_4$ products based on XPS valence band spectra and calculated band gaps.51Figure 3.1 - Method for introducing porosity into $C_3N_4$ by hard templating with salts.54Figure 3.2 - XRD patterns of TCM-KBr mixtures prepared by (a) mortar and 	Figure 2.14 – UV lig degrad	ght degradation of methyl orange with C <sub>3</sub> N <sub>4</sub> products, (A) lation plots and (B) linear rate fits	41
<ul> <li>Figure 2.16 – XRD TCM-C<sub>3</sub>N<sub>4</sub>(P/A) with photodeposited (a) Pt, (b), Pd, and (c) Ag</li></ul>	Figure 2.15 – Visibl degrad	le light degradation of methyl orange with C <sub>3</sub> N <sub>4</sub> products, (A) lation plots and (B) linear rate fits	43
<ul> <li>Figure 2.17 – Photocatalytic hydrogen evolution using TCM-C<sub>3</sub>N<sub>4</sub>(P) with and without Pt co-catalyst.</li> <li>Figure 2.18 – Valence band XPS spectra of (A) TCM-C<sub>3</sub>N<sub>4</sub>(S), (B) TCM-C<sub>3</sub>N<sub>4</sub>(P), (C) TCM-C<sub>3</sub>N<sub>4</sub>(T), and (D) TCM-C<sub>3</sub>N<sub>4</sub>(P/A).</li> <li>Figure 2.19 – Band structure diagram for TCM-C<sub>3</sub>N<sub>4</sub> products based on XPS valence band spectra and calculated band gaps.</li> <li>Figure 3.1 – Method for introducing porosity into C<sub>3</sub>N<sub>4</sub> by hard templating with salts.</li> <li>S4</li> <li>Figure 3.2 – XRD patterns of TCM-KBr mixtures prepared by (a) mortar and pestle, (b) solution mixture, and (c) ball milling.</li> <li>Figure 3.3 – XRD patterns of TCM-KSCN mixtures prepared by (a) mortar and pestle, (b) solution mixture, and (c) ball milling.</li> <li>Figure 3.4 – XRD patterns of TCM-KSCN mixtures prepared by (a) mortar and pestle, (b) solution mixture, and (c) ball milling.</li> <li>Figure 3.5 – IR spectra of C<sub>3</sub>N<sub>4</sub> precursors (a) TCM, (b) TCM-NaCl, (c) TCM-KBr, (d) TCM-KI, (e) TCM-KSCN, and (f) TCM-NH<sub>4</sub>SCN.</li> <li>Figure 3.6 – C<sub>3</sub>N<sub>4</sub> products prepared from (a) 3:1, (b) 2:1, (c) 1:1, and (d) 1:2 mixtures of TCM:NaCl.</li> <li>Figure 3.7 – TGA plots of C<sub>3</sub>N<sub>4</sub> products from 1:1 TCM:NaCl mixtures under air flow and a 10 °C/min ramp rate.</li> </ul>	Figure 2.16 – XRD Ag	TCM-C <sub>3</sub> N <sub>4</sub> (P/A) with photodeposited (a) Pt, (b), Pd, and (c)	44
<ul> <li>Figure 2.18 – Valence band XPS spectra of (A) TCM-C<sub>3</sub>N<sub>4</sub>(S), (B) TCM-C<sub>3</sub>N<sub>4</sub>(P), (C) TCM-C<sub>3</sub>N<sub>4</sub>(T), and (D) TCM-C<sub>3</sub>N<sub>4</sub>(P/A)</li></ul>	Figure 2.17 – Photo withou	catalytic hydrogen evolution using TCM- $C_3N_4(P)$ with and at Pt co-catalyst.	47
<ul> <li>Figure 2.19 – Band structure diagram for TCM-C<sub>3</sub>N<sub>4</sub> products based on XPS valence band spectra and calculated band gaps</li></ul>	Figure 2.18 – Valen (C) TC	tee band XPS spectra of (A) TCM- $C_3N_4(S)$ , (B) TCM- $C_3N_4(P)$ , CM- $C_3N_4(T)$ , and (D) TCM- $C_3N_4(P/A)$	50
<ul> <li>Figure 3.1 – Method for introducing porosity into C<sub>3</sub>N<sub>4</sub> by hard templating with salts</li></ul>	Figure 2.19 – Band valenc	structure diagram for TCM- $C_3N_4$ products based on XPS se band spectra and calculated band gaps	51
<ul> <li>Figure 3.2 – XRD patterns of TCM-KBr mixtures prepared by (a) mortar and pestle, (b) solution mixture, and (c) ball milling.</li> <li>Figure 3.3 – XRD patterns of TCM-KI mixtures prepared by (a) mortar and pestle, (b) solution mixture, and (c) ball milling.</li> <li>Figure 3.4 – XRD patterns of TCM-KSCN mixtures prepared by (a) mortar and pestle, (b) solution mixture, and (c) ball milling.</li> <li>Figure 3.5 – IR spectra of C<sub>3</sub>N<sub>4</sub> precursors (a) TCM, (b) TCM-NaCl, (c) TCM-KBr, (d) TCM-KI, (e) TCM-KSCN, and (f) TCM-NH<sub>4</sub>SCN.</li> <li>Figure 3.6 – C<sub>3</sub>N<sub>4</sub> products prepared from (a) 3:1, (b) 2:1, (c) 1:1, and (d) 1:2 mixtures of TCM:NaCl.</li> <li>Figure 3.7 – TGA plots of C<sub>3</sub>N<sub>4</sub> products from 1:1 TCM:NaCl mixtures under air flow and a 10 °C/min ramp rate.</li> </ul>	Figure 3.1 – Method salts	d for introducing porosity into $C_3N_4$ by hard templating with	54
<ul> <li>Figure 3.3 – XRD patterns of TCM-KI mixtures prepared by (a) mortar and pestle, (b) solution mixture, and (c) ball milling.</li> <li>Figure 3.4 – XRD patterns of TCM-KSCN mixtures prepared by (a) mortar and pestle, (b) solution mixture, and (c) ball milling.</li> <li>Figure 3.5 – IR spectra of C<sub>3</sub>N<sub>4</sub> precursors (a) TCM, (b) TCM-NaCl, (c) TCM-KBr, (d) TCM-KI, (e) TCM-KSCN, and (f) TCM-NH<sub>4</sub>SCN.</li> <li>Figure 3.6 – C<sub>3</sub>N<sub>4</sub> products prepared from (a) 3:1, (b) 2:1, (c) 1:1, and (d) 1:2 mixtures of TCM:NaCl.</li> <li>Figure 3.7 – TGA plots of C<sub>3</sub>N<sub>4</sub> products from 1:1 TCM:NaCl mixtures under air flow and a 10 °C/min ramp rate.</li> </ul>	Figure 3.2 – XRD p pestle,	atterns of TCM-KBr mixtures prepared by (a) mortar and (b) solution mixture, and (c) ball milling	64
<ul> <li>Figure 3.4 – XRD patterns of TCM-KSCN mixtures prepared by (a) mortar and pestle, (b) solution mixture, and (c) ball milling</li></ul>	Figure 3.3 – XRD p (b) sol	atterns of TCM-KI mixtures prepared by (a) mortar and pestle, aution mixture, and (c) ball milling.	65
<ul> <li>Figure 3.5 – IR spectra of C<sub>3</sub>N<sub>4</sub> precursors (a) TCM, (b) TCM-NaCl, (c) TCM-KBr, (d) TCM-KI, (e) TCM-KSCN, and (f) TCM-NH<sub>4</sub>SCN</li></ul>	Figure 3.4 – XRD p pestle,	atterns of TCM-KSCN mixtures prepared by (a) mortar and (b) solution mixture, and (c) ball milling	65
<ul> <li>Figure 3.6 - C<sub>3</sub>N<sub>4</sub> products prepared from (a) 3:1, (b) 2:1, (c) 1:1, and (d) 1:2 mixtures of TCM:NaCl.</li> <li>Figure 3.7 - TGA plots of C<sub>3</sub>N<sub>4</sub> products from 1:1 TCM:NaCl mixtures under air flow and a 10 °C/min ramp rate.</li> </ul>	Figure 3.5 – IR spec KBr, (	ctra of C <sub>3</sub> N <sub>4</sub> precursors (a) TCM, (b) TCM-NaCl, (c) TCM- d) TCM-KI, (e) TCM-KSCN, and (f) TCM-NH <sub>4</sub> SCN	67
Figure 3.7 – TGA plots of C <sub>3</sub> N <sub>4</sub> products from 1:1 TCM:NaCl mixtures under air flow and a 10 °C/min ramp rate	Figure 3.6 – C <sub>3</sub> N <sub>4</sub> p mixtur	roducts prepared from (a) 3:1, (b) 2:1, (c) 1:1, and (d) 1:2 res of TCM:NaCl.	69
	Figure 3.7 – TGA p flow a	lots of C <sub>3</sub> N <sub>4</sub> products from 1:1 TCM:NaCl mixtures under air nd a 10 °C/min ramp rate	71
Figure 3.8 – TGA plots of $C_3N_4$ products from salt exchange reactions under air flow at a 10 °C/min ramp rate	Figure 3.8 – TGA p flow a	lots of C <sub>3</sub> N <sub>4</sub> products from salt exchange reactions under air t a 10 °C/min ramp rate	72

Figure 3.9 –	IR spectra of water washed $C_3N_4$ products prepared from 500 °C tube decompositions of (a) TCM, (b) TCM-NaCl, (c) TCM-KBr, (d) TCM-KI, (e) TCM-KSCN, and (f) as-synthesized TCM-NH <sub>4</sub> SCN	'4
Figure 3.10 -	- XRD patterns of C <sub>3</sub> N <sub>4</sub> products from TCM-KBr (a) mortar and pestle mixtures, (b) solution mixtures, (c) ball milled mixtures, and (d) washed product	6
Figure 3.11 -	- XRD patterns of C <sub>3</sub> N <sub>4</sub> products from TCM-KI (a) mortar and pestle mixtures, (b) solution mixtures, (c) ball milled mixtures, and (d) washed product	6
Figure 3.12 -	- SEM images of (A) 3:1 TCM:NaCl product and its (B) water washed $C_3N_4$ product, (C) 1:1 TCM:NaCl product and its (D) water washed $C_3N_4$ product, (E) 1:2 TCM:NaCl product and its (F) water washed $C_3N_4$ product.	'8
Figure 3.13 -	<ul> <li>SEM images of washed C<sub>3</sub>N<sub>4</sub> products from (A) TCM-KBr, (B)</li> <li>TCM-KI, (C) TCM-KSCN, and (D) TCM-NH<sub>4</sub>SCN.</li> </ul>	'9
Figure 3.14 -	<ul> <li>Optical microscope images of C<sub>3</sub>N<sub>4</sub> products from 500 °C tube decomposition of (A) TCM-NaCl, (B) TCM-KBr, (C) TCM-KI, (D) TCM-KSCN, and (E) TCM-NH<sub>4</sub>SCN. Scale bar represents 2 mm</li></ul>	31
Figure 3.15 -	- UV-vis DRS data for TCM-salt products (A) reflectance data and (B) Tauc plots	52
Figure 3.16 -	- UV-vis analysis of the photodegradataion of MO dye using C <sub>3</sub> N <sub>4</sub> products from TCM-NaCl mixtures (A) UV-light data (B) linear rate data	34
Figure 3.17 -	<ul> <li>UV-vis analysis of the photodegradation of MO dye using washed</li> <li>C<sub>3</sub>N<sub>4</sub> products from TCM-salt mixtures (A) UV-light degradation</li> <li>data (B) linear rate data</li></ul>	86
Figure 3.18 -	- Stacked plots of photocatalytic hydrogen evolution using TCM-salt products. Each catalyst contained a targeted 1 wt% Pt co-catalyst. The UV lamp was turned on for 1 hour cycles, and the reactor was purged with argon for 10 minutes between each cycle	s9
Figure 4.1 –	Optical microscope images of (A) $C_3N_4@SiO_2$ (Schlenk), (B) $C_3N_4@SiO_2$ (Tube), (C) $C_3N_4@Al_2O_3$ (Schlenk), (D) $C_3N_4@Al_2O_3$ (Tube), (E) $C_3N_4@TiO_2$ (Schlenk), and (F) $C_3N_4@TiO_2$ (Tube). Scale bar represents 2 mm. 10	)1
Figure 4.2 –	TGA plots of $C_3N_4@SiO_2$ products under air flow, heated at a rate of 10 °C/min	)3

Figure 4.3 –	- TGA plots of C <sub>3</sub> N <sub>4</sub> @Al <sub>2</sub> O <sub>3</sub> products under air flow, heated at a rate of 10 °C/min.	103
Figure 4.4 –	- TGA plots of C <sub>3</sub> N <sub>4</sub> @TiO <sub>2</sub> under air flow, heated at a rate of 10 °C/min.	104
Figure 4.5 –	- Optical microscope images of (A) $C_3N_4@WO_3$ (Schlenk), (B) $C_3N_4@WO_3$ (Anneal), (C) $C_3N_4@WO_3$ (Tube), and (D) $WO_3@C_3N_4$ . Scale bar represents 2 mm.	105
Figure 4.6 –	- TGA plots of C <sub>3</sub> N <sub>4</sub> @WO <sub>3</sub> and WO <sub>3</sub> @C <sub>3</sub> N <sub>4</sub> under air flow, heated at a rate of 10 °C/min.	106
Figure 4.7 –	- IR spectra of (a) $C_3N_4@SiO_2(Schlenk)$ and (b) $C_3N_4@SiO_2(Tube)$ . The large broad peak centered at 1100 cm <sup>-1</sup> is characteristic of SiO <sub>2</sub>	109
Figure 4.8 –	- IR spectra of $C_3N_4$ @TiO <sub>2</sub> (Schlenk) and (b) $C_3N_4$ @TiO <sub>2</sub> (Tube). The large broad peak centered at 800 cm <sup>-1</sup> is characteristic of TiO <sub>2</sub>	110
Figure 4.9 –	- IR spectra of (a) $C_3N_4@$ (Schlenk) and (b) $C_3N_4@$ (Tube). The broad absorptions centered at 700 cm <sup>-1</sup> are characteristic of $Al_2O_3$	111
Figure 4.10	- IR spectra of (a) $C_3N_4@WO_3$ (Schlenk) (b) $C_3N_4@WO_3$ (anneal) (c) $C_3N_4@WO_3$ (Tube), and (d) $WO_3@C_3N_4$	112
Figure 4.11	- XRD patterns of (a) $C_3N_4$ @TiO <sub>2</sub> (Tube), (b) $C_3N_4$ @SiO <sub>2</sub> (Tube), and (c) $C_3N_4$ @Al <sub>2</sub> O <sub>3</sub> (Tube).	113
Figure 4.12	- XRD patterns of (a) $C_3N_4@WO_3(Schlenk)$ , (b) $C_3N_4@WO_3(anneal)$ , (c) $C_3N_4@WO_3(Tube)$ , and (d) $WO_3$ - $C_3N_4$	115
Figure 4.13	- SEM images of (A) $C_3N_4@SiO_2(Tube)$ , (B) $C_3N_4@Al_2O_3(Tube)$ , (C, D) $C_3N_4@TiO_2(Tube)$ , (E) WO <sub>3</sub> @C <sub>3</sub> N <sub>4</sub> , and (F) $C_3N_4@WO_3(Tube)$	117
Figure 4.14	– SEM images of ball milled (A, B) $C_3N_4$ -TiO <sub>2</sub> and (C, D) $C_3N_4$ -WO <sub>3</sub>	118
Figure 4.15	– UV-vis DRS data for C <sub>3</sub> N <sub>4</sub> , TiO <sub>2</sub> , and WO <sub>3</sub> (A) reflectance spectra and (B) Tauc plots.	120
Figure 4.16	<ul> <li>UV-vis DRS data for C<sub>3</sub>N<sub>4</sub>-metal oxide composites (A) reflectance spectra and (B) Tauc plots</li> </ul>	121
Figure 4.17	<ul> <li>UV-vis analysis of photodegradation of MO dye (A) UV light data and (B) UV linear rate data.</li> </ul>	123
Figure 4.18	<ul> <li>UV-vis analysis of photodegradation of MO dye (A) visible light data and (B) visible linear rate data</li> </ul>	125

Figure 4.19	<ul> <li>UV-vis analysis of photodegradation of MO dye using milled C<sub>3</sub>N<sub>4</sub>- TiO<sub>2</sub> and C<sub>3</sub>N<sub>4</sub>-WO<sub>3</sub> composites (A) UV light data and (B) UV linear rate data.</li> </ul>	27
Figure 4.20	<ul> <li>UV-vis analysis of photodegradation of MO dye using C<sub>3</sub>N<sub>4</sub>- TiO<sub>2</sub>(milled) and C<sub>3</sub>N<sub>4</sub>-WO<sub>3</sub>(milled) composites (A) visible light data and (B) visible linear rate data.</li> </ul>	28
Figure 4.21	- Photocatalytic hydrogen evolution using $C_3N_4$ @TiO <sub>2</sub> (Tube) with 1% Pt co-catalyst. The UV lamp was turned on for 1 hour cycles and the reactor was purged with argon for 10 minutes between each cycle	30
Figure 5.1 –	XRD patterns for select M(2%):TiO <sub>2</sub> products, rutile phase TiO <sub>2</sub> is marked with an asterisk	10
Figure 5.2 –	SEM images of (A) P25 TiO <sub>2</sub> and (B) Cu(1%):TiO <sub>2</sub> photoreaction product	41
Figure 5.3 –	Representative XPS spectra for 2p doublets for Ti from Cu(2%):TiO <sub>2</sub> , Mn from Mn(2%):TiO <sub>2</sub> , Fe from Fe(2%):TiO <sub>2</sub> , Co from Co(2%):TiO <sub>2</sub> , Ni from Ni(2%):TiO <sub>2</sub> , and Cu from Cu(2%)TiO <sub>2</sub> 14	14
Figure 5.4 –	UV-vis DRS data on M(2%):TiO <sub>2</sub> samples (A) reflectance spectra and (B) Tauc plots	18
Figure 5.5 –	UV-vis analysis of the photodegradation of MB dye with M(2%):TiO <sub>2</sub> samples (A) UV light data (B) linear rate data	50
Figure 5.6 –	UV-vis analysis of the photodegradation of MB dye with $M(1\%)M'(1\%)$ :TiO <sub>2</sub> samples (A) UV light data (B) linear rate data 15	52
Figure 5.7 –	Real-time hydrogen gas evolution under UV light illumination for (A) $M(1\%)$ :TiO <sub>2</sub> and (B) $M(1\%)$ ,M'(1%):TiO <sub>2</sub> . H <sub>2</sub> production was measured during one-hour illumination cycles followed by a 10 minute argon flush. The sequential illumination cycles began at the 30, 100, 170, and 240 minute marks	54
Figure 5.8 –	<i>In-situ</i> metal deposition and hydrogen evolution data, average rates for all 4 cycles ( $\mu$ mol/h) are Co 12.2, Ni 15.8, and Cu 28.3. H <sub>2</sub> production was measured during 1 hour illumination cycles followed by a 10 minute argon flush. Each illumination cycle began at the 30, 100, 170, and 240 minute marks	51
Figure 5.9 –	<i>In-situ</i> copper UV photodeposition and hydrogen evolution in 18 M $\Omega$ water. First UV lamp on cycle began near 30 min and ended at 90 min. Reactor headspace was flushed with argon between UV illumination runs. Second UV lamp on cycle began at 100 min and	

	ended at 160 min. Methanol is spiked into solution during the argon flush prior to the third cycle.	162
Figure 6.1 –	XRD pattern of $Ni_3N$ on $C_3N_4$ prepared from annealing $NiCl_2 \cdot 6H_2O$ on $C_3N_4$ at 500 °C under argon.	170
Figure 6.2 –	IR spectra of (a) $C_3N_4@SiO_2$ and (b) the 6M NaOH washed product to etch the SiO <sub>2</sub> .	173
Figure 6.3 –	SEM images of $C_3N_4@SiO_2$ after etching with NaOH.	174
Figure 6.4 –	Targeted photocatalytic oxidations of small organic molecules using $C_3N_4$ -based catalysts.	175
Figure 6.5 –	IR spectra of benzene vapor before and after 4 hour UV light (450 W Hg) illumination in the presence of $C_3N_4@TiO_2$	176
Figure A.1 –	Linear calibration for the quantification of hydrogen using RGA-MS	181
Figure A.2 –	Photo of Schlenk flask inside the photoreactor cabinet. The hose connected to the gas inlet adapter is connected to an argon Schlenk line in the adjacent hood. The side-arm is connected to the sampling capillary from the RGA-MS. The flask is clamped above the stir plate, positioned 20 cm from the UV lamp.	183
Figure A.3 –	Photographs of (A) prepared reduction tube and (B) prepared combustion tube.	188

#### CHAPTER 1

#### INTRODUCTION AND BACKGROUND

#### 1.1 Introduction

This thesis will describe the synthesis and modification of carbon nitrides ( $C_3N_4$ ) and titanium dioxide (TiO<sub>2</sub>) and their uses in photocatalysis. It is important to first give an appropriate background to the history of the development of these materials, including synthesis, properties, and applications. The general mechanism for photocatalytic processes will also be discussed.

#### 1.2 Carbon Nitrides

The history of carbon nitride materials dates back to 1834 and the synthesis of the polymer melon by Justus Liebig.<sup>1</sup> By heating mercury (II) thiocyanate, a brown polymeric solid formed with the elimination of mercury sulfide and carbon disulfide, a reaction commonly known as the Pharaoh's serpent. In the years following, reports of the composition of this solid varied around the ratio in the formula  $H_3C_6N_9$ .<sup>1-2</sup> It was some time before the structure of melon was further investigated. In 1922, work by Edward Franklin looked at the condensation of the class of ammono carbonic acids.<sup>3</sup> Starting with guanidine, heating the solid leads to a chain of ammonia elimination reactions. With each step the structure condensed, forming dimers and eventually cyclizing and expanding to the melon polymer (Figure 1.1). He also proposed the as yet-to-be-isolated "carbonic nitride" as the final stage of the deammonolysis reactions before complete thermal decomposition. In 1937, Linus Pauling followed up on Franklin's work, identifying the heptazine or tri-s-triazine unit ( $C_6N_7$ ) as the core of several

cyameluric salts.<sup>4</sup> Work on derivatives of this core structure continued through the years, but the investigation of the extended C-N structures beyond melon had stagnated.<sup>5-8</sup>



Figure 1.1 – Franklin's proposed pathway of deammonation products, which lead to the melon polymer.

Focus had shifted by the late 1980's, with Kouvetakis preparing nitrogencontaining graphite thin-film structures from the gas-phase reaction between pyridine and chlorine.<sup>9</sup> Sekine and co-workers studied high pressure pyrolysis of tertracyanoethylene giving a product with the formula  $C_{4.66}N$ .<sup>10</sup> This led to the consideration of nitrogen in diamond-like structures with carbon. In 1989, a theoretical study by Cohen and Liu proposed the  $\beta$ -C<sub>3</sub>N<sub>4</sub> phase based on the structure of  $\beta$ -Si<sub>3</sub>N<sub>4</sub>.<sup>11</sup> Their study predicted this sp<sup>3</sup> covalent structure to have a bulk modulus higher than diamond.<sup>11</sup> They proposed using high pressure and temperature reaction conditions to force a phase transition from recently published-amorphous carbon nitride films with the formula C<sub>3</sub>N<sub>4</sub>.<sup>12</sup> Early attempts at the formation of  $\beta$ -C<sub>3</sub>N<sub>4</sub> found only amorphous, sp<sup>2</sup> structures.<sup>13-14</sup> Further computational studies proposed that trying to fit the composition of  $C_3N_4$  into the dense 3-D  $\beta$ -C<sub>3</sub>N<sub>4</sub> structure was unfavorable due to N-N repulsions, which could only be relieved at higher C:N ratios (C<sub>4</sub>N<sub>3</sub>).<sup>15</sup>

While research continued on attempts at forming  $\beta$ -C<sub>3</sub>N<sub>4</sub>, attention turned to the amorphous  $sp^2 C_3N_4$  products. In a study by Kawaguchi and Nozaki, the polymerization of cyanuric trichloride ( $C_3N_3Cl_3$ ) by itself or mixed with melamine ( $C_3N_6H_6$ ) formed insoluble vellow powders with high thermal stability.<sup>16</sup> They noted similarities in the Xray diffraction peaks to those of graphite and hexagonal boron nitride, and proposed a structure of linked triazine rings. This graphitic carbon nitride (g-C<sub>3</sub>N<sub>4</sub>) was prepared from triazines<sup>17-21</sup> as well as the heptazine derivatives that followed from Pauling's work.<sup>22-23</sup> As  $g-C_3N_4$  could be formed from both triazine and heptazine-based precursors, more studies of the core unit and arrangement of these extended structures were conducted.<sup>24-25</sup> Solid state NMR studies by the Schnick group were able to show the change of the smaller linear molecules or triazine rings to heptazine from the isolation of the melem ( $C_6N_7(NH_2)_3$ ) intermediate.<sup>26-27</sup> Further supporting this, Holst in our group showed the base hydrolysis product of  $g-C_3N_4$ , prepared from a triazine precursor, was a crystalline heptazine salt. One possible ideal form of an ordered  $g-C_3N_4$  structure is shown in Figure 1.2, though actual products contain low amounts of hydrogen and are likely to have some degree of defects.

With the generally accepted structure now better understood, studies turned to examinations of chemical and physical properties. In addition to its robust chemical and thermal stability,  $g-C_3N_4$  exhibited interesting optical and luminescent properties.<sup>16, 28-30</sup> Furthermore, there are aspects of the structure that feature electronic properties and

different types of coordinating sites that have potential in catalytic or coordination applications.<sup>31</sup> Carbon nitride has a reported band gap of 2.7 eV, though this has been shown to vary slightly depending on the precursors and synthesis conditions.<sup>32-33</sup> It has grown in use in photocatalytic reactions, including water-splitting and oxidation of dyes and other organic compounds.<sup>34-36</sup> The principles of photocatalysis will be discussed in more detail near the end of this chapter.



Figure 1.2 – The generally accepted ideal structure of  $g-C_3N_4$ .

#### 1.3 Titanium Dioxide

Titanium dioxide (TiO<sub>2</sub>) is a white, semiconducting solid, with uses as a catalyst, component in batteries, or pigment.<sup>37-38</sup> One of the most significant contributions to the study of photocatalysis was the work of Fujishima and Honda, who demonstrated the photolysis of water into hydrogen and oxygen using TiO<sub>2</sub> in an electrochemical cell.<sup>39</sup>

TiO<sub>2</sub> has two common structures, anatase (Figure 1.3A) and rutile (Figure 1.3B). The band gap for these structures differ between 3.2 eV for anatase and 3.0 eV for rutile. Though different, both structures are restricted to the absorption of UV light in order to promote electrons from the valence band to the conduction band and utilize this charge separation for a photocatalytic process. To overcome these challenges, structural doping, dye sensitization, or compositing with other light-absorbing semiconductors or metal cocatalysts are methods used.<sup>40-42</sup>



Figure  $1.3 - \text{Unit cell structures of TiO}_2$  (A) anatase and (B) rutile. Ti atoms are in cyan and O atoms are in red.

#### 1.4 Principles of Photocatalysis

Following the work of Fujishima and Honda, many studies have gone into the design of catalysts for the photocatalytic production of hydrogen as a clean, renewable fuel source.<sup>43-44</sup> However, this is only one class of photocatalytic reaction. Photochemical reactions have been used for many different redox processes.<sup>45</sup> With a suitable semiconducting substrate, platinum and other precious metals can be photo-reduced onto the surface without the need of a reducing agent such as NaBH<sub>4</sub> or flowing hydrogen.<sup>46-47</sup> This photoreduction method has also be used as a way for removing metal ions from wastewater.<sup>48-50</sup> So far, all of the reactions described have been reduction processes, but photocatalytic oxidations are just as prominent. Similar to the removal of metal ions, the removal of organic dyes and industrial or pharmaceutical waste can be done with photocatalysis.<sup>51-54</sup> The oxidative degradation of this waste to more environmentally-benign products presents an alternate method of water remediation.

The requirements for a photocatalyst are a semiconductor (SC) with a band gap suitable to the light source, stability in the reaction media, and favorable chargeseparation kinetics. Band gaps for the semiconductors studied in the following chapters are summarized in Table 1.1.

Semiconductor	Bandgap (eV)	Absorption edge (nm)
$C_3N_4$	2.7	460
TiO <sub>2</sub>	3.2 (A), 3.0 (R)	388 (A), 413 (R)
WO <sub>3</sub>	2.6-2.8	443

Table 1.1. Band gaps and corresponding absorption edges for C<sub>3</sub>N<sub>4</sub>, TiO<sub>2</sub>, and WO<sub>3</sub>.

The general mechanism for the function of the catalyst is shown in Figure 1.3. First, the catalyst absorbs light with energy larger than its band gap. This excites an electron from the full valence band (VB) to the empty conduction band (CB). The promoted electron can then be transferred to target chemical reduction to form the intended product. The hole remaining in the valence band is used to oxidize another component, accepting an electron and allowing the cycle to continue. In most cases either the reduction or oxidation is the intended process, and a sacrificial oxidant or reductant will be added to the system to suppress the reverse reaction.



## Figure 1.4 – Generalized mechanism for photocatalysis at a semiconductor particle surface.

The photocatalytic oxidative degradation of organic dyes is often attributed to a radical hydroxide mechanism (Equations 1.1-1.4).<sup>55</sup>

$$hv + SC \rightarrow e^- + h^+$$
 (1.1)

$$H_2O + h^+ \rightarrow H^+ + OH^{\bullet}$$
(1.2)

$$OH' + R - H \rightarrow H_2O + R'$$
(1.3)

$$\mathbf{R}^{\bullet} + \mathbf{O}_2 \rightarrow \mathbf{ROO}^{\bullet} \rightarrow \rightarrow \mathbf{CO}_2 + \mathbf{H}_2\mathbf{O} + \text{salts}$$
(1.4)

The hole produced when the semiconductor is illuminated produces a hydroxyl radical from water. The hydroxyl radical abstracts hydrogen from the organic molecule, forming the radical organic compound. This interacts with atmospheric oxygen to form an organic peroxyl radical, and the process continues until complete oxidation to carbon dioxide, with aqueous inorganic salts as byproducts. In the case of the dyes to be studied in this thesis, methyl orange (Figure 1.5A) and methylene blue (Figure 1.5B), these salts will be nitrates and sulfates.



Figure 1.5 – Chemical structures of (A) methyl orange and (B) methylene blue dyes.

The water-splitting reaction is comprised of the reduction of protons to hydrogen and oxidation of water to oxygen (Equations 1.5 and 1.6). Based on the reduction potentials for each half reaction, the semiconductor used as catalyst must have a band gap greater than 1.23 eV.<sup>44</sup> While the semiconductors described in Table 1.1 all have large enough band gaps, only  $C_3N_4$  and TiO<sub>2</sub> have suitable CB levels to perform the reduction of hydrogen.<sup>56</sup> Methanol or triethanolamine are commonly used as the sacrificial oxidant for hydrogen evolution.<sup>44</sup> This limits the production of oxygen during the reaction, instead forming the higher organic oxidation products leading to carbon dioxide.

$$2 H^+ + 2 e^- \rightarrow H_2$$
  $E^0 = 0 V (vs. NHE)$  (1.5)

$$H_2O \rightarrow \frac{1}{2}O_2 + 2H^+ + 2e^ E^0 = 1.23 V (vs. NHE)$$
 (1.6)

#### 1.5 Thesis Overview

The remaining chapters of this thesis are as follows. Chapter 2 will detail the synthesis of  $C_3N_4$  from a trichloromelamine (TCM) precursor under several synthesis temperatures and reactor designs. Chapter 3 will detail the modification of the TCM precursor by incorporating different salts as hard templates to increase porosity in the product structure. Chapter 4 will detail the use of different oxides as supports for the formation of  $C_3N_4$ . Chapter 5 will detail the modification of TiO<sub>2</sub> by photoreduction of first row transition metals onto its surface. These chapters will feature characterization of the solids by infrared (IR) spectroscopy, thermogravimetric analysis (TGA), elemental analysis, powder X-ray diffraction (XRD), nuclear magnetic resonance (NMR), X-ray photoelectron spectroscopy (XPS), scanning electron microscopy (SEM), and UV-vis diffuse reflectance spectroscopy (UV-vis DRS). Each chapter will also examine the

photocatalytic degradation of dyes and the water-splitting reaction for hydrogen production. Chapter 6 will discuss overall conclusions and research areas with preliminary results that may be starting points for future projects.

#### **CHAPTER 2**

### SYNTHESIS AND ACTIVATION OF CARBON NITRIDE FROM TRICHLOROMELAMINE DECOMPOSITION

#### 2.1 Introduction

Graphitic carbon nitride  $(g-C_3N_4)$  has been extensively studied as a metal-free photocatalyst.<sup>55, 57-62</sup> It consists of a conjugated nitrogen-rich network that is typically formed through the thermal condensation and ring expansion of nitrogen containing precursors, with cyanamide (NC-NH<sub>2</sub>) being one of the simplest precursors.<sup>3, 31, 58</sup> Early synthetic work to obtain carbon nitrides utilized film growth from triazine-based precursors.<sup>14, 63</sup> Two precursors commonly used for C<sub>3</sub>N<sub>4</sub> growth are dicyandiamide (DCDA,  $C_2H_4N_4$ ) and melamine (triaminotriazine,  $C_3N_3(NH_2)_3$ ).<sup>64-66</sup> The generallyaccepted local structure of sp<sup>2</sup> hybridized carbon nitride materials is composed of linked tri-s-triazine (heptazine,  $C_6N_7$ ) units. The heptazine structural model is supported by Xray and neutron diffraction, IR, NMR, and XPS analysis.<sup>21, 26, 67-70</sup> These structurallydisordered carbon nitride materials are akin to condensed and crosslinked polymers, and early molecules to materials syntheses studied heptazine polymers as  $C_3N_4$ intermediates.<sup>22-23</sup> As described in Chapter 1, even C<sub>3</sub>N<sub>4</sub> grown from triazine-based precursors shows properties consistent a tri-s-triazine extended structure rather than a triazine network. The heptazine core structure is supported by the fact that  $C_3N_4$ chemical degradation in hot KOH yields only the heptazine oxyanion, and this anion has strong luminescent properties.<sup>69, 71</sup>

Carbon nitride materials have recently found extensive use in photocatalytic oxidation and reduction reactions, including environmental applications such as the

oxidative degradation of aqueous organic model pollutants<sup>72-73</sup> or gas phase photolysis of NO.<sup>74</sup> Applications in organic reactions include use as a photoinitiator in radical polymerization<sup>75</sup> and selective oxidation of alcohol or sulfonate functional groups.<sup>76-77</sup> The photocatalytic production of hydrogen from aqueous systems has been demonstrated with carbon nitrides decorated with Pt co-catalysts.<sup>34, 59, 78-79</sup> This has also extended to the use of  $C_3N_4$  in electrode design for photoelectrochemical hydrogen evolution reactions (HER).<sup>61, 80</sup> Silica templates have been utilized to generate mesoporous C<sub>3</sub>N<sub>4</sub> products with higher surface area that leads to greater photocatalytic activity.<sup>81-82</sup> Thermal or chemical exfoliation of bulk C<sub>3</sub>N<sub>4</sub> produces colloidal nanoflakes that lead to improved photocatalytic activity.<sup>83-84</sup> The level of structural order in the  $\pi$ -stacking direction is often on the order of a few nanometers and can vary with different synthetic methods. Such nanoscale changes influence physical properties of the carbon nitride materials. Given the typical structural disorder associated with these polymer-like structures, recent work has identified and exploited important carbon nitride defects to enhance its catalytic properties.<sup>85-87</sup>

This Chapter describes the structure and photocatalytic behavior of  $C_3N_4$ produced from the rapid and exothermic self-decomposition of a thermochemicallyunstable trichloromelamine (TCM,  $(C_3N_3(NHCl)_3)$  and compares its properties with carbon nitride materials produced from melamine and DCDA. Our previous work established the thermochemically-reactive nature of the TCM precursor under external heating (<250 °C) or in contact with a resistively-heated nichrome wire.<sup>21, 69</sup> These reports also showed that while TCM-derived  $C_3N_4$  is very stable in organic solvents, acids, and bases, it undergoes base hydrolysis when refluxed in 3 M KOH. The local  $C_3N_4$  structural information was consistent with a network structure built up of interconnected heptazine ( $C_6N_7$ ) rings. Though TCM has the advantage of a low reaction temperature, higher synthesis temperatures and post-reaction annealing were explored to compare their effect on photocatalytic activity. The rapid combustion-like decomposition and condensation of TCM leads to carbon nitrides that form as large nanoscale to small micrometer-sized plate-like particles made up of smaller fused regions. The TCMderived  $C_3N_4$  materials were utilized as photocatalysts in the oxidative degradation of methyl orange dye under UV and filtered visible light compared to  $C_3N_4$  from DCDA and melamine. Solution photoreduction reactions were used to add Pt, Pd, or Ag nanoparticle co-catalysts onto the  $C_3N_4$  surface for HER. HER photocatalysis was successfully performed in aqueous solution using triethanolamine as a sacrificial oxidant, and hydrogen gas evolution was conveniently measured in real-time using a portable mass spectrometer.

#### 2.2 Experimental

#### Reagents

Trichloromelamine ( $C_3N_3(NHCl)_3$ , TCM, 98%) was purchased from GFS Chemical and stored in an argon atmosphere glovebox. Other reagents used as received were melamine ( $C_3H_6N_6$ , MA, Aldrich 99+%), dicyandiamide ( $C_2H_4N_4$ , DCDA, Alfa Aesar, 99%), methanol (ACS certified, Fisher), triethanolamine (Alfa Aesar, 98+%), H<sub>2</sub>PtCl<sub>6</sub>·6H<sub>2</sub>O (99.9%, Strem), K<sub>2</sub>PdCl<sub>6</sub> (Baker & Co. Inc.), AgNO<sub>3</sub> (Aldrich, 99+%), methylene blue (high purity, Alfa Aesar), and methyl orange (85% dye content, Sigma Aldrich). Deionized water (18 MΩ) was prepared using a Photronix MiniQuad system.
## Thermal decomposition of TCM

Safety Warning: Most  $C_3N_4$  formation reactions involve the condensation of C-N bonded precursors that can liberate hazardous byproduct gases. Careful hood ventilation must be utilized and pressure calculations should be performed on closed systems that can rapidly increase in pressure during synthesis. Appropriate hood shields and PPE should be employed during synthesis and product workup and isolation.

## Rapid heating in Schlenk reactor

TCM (600 mg, 2.6 mmol) was loaded into a 100 mL round bottom flask and connected to a reflux condenser fitted with a gas inlet adapter. The reactor was purged with nitrogen on a Schlenk line prior to heating. The reactor was heated rapidly on a heating mantle to a set temperature of 300 °C. Typically, by 240 °C the decomposition will occur, noted by the color change of the solid in the flask and vigorous gas evolution. The byproduct gases were first sampled via syringe to transfer to a gas IR cell for characterization, and then the remaining gas in the reactor was vented into the fume hood. The bulk tan-orange product was recovered (~23% mass yield). This Schlenk synthesized product from TCM will be referred to as TCM-C<sub>3</sub>N<sub>4</sub>(S).

### Rapid heating in a Parr reactor

Inside a glovebox, TCM (5.0 g, 21.8 mmol) was loaded into a glass-lined stainless steel reactor (Parr Instruments, 125 mL, model 4752). The reactor was sealed using a wrench on the six hex bolts before removal from the glovebox. The reactor was put into a fitted Glass-col heating mantle and heated to 250 °C at 10 °C/min. The external temperature, internal temperature, and internal pressure from the thermocouple and pressure gauge readings were written down at timed intervals as the system was heated.

Near 220 °C there was a spike in internal temperature (from 120 °C to 250 °C) and pressure (1000 psi). As the system cooled, the pressure equilibrated to 350 psi. The reactor was slowly vented into the fume hood and then disassembled. A bright orange, sponge-like solid was collected from the inside of the reactor (20% mass yield). This Parr-synthesized product from TCM will be referred to as TCM-C<sub>3</sub>N<sub>4</sub>(P).

## Single-step high temperature heating in a Schlenk tube

TCM (400 mg, 1.7 mmol) was loaded into a 1.5 cm diameter test tube cut to 5 cm long, and put into a Schlenk tube with and fitted with a gas inlet adapter. An oil bubbler was connected to the side arm for pressure venting and degassed with nitrogen. The tube was clamped in position above a vertically oriented furnace, set to heat to 500 °C at a rate of 10 °C/min and held for 1 hour. Gas evolution was observed from the bubbler near 180°C. After the hold, the tube was cooled and vented of byproduct gases. A porous orange pellet was recovered from the insert (~20% mass yield). This tube-synthesized product from TCM will be referred to as TCM-C<sub>3</sub>N<sub>4</sub>(T).

# Annealing of carbon nitrides

As synthesized  $C_3N_4$  (Schlenk or Parr methods) was treated at higher temperatures to further anneal the structure. Typically, 100-300 mg of product was loaded into an alumina boat, which was put inside a Pyrex tube closed on one end. The tube was connected to a Schlenk line, evacuated, and back-filled with argon. The tube was placed into a clamshell furnace and heated to between 450 °C and 500 °C at a rate of 5 °C/min and held for 1-3 hours. Following annealing, 60-90% of the starting mass was recovered from the boat and a transported film of white material was observed at the cool end of the tube. These anneals of Schlenk and Parr synthesized products from TCM will be referred to as TCM- $C_3N_4(S/A)$  and TCM- $C_3N_4(P/A)$ .

## Synthesis of reference carbon nitrides

Melamine (MA) or dicyandiamide (DCDA) (400 mg, 3.2 mmol MA, 4.75 mmol DCDA) was loaded into a 1.5 cm diameter test tube cut to 5 cm long, and put into a Schlenk tube with and fitted with a gas inlet adapter. An oil bubbler was connected to the side arm for pressure venting and degassed with nitrogen. The tube was clamped in position above a vertically-oriented furnace, set to heat to 500 °C at a rate of 10 °C/min and held for 1 hour. After the hold, the tube was cooled and vented of byproduct gases. A hard yellow solid was recovered from the insert (~34% mass yield from MA, ~45% mass yield from DCDA). This tube synthesized product from TCM will be referred to as MA-C<sub>3</sub>N<sub>4</sub>(T) and DCDA-C<sub>3</sub>N<sub>4</sub>(T).

## Ball milling of carbon nitrides

To examine the effect of macroscopic surface area changes on carbon nitride properties, selected products were ball milled using a high-energy ball mill (FormTech FTS 1000). Solid (100 mg) was loaded into a 5 mL stainless steel milling jar with 2x 5 mm steel balls. Milling was performed at 1200 rpm for a duration of 5 minutes.

## **Characterization**

FT-IR spectra were collected on a Nicolet Nexus 670. Solid samples were prepared as KBr pellets for data collection. Gas samples were analyzed by transferring 5 mL of gas from the reaction vessel into a 10 cm long gas IR cell with KBr windows. Elemental analysis was performed on a Perkin Elmer 2400 Series II CHNS/O Analyzer,

with 1-2 mg of sample in crimped tin capsules. Powder X-ray diffraction was collected on a Bruker D8 system (source Cu Ka, 5-80°, 0.050 °/step) with samples deposited on glass slides. Solid state <sup>13</sup>C and <sup>15</sup>N CP-MAS-NMR spectra were collected on a 500 MHz Bruker spectrometer. Thermogravimetric analysis was performed on a Seiko Exstar 6300 TGA-DTA. Samples were heated up to 1000°C at a rate of 10°C/min under air or argon flow. Scanning electron microscopy images were collected on a Hitachi S-4800 FE-SEM ( $v_{acc}$ =1.8kV). X-ray photoelectron spectra for samples pressed onto indium foil were collected on a Kratos Axis Ultra DLD XPS. Diffuse reflectance spectra were collected on a Cary 5000 Series UV-Vis-NIR Spectrophotometer (300-800 nm, 10 nm/s). Solid samples were loaded into a round compression sample holder with a 1 cm diameter sampling area. Kubelka-Munk conversions were calculated from the Cary software, and Tauc plots were generated using  $[F(R)hv]^{1/2}$ . Indirect band gap values were approximated by extrapolating the linear portion of the absorption edge. BET surface area measurements were collected on a Quantachrome Nova 1200, using 100-200 mg of ground solid sample. ICP-OES data was collected using a Varian 720-ES. Samples (5-10 mg) were dissolved in a heated mixture of 5 mL concentrated  $H_2SO_4$  and 1 mL concentrated HNO<sub>3</sub> and diluted in 25 mL volumetric flasks with 5% HNO<sub>3</sub>.

#### Photocatalytic reactions

## Oxidative degradation of organic dyes

Dyes studied for photochemical reactions were methyl orange ( $C_{14}H_{14}N_3NaO_3S$ ) and methylene blue ( $C_{16}H_{18}ClN_3S$ ). A stock  $10^{-3}$  M dye solution was prepared by dissolving the solid dye in 18 M $\Omega$  water. The stock solutions were diluted to  $4 \cdot 10^{-5}$  M for methyl orange and  $2 \cdot 10^{-5}$  M for methylene blue to fit the peak at  $\lambda_{max}$  in the range ( $\lambda_{MO}$ =504 nm,  $\lambda_{MB}$ =665 nm, A<1.50) of the UV-vis spectrophotometer (Agilent 8453). To 30 mL beakers, 10 mL of dye solution and 10 mg of catalyst was added. An additional beaker containing only dye was used as to evaluate self-degradation in the absence of catalyst. Samples were stirred a minimum of 30 minutes in the dark to allow dye absorption equilibrium to be reached. Samples were exposed to UV light (Ace-Hanovia, 450 W Hg) in 15-30 minute intervals in air. After each irradiation period, the catalysts were separated from the dye solution by centrifugation and the UV-vis spectrum of each solution was collected to quantify the remaining concentration of dye. Experiments were typically carried out for 2 hours of total UV irradiation. For visible light experiments, 400 nm long pass filters (Edmund Optics) were placed between the samples and the lamp to filter light below 400 nm, and total illumination times were extended.

### Photocatalytic hydrogen evolution via water-splitting

Catalysts were used as synthesized or with Pt, Pd, or Ag co-catalysts. Metals were photoreduced onto the  $C_3N_4$  surface under UV light using H<sub>2</sub>PtCl<sub>6</sub>, K<sub>2</sub>PdCl<sub>6</sub>, or AgNO<sub>3</sub> as the metal source.  $C_3N_4$  was suspended in a 50% aqueous MeOH solution containing an amount of the metal salt to target 1-3 wt% of reduced metal on  $C_3N_4$ . The solution was degassed with argon and then exposed to UV light for 2 hours. The solid was collected by centrifugation, rinsed with MeOH, and dried in air.

For hydrogen evolution tests, 10 mg of catalyst ( $C_3N_4$  or M(x%): $C_3N_4$ ) was dispersed in a 10 mL solution of 10% aqueous triethanolamine ( $C_6H_{15}NO_3$ ) contained in a 50 mL Schlenk flask. The solution was degassed with argon and connected to a Residual Gas Analysis Mass Spectrometer (RGA-MS, Stanford Research Systems QMS 300 series) by capillary to the side arm. Data were collected as a pressure vs. time scan, monitoring the masses for nitrogen, argon, hydrogen, water, and oxygen. The reactor was exposed to UV light in 1 hour intervals and was vented with an argon purge between each UV run in order to return hydrogen concentrations to baseline levels. The molar amounts of hydrogen produced were determined using a pressure signal calibration based on known concentration mixtures of  $H_2/Ar$  (addition details in the Appendix).<sup>88</sup>

#### 2.3 Results and Discussion

## Formation of carbon nitride from the decomposition of TCM

The thermal decomposition of trichloromelamine was studied using three different reaction methods, each suited for different scales or heating methods. In the glass Schlenk reactor, the physical changes were observed directly (e.g. color change and gas evolution) at reaction temperature. In the Parr reactor the internal temperature and pressure were monitored. As the heating mantle reached 250 °C, the interior temperature was 105 °C immediately before the reaction occurred. The internal temperature jumped to 230 °C, but quickly cooled back to 150 °C within a minute, indicating a sudden, brief exothermic decomposition. This temperature change also paired with a pressure increase to 1000 psi at the instant of decomposition, but dropped to 450 psi after the initial shock. Once the reactor had cooled back to room temperature, the internal pressure had equilibrated to 350 psi. Thermogravimetric analysis of TCM under argon flow (Figure 2.1) supports the observations from these reaction methods, showing the first major mass loss paired with an exothermic event near 200 °C. Rapid mass loss continues for the next

20-30 °C as the stable carbon nitride product forms. Mass loss is gradual at higher temperatures due to the slow volatilization of  $C_x N_y$  fragments from the solid, until complete decomposition by 700 °C. The thermal behavior of TCM from the TGA data helped to determine suitable conditions for annealing experiments.



Figure 2.1 – TGA and DTA plots of TCM decomposition under argon flow and a 10 °C/min ramp rate.

The exothermic peak in the DTA matches with the gas evolution that occurs during the decomposition reaction. By transferring this gas to an IR cell, the IR signatures of the gases present could be identified. From the gas phase IR spectrum (Figure 2.2) several sets of peaks were observed. The broad peaks from 3245-3380 cm<sup>-1</sup> correspond to HCN, the sharp peaks from 2600-3070 cm<sup>-1</sup> correspond to HCl, the peaks at 2200 cm<sup>-1</sup> correspond to  $C_2N_2$ , and the peaks at 2100 cm<sup>-1</sup> correspond to CNCl. Further analysis of the product gases by RGA-MS also showed the presence of  $Cl_2$  (m/z 70 for  ${}^{35}\text{Cl}{}^{35}\text{Cl}$ , m/z 72 for  ${}^{35}\text{Cl}{}^{37}\text{Cl}$ ) which could not be observed by IR spectroscopy and also confirmed the presence of the gases observed by IR.



Figure 2.2 – FT-IR spectrum of gas phase products from TCM decomposition.

For the Parr reaction where 22 mmol of TCM was used, ~100 mmol of gaseous products were generated based on the final pressure of the reactor. This mmol estimate is consistent with the expected moles of gaseous C/N/H/Cl containing byproducts shown in Figure 2.3.



Figure 2.3 – Decomposition of TCM to an ideal  $C_3N_4$  structure of condensed hepatazines.

# Solid state characterization of carbon nitride from TCM decomposition

Infrared analysis of the different precursors used to prepare carbon nitride show distinct characteristics in some regions (Figure 2.4). Each of the precursors have N-H stretches above 3000 cm<sup>-1</sup> as well as C-N stretches from 1200-1700 cm<sup>-1</sup>. Unique to DCDA is the peak at 2100 cm<sup>-1</sup> for the C-N triple bond of a nitrile group.



Figure 2.4 – FT-IR spectra of precursors used to prepare  $C_3N_4$  (a) TCM, (b) MA, and (c) DCDA.

The IR spectra of the  $C_3N_4$  products (Figure 2.5) show many similarities between the different decomposition methods and precursors used. The broad band from 3200-3500 cm<sup>-1</sup> corresponds to N-H stretches from terminal groups in the  $C_3N_4$  structure. The bands from 1200-1700 cm<sup>-1</sup> correspond to C-N (1200-1480 cm<sup>-1</sup>) and C=N (1570-1640 cm<sup>-1</sup>) stretches in the ring structure. The sharp peak near 810 cm<sup>-1</sup> is characteristic of the ring breathing mode for conjugated heptazine heterocycles. These peaks are consistent with those observed for  $C_3N_4$  produced from TCM using rapid, low temperature methods as well as products produced from melamine or cyanamide in air.<sup>21, 89-90</sup> As compared to Parr TCM, DCDA, and melamine products, the TCM- $C_3N_4(T)$  product has less welldefined peaks, particularly near 1200 cm<sup>-1</sup> region, which may indicate some lower structural ordering in this  $C_3N_4$  product.



Figure 2.5 – FT-IR spectra of (a) TCM- $C_3N_4(S)$ , (b) TCM- $C_3N_4(P)$ , (c) TCM- $C_3N_4(T)$ , (d) TCM- $C_3N_4(P/A)$ , (e) DCDA- $C_3N_4(T)$ , and (f) MA- $C_3N_4(T)$ .

The similarities in the product spectra are consistent with the literature proposed condensation pathway of the small linear or cyclic molecular precursors. The heptazine fragment, as the core repeat unit of the idealized  $C_3N_4$  structure, is supported by the IR spectra.

Solid state NMR spectra were collected on TCM-C<sub>3</sub>N<sub>4</sub>(S) for both <sup>13</sup>C and <sup>15</sup>N nuclei (Figure 2.6). In the <sup>13</sup>C spectrum two resonances were present at 156 ppm and 163 ppm. Assigning these signals to the core heptazine unit fits the peak at 156 ppm to the internal ring carbon and the peak at 163 ppm to the external ring carbon. The <sup>15</sup>N spectrum showed one well resolved peak at -200 ppm and two overlapping peaks at -250 ppm and -280 ppm. The single peak at -200 ppm can be assigned to the bridging nitrogen (C-N=C) in the heptazine ring. The peak at -250 ppm is assigned to the nitrogen located in the center of the ring, while the remaining peak at -280 ppm is assigned to the nitrogen that links heptazine units. The nitrogen atoms in these last two peaks share a similar sp<sup>2</sup> bonding environment to three carbon atoms, fitting with the proximity of the chemical shifts. Peak assingments with local heptazine repeat unit are summarized in Table 2.1.

Local structure	Position	Chemical shift (ppm)	Reference
N <sub>3</sub>	C1	163	164.6 <sup>69</sup>
	C2	156	156.6 <sup>69</sup>
	N1	-200	-197 to -205 <sup>26</sup>
$ \begin{bmatrix} 1 \\ 1 \\ 1 \end{bmatrix} \begin{bmatrix} 1 \\ 1 \\ 1 \end{bmatrix} $	N2	-250	-234 <sup>26</sup>
N3 N1 N1 N3	N3	-280	$-267 \text{ to } -281^{26}$

Table 2.1. NMR assignments and shifts for TCM- $C_3N_4(S)$ .



Figure 2.6 – CP-MAS-NMR spectra for TCM-C<sub>3</sub>N<sub>4</sub>(S), (a)  $^{13}$ C, (b)  $^{15}$ N.

Powder X-ray diffraction of the  $C_3N_4$  products from the different methods showed the same level of similarity as the IR spectra (Figure 2.7). The major peak at 27.5° (d=3.24 Å) is attributed to the spacing between the tri-s-triazine linked layers or sheets in the  $C_3N_4$  structure. This peak appears sharper in most of the products heated to 500 °C, though it was consistently broader for TCM- $C_3N_4$ (T). A weaker secondary peak at 13° (d=6.81 Å) is sometimes detectable and can be assigned to the in-plane repeat of tri-striazine units. The lower intensity or absence of this peak in some samples may indicate that the in-plane ordering is more random during the condensation reactions to form the  $C_3N_4$  structure.



Figure 2.7 – Powder X-ray diffraction of  $C_3N_4$  products (a) TCM- $C_3N_4(S)$ , (b) TCM- $C_3N_4(P)$ , (c) TCM- $C_3N_4(T)$ , (d) TCM- $C_3N_4(P/A)$ , (e) MA- $C_3N_4(T)$ , and (f) DCDA- $C_3N_4(T)$ .

# Compositional analysis of carbon nitride

Though carbon nitride is referred to with the formula  $C_3N_4$ , the exact C:N ratio may vary, and some amount of hydrogen usually remains in the product. CHN combustion analysis showed the composition data as shown in Table 2.2. All products are more nitrogen rich than the general empirical formula for  $C_3N_4$ , instead fitting  $C_3N_{4+x}$  (0.2<x<0.8), consistent with both our previous TCM- $C_3N_4$  products and  $C_3N_4$  prepared from other percursors.<sup>21, 69, 83-84, 91</sup> The products also contained hydrogen, likely bound to prevalent NH<sub>2</sub> terminal sites in the structure. The distribution of hydrogen could also cause disruption in the in-plane ordering or conjugation in the long-range structure through formation of secondary NH bridges between heptazine units. The higher-temperature decomposition methods and annealing processes led to  $C_3N_4$  product with a higher total CHN content, indicating that volatile fragments eliminated during heating may contain other elements such as O or Cl. This is consistent with the observed transport of NH<sub>4</sub>Cl during annealing experiments, removing chlorine from the low temperature TCM- $C_3N_4$ (P) product. The remaining mass content may be oxygen from either adsorbed water or surface oxidation.

Product (Method)	C wt%	N wt%	H wt%	Total wt % CHN	Formula
TCM-C <sub>3</sub> N <sub>4</sub> (S)	30.47	57.15	2.41	90.03	$C_{3}N_{4.8}H_{2.8}$
TCM-C <sub>3</sub> N <sub>4</sub> (S/A)	33.55	58.66	1.81	94.02	$C_3N_{4.5}H_{1.9}$
TCM-C <sub>3</sub> N <sub>4</sub> (P)	31.43	55.07	1.93	88.44	$C_{3}N_{4.5}H_{2.2}$
TCM-C <sub>3</sub> N <sub>4</sub> (P/A)	34.27	61.89	1.76	97.92	$C_{3}N_{4.6}H_{1.8}$
TCM-C <sub>3</sub> N <sub>4</sub> (T)	34.41	58.65	1.84	94.90	$C_3N_{4.4}H_{1.9}$
MA-C <sub>3</sub> N <sub>4</sub> (T)	33.85	2.16	60.61	96.62	$C_3N_{4.6}H_{2.2}$
DCDA-C <sub>3</sub> N <sub>4</sub> (T)	34.03	2.41	60.02	96.46	$C_{3}N_{4.5}H_{2.5}$

Table 2.2. CHN elemental analysis of  $C_3N_4$  products.

X-ray photoelectron spectroscopy was used to gain additional compositional data as well as information about the bonding environment around carbon (C 1s) and nitrogen (N 1s) as well as oxygen (O 1s) and chlorine (Cl 2p) regions (Figure 2.8). Two peaks were present in the C 1s spectrum at 284.5 eV and 288 eV. The peak at 284.5 eV had a lower intensity and is likely representative of adventitious surface carbon contamination commonly detected in XPS samples. The primary peak at 288 eV agrees with reported  $C_3N_4$  values and can be assigned to carbon in an sp<sup>2</sup> coordination environment as would be expected of an interior or exterior  $sp^2$  carbon in the heptazine subunit (either C1 or C2) from Table 2.1).<sup>21, 67</sup> The N 1s spectrum shows a major peak at 398 eV with a shoulder at higher energy that is deconvoluted into two peaks at 400 eV and 401 eV. The peak at 398 eV can be assigned to the bridging C-N-C on the exterior of the heptazine ring. The higher energy shoulder peaks fit to N-(C)<sub>3</sub>, H-N-(C)<sub>2</sub>, or (H)<sub>2</sub>-N-C and potentially some surface N-O further at 404.5 eV.<sup>21,67</sup> There are two different three-coordinate nitrogen sites in the ideal layered  $C_3N_4$  structure as shown in Table 2.1, one in the center of the heptazine ring (N2) and the second acting as the linker between heptazine units (N3).



Figure 2.8 – XPS C 1s, N 1s, O 1s, and Cl 2p spectra for TCM-C<sub>3</sub>N<sub>4</sub>(P/A).

While the bulk elemental analysis data above shows that the  $C_3N_4$  materials are mainly C, N, and H, XPS also showed the presence of surface O and Cl. Products prepared from all methods have some degree of oxygen present on the surface. Trace chlorine was mainly present in TCM- $C_3N_4(S)$  and TCM- $C_3N_4(P)$  products and was generally smaller or absent from the annealed  $C_3N_4$  or product prepared with extended heating at 500 °C. This agrees with the increase of total CHN content in annealed products and observation of NH<sub>4</sub>Cl transport. The surface oxygen content also decreased significantly in the TCM- $C_3N_4(P/A)$  product. The surface compositions measured by XPS fit a formula of  $C_3N_{3+x}$  (0.4<x<0.6) that has a higher C:N ratio than the bulk elemental analysis (~C<sub>3</sub>N<sub>4+x</sub> with x  $\ge$  0.2). Since these compositional analysis methods measure different parts of the sample (bulk versus surface), it is not surprising that these values differ. The higher amounts of total carbon on the surface include the small adventitious carbon peak at ~284 eV and, if it is removed from the XPS surface composition analysis, then the XPS and CHN data are similar (*e.g.*, for TCM-C<sub>3</sub>N<sub>4</sub>(P/A) C<sub>3</sub>N<sub>4.2</sub> vs. C<sub>3</sub>N<sub>3.4</sub>). The low levels of oxygen and chlorine content are also slightly higher than our previously reported values; however, this also represents a difference between bulk and surface compositional analysis.<sup>21</sup>

Sample	C1s	N1s	O1s	Cl2p	XPS Composition
TCM-C <sub>3</sub> N <sub>4</sub> (S)	284.33, 286.04, 288.09	398.53, 399.73	531.5	197.14, 198.42, 199.81	C <sub>3</sub> N <sub>3.5</sub> O <sub>.4</sub> Cl <sub>0.3</sub>
TCM-C <sub>3</sub> N <sub>4</sub> (S/A)	284.5, 286.03, 288.02	398.49, 399.86, 401.07, 404.6	532.04	196.85, 198.4, 200.31	$C_3N_{3.4}O_{0.2}Cl_{0.01}$
TCM-C <sub>3</sub> N <sub>4</sub> (P)	284.34, 288.18	398.34, 399.92	531.38	196.94, 198.34, 199.49	$C_3N_{3.6}O_{0.8}Cl_{0.2}$
TCM-C <sub>3</sub> N <sub>4</sub> (P/A)	284.62, 286.22, 288.05	398.53, 399.86, 404.5, 401.1	531.88	Not detected	$C_3N_{3.4}O_{0.2}$
TCM-C <sub>3</sub> N <sub>4</sub> (T)	284.68, 287.94, 288.07, 291.46	398.52, 399.94	532.05, 534.49	197.15, 198.71, 200.83	$C_3N_{2.6}O_{0.8}Cl_{0.05}$

Table 2.3. XPS signal positions and surface composition for C<sub>3</sub>N<sub>4</sub>.

## Microstructure and morphology

The analysis of carbon nitride morphologies by SEM shows that TCM-derived  $C_3N_4$  products form large aggregates above 1  $\mu$ m (Figure 2.9). These appear to have smooth faces but have fractures along the edges that reveal large nanometer-size features. The produced carbon nitride appears to grow in an aggregated mass of irregularly-shaped plate or sheet-like structures. The TCM- $C_3N_4(S)$  products have a coral-like appearance of interconnected fragments. The surfaces also appear to have shallow pores formed from the gas evolution during the decomposition reaction. When compared with TCM- $C_3N_4(P)$ , TCM- $C_3N_4(P/A)$  exhibits many of the same features, including ~100-200 nm sized plate-like particles. There is some opening up of pores and perhaps some smoothing of surface features on some particles. The TCM- $C_3N_4(T)$  product prepared by direct inert tube heating at 500 °C shows larger, more extended structures on the order of 10-40  $\mu$ m that contain large pores or depressions on the order of ~500 nm, likely a consequence of rapid gas evolution with precursor condensation to form the carbon nitride. The surfaces of these larger structures contain smooth 1-5 µm fragments with rough edges that are fairly thin (~100 nm).



Figure 2.9 – SEM of (A,B) TCM-C<sub>3</sub>N<sub>4</sub>(S), (C,D) TCM-C<sub>3</sub>N<sub>4</sub>(P), (E,F) TCM-C<sub>3</sub>N<sub>4</sub>(T), (G,H) TCM-C<sub>3</sub>N<sub>4</sub>(P/A).

BET surface area analysis showed that TCM-C<sub>3</sub>N<sub>4</sub>(P) has a low surface area of  $3.3 \text{ m}^2 \text{ g}^{-1}$ , comparable to our previously reported  $5 \text{ m}^2 \text{ g}^{-1}$  for a TCM product decomposed via hot wire initiation, and consistent with the packed and aggregated flakes observed.<sup>21</sup> Following ball-milling, this increased to  $6.2 \text{ m}^2 \text{ g}^{-1}$ . Higher temperature treatment resulted in a slight increase to  $4.2 \text{ m}^2 \text{ g}^{-1}$  for TCM-C<sub>3</sub>N<sub>4</sub>(P/A) and  $4.1 \text{ m}^2 \text{ g}^{-1}$  for TCM-C<sub>3</sub>N<sub>4</sub>(T). The surface area of TCM-C<sub>3</sub>N<sub>4</sub> products are lower than that measured for DCDA-C<sub>3</sub>N<sub>4</sub>(T) formed in the same experimental setup at 7.6 m<sup>2</sup> g<sup>-1</sup>, which is comparable to reported values.<sup>79</sup>

# **Optical properties**

The color of carbon nitride ranges from light yellow to darker orange or tan depending on the synthesis conditions (Figure 2.10). Diffuse reflectance UV-vis was used to determine the optical absorption properties and band gaps of the C<sub>3</sub>N<sub>4</sub> products. The reflectance data (Figure 2.11A) can be converted to absorbance data using the Kubelka-Munk function, F(R), and modified to fit a Tauc plot (Figure 2.11B). Plotting this against energy (eV), the band gap of a material can be extracted from an extrapolation of linear regions of the data giving  $E_g$  values of 2.2, 2.45, 2.35, 2.42, 2.62, and 2.65 eV for TCM-C<sub>3</sub>N<sub>4</sub>(S), TCM-C<sub>3</sub>N<sub>4</sub>(P), TCM-C<sub>3</sub>N<sub>4</sub>(P/A), TCM-C<sub>3</sub>N<sub>4</sub>(T), MA-C<sub>3</sub>N<sub>4</sub>(T), and DCDA-C<sub>3</sub>N<sub>4</sub>(T), respectively. The band gaps of the DCDA and melamine products are very similar and are comparable to values reported for carbon nitrides,<sup>32</sup> and they share a light yellow color as shown in Figure 2.10E and 2.10F. In contrast, the products formed from TCM are consistently darker in color, ranging from bright orange to dark tan. The absorption edges of the TCM products are not as sharp and extend

further into the visible region with low absorbance continuing to ~600 nm, consistent with the darker orange colorations of the TCM- $C_3N_4$  products. A second lower energy absorption event near 2 eV is also observed for the TCM products, which may arise from structural defects giving rise to intermediate band absorption. Similar secondary absorption edges have been reported for annealed carbon nitrides.<sup>84</sup>



Figure 2.10 – Optical microscope images of (a) TCM- $C_3N_4(S)$ , (b) TCM- $C_3N_4(P)$ , (c) TCM- $C_3N_4(T)$ , (d) TCM- $C_3N_4(P/A)$ , (e) MA- $C_3N_4(T)$ , and (f) DCDA- $C_3N_4(T)$ .



Figure 2.11 – DRS UV-vis of  $C_3N_4$  products (a) raw reflectance spectra and (b) Tauc plots.

# Photocatalytic oxidation of organic dyes

The degradation of methylene blue was studied to compare the activity of TCM- $C_3N_4$  products. Following the dark stir, there was a 20-40% lost in initial absorbance intensity (Figure 2.12A). This was also evident in the  $C_3N_4$  powders that developed a green tint (blue dye mixing with the bulk orange color). Upon UV illumination, the carbon nitrides steadily degrade methylene blue, evidenced by solution decolorization. After one hour of total irradiation, the as-synthesized Schlenk and Parr products reduced the dye to below 20% of its initial concentration. The annealed product had higher dark adsorption and a slower initial degradation rate, but reached the same level of total degradation at the end of 100 minutes. The linear rate fits (Figure 2.12B) and calculated rate constants (Table 2.4) show that TCM- $C_3N_4(P)$  was most effective at degradation of MB.

TCM-C3N4(S)TCM-C3N4(P)TCM-C3N4(P/A)UV light0.04030.05280.0175(R<sup>2</sup> = 0.982)(R<sup>2</sup> = 0.991)(R<sup>2</sup> = 0.999)

Table 2.4. Rate constants (min<sup>-1</sup>) for UV light degradation of MB dye using TCM- $C_3N_4$  products.



Figure 2.12 - UV light degradation of methylene blue with TCM-C<sub>3</sub>N<sub>4</sub> products, (A) degradation plots and (B) linear rate fits.

The TCM-C<sub>3</sub>N<sub>4</sub>(P) product, being most active under UV light, was tested against MA-C<sub>3</sub>N<sub>4</sub>(T) and DCDA-C<sub>3</sub>N<sub>4</sub>(T) under visible light illumination using 400 nm long pass filters. As in the unfiltered experiment, there was some degree of dark adsorption of the dye on the catalyst surface. Activity was lower in the absence of UV light, with all C<sub>3</sub>N<sub>4</sub> samples reaching 40-60% of initial dye concentration after 150 minutes (Figure 2.13A). TCM-C<sub>3</sub>N<sub>4</sub>(P) showed very similar MB dye degradation (~30%) as compared to the DCDA-C<sub>3</sub>N<sub>4</sub>(T) and MA-C<sub>3</sub>N<sub>4</sub>(T) products, after account for their higher initial dark adsorption. The self-degradation of methylene blue is minimal here compared to the unfiltered experiment. Estimated rate constants (Table 2.5) based on linear fits (Figure 2.13B) of first order kinetic data for the degradation of methylene blue are between 6.3% of the unfiltered UV experiments for TCM-C<sub>3</sub>N<sub>4</sub>(P).

Table 2.5. Rate constants (min<sup>-1</sup>) for visible light degradation of MB dye using TCM, MA, and DCDA- $C_3N_4$  products.

	TCM-C <sub>3</sub> N <sub>4</sub> (P)	MA-C <sub>3</sub> N <sub>4</sub> (T)	DCDA-C <sub>3</sub> N <sub>4</sub> (T)
Visible (>400 nm)	0.0033	0.0038	0.0045
	(R <sup>2</sup> = 0.962)	(R <sup>2</sup> = 0.993)	(R <sup>2</sup> = 0.995)



Figure 2.13 – Visible light degradation of methylene blue with C<sub>3</sub>N<sub>4</sub> products, (A) degradation plots and (B) linear rate fits.

The activities for the oxidative degradation of methyl orange (MO) dye were compared for the ball-milled TCM- $C_3N_4$  and DCDA- $C_3N_4$  products. MO acts as a pH indicator, and its  $\lambda_{max}$  will change when the solution pH is below 4. Initial photooxidation tests with C<sub>3</sub>N<sub>4</sub> prepared from TCM were performed in neutral conditions, however, after the dark stir period, the dye solution for some samples changed to a darker red-orange color indicating an increase in solution acidity. The Parr products may contain some residual HCl bound to the surface during the high pressure reactor synthesis, and after stirring in the MO dye it is released and causes the dye to change color. This is consistent with previous tests showing that suspensions of TCM-C<sub>3</sub>N<sub>4</sub> in water reduced the pH from 7 to  $3^{21}$  In order to insure that the dye absorption maximum did not change or move during the oxidation studies, MO dye experiments were performed in a pH 3 solution environment. The MO dye is very stable under UV light illumination in the absence of catalyst, staying above 95% of its initial concentration after 2 hours of UV exposure (Figure 2.14A). The amount of dark adsorption onto the catalyst surfaces varied, with most being less than 10% except for TCM- $C_3N_4(P)$ , which adsorbed near 20% of the starting dye concentration.  $C_3N_4$  samples prepared from the different methods performed similarly, with < 30% dye remaining. Both TCM-C<sub>3</sub>N<sub>4</sub>(P) and  $DCDA-C_3N_4(T)$  reached complete decolorization of the solution after 2 hours. TCM- $C_3N_4(P)$  was consistently lower in concentration at each measurement point, though some of this difference is due to the additional amount of dye absorbed during the dark stir. The linearized rates of each catalyst are compared in Figure 2.14B, and calculated rate constants are in Table 2.6.



Figure 2.14 - UV light degradation of methyl orange with  $C_3N_4$  products, (A) degradation plots and (B) linear rate fits.

Under visible light illumination, the  $C_3N_4$  photoactivity was lower compared to the unfiltered UV illumination (Figure 2.15A). The remaining MO dye concentration reached 50-80% in 150 minutes. TCM-C<sub>3</sub>N<sub>4</sub>(P) still had near 20% initial concentration loss due to dark dye surface adsorption, as well as the lowest concentration of dye remaining at the end of the illumination periods. The estimated rates based on Figure 2.16B for visible light degradation of methyl orange ranged from 8-25% of unfiltered UV light activity. The results also showed that after accounting for the concentration loss due to dark adsorption, TCM- $C_3N_4(P)$  still showed the highest rate of degradation for methyl orange for both visible (0.0037 min<sup>-1</sup>) and UV light (0.024 min<sup>-1</sup>) (Table 2.6). These values are comparable to previously reported visible light  $C_3N_4$  degradation rates for methyl orange.<sup>92</sup> The approximate photon flux reaching the  $C_3N_4$  catalysts in the dye solution is  $3.2 \times 10^{17}$  photons s<sup>-1</sup> for UV light and  $1.1 \times 10^{17}$  photons s<sup>-1</sup> for visible light based on our previous work.<sup>88</sup> Based on the initial degradation rates, there is a loss of  $3.05 \times 10^{-4}$  dye molecules per UV light photon and  $1.36 \times 10^{-4}$  dye molecules per visible light photon. Given that the carbon nitride band gaps are in the near UV region, cutoff filters limit both flux intensity and photons of appropriate wavelength for  $C_3N_4$ electron/hole formation necessary for MO photooxidation.

Table 2.6. Rate constants  $(min^{-1})$  for UV and visible light degradation of methyl orange using TCM-C<sub>3</sub>N<sub>4</sub> and DCDA-C<sub>3</sub>N<sub>4</sub> products.

	TCM-	TCM-	TCM-	DCDA-
	$C_3N_4(P)$	$C_3N_4(P/A)$	$C_3N_4(T)$	$C_3N_4(T)$
I IV	0.0242	0.0096	0.0112	0.0152
UV	$(R^2 = 0.977)$	$(R^2 = 0.978)$	$(R^2 = 0.983)$	$(R^2 = 0.993)$
Visible (>400	0.0037	0.0024	0.0015	0.0018
nm)	$(R^2 = 0.971)$	$(R^2 = 0.977)$	$(R^2 = 0.980)$	$(R^2 = 0.951)$



Figure 2.15 – Visible light degradation of methyl orange with  $C_3N_4$  products, (A) degradation plots and (B) linear rate fits.

# Photodeposition of metals onto the surface of C<sub>3</sub>N<sub>4</sub>

Precious metals were photoreduced onto  $C_3N_4$  by exposing a suspension of  $C_3N_4$ containing a metal salt in 50:50 H<sub>2</sub>O:methanol to UV light. In reactions using the melamine or DCDA products, a clear change in color was seen from light yellow to grey. This color change was less notable in metal photodepositions using the darker orange TCM products. Some of these products were more brown in color or had a slight grey tint, but most remained close to their starting orange color. At 1-2 wt% loadings, it is not surprising that a visible change might not be prominent. For some of the 2 wt% samples, the reduced metal was detected using powder XRD (Figure 2.16).



Figure 2.16 – XRD TCM-C<sub>3</sub>N<sub>4</sub>(P/A) with photodeposited (a) Pt, (b), Pd, and (c) Ag.

The peaks for these metals, platinum in particular, were very broad and low in intensity. The broad peaks can be attributed to small crystallite sizes (~4-9 nm) for the deposited metals. The low intensity, or absence of peaks in the case of samples with lower metal loadings, is due to being near detection limits. How uniformly the metals deposited on the carbon nitride surface could also impact how well the metals are able to diffract.

The metals were detected on the surface of carbon nitride using XPS (Table 2.7). The XPS data was able to give further information about the oxidation state of the metal. The Pd 4d spectrum showed the major peak at 335.26 eV, with a shoulder at 337.45 eV, indicating the presence of both Pd(0) and Pd(II). Both Ag and Pt spectra showed a single primary peak fitting the position of the reduced metal.

Metal	C (at.%)	N (at.%)	M (at.%)	M (eV)
Pt	50.5	42.1	7.4	71.29 (Pt <sup>0</sup> 71.1-3)
Pd	55.4	37.9	6.7	335.26 (Pd <sup>0</sup> 335.1-6) 337.45 (PdO 337.1)
Ag	46.3	53.2	0.5	368.05 (Ag <sup>0</sup> 368.1-2)

Table 2.7. XPS surface analysis data for metals deposited on TCM- $C_3N_4(P/A)$ .

## Photocatalytic hydrogen evolution reactions

In many previous hydrogen evolution studies with  $C_3N_4$  materials, the UV photodeposition of a noble metal co-catalyst is performed *in-situ*, followed by H<sub>2</sub> evolution analysis. This can lead to unknown solution environments or redox-active metal species remaining in solution. Instead, sequential metal (Pt, Pd, Ag) photodeposition and isolation was performed in order to analyze the metal-deposited  $C_3N_4$  catalyst using nominally 1 wt% metal in solution. At nominally 1 wt% concentrations, the crystalline metals were not detectable by XRD, but XPS identified the presence of each metal on the carbon nitride surface. Quantification of the deposited metal co-catalyst by ICP-OES determined that the deposited Pt metal varies widely from ~0.13 to 0.5 wt% (Table 2.8), demonstrating that a nominally 1 wt% metal-coated product contains much less than that amount on its surface, which would leave the remaining metal cations in solution when the catalyst is not isolated before conducting photocatalytic hydrogen evolution studies.

The photocatalytic hydrogen evolution reactions were monitored in real time using the RGA-MS system; four cycles for TCM-C<sub>3</sub>N<sub>4</sub>(P) are shown in Figure 2.17. Under this system, the first cycle sometimes showed lower activity than the subsequent cycles. Venting the reactor after the first illumination cycle increased the rate, which then remained stable for at least three additional one-hour cycles. This activating first UV cycle was effective with as little as 10 minutes of UV light. The nature of this photoactivation step is unclear, but it may involve some surface-adsorbed species on the platinum resulting from the methanol photodeposition step that the TEOA solution removes under UV illumination.



 $\label{eq:Figure 2.17-Photocatalytic hydrogen evolution using TCM-C_3N_4(P) \ with and without Pt co-catalyst.$ 

While rates for the oxidative degradation of methyl orange above were comparable for the TCM, DCDA, and MA derived  $C_3N_4$  photocatalysts, there were clear differences in the hydrogen production rates (Table 2.8), with DCDA- $C_3N_4(T)$  having the highest mass averaged rate of 453 µmol h<sup>-1</sup> g<sup>-1</sup>. Some of these differences can be attributed to catalyst surface area (DCDA- $C_3N_4$  being highest) and also variations in the amount of Pt deposited on the surface (TCM- $C_3N_4(T)$  being lowest) and, finally, there are differences in  $C_3N_4$  band gaps and band edges (which may relate to structural order/defects). There is a notable improvement in H<sub>2</sub> evolution rates when milling the assynthesized Parr product before photodeposition (doubles surface area with similar amount of Pt deposited), but less improvement when treating the annealed product in the same manner (see Table 2.8) even though metal deposition amount increases.

Sample	Target wt% M (ICP wt%)	H <sub>2</sub> Rate ( $\mu$ mol h <sup>-1</sup> g <sup>-1</sup> )
TCM-C <sub>3</sub> N <sub>4</sub> (P) - milled	None	9 (3)
TCM-C <sub>3</sub> N <sub>4</sub> (P)	1% Pt (0.39)	43 (4)
TCM-C <sub>3</sub> N <sub>4</sub> (P) - milled	1% Pt (0.40)	260 (30)
TCM-C <sub>3</sub> N <sub>4</sub> (P/A)	1% Pt (0.28)	153 (35)
TCM-C <sub>3</sub> N <sub>4</sub> (P/A)	1.5% Pt (0.36)	210 (15)
TCM- $C_3N_4(P/A)$ - milled	1% Pt (0.49)	193 (12)
TCM-C <sub>3</sub> N <sub>4</sub> (S/A)	1.5% Pt (0.38)	159 (17)
TCM-C <sub>3</sub> N <sub>4</sub> (T)	1% Pt (0.13)	68 (8)
TCM-C <sub>3</sub> N <sub>4</sub> (T)	2% Pt (0.46)	42 (4)
DCDA-C <sub>3</sub> N <sub>4</sub> (T)	1% Pt (0.33)	453 (67)
MA-C <sub>3</sub> N <sub>4</sub> (T)	1% Pt (0.28)	283 (50)
TCM- $C_3N_4(P)$ - milled	1% Pd (0.37)	127 (15)
TCM- $C_3N_4(P)$ - milled	1% Ag (0.06)	63 (10)
TCM-C <sub>3</sub> N <sub>4</sub> (P)	0.75% Ag (0.158) 0.25% Pd (0.01)	85 (17)

Table 2.8. Average hydrogen production rates (with standard deviation) for Pt, Pd, and Ag deposited  $C_3N_4$  products with ICP-OES quantification of metal loading.

The effect of the co-catalyst metal was also studied and, moving from Ag<Pd<Pt, resulted in roughly doubled hydrogen evolution rates for nominally 1 wt% metal deposition (Table 2.8). The difference in activities for these metals shows similar trends to previously reported data for other precious metal co-catalysts on cyanamide based

products.<sup>93</sup> For example, increasing the Pt ion solution concentration to 2 wt% for a TCM-C<sub>3</sub>N<sub>4</sub>(T) sample increased the metal loading to 0.46 wt% Pt on the C<sub>3</sub>N<sub>4</sub> surface, but its mass-averaged HER was ~42  $\mu$ mol h<sup>-1</sup> g<sup>-1</sup>, similar to that for the nominally 1 wt% sample in Table 2.8. The variability of actual metal deposited on different carbon nitride structures observed here (and corresponding differences in HER photocatalytic activity) highlights the importance of quantitative metal analysis on the exact carbon nitride samples used for catalytic studies, to allow for useful comparisons across different literature reports.

### Band structure

As noted above, the extrapolated band gap absorptions (HOMO-LUMO) for the TCM derived  $C_3N_4$  network structures are ~0.3 eV lower than that for DCDA or MA products. In addition, low energy valence band spectra from XPS shows that the TCM products have valence band onset levels near 1.4-1.9 eV, which is similar to those reported for DCDA and MA derived  $C_3N_4$  (Figure 2.18).<sup>94-95</sup> Combining the  $E_g$  and VB results leads to approximate simplified energy diagrams for TCM derived  $C_3N_4$  materials (Figure 2.19). Even though there are subtle variations in the band gaps and conduction band positions, the TCM- $C_3N_4$  products are still in a position where the  $2H^+ \rightarrow H_2$  reduction is possible.


Figure 2.18 – Valence band XPS spectra of (A) TCM- $C_3N_4(S)$ , (B) TCM- $C_3N_4(P)$ , (C) TCM- $C_3N_4(T)$ , and (D) TCM- $C_3N_4(P/A)$ .

As expected, the deposition of Pt onto the TCM-C<sub>3</sub>N<sub>4</sub> surface leads to electron emission near 0 eV, consistent with facile electron ejection from the surface-deposited metal. This would also place Pt metal states below the C<sub>3</sub>N<sub>4</sub> conduction band to readily accept photoexcited electrons from the CB to subsequently use in proton reduction on its surface. The lower structure-ordering of the TCM-C<sub>3</sub>N<sub>4</sub>(T) product versus TCM-C<sub>3</sub>N<sub>4</sub>(P) product may be somehow influenced by the latter product forming under moderate pressure in the Parr reactor in the presence of hot HCl gas. This structural disorder for TCM-C<sub>3</sub>N<sub>4</sub>(T) may influence the lower photocatalytic activity for both Pt metal deposition and hydrogen production. It is expected that in these non-crystalline C<sub>3</sub>N<sub>4</sub> structures, solution photodeposition of metal co-catalysts will be very dependent on accessible surface structures that can serve as "antennas" to shuttle photoexcited conduction band electrons from  $C_3N_4$  to the Pt cations for surface reduction. These deposited metal regions are then likely the most readily photoreducing sites for subsequent  $H^+ \rightarrow H_2$  photoreduction reactions. As noted earlier, tuning and controlling defects in  $C_3N_4$  can greatly influence its catalytic activity.<sup>85-87</sup>



Figure 2.19 – Band structure diagram for TCM-C<sub>3</sub>N<sub>4</sub> products based on XPS valence band spectra and calculated band gaps.

## 2.4 Conclusions

This chapter has described the preparation of semiconducting  $C_3N_4$  networks from a trichloromelamine (TCM) precursor. TCM has a unique precursor feature to undergo a facile and energetic thermochemical decomposition that produces multiple gaseous byproducts, driving the reaction to condensation and completion at lower temperatures than other precursors via self-heating. Rapid self-heating from this triazine precursor leads to effective ring expansion into heptazine-based  $C_3N_4$  networks with large several hundred nanometer sized plate or sheet-like morphologies, but with only a few nanometers of interlayer ordering. The presence of chlorine in the precursor provides a site for potential functionalization of the precursor that may alter the C<sub>3</sub>N<sub>4</sub> product and positively-impact interaction with a Pt cocatalyst.<sup>86</sup> Though starting with a different precursor, structural analysis shows TCM- $C_3N_4$  has key features that mirror  $C_3N_4$ products prepared from other precursors. One notable difference is its darker orange product color and smaller band gap. Its photocatalytic activity for oxidative degradation of methyl orange dye was comparable or exceeding that of DCDA-C<sub>3</sub>N<sub>4</sub>. Photocatalytic hydrogen evolution rates for TCM-C<sub>3</sub>N<sub>4</sub> materials approach those from DCDA and MA precursors, with differences influenced by surface area, metal co-catalyst content, and TCM-C<sub>3</sub>N<sub>4</sub> optical absorption properties. Since carbon nitrides are non-crystalline network structures, they can vary in subtle but important structural and electronic ways, and so careful attention should be paid to analysis of the actual amount of metal cocatalyst deposition on catalytically-accessible surface sites as it may vary greatly for different C<sub>3</sub>N<sub>4</sub> structures, which can influence catalytic properties.

#### **CHAPTER 3**

# POROUS ENGINEERING BY SALT MODIFICATIONS TO CARBON NITRIDE SYNTHESIS

## 3.1 Introduction

The previous chapter has described the synthesis and properties of  $C_3N_4$  prepared from the thermal decomposition of TCM. Though chemically and structurally very similar to carbon nitrides produced from DCDA or melamine, there were differences in the photocatalytic activity for these darker-orange powders. The subtle differences in the band structures play a role in some of the change, so this property would be difficult to tune controllably without the use of dopants or control of defects, structure, or composition. Surface area is a physical property that influences catalytic activity through modification of the amount of accessible surface sites that can be adjusted without making changes to the chemical nature of  $C_3N_4$ .

Hard templating is a common method of producing higher surface area, more porous materials.<sup>96</sup> By mixing a precursor with a hard inert support that can be chemically separated from the product with a wash step, one can obtain a product containing pores with the negative image of the template. SiO<sub>2</sub> particles are a common template choice available in different sizes and frequently used in the synthesis of mesoporous carbon nitrides.<sup>82, 97-99</sup> The catalytic activity of these higher surface area carbon nitrides has been shown to improve by factors of 5-10.<sup>58, 81</sup>

This chapter will describe the synthesis of  $C_3N_4$  from mixtures of TCM with different alkali salt templates (NaCl, KBr, KI). In contrast to SiO<sub>2</sub>, salts can be readily removed at room temperature using water as the wash solvent. As TCM is soluble in methanol, it can be co-precipitated with the salt prior to reaction for a more intimate mixture. In addition to solution mixtures, TCM was also mixed with the salts using solid state methods using mortar and pestle or ball milling. The bromide and iodide salts can also exchange with the chlorine in TCM and change the reactivity of the precursor and its decomposition, potentially leading to some differences in the product's reactivity. The general scheme for the synthetic steps is shown in Figure 3.1.



Figure 3.1 – Method for introducing porosity into  $C_3N_4$  by hard templating with salts.

Previous work has treated melon with KSCN as a method for introducing structure defects and modifying catalytic activity.<sup>86-87</sup> Mixtures of TCM with KSCN and also NH<sub>4</sub>SCN were also investigated. The addition of extra C-N linkages could change the connectivity of the heptazine units and degree of condensation within the  $C_3N_4$  structure.

### 3.2 Experimental

# Reagents

Trichloromelamine (C<sub>3</sub>N<sub>3</sub>(NHCl)<sub>3</sub> ,TCM, 98%) was purchased from GFS Chemical and stored in an argon atmosphere glovebox. Other reagents used as received were NaCl (EM Science), KBr (Acros, 99+%), KI (J.T. Baker Chemical, reagent grade), KSCN (Aldrich, 98%), NH<sub>4</sub>SCN (Sigma Aldrich, 97.5+%), H<sub>2</sub>PtCl<sub>6</sub>· 6H<sub>2</sub>O (99.9%, Strem), methanol (ACS certified, Fisher), methyl orange (85% dye content, Sigma Aldrich), and triethanolamine (Alfa Aesar, 98+%). Deionized water (18 M $\Omega$ ) was obtained from a Photronix MiniQuad system.

#### $C_3N_4$ synthesis from alkali halide mixtures with TCM

#### NaCl Mixtures

TCM (300 mg, 1.31 mmol) was dissolved in 40 mL of methanol in a 100 mL Schlenk flask. The flask was purged by bubbling nitrogen through the side arm from a Schlenk line and then capped with a stopper. While maintaining nitrogen flow, the stopper was removed and 0.3, 0.5, 1 or 2 mass equivalents of NaCl were added to the dissolved TCM. The mixture was allowed to stir overnight. The methanol was removed under vacuum by a liquid nitrogen cold trap. The mixed solid was transferred to the Schlenk style reactor described in Chapter 2. The reactor was placed on the heating mantle and set to 300 °C. Unlike the reactions using only TCM, the temperature often reached 300 °C before visible decomposition occurred, as indicated by gas evolution and color change. In the reactions with higher salt content, gas evolution was significantly less vigorous and the color change was observed to be more gradual. A tan to light brown solid was collected from the reactor.

TCM-NaCl mixtures were also reacted using the tube method. The mixed solid was placed into a 1.5 cm diameter test tube cut to 5 cm length and loaded into a Schlenk tube. An oil bubbler was connected to the side arm and the tube was purged with nitrogen. The tube was immersed in a vertically-oriented furnace and heated to 500 °C at a rate of 10 °C/min and held for 1-2 hours. Gas evolution was observed near 180 °C.

#### KBr Mixtures

TCM (400 mg, 1.75 mmol) was dissolved in 40 mL of methanol in a 100 mL beaker. While stirring, KBr (630 mg, 5.25 mmol) was added to the solution. Immediately upon addition, a bright yellow color developed, indicating possible reaction between TCM and KBr. The mixture was stirred for 1 hour, then left open in air to allow the methanol to evaporate overnight and deposit the mixed solid. A pale yellow solid was collected and placed into a 1.5 cm diameter test tube cut to 5 cm length, which was loaded into a Schlenk tube. A bubbler was connected to the side arm and the tube was purged with nitrogen. The tube was immersed in a vertically-oriented furnace and heated to 500 °C at a rate of 10 °C/min and held for 1 hour. An orange solid was collected from the tube insert. Evolution of a yellow-brown gas was observed between 160 and 180 °C.

## KI Mixtures

TCM (400 mg, 1.75 mmol) was dissolved in 40 mL of methanol in a 100 mL beaker. While stirring, KI (880 mg 5.25 mmol) was added to the solution. Immediately upon addition, a dark red-brown solution color developed. The mixture was stirred for 1 hour, then left open in air overnight to allow the methanol to evaporate and deposit the

mixed solid. A rust red solid was collected and placed into a 1.5 cm diameter test tube cut to 5 cm long, which was loaded into a Schlenk tube. A bubbler was connected to the side arm and the tube was purged with nitrogen. The tube was immersed in a vertically-oriented furnace and heated to 500 °C at a rate of 10 °C/min. As the reaction was heated from 200-300 °C, purple vapors filled the tube and discolored the oil in the bubbler. A dark-purple film deposited around the top of the tube that remained outside the furnace, and some white solids deposited just under the top. The reaction was held at 500 °C for 1 hour, and then the tube was removed from the furnace and allowed to cool. A light brown solid was collected from the tube insert.

#### $C_3N_4$ synthesis from thiocyanate mixtures with TCM

#### KSCN Mixtures

TCM (300 mg, 1.31 mmol) was first dissolved in 40 mL of methanol in a 100 mL beaker. While stirring, KSCN (390 mg 3.93 mmol) was added to the solution. Immediately, upon addition a bright yellow solution color developed. The mixture was stirred for 1 hour, then left open in air overnight to allow the methanol to evaporate and deposit the mixed solid. A bright-yellow solid was collected and placed into a 1.5 cm diameter test tube cut to 5 cm long, which was loaded into a Schlenk tube. The tube was immersed in a vertically-oriented furnace and heated to 500 °C at a rate of 10 °C/min and held for 1 hour. Gas evolution was observed at 160 °C.

#### <u>NH<sub>4</sub>SCN Mixtures</u>

TCM (300 mg, 1.31 mmol) was dissolved in 40 mL of methanol in a 100 mL beaker. While stirring, NH<sub>4</sub>SCN (300 mg 3.93 mmol) was added to the solution.

Immediately, upon addition a bright yellow color developed. The mixture was stirred for 1 hour, then left open in air overnight to allow the methanol to evaporate and deposit the mixed solid. A vibrant-yellow solid was collected and placed into a 1.5 cm diameter test tube cut to 5 cm long, which was loaded into a Schlenk tube. The tube was immersed in a vertically-oriented furnace and heated to 500 °C at a rate of 10 °C/min and held for 1 hour. Gas evolution was observed by 200 °C. When the furnace temperature had reached 500 °C, a layer of white solid deposited at the top of the tube that remained outside the furnace.

## Solid state mixtures with salts

Mixtures between TCM and the previously-outlined salts were also prepared using a mortar and pestle. The same 1:3 molar ratio for TCM versus KBr, KI, KSCN, NH<sub>4</sub>SCN were used. The solids were ground and mixed for 5-10 minutes and then loaded into a Schlenk tube with a glass insert. Solids were also mixed using an addition of 1 mL methanol to aid in grinding or pre-reaction. In these methanol-assisted mixtures, color changes similar to the solution mixtures were observed immediately upon addition of methanol. Color changes during dry mixing were lighter shades than the respective solution mixtures when shorter grinding times were used. The mixed reagents were heated in a Schlenk tube in the same manner as the solution mixed precursors.

Mixtures of TCM with these salts were also prepared using a ball mill. TCM and the additive salt were loaded into a 3 mL screw cap Nalgene bottle with 2 x 5 mm  $ZrO_2$ balls to avoid potential reactivity with the milling jar. This bottle was placed into a 15 mL steel milling jar and milled for various timing intervals at 1000-1200 rpm. The solid precursor mixtures exhibited color changes similar to their solution-mixed counterparts, typically in under 5 minutes of milling time. The milled solids were loaded into a Schlenk tube purged with nitrogen and heated in the same manner as previous mixtures. **Safety note:** The physical mixtures between TCM and the thiocyanate salts, by grinding or milling, occasionally exhibited a violent exothermic reaction. It is important to prepare these mixtures at lower scales and in a fume hood. Always wear gloves when handling thiocyanate salts.

## Water wash of $C_3N_4$ -salt composites

All salt-composite products were washed with water to remove the salt, leaving the enhanced porosity  $C_3N_4$  products. The products were lightly ground and placed into a 100 mL beaker with 50 mL of H<sub>2</sub>O and sonicated in a BioSonic UC100 for 1 hour. Then the solid was stirred for 1-2 hours following sonication. The suspension was transferred to a centrifuge tube and spun at 4000 rpm to separate the remaining  $C_3N_4$ . The solids were washed with 20 mL H<sub>2</sub>O and separated by centrifugation again. The solids were rinsed with methanol and left to dry in air.

## *Characterization*

FT-IR spectra were collected on a Nicolet Nexus 670. Solid samples were prepared as KBr pellets for data collection. Gas samples were analyzed by transferring 5 mL of gas from the reaction vessel into a 10 cm long gas IR cell with KBr windows. Elemental analysis was performed on a Perkin Elmer 2400 Series II CHNS/O Analyzer, with 1-2 mg of sample in crimped tin capsules. Powder X-ray diffraction was collected on a Bruker D8 system (source Cu K $\alpha$ , 5-80°, 0.050 °/step) with samples deposited on glass slides. Thermogravimetric analysis was performed on a Seiko Exstar 6300 TGA-DTA. Samples were heated up to 1000°C at a rate of 10°C/min under air or argon flow. Scanning electron microscopy images were collected on a Hitachi S-4800 FE-SEM ( $v_{acc}$ =1.8kV). Diffuse reflectance spectra were collected on a Cary 5000 Series Spectrophotometer (200-800 nm, 10 nm/s). Solid samples were loaded into a round compression sample holder with a 1 cm diameter sampling area. Kubelka-Munk conversions were calculated from the Cary software, and Tauc plots were generated using [F(R)hv]<sup>1/2</sup>. BET surface area measurements were collected on a Quantachrome Nova 1200 using 100-200 mg of ground solid sample. Quantification of deposited platinum was determined using a Varian 720-ES ICP-OES. Samples (5-10 mg) were dissolved in a heated mixture of 5 mL concentrated H<sub>2</sub>SO<sub>4</sub> and 1 mL concentrated HNO<sub>3</sub> and diluted in 25 mL volumetric flasks with 5% HNO<sub>3</sub>.

## Photocatalytic reactions

#### Oxidative degradation of organic dyes

The oxidative degradation of methyl orange dye ( $C_{14}H_{14}N_3NaO_3S$ ) was studied. A stock 10<sup>-3</sup> M dye solution was prepared by dissolving the solid dye in 18 M $\Omega$  water. The stock solutions were diluted to 4 · 10<sup>-5</sup> M for methyl orange in order to fit the peak at  $\lambda_{max}$  in the range ( $\lambda_{MO}$ =504 nm, A<1.50) of the UV-vis spectrophotometer (Agilent 8453). In 30 mL beakers 10 mL of dye solution and 10 mg of catalyst were added. An additional beaker containing only dye was used to evaluate self-degradation in the absence of catalyst. Samples were stirred in the dark for a minimum of 30 minutes to equilibrate dye absorption. Samples were exposed to UV light (Ace-Hanovia, 450 W Hg) in 15-30 minute intervals in air. After each irradiation period, the catalysts were separated from the dye solution by centrifugation, and the UV-vis spectrum of each solution was collected to quantify the remaining concentration of dye. Experiments were typically carried out for 2 hours of total UV irradiation.

#### Hydrogen evolution via water-splitting

The water washed products were used as catalysts with photodeposited Pt. Pt was photoreduced onto the  $C_3N_4$  surface under UV light (450 W Hg) using H<sub>2</sub>PtCl<sub>6</sub> as the metal source.  $C_3N_4$  was suspended in a 50% aqueous MeOH solution containing an amount of the metal salt to target 0.5-2 wt% of reduced metal on  $C_3N_4$ . The solution was degassed with argon and exposed to UV light for 2 hours. The solid was collected by centrifugation, rinsed with MeOH, and dried in air.

For hydrogen evolution tests, 10 mg of catalyst ( $C_3N_4$  or  $M(x\%):C_3N_4$ ) was dispersed in a 10 mL solution of 10% aqueous triethanolamine ( $C_6H_{15}NO_3$ ) contained in a 50 mL Schlenk flask. The solution was degassed with argon and connected to a Residual Gas Analysis Mass Spectrometer (RGA-MS, Stanford Research Systems QMS 300 series) by capillary to the side arm. Data was collected as a pressure vs. time scan, monitoring the masses for nitrogen, argon, hydrogen, water, and oxygen. The reactor was exposed to UV light in 1 hour intervals and was vented with an argon purge between each UV run in order to return hydrogen concentrations to baseline levels. The amount of hydrogen was quantified using a calibration curve referencing known quantities of hydrogen to the argon background.<sup>88</sup> Additional details about calibration and experimental set-up can be found in the appendix.

#### 3.3 Results and Discussion

Initial reactions between TCM and salt additives

In the solution precursor-salt mixtures, evidence of reaction was immediately clear when using the non-chloride salts in solution. Bright-yellow solution colors for KBr, KSCN, and NH<sub>4</sub>SCN and dark-red for KI were observed upon addition of the salt to dissolved TCM. Analysis of the isolated solids by X-ray diffraction showed the formation of KCl from the exchange of the chlorine on TCM with the anion of the salt (Figures 3.2b, 3.3b, and 3.4b).

In the solid mixtures prepared with a mortar and pestle, the color change of the precursor was not as substantial, instead being a lighter shade of the yellow or red color achieved from solution mixtures. Longer grinding times or the addition of 1 mL of methanol to grind as a slurry produced a mixed precursor more-closely resembling the halide-exchanged solution mixture. The X-ray diffraction showed complete exchange of the chlorine in most cases. The incomplete exchange was most notable for KBr physical mixtures that retain more of the starting KBr salt (Figure 3.2a), though the TCM-KI mixture had a small amount of residual KI (Figure 3.3a).

Ball milling provided an alternate, facile approach to the solid state precursor mixtures. As noted, the solids were placed into Nalgene bottles inside the milling jars due to uncertainty of the reactivity between TCM and the different salts under the more energetic conditions. Color change consistent with the solution mixtures were observed between 1 and 5 minutes of total milling time at 1200 rpm. As with the mortar and pestle mixture, the TCM-KBr mixture exhibited the presence of some KCl formation in the

XRD but remained predominantly KBr (Figure 3.2c). The KI and thiocyanate salt mixtures showed complete exchange and KCl formation by XRD (Figure 3.3c and 3.4c)

As to be expected, there was no reaction between TCM and NaCl under any of the mixing methods, as the anions are the same, though chloride exchange could occur. For the remaining salts tested where formation of KCl is observed, the exchange was thermodynamically-favorable based on the heats of formation of the halide salts (Table 3.1). Of the potassium halide salts, KF has the most negative value for  $\Delta H_f$  and would be unlikely to complete the exchange reaction with TCM. Though the decomposition reaction of a TCM-KF mixture was not investigated for concerns of gaseous fluorinated byproducts, the precursor mixture was prepared in methanol. XRD analysis of the isolated powder showed no formation of KCl and only KF, consistent with the thermochemistry data. In addition to the KX conversion to KCl, the other consideration is the thermodynamics of the formation of N-X bonds in place of N-Cl bonds in TCM.

Salt	$\Delta H_{f}$ (kJ/mol)	
KF	-567.3	
KCl	-436.5	
KBr	-393.8	
KI	-327.9	
KSCN	-200.2	
NH <sub>4</sub> Cl	-314.4	
NH <sub>4</sub> SCN	-83.4	

Table 3.1. Heats of formation for tested additive salts.<sup>100</sup>



Figure 3.2 – XRD patterns of TCM-KBr mixtures prepared by (a) mortar and pestle, (b) solution mixture, and (c) ball milling.



Figure 3.3 – XRD patterns of TCM-KI mixtures prepared by (a) mortar and pestle, (b) solution mixture, and (c) ball milling.



Figure 3.4 – XRD patterns of TCM-KSCN mixtures prepared by (a) mortar and pestle, (b) solution mixture, and (c) ball milling.

The IR spectra of the precursor mixtures are shown in Figure 3.5. Compared with unmodified TCM, the halide mixtures show fewer distinct differences than the thiocyanate mixtures. Nitrogen halide stretches occur at low frequencies near 686 cm<sup>-1</sup> for an N-Cl stretch in chloro amine functional groups.<sup>101</sup> The TCM and TCM-NaCl mixture have a small absorption at 680  $\text{cm}^{-1}$ , just below this frequency. As indicated by the XRD data, the exchange of chlorine with the halogen of the KX salt would yield a new N-X bond. The presence of the heavier halogen atom would shift this stretch to an even lower frequency. The TCM-KBr and TCM-KI mixtures show small absorptions below 556 cm<sup>-1</sup> and at 474 cm<sup>-1</sup> instead of the absorption near 686 cm<sup>-1</sup> in support of this exchange. The precursors formed from thiocyanate mixtures show new distinct features. Both the KSCN and NH<sub>4</sub>SCN show the thiocyanate stretch near 2050 cm<sup>-1</sup>, though notably weaker than for the starting salts. The NH<sub>4</sub>SCN mixture had more prominent features from 2900-3200 cm<sup>-1</sup> for N-H stretches of the ammonium cation. The C-N stretches in the 1400-1700 cm<sup>-1</sup> range are overlapped with features from the thiocyanate salts.



Figure 3.5 – IR spectra of C<sub>3</sub>N<sub>4</sub> precursors (a) TCM, (b) TCM-NaCl, (c) TCM-KBr, (d) TCM-KI, (e) TCM-KSCN, and (f) TCM-NH<sub>4</sub>SCN.

# Carbon nitride formation from decomposition of TCM-salt mixtures

As described in the methods section, the synthesis of carbon nitride using salt mixtures with TCM differed from the decomposition of TCM alone. The early tests using NaCl mixtures in the Schlenk reactor saw higher initiation temperatures, less vigorous decompositions than TCM only reactions, or no reaction at all. Reactions were tested in 3:1, 2:1, 1:1, and 1:2 mass ratios of TCM:NaCl. Initially, the size of the flask was reduced from 250 mL to 50 mL to fit the powder in a more focused area on the

heating mantle. Even with this change, the NaCl-rich reactions had to dwell at 300 °C before reaction would visibly occur. The color of these products varied, typically being a lighter-yellow shade than the orange TCM only product. IR analysis of the TCM-NaCl products is shown in Figure 3.6. Significant differences were seen in the 1200-1700 cm<sup>-1</sup> region with increasing proportions of NaCl. The 3:1 starting ratio of TCM:NaCl has an even intensity across this IR region, most similar to a TCM only product. As more NaCl is added, the lower-energy end of these bands decreases in intensity. While the peaks are broadened, indicating some degree of the decomposition reaction has taken place, the intensity distribution is closer to that of the TCM precursor. This may indicate the formation of a smaller ring triazine-based network if the typical heptazine-based  $C_3N_4$  could not fully form by thermal rearrangement in the presence of the thermally-insulating salt.



Figure  $3.6 - C_3N_4$  products prepared from (a) 3:1, (b) 2:1, (c) 1:1, and (d) 1:2 mixtures of TCM:NaCl.

As a next step, small scale reactions in test tubes were used to see if the reactions proceeded to completion at higher temperatures. This was first done using a natural gas torch to heat the bottom of the tube until decomposition occurred. This produced dark brown powders with IR spectra more characteristic of fully formed  $C_3N_4$ , with an additional peak near 2150 cm<sup>-1</sup> consistent with C=N or cumulated double bond -N=C=Nstretching frequencies. This method was challenging in order to avoid melting the tube in the presence of the flame and the heat and gas released from the TCM decomposition. To perform this reaction in a more controlled manner and in larger scales, the Schlenk tube method described in Chapter 2 was developed. By clamping the tube in place inside a vertically-oriented furnace, the decomposition was tested from 400-500 °C. Under this method, the tube reactor was heated from all sides rather than just the base of a flask.

TGA was used to compare the thermal stabilities of the different NaCl-containing products produced in the different reactor set-ups. Figure 3.7 shows the TGA plots for products from reactions starting with 1:1 mass mixtures of TCM:NaCl prepared below 300 °C in 250 mL and 50 mL Schlenk flask reactors, and at 500 °C in the Schlenk tube reactor. The 250 mL product, which received the least-focused heating on the solid, showed mass loss just above 200 °C and continues until 550 °C. The 50 mL product lost mass at a similar temperature, but more gradually. The tube product, having been prepared at 500 °C, shows high stability until after 500 °C where  $C_3N_4$  undergoes complete combustion. The residual mass in each sample is the NaCl, which melts at 801 °C and is volatile enough to be carried away by the air flow. The DTA for these products showed broad exothermic peaks at the end of the C<sub>3</sub>N<sub>4</sub> mass loss and a sharp endothermic peak corresponding to the melting of NaCl. The lower-temperature Schlenk products had less relative NaCl mass, this is due to having a less complete condensation to a more stable final  $C_3N_4$  product. From a starting 1:1 mixture, the tube product contains approximately 25%  $C_3N_4$  or 1:3  $C_3N_4$ :NaCl by mass. Since the mass of NaCl should be unchanged in the reaction, the mass yield of  $C_3N_4$  from TCM should be roughly 33%.



Figure 3.7 – TGA plots of  $C_3N_4$  products from 1:1 TCM:NaCl mixtures under air flow and a 10 °C/min ramp rate.

The remaining TCM-salt products were produced at 500 °C using the tube reactor; the TGA data for these is shown in Figure 3.8. With the exception of the TCM-KI product, the remaining products showed high stability with no significant mass loss until above 500 °C. The TCM-KI product has a gradual 10% mass loss until 500 °C, and after this point showed rapid weight loss like the others. The amount of residual salt mass varies between these samples due to the different starting mass ratios to target the 1:3 TCM:salt molar ratio for complete exchange of chlorine. The decomposition of  $C_3N_4$ in each case is paired with a large broad exothermic peak in the DTA. Since these products formed KCl, the final mass loss occurs near the melting point of KCl at 770 °C. The endothermic peak observed for these in the DTA is broader and less intense that what was observed for the melting of NaCl. The earlier loss of mass for the TCM-KBr and TCM-KSCN products could be from residual amounts of the original, lower melting point salts. In the case of the TCM-NH<sub>4</sub>SCN product, there is no residual salt mass since NH<sub>4</sub>Cl is produced as the exchange product (observed from XRD) and sublimes out of the  $C_3N_4$  product during the synthesis.



Figure 3.8 – TGA plots of  $C_3N_4$  products from salt exchange reactions under air flow at a 10 °C/min ramp rate.

# Chemical and structural characterization of carbon nitride salt composites

Elemental analysis data for the TCM-salt products is shown in Table 3.2. While data was collected on the water-washed products, the total CHN percentages determined were lower than for  $C_3N_4$  produced from TCM only. These differences may indicate presence of some residual salts, which contain elements not quantifiable by this method. Insufficient drying of the samples could also contribute to lower values, as the oxygen present in water or methanol would also go unanalyzed. The exception of this set was the TCM-NH<sub>4</sub>SCN product, which was not washed since the NH<sub>4</sub>Cl transports out of the product during its synthesis. Some of the unanalyzed mass in the KSCN and NH<sub>4</sub>SCN products could also result from incorporated sulfur. Alternative methods of combustion analysis could confirm and quantify the amount of sulfur present. However the relative atomic C:N ratios are very consistent for the different TCM-salt products, fitting  $C_3N_{4+x}$  (0.2<x<0.4). This formula ratio is also consistent with those reported in Chapter 2 for the TCM, melamine, and DCDA products, being only slightly-less nitrogen rich on average.

Product	C wt%	N wt%	H wt%	Total wt% CHN	Formula
C <sub>3</sub> N <sub>4</sub> (TCM-NaCl)	28.39	48.56	1.87	78.76	$C_3N_{4.4}H_{2.4}$
C <sub>3</sub> N <sub>4</sub> (TCM-KBr)	30.46	1.74	51.67	83.87	$C_3N_{4.4}H_{2.0}$
C <sub>3</sub> N <sub>4</sub> (TCM-KI)	27.94	46.70	1.60	76.24	$C_{3}N_{4.3}H_{2.1}$
C <sub>3</sub> N <sub>4</sub> (TCM-KSCN)	27.11	44.67	2.14	73.92	$C_{3}N_{4.2}H_{2.8}$
C <sub>3</sub> N <sub>4</sub> (TCM-NH <sub>4</sub> SCN)	35.15	58.43	1.55	95.13	$C_3N_{4.3}H_{1.6}$

Table 3.2. CHN elemental analysis for selected washed TCM-salt  $C_3N_4$  products from 500 °C tube reactions.

The IR spectra of the TCM-salt products shown in Figure 3.9 are representative of the solution-mixed products for each salt. As with the TCM-only  $C_3N_4$  products in Chapter 2, these salt products have a broad absorption above 3100 cm<sup>-1</sup> for N-H stretches, absorptions from 1200-1700 cm<sup>-1</sup> for C-N and C=N stretches, and a sharp ring breathing mode at 810 cm<sup>-1</sup>. There are notable differences in the relative intensities of the absorptions in the 1200-1700 cm<sup>-1</sup> region. The TCM-KI product, in particular, has much sharper absorptions in this region, which could indicate a more-ordered local structure.

With the exception of the TCM-NH<sub>4</sub>SCN product, these products also show a new absorption at 2150 cm<sup>-1</sup>, fitting a nitrile stretch. This feature has been observed in  $C_3N_4$  products grown in eutectic mixtures of LiCl, NaCl, and KCl.<sup>102</sup> The IR spectra of these products showed little to no variation regardless of the precursor mixing methods.



Figure 3.9 – IR spectra of water washed C<sub>3</sub>N<sub>4</sub> products prepared from 500 °C tube decompositions of (a) TCM, (b) TCM-NaCl, (c) TCM-KBr, (d) TCM-KI, (e) TCM-KSCN, and (f) as-synthesized TCM-NH<sub>4</sub>SCN.

Powder XRD was used on the products before washing with water to identify which salt remained after the reaction and, also, after washing to confirm its removal. While most mixing methods between TCM and the different salts showed an exchange to form KCl or NH<sub>4</sub>Cl, some of the products after the thermal decomposition reaction showed the exchange partially reversed. In the case of the TCM-KBr products (Figure 3.10), the solution mix contained primarily KCl, and the mortar and pestle and ball milled products showed mostly KBr. Each of these products also showed different intensities of a peak fitting a solid solution of KBr-KCl rather that the other salt. The KI showed a partial reversal of the halide exchange only for the mortar and pestle and ball milled products (Figure 3.11). Unlike the case with KBr, these did not form a KCl-KI solid solution, but instead reformed KI in close to a 1:1 ratio with KCl based on the most intense peak. The difference in radii between Cl, Br, and I can be used to rationalize why a solid solution was only observed in the KBr-KCl mixtures and not the KI-KCl mixtures. This agrees with phase diagrams for KI-KCl, which shows that solid solutions for a 1:1 ratio can form, but only at temperatures higher than the 500 °C used in tube synthesis.<sup>103</sup> None of the TCM-KSCN showed the exchange reverse, only having KCl in the XRD. As described earlier, the TCM-NH<sub>4</sub>SCN products transported NH<sub>4</sub>Cl out of the  $C_3N_4$  product during the decomposition reaction. This product as well as the washed products showed the main peak at 27 ° for spacing between C<sub>3</sub>N<sub>4</sub> layers. A summary of the identified crystalline phases is listed in Table 3.3. Though KCl was formed from the different mixing methods between TCM and the different salts, solid state mixtures for the KBr and KI products showed partial re-formation of the original salt. It is possible the hot halide volatiles generated during the synthesis could react with the KCl.



Figure 3.10 - XRD patterns of  $C_3N_4$  products from TCM-KBr (a) mortar and pestle mixtures, (b) solution mixtures, (c) ball milled mixtures, and (d) washed product.



Figure 3.11 - XRD patterns of  $C_3N_4$  products from TCM-KI (a) mortar and pestle mixtures, (b) solution mixtures, (c) ball milled mixtures, and (d) washed product.

Incorporated salt	Mixture method	XRD phases before	XRD phases after	
KBr	Methanol solution	KCl	KCl	
	Mortar and pestle	KBr	<b>KBr</b> , KCl <sub>0.4</sub> Br <sub>0.6</sub>	
	Ball milling	<b>KBr</b> , KCl	KBr, KCl <sub>0.4</sub> Br <sub>0.6</sub>	
KI	Methanol solution	KCl	KCl	
	Mortar and pestle	tar and pestle <b>KCI</b> , KI		
	Ball milling	KCl	KI, KCl	
KSCN	Methanol solution	KCl	KCl	
	Mortar and pestle	KCl	KCl	
	Ball milling	KCl	KCl	
NH <sub>4</sub> SCN	Methanol solution	NH <sub>4</sub> Cl	No salt	
	Mortar and pestle	NH <sub>4</sub> Cl	No salt	
	Ball milling	NH4Cl, NH4SCN	No salt	

Table 3.3. Summary of XRD phases observed in different salt mixtures before and after decomposition reaction (major phase in bold).

# Morphology and surface area

Expectations are that salt inclusions will serve to spread out  $C_3N_4$  products into a more-accessible solid form for enhanced catalytic activity. Analysis of the product morphologies by SEM showed some differences depending on the quantity and type of salt mixed with TCM prior to the decomposition reaction. The unwashed and washed 3:1, 1:1, and 1:2 products are shown in Figure 3.12. The TCM-rich 3:1 product showed similar aggregates above 10  $\mu$ m, with smooth surfaces and rough, porous edges. Washing added some extra surface roughness, but not in the form of any ordered porosity. This rougher surface was more notable compared to  $C_3N_4$  prepared in the absence of salt. The 1:1 product has shorter regions with smooth surface features and

still had some edge nanoscale roughness. The 1:2 product aggregates were still above 10  $\mu$ m, but discrete 2  $\mu$ m chunks could be identified. The washed product has a rough surface with pockets 500 nm and smaller.



Figure 3.12 – SEM images of (A) 3:1 TCM:NaCl product and its (B) water washed C<sub>3</sub>N<sub>4</sub> product, (C) 1:1 TCM:NaCl product and its (D) water washed C<sub>3</sub>N<sub>4</sub> product, (E) 1:2 TCM:NaCl product and its (F) water washed C<sub>3</sub>N<sub>4</sub> product.

Figure 3.13 shows SEM images of the morphologies of the other salt products. These also show large aggregates, but with very distinct fine features. The TCM-KBr product has a very moss-like appearance, with some walls thinner than 200 nm. The TCM-KI product resembled coral, made up of short interconnected rods with 200-400 nm channels. The KSCN and NH<sub>4</sub>SCN products were similar extended aggregates without distinct particle shapes. These did have very rough surfaces with small <100 nm pores.



Figure 3.13 – SEM images of washed C<sub>3</sub>N<sub>4</sub> products from (A) TCM-KBr, (B) TCM-KI, (C) TCM-KSCN, and (D) TCM-NH<sub>4</sub>SCN.

The BET surface areas for the tube reactor products from solution mixed precursors were measured and summarized in Table 3.4. The NaCl and KBr products had similar surface area compared to the equivalent TCM product. The surface areas of the KI, KSCN, and NH<sub>4</sub>SCN products increased 3-4 times. Though not a significant increase, this was higher than the surface area of as-synthesized  $C_3N_4$  products from DCDA or melamine, two commonly used  $C_3N_4$  precursors.

Product	BET Surface Area (m <sup>2</sup> g <sup>-1</sup> )		
C <sub>3</sub> N <sub>4</sub> (TCM)	4.1		
C <sub>3</sub> N <sub>4</sub> (TCM-NaCl)	3.22		
C <sub>3</sub> N <sub>4</sub> (TCM-KBr)	3.59		
C <sub>3</sub> N <sub>4</sub> (TCM-KI)	12.02		
C <sub>3</sub> N <sub>4</sub> (TCM-KSCN)	10.14		
C <sub>3</sub> N <sub>4</sub> (TCM-NH <sub>4</sub> SCN)	16.67		

Table 3.4. BET surface areas of C<sub>3</sub>N<sub>4</sub> products from TCM and TCM-salt mixtures.

# **Optical properties**

The colors of the TCM-salt C<sub>3</sub>N<sub>4</sub> products vary from the orange products described in Chapter 2 (Figure 3.14). Depending on the starting ratio and synthesis method, the TCM-NaCl products ranged from light-tan to orange to very dark-brown. The TCM-KBr products were tan-orange in color, generally close to that of the TCM products. The TCM-KI products were lighter-yellow in color, similar to C<sub>3</sub>N<sub>4</sub> produced from DCDA. The KSCN and NH<sub>4</sub>SCN products were brown and dark-tan colors, respectively. Diffuse reflectance UV-vis was used to quantify these colors as band gap values. Converting the raw reflectance data (Figure 3.15A) into absorbance data using the Kubelka-Munk function yielded a Tauc Plot (Figure 3.15B). The indirect band gaps ( $E_{e}$ ) were approximated by extrapolating the linear region of the absorption edge to the

intercept with the energy axis (in eV). The  $E_g$  values for TCM-NaCl, TCM-KBr, TCM-KI, TCM-KSCN, and TCM-NH<sub>4</sub>SCN were 2.23, 2.42, 2.7, 2.45, and 2.58 eV respectively. Except for the TCM-KI product, the rest appeared to have a secondary absorption edge at lower energy.



 $\begin{array}{l} \mbox{Figure 3.14-Optical microscope images of $C_3N_4$ products from 500 °C tube} \\ \mbox{decomposition of (A) TCM-NaCl, (B) TCM-KBr, (C) TCM-KI, (D) TCM-KSCN, and (E) TCM-NH_4SCN. Scale bar represents 2 mm.} \end{array}$ 



Figure 3.15 – UV-vis DRS data for TCM-salt products (A) reflectance data and (B) Tauc plots.

## Photocatalytic dye oxidation

The degradation of methyl orange dye was studied with the different ratio TCM:NaCl products and the different salt mixtures. Figure 3.16A shows the UV light oxidative degradation in air of MO with  $C_3N_4$  products, for increasing amounts of NaCl had been mixed with TCM. All samples had less than 10% adsorption of dye during the dark stir. At the end of the test, the base  $C_3N_4$  materials (TCM, tube reactor) reached 40% of dye remaining. The 1:1 starting TCM:NaCl product had improved activity to reach under 20% dye remaining. The 2:3 and 1:2 TCM:NaCl products performed nearly identically, the 2:3 product leading by 5% at most time points, but both reached complete degradation by the end of the experiment. The first-order rate fit (Figure 3.16B) and calculated rate constants (Table 3.5) also showed that both the 2:3 and 1:2 TCM:NaCl products were most efficient at degrading MO.

Table 3.5. Rate constants (min<sup>-1</sup>) for UV light degradation of methyl orange dye using washed  $C_3N_4$  products from TCM-NaCl mixtures.

Sample	C <sub>3</sub> N <sub>4</sub> (no	C <sub>3</sub> N <sub>4</sub> (1:1	C <sub>3</sub> N <sub>4</sub> (2:3	C <sub>3</sub> N <sub>4</sub> (1:2
	NaCl)	TCM-NaCl)	TCM-NaCl)	TCM-NaCl)
UV Rate	0.007	0.012	0.033	0.032
	( $\mathbf{R}^2 = 0.997$ )	(R <sup>2</sup> = 0.992)	(R <sup>2</sup> = 0.95)	(R <sup>2</sup> = 0.96)



Figure 3.16 – UV-vis analysis of the photodegradataion of MO dye using C<sub>3</sub>N<sub>4</sub> products from TCM-NaCl mixtures (A) UV-light data (B) linear rate data.

The degradation of methyl orange was also studied with the tube products of each TCM-salt mixture. Figure 3.17A shows the oxidative degradation of MO in air under UV light. Each sample had some degree of dye adsorption during the 60 minute dark stir, the TCM-KI product being the highest near 20%, and the remaining samples at 10% or less. The TCM-KI product was most active reaching near 20% dye remaining by 45 minutes. The TCM-KSCN and TCM-KBr products were different at the first measurement, but from 30 minutes onward had near-identical values. Each of these samples reached 10% of dye remaining by the end of the experiment. The TCM-NH<sub>4</sub>SCN and TCM-NaCl products reached 25% and 35% by the end of the experiment, with the TCM-NH<sub>4</sub>SCN product leading by 10% and each measurement time. A comparison of the linear rate fits (Figure3.17B) and first order rate constants (Table 3.6) showed that the TCM-KSCN, TCM-KBr, and TCM-NH<sub>4</sub>SCN products were most effective within a close margin. The TCM-NaCl and TCM-NH<sub>4</sub>SCN products had close rates to each other and were between 34-46% as effective as the other TCM-salt products.

Table 3.6. Rate constants (min<sup>-1</sup>) for UV light degradation of methyl orange dye using washed  $C_3N_4$  produced from different TCM-salt mixtures.

Sample	C <sub>3</sub> N <sub>4</sub> (TCM- NaCl)	C <sub>3</sub> N <sub>4</sub> (TCM-KBr)	C <sub>3</sub> N <sub>4</sub> (TCM-KI)	C <sub>3</sub> N <sub>4</sub> (TCM- KSCN)	C <sub>3</sub> N <sub>4</sub> (TCM- NH <sub>4</sub> SCN)
UV Rate	0.0072	0.0224	0.0213	0.0236	0.0099
	( $\mathbf{R}^2 = 0.979$ )	( $\mathbf{R}^2 = 0.979$ )	(R <sup>2</sup> = 0.962)	(R <sup>2</sup> = 0.972)	( $R^2 = 0.999$ )


Figure 3.17 – UV-vis analysis of the photodegradation of MO dye using washed C<sub>3</sub>N<sub>4</sub> products from TCM-salt mixtures (A) UV-light degradation data (B) linear rate data.

# Photocatalytic hydrogen evolution via water-splitting

In the hydrogen evolution studies using the TCM-salt decomposition products, only platinum was studied as a co-catalyst. The target amount was typically 1 wt%, however some tests used between 0.5-1.5 wt% as a target loading. Table 3.7 contains the quantification of Pt content for selected samples by ICP-OES. As in Chapter 2, the actual amount of Pt successfully reduced on the surface was a fraction of the target, typically less than a third. Again, this stresses the importance of proper quantification of the amount of co-catalyst loaded in these experiments.

Table 3.7. Average hydrogen production rates (with standard deviation) for Pt deposited  $C_3N_4$  and ICP-OES quantification of metal loading.

Product source	Target wt% M (ICP wt%)	H <sub>2</sub> Rate (µmol h <sup>-1</sup> g <sup>-1</sup> )
TCM (Ch.2)	1% Pt (0.13)	68 (8)
TCM-KBr	1% Pt (0.29)	147 (25)
TCM-KBr	1.52% Pt (0.42)	133 (29)
TCM-KI	0.87% Pt (0.18)	520 (254)
TCM-NH <sub>4</sub> SCN	0.54% Pt (0.33)	300 (10)
TCM-KSCN	0.89% Pt (0.15)	80 (32)

The photocatalytic production of hydrogen was monitored in real time using the RGA-MS system. Figure 3.18 shows overlays of 1 hour UV illumination cycles for each of the TCM-salt precursor  $C_3N_4$  products. Most notable is the product from the TCM-KI mixture, showing 2-3 times as much hydrogen products as the next most-active sample.

This product has consistently shown a drop in activity after the first cycle, after which the activity stabilizes. While some of this enhanced activity can be attributed to the increased surface area of this product, there are reports of iodine as a dopant in  $C_3N_4$  increasing activity for photocatalytic hydrogen evolution.<sup>104</sup> Since the XRD analysis of the precursor showed the exchange between chloride and iodide, it is possible that some amount of iodide remained in the structure after the decomposition. If this incorporated iodide contributed in some part to this enhanced activity, the decrease after the first cycle could indicate that some may have leached out during the reaction.

The TCM-KBr and TCM-NH<sub>4</sub>SCN products performed similarly in the data as shown in Figure 3.18, though the TCM-KBr product exhibited the first-cycle activation step described for some catalysts in Chapter 2. TCM-NH<sub>4</sub>SCN products showed fairly consistent activity, even in cases with low target Pt amounts (Table 3.7). The activity of TCM-KBr products was more variable. The rate activity for the test in Figure 3.17 was 293 ( $\pm$ 31) µmol h<sup>-1</sup> g<sup>-1</sup>, double that of previous data for the samples with different amounts of platinum in Table 3.7. Since the total amount of platinum was not quantified on this particular sample, differences in the loading may contribute to the increase in activity. However, even with a difference of 0.13% platinum, the activities of the samples in Table 3.7 were within the standard deviation of each other.

The TCM-NaCl and TCM-KSCN products did not show a significant enhancement as compared to the TCM-C<sub>3</sub>N<sub>4</sub>(T) product (see Chapter 2), prepared under analogous conditions. This is understandable for the TCM-NaCl product, which did not have a significantly-different surface area. The specific TCM-KSCN product in Figure 3.17 had a hydrogen evolution rate of 143  $\mu$ mol h<sup>-1</sup> g<sup>-1</sup>. The activity of this sample decreased over the four cycles tested, stating at 240  $\mu$ mol h<sup>-1</sup> g<sup>-1</sup> and ending at 77  $\mu$ mol h<sup>-1</sup> g<sup>-1</sup>, indicating some chemical change, similar to the TCM-KI product catalysts. A possibility is if residual loosely SCN<sup>-</sup> or CN<sup>-</sup> fragments remained in the final C<sub>3</sub>N<sub>4</sub> product, as these could strongly coordinate to Pt, effectively poisoning the co-catalyst.<sup>105-106</sup>



Figure 3.18 – Stacked plots of photocatalytic hydrogen evolution using TCM-salt products. Each catalyst contained a targeted 1 wt% Pt co-catalyst. The UV lamp was turned on for 1 hour cycles, and the reactor was purged with argon for 10 minutes between each cycle.

In regards to surface area, the order from highest to lowest follows TCM-NH<sub>4</sub>SCN > TCM-KI > TCM-KSCN > TCM-KBr > TCM-NaCl products. The products of TCM-NH<sub>4</sub>SCN and TCM-KI have the highest surface areas and generally the highest rate of hydrogen production. As previously discussed, an iodide-doping effect may be a cause of the further increased activity for the TCM-KI product. The TCM-KSCN product has the next-highest surface area, but even at its best activity in the first cycle is comparable to rates for TCM-KBr products. The proposed SCN<sup>-</sup> or CN<sup>-</sup> poisoning may take effect as early as the first cycle in this case. Though the TCM-KBr product does not have a significantly-different surface area from the TCM-C<sub>3</sub>N<sub>4</sub>(T) product, it does have increased activity. The differences in the band gaps for these products could also contribute to the changes in activity.

## 3.4 Conclusions

This chapter has described the preparation of  $C_3N_4$  products from different mixtures of TCM with halide or thiocyanate salts, with the goal of using the salts as easily removable templates to increase product surface area. The TCM precursor readily reacted with most of these salts, exchanging chlorine for bromine, iodine, or thiocyanate. Use of the low temperature Schlenk reactor set-up for the synthesis of  $C_3N_4$  was met with difficulty in fully propagating decomposition, due to the presence of the thermally insulating salt. The tube reactor decompositions at 500 °C worked for achieving fullyformed products from each TCM-salt mixture. These products formed a more diverse array of colored product than the typical orange product prepared from TCM only, with band gaps ranging from 2.23 to 2.7 eV. Surface areas increased the most for the TCM-KI, TCM-NH<sub>4</sub>SCN, and TCM-KSCN products, though SEM displayed new rougher, porous surfaces for the other products. Photocatalytic activity was evaluated for both the degradation of methyl orange dye and hydrogen evolution.

## **CHAPTER 4**

# DEPOSITION OF CARBON NITRIDE ONTO METAL OXIDE SUPPORTS 4.1 Introduction

In contrast to the previous chapter where the goal was to utilize different salts as templates to modify and improve the surface area of carbon nitride ( $C_3N_4$ ), this chapter aims to utilize high surface-area materials as supports for the growth of carbon nitride. Previous work utilized a low melting point cyanamide precursor to infiltrate a silica (SiO<sub>2</sub>) host matrix prior to carbon nitride formation.<sup>107-108</sup> A similar approach has been used to incorporate dicyandiamide into alumina (Al<sub>2</sub>O<sub>3</sub>) foams heated at different temperatures.<sup>109</sup>

In addition to utilizing high surface-area supports, using other known metal oxide semiconductors as supports for semiconducting  $C_3N_4$  could enable new synergistic photocatalytic benefits from different absorption properties or differences in band edge positions more suitable to other oxidation or reduction processes.<sup>55, 110</sup> Titanium dioxide (TiO<sub>2</sub>) has a wider band gap ( $E_g$ =3.2 eV) than  $C_3N_4$  ( $E_g$ =2.6-2.7 eV), and while limited to using UV light, it can function for a wide variety of photocatalytic reactions including organic molecule oxidation, sterilization, and water-splitting.<sup>38, 40</sup> C<sub>3</sub>N<sub>4</sub> composites with TiO<sub>2</sub> have been prepared by hydrothermal synthesis and calcination of mixed sol-gel precursors,<sup>111</sup> by electrodeposition of C<sub>3</sub>N<sub>4</sub> on a TiO<sub>2</sub> nanotube substrate,<sup>112</sup> or by simple solution mixing and evaporation of C<sub>3</sub>N<sub>4</sub> and is capable of using low wavelength visible light. In contrast to TiO<sub>2</sub>, applications of WO<sub>3</sub> are primarily used for photocatalytic oxidation reactions, rather than both oxidation and reduction.<sup>114-116</sup> C<sub>3</sub>N<sub>4</sub> composites with WO<sub>3</sub> have been prepared by hydrothermal synthesis with  $C_3N_4$  in a WO<sub>3</sub> precursor solution<sup>117</sup> or annealing of dried solution mixtures of  $C_3N_4$  with WO<sub>3</sub>.<sup>118-119</sup> Low-level doping or composite formation was also achieved by intercalation of WO<sub>4</sub><sup>2-</sup> between g- $C_3N_4$  layers.<sup>120</sup>

The work in this chapter explores the use of inert and photoactive oxide supports as a means of improving the photocatalytic activity of TCM-derived carbon nitride. Commercial SiO<sub>2</sub> and Al<sub>2</sub>O<sub>3</sub> act as high surface area (300 and 100 m<sup>2</sup>/g) supports. Composites with TiO<sub>2</sub> and WO<sub>3</sub> are supports with larger and smaller band gaps, respectively, than C<sub>3</sub>N<sub>4</sub>. The methanol-soluble TCM was coated onto the oxide by suspending the oxide in a solution of TCM and evaporating the solvent to deposit TCM on the surface. The coated metal oxide was heated to decomposition conditions in the previously-used reactors. Alternate hybrid composites using C<sub>3</sub>N<sub>4</sub> powders were also produced.

#### 4.2 Experimental

# Reagents

Trichloromelamine (C<sub>3</sub>N<sub>3</sub>(NHCl)<sub>3</sub> ,TCM, 98%) was purchased from GFS Chemical and stored in an argon atmosphere glovebox. Other reagents used as received were; SiO<sub>2</sub> Aerosil 300 (Degussa), Al<sub>2</sub>O<sub>3</sub> Aluminoxid (Degussa), P25 TiO<sub>2</sub> Titanoxid (Degussa), H<sub>2</sub>WO<sub>4</sub> (Alfa Aesar), Na<sub>2</sub>WO<sub>4</sub>·2H<sub>2</sub>O (99.0%, Alfa Aesar), H<sub>2</sub>PtCl<sub>6</sub>·6H<sub>2</sub>O (99.9%, Strem), methanol (ACS certified, Fisher), methyl orange (85% dye content, Sigma Aldrich), and triethanolamine (Alfa Aesar, 98+%). Deionized water (18MΩ) was prepared using a Photronix MiniQuad system.

## Coating of $C_3N_4$ onto SiO<sub>2</sub>, TiO<sub>2</sub> and Al<sub>2</sub>O<sub>3</sub> oxide supports

The mixture of the TCM precursor and the oxide support was prepared both under inert conditions and in air. TCM (300 mg, 1.31 mmol) was dissolved in 40 mL of methanol in a 100 mL Schlenk flask. The flask was purged with nitrogen by connecting the side arm to a Schlenk line and capped with a stopper. While maintaining nitrogen flow, the stopper was removed, and 0.5 or 1 mass equivalents of SiO<sub>2</sub>, TiO<sub>2</sub> or Al<sub>2</sub>O<sub>3</sub> was added to the dissolved TCM. The mixture was stirred overnight. The methanol was then removed under vacuum by a liquid nitrogen cold trap. The mixed precursor remained white in color but was notably denser than the bare metal oxide. Alternatively, the precursor mixture was prepared in air by dissolving TCM (300-500 mg, 1.31-2.18 mmol) in 50 mL of methanol in a 100 mL beaker. Once completely dissolved, 0.5 or 1 mass equivalent of either oxide was added to the solution. The solution was stirred for two hours and then left to evaporate in air overnight to deposit the mixed solid.

## Schlenk heating

The mixed solid was transferred to the Schlenk style reactor described in Chapter 2. The reactor was placed on the heating mantle and set to 300 °C. The decomposition reaction was less vigorous than for TCM only, with the color change not occurring until higher temperatures and lower or slower gas evolution. Depending on the reaction, the reactor was allowed to dwell at 300 °C for up to 5 minutes. Products from this method will be identified as  $C_3N_4@SiO_2(Schlenk), C_3N_4@Al_2O_3(Schlenk), and C_3N_4@TiO_2(Schlenk).$ 

# Tube heating

The decomposition reaction was also performed using the tube reactor set-up. The TCM deposited oxide precursor was loaded into a Schlenk tube with a test tube insert. An oil bubbler was connected to the side arm to allow pressure release, and the tube was purged with nitrogen. The tube was clamped in place above a vertical tube furnace and heated to 500 °C at a rate of 10 °C with a 1 hour dwell time. Gas evolution typically occurred between 180 and 200 °C. Products from this method will be identified as  $C_3N_4@SiO_2(Tube)$ ,  $C_3N_4@Al_2O_3(Tube)$ , and  $C_3N_4@TiO_2(Tube)$ .

# *Preparation of* $C_3N_4$ -WO<sub>3</sub> *composite materials*

 $C_3N_4$ -WO<sub>3</sub> composites were prepared using two different approaches; coating TCM onto WO<sub>3</sub> and performing the decomposition, or depositing a soluble WO<sub>3</sub> precursor onto already made  $C_3N_4$  and completing the metal oxide formation at lower temperature.

## C<sub>3</sub>N<sub>4</sub> formation on WO<sub>3</sub>

For the first method, crystalline WO<sub>3</sub> was first prepared from tungstic acid  $(H_2WO_4)$ .  $H_2WO_4$  (1 g, 4 mmol) was placed into a crucible, which was loaded into a muffle furnace. The furnace was heated to 300 °C at a rate of 10 °C min<sup>-1</sup> in air and held overnight. The conversion of  $H_2WO_4$  to  $WO_3$  was verified by XRD. TCM (200 mg, 0.87 mmol) was dissolved in 40 mL of methanol in a 100 mL beaker, then 0.3 or 0.5, mass equivalents of the prepared WO<sub>3</sub> was added to the solution. The mixture was briefly sonicated and then stirred for 2 hours. The beaker was left open in air to allow the methanol to evaporate overnight and deposit the mixed solid.

#### Schlenk heating

The mixed precursor was loaded into the Schlenk style reactor and put under nitrogen. The reactor was placed onto a heating mantle and set to heat to 300 °C. The decomposition initiated between 240 and 265 °C. An orange to light-brown powder was collected after cooling the reactor (~42-53% mass yield from total precursor mass). Products from this method will be identified as  $C_3N_4@WO_3(Schlenk)$ .

# Tube heating

The decomposition reaction was performed using the tube reactor set-up. The TCM deposited oxide precursor was loaded into a Schlenk tube with a test tube insert. An oil bubbler was connected to the side arm to allow pressure release, and the tube was purged with nitrogen. The tube was clamped in place above a vertical tube furnace, heated to 500 °C at a rate of 10 °C, and held for 1 hour. Gas evolution typically occurred between 180-200 °C. After cooling and venting the tube, a dark brown-black powder was collected from the insert (~18% mass yield from total precursor mass). Products from this method will be identified as  $C_3N_4@WO_3(Tube)$ .

# WO<sub>3</sub> formation on C<sub>3</sub>N<sub>4</sub>

In the alternate method of preparing  $C_3N_4$ -WO<sub>3</sub>, Na<sub>2</sub>WO<sub>4</sub> 2H<sub>2</sub>O (200 mg, 0.61 mmol) was dissolved in 10 mL of water.  $C_3N_4$  product (150 mg) from a Parr TCM decomposition reaction (Chapter 2) was added into the solution and the mixture stirred. The solution was acidified with 2 mL of concentrated HCl, after which a yellow color developed indicating the formation of H<sub>2</sub>WO<sub>4</sub>. After precipitation, the solids were isolated by centrifugation. The yellow solids were then rinsed over a Büchner funnel to ensure complete removal of by-product NaCl. The solid was allowed to dry in air. After

drying, the solid was placed into a covered crucible. The crucible was placed into a muffle furnace, heated to 300 °C in air at a rate of 10 °C/min, and held for 3 hours. A yellow-orange powder was collected from the crucible (~91% mass yield based on dry composite before heating). Products from this method will be identified as WO<sub>3</sub>@C<sub>3</sub>N<sub>4</sub>.

# Annealing of $C_3N_4$ -metal oxide composites

Higher-temperature treatments of the Schlenk reactor products were investigated. Typically, 100-200 mg of the solid was placed into an alumina boat that was then placed inside a Pyrex tube closed on one end. The tube was connected to a Schlenk line, evacuated, and refilled with argon. The tube was placed into a clamshell furnace, heated to 450-500 °C at a rate of 10 °C/min, and held for 1-2 hours. Mass recovery ranged from 37-76% depending on the initial ratio of  $C_3N_4$  to oxide support.

## *Reaction of a physical mixture of* $C_3N_4$ *and metal oxide support*

Physical mixtures of  $C_3N_4$  (Parr TCM product) and  $TiO_2$  or  $WO_3$  were prepared via ball-milling. 100 mg of  $C_3N_4$  and 50 mg of  $TiO_2$  or  $WO_3$  were loaded into 5 mL stainless steel milling jars with 2 x 5 mm steel balls. The solids were milled for 5 minutes at 1200 rpm in a FormTech FTS 1000. The composites were used without any additional heat treatment. Products from this method will be identified as  $C_3N_4$ -TiO<sub>2</sub>(mill) and  $C_3N_4$ -WO<sub>3</sub>(mill).

# **Characterization**

FT-IR spectra were collected on a Nicolet Nexus 670. Solid samples were prepared as KBr pellets for data collection. Gas samples were analyzed by transferring 5 mL of gas from the reaction vessel into a 10 cm long gas IR cell with KBr windows. Elemental analysis was performed on a Perkin Elmer 2400 Series II CHNS/O Analyzer, with 1-2 mg of sample in crimped tin capsules. Powder X-ray diffraction was collected on a Bruker D8 system (source Cu Ka, 5-80°, 0.050 °/step) with samples deposited on glass slides. Thermogravimetric analysis was performed on a Seiko Exstar 6300 TGA-DTA. Samples were heated up to 1000°C at a rate of 10°C/min under air or argon flow. Scanning electron microscopy images were collected on a Hitachi S-4800 FE-SEM (v<sub>acc</sub>=1.8 kV). Diffuse reflectance spectra were collected on a Cary 5000 Series Spectrophotometer (200-800 nm, 10 nm/s). Solid samples were loaded into a round compression sample holder with a 1 cm diameter sampling area. Kubelka-Munk conversions were calculated from the Cary software, and Tauc plots were generated using  $[F(R)hv]^{1/2}$ . BET surface area measurements were collected on a Quantachrome Nova 1200 using 100-200 mg of ground solid sample.

### Photocatalytic reactions

#### Oxidative degradation of organic dyes

The degradation of methyl orange ( $C_{14}H_{14}N_3NaO_3S$ ) dye was studied using the materials prepared in this chapter. A stock 10<sup>-3</sup> M dye solution was prepared by dissolving the solid dye in 18 M $\Omega$  water. The stock solution was diluted to 4 · 10<sup>-5</sup> M for methyl orange to fit the peak at  $\lambda_{max}$  in the range ( $\lambda_{MO}$ =504 nm, A<1.50) of the UV-vis

spectrophotometer (Agilent 8453). To 30 mL beakers, 10 mL of dye solution and 10 mg of catalyst were added. An additional beaker containing only dye was used as to evaluate self-degradation in the absence of catalyst. Samples were stirred a minimum of 30 minutes in the dark to equilibrate dye absorption. Samples were exposed to UV light (Ace-Hanovia, 450 W Hg) in 15-30 minute intervals. After each illumination period, the catalysts were separated from the dye solution by centrifugation (3000 rpm), and the UV-vis spectrum of each solution was collected to quantify the remaining concentration of dye. Experiments were typically carried out for 2 hours of total UV irradiation. For visible light experiments, 400 nm long pass filters (Edmund Optics) were placed between the lamps and samples and irradiation times were extended.

## Photocatalytic hydrogen evolution via water-splitting

The composite products were examined as HER catalysts. Pt was photoreduced onto the  $C_3N_4$  surface under UV light using  $H_2PtCl_6$  as the metal source.  $C_3N_4$  was suspended in a 50% aqueous MeOH solution containing the metal salt to target 1-2 wt% of reduced metal on  $C_3N_4$ . The solution was degassed with argon and exposed to UV light for 2 hours. The solid was collected by centrifugation, rinsed with MeOH, and dried in air.

For hydrogen evolution tests, 10 mg of catalyst ( $C_3N_4$  or M(x%): $C_3N_4$ ) was dispersed in a 10 mL solution of 10% aqueous triethanolamine ( $C_6H_{15}NO_3$ ) contained in a 50 mL Schlenk flask. The solution was degassed with argon and connected to a Residual Gas Analysis Mass Spectrometer (RGA-MS, Stanford Research Systems QMS 300 series) by capillary to the side arm. Data were collected as a pressure vs. time scan, monitoring the masses for nitrogen, argon, hydrogen, water, and oxygen. The reactor was exposed to UV light in 1 hour intervals and was vented with an argon purge between each UV run in order to return hydrogen concentrations to baseline levels. The moles of hydrogen produced were determined using a pressure signal calibration based on known concentration mixtures of  $H_2/Ar$ .<sup>88</sup> Additional details about calibration and experimental set-up can be found in the appendix.

## 4.3 Results and Discussion

## Decomposition reaction and thermal stability of $C_3N_4$ in oxide composites

## $\underline{C_3N_4}$ with SiO<sub>2</sub>, Al<sub>2</sub>O<sub>3</sub>, and TiO<sub>2</sub>

Prior to decomposition reactions, the TCM precursor was mixed with commercial  $SiO_2$ ,  $Al_2O_3$ , and  $TiO_2$  in solution to coat the reactive precursor onto the oxide surface. The oxides used were very fine, high surface area powders ( $SiO_2 \sim 300 \text{ m}^2/\text{g}$ ,  $Al_2O_3 \sim 100 \text{ m}^2/\text{g}$ ,  $TiO_2 \sim 45 \text{ m}^2/\text{g}$ ), and were highly dispersible. Once coated in TCM, these oxides appeared denser and less susceptible to static cling. The color remained unchanged, unlike the reactive mixtures between TCM and the halide and thiocyanate salts described in Chapter 3. The IR spectra showed no change in the signature absorptions of TCM. The only new feature was the appearance of a large broad absorption for the respective M-O stretches (M = Si, Al, Ti).

Using the Schlenk flask and heating mantle reactor, visible decomposition did not take place until reaching higher temperatures than TCM without an additive. The SiO<sub>2</sub> products reached 250-270 °C before decomposition occurred, as indicated by gas evolution and a color change of the solid to a sandy, light-yellow-orange color (Figure 4.1A). The Al<sub>2</sub>O<sub>3</sub> composite reactions were unique in that the decomposition occurred without violent gas evolution and a more gradual color change to an orange solid by 260 °C (Figure 4.1C). These reactions were allowed to dwell at this temperature for 10 minutes after the color changed, but there were no indications of further reaction. The TiO<sub>2</sub> products behaved similarly, reaching 250 °C to yield a light-sand colored solid (Figure 4.1E). When SiO<sub>2</sub> reactions were tested at larger scales, the temperature was allowed to reach 300 °C with no gas evolution, but some change in product color was observed. As with the NaCl reactions described in Chapter 3, reducing the flask size used for the reactor, from 250 mL to 50 mL, helped to focus the heat from the mantle on the solids. This resulted in gas evolution from the decomposition by 290 °C. The presence of the inert oxides was thermally insulating, reducing the reactivity of TCM and allowing decomposition to fully propagate.

Higher temperature annealing of the  $C_3N_4@SiO_2(Schlenk)$  products was explored to see if the reaction would continue after a second heating step. Heating to 450 °C under argon did result in a darker product; however some exposed SiO<sub>2</sub> surfaces could begin to be seen. While  $C_3N_4@SiO_2(Schlenk)$  composites would contain up to 40%  $C_3N_4$  by mass, following annealing this decreased to 10%. While TGA-DTA analysis of  $C_3N_4$ from TCM only showed that the product could be stable up to 600 °C, higher temperature heating of the poorly-formed product in the oxide composite resulted in a greater breakdown of the residual  $C_3N_4$  components.

The formation of  $C_3N_4$  composite products from single-step high temperature heating in the tube reactor led to reactions that behaved similarly to the decomposition of TCM in the absence of additive. Gas evolution was observed in the lower temperature range of 180-200 °C. The products were darker than their respective Schlenk versions

(Figure 4.1 B, D, and F). The more uniform and encompassing heating method using a furnace rather than a heating mantle resulted in a more typical TCM decomposition reaction.



 $\begin{array}{l} \mbox{Figure 4.1-Optical microscope images of (A) $C_3N_4@SiO_2$ (Schlenk), (B) $C_3N_4@SiO_2$ (Tube), (C) $C_3N_4@Al_2O_3$ (Schlenk), (D) $C_3N_4@Al_2O_3$ (Tube), (E) $C_3N_4@TiO_2$ (Schlenk), and (F) $C_3N_4@TiO_2$ (Tube). Scale bar represents 2 mm. } \end{array}$ 

Comparisons between the thermal stabilities of the Schlenk and tube synthesis methods are shown in the TGA plots in Figures 4.2. 4.3 and 4.4. The  $C_3N_4$ @SiO<sub>2</sub>(Schlenk) composite has a low rate of mass loss until 225 °C, where an increased rate of mass loss begins and continues steadily until 550 °C; at this point, the  $C_3N_4$  has fully decomposed leaving only SiO<sub>2</sub> (57% residual mass). The tube product (heated to 500 °C during synthesis) started with mass loss at the same low rate as the Schlenk products, but this rate continued until 425 °C before the rapid mass loss occurred, leaving only SiO<sub>2</sub> (Figure 4.2). The  $C_3N_4@SiO_2$  products had weak broad exothermic peaks near 550 °C in the DTA. The C<sub>3</sub>N<sub>4</sub>@Al<sub>2</sub>O<sub>3</sub>(Schlenk) product showed an increased mass loss just above 200 °C, until a slight inflection point near 250 °C where the rate was steady, but slower until reaching 450 °C. This product had a broad exothermic peak in the DTA near 500 °C. The  $C_3N_4@Al_2O_3$ (Tube) product had a low rate of mass loss until 430  $^{\circ}$ C, leading to complete decomposition of the C<sub>3</sub>N<sub>4</sub> component by 550 °C (Figure 4.3), and showed a sharper exothermic peak near 500 °C in the DTA. Both the C<sub>3</sub>N<sub>4</sub>@TiO<sub>2</sub>(Schlenk) and C<sub>3</sub>N<sub>4</sub>@TiO<sub>2</sub>(Tube) products exhibit very similar behavior, each having a steady mass loss until 420 °C, followed by complete decomposition of  $C_3N_4$  by 550 °C (Figure 4.4). Both  $C_3N_4$ @TiO<sub>2</sub> products had broad exothermic peaks near 480 °C, and the C<sub>3</sub>N<sub>4</sub>@TiO<sub>2</sub>(Tube) product had a clear second peak at 520 °C. The differences in the amount of mass remaining after 600 °C were due to different amounts of  $TiO_2$  mixed with TCM in the composite.



Figure 4.2 – TGA plots of  $C_3N_4@SiO_2$  products under air flow, heated at a rate of 10  $^{\circ}C/min$ .



Figure 4.3 – TGA plots of  $C_3N_4@Al_2O_3$  products under air flow, heated at a rate of 10  $^{\circ}C/min$ .



Figure 4.4 – TGA plots of C<sub>3</sub>N<sub>4</sub>@TiO<sub>2</sub> under air flow, heated at a rate of 10 °C/min.

# $\underline{C_3N_4 \text{ with WO}_3}$

The synthesis of the  $C_3N_4@WO_3$  composites differed from those of the other oxides. Starting with the Schlenk decomposition reactions of TCM coated on WO<sub>3</sub>, gas evolution and product color change occurred in the range of 240-265 °C. These reactions used lower relative amounts of WO<sub>3</sub> to TCM compared to the other oxides, and were put in the smaller 50 mL volume reactor to receive more focused heat from the mantle. Though the low temperature formation of these products seemed well behaved, higher temperature annealing was explored. When the starting orange product (Figure 4.5A) was heated to 500 °C under argon, a very dark brown, nearly black solid formed (Figure 4.5B). This dark product also formed from annealing a physical mixture of  $C_3N_4$  and WO<sub>3</sub> under the same conditions. The synthesis was also tested in the tube reactor, starting with TCM on WO<sub>3</sub> and heating to 500  $^{\circ}$ C in a single step. This also produced the dark product (Figure 4.5C).

Given that heating WO<sub>3</sub> at high temperatures in the presence of  $C_3N_4$ , whether from TCM or pre-formed, led to an unusual dark product, an alternate approach was taken to form the composite. Starting with Na<sub>2</sub>WO<sub>4</sub> dissolved in water, HCl was used to precipitate hydrated WO<sub>3</sub> onto pre-formed  $C_3N_4$ . This was then heated at 300 °C to dehydrate and ensure complete oxidation in air, without losing significant amounts of  $C_3N_4$ . The products from this method were the more typical yellow-orange powders (Figure 4.5D).



Figure 4.5 – Optical microscope images of (A)  $C_3N_4@WO_3$  (Schlenk), (B)  $C_3N_4@WO_3$  (Anneal), (C)  $C_3N_4@WO_3$  (Tube), and (D)  $WO_3@C_3N_4$ . Scale bar represents 2 mm.

Figure 4.6 shows the TGA plots of the final products from these two approaches. The dark  $C_3N_4@WO_3$  product exhibits continuous mass loss at lower temperatures followed by rapid loss after 450 °C. There was a very weak, broad exothermic peak in the DTA near 480 °C, fitting the more gradual decomposition of the  $C_3N_4$  component of the composite. The WO<sub>3</sub>@C<sub>3</sub>N<sub>4</sub> product is more stable, only beginning to show some mass loss after 450 °C and full decomposition of the C<sub>3</sub>N<sub>4</sub> component by 600 °C. This product showed a sharp exothermic peak near 550 °C in the DTA.



Figure 4.6 – TGA plots of  $C_3N_4@WO_3$  and  $WO_3@C_3N_4$  under air flow, heated at a rate of 10 °C/min.

*Chemical and structural characterization of carbon nitride and oxide composites* 

Elemental analyses of the C<sub>3</sub>N<sub>4</sub>-metal oxide composites are shown in Table 4.1.

Data were collected for both the Schlenk and tube products to compare the compositions

and find correlation with the less-complete reactions. The  $C_x N_y H_z$  components of  $C_3N_4$ @TiO<sub>2</sub> products from each method had nearly identical compositions. As described earlier, compared to the other composites the  $C_3N_4$ @TiO<sub>2</sub> composites had similar thermal stability and likely achieved the same degree of structural condensation. The  $C_{3}N_{4}@Al_{2}O_{3}(Schlenk)$  composite had a higher N:C ratio than the tube counterpart. This is closer to the starting  $C_3N_6$  of TCM, indicating less condensation to the ideal  $C_3N_4$ structure by retaining extra nitrogen. The  $C_3N_4@SiO_2$  composites had similar compositions to each other but with slightly higher N:C ratios than other products. Both preparation methods of the C<sub>3</sub>N<sub>4</sub>@Al<sub>2</sub>O<sub>3</sub> and C<sub>3</sub>N<sub>4</sub>@SiO<sub>2</sub> composites had higher relative hydrogen content than other products. Since the hydrogen is typically located in terminal amine sites, these products may be smaller oligomers rather than a more extended structure, in order to incorporate more of these sites, or have high surface adsorption of  $H_2O$  due to their high surface areas. The WO<sub>3</sub>@C<sub>3</sub>N<sub>4</sub> composite maintained its starting composition of the  $C_3N_4$  used as the support. Heating to  $C_3N_4$  300 °C in the presence of a WO<sub>3</sub> precursor had no impact on its composition. The C<sub>3</sub>N<sub>4</sub>@WO<sub>3</sub> composite from heating TCM with WO<sub>3</sub> showed higher relative amounts of both nitrogen and hydrogen. Unlike the  $Al_2O_3$  or SiO<sub>2</sub> composites, this is not due to incomplete lower temperature reaction as it was heated and held at 500 °C. There is, instead, some other reaction between the  $C_3N_4$  and  $WO_3$ , as evidenced by the black solid formed.

Sample	C wt%	N wt%	H wt%	Total wt% CHN (% loss by TGA)	Formula
C <sub>3</sub> N <sub>4</sub> @TiO <sub>2</sub> (Schlenk)	12.13	21.94	1.49	35.56 (32.1)	$C_3N_{4.6}H_{4.4}$
C <sub>3</sub> N <sub>4</sub> @TiO <sub>2</sub> (Tube)	14.45	25.98	1.78	42.21 (49.9)	$C_3N_{4.6}H_{4.3}$
$C_3N_4@Al_2O_3$ (Schlenk)	9.06	18.68	1.09	28.83 (43.8)	$C_3N_{5.3}H_{4.3}$
$C_3N_4@Al_2O_3$ (Tube)	10.47	19.07	1.78	31.32 (43.4)	$C_3N_{4.7}H_{6.1}$
C <sub>3</sub> N <sub>4</sub> @SiO <sub>2</sub> (Schlenk)	9.65	18.08	1.80	29.53 (43.1)	$C_{3}N_{4.8}H_{6.7}$
C <sub>3</sub> N <sub>4</sub> @SiO <sub>2</sub> (Tube)	8.65	16.53	2.01	27.19 (35.4)	$C_3N_{4.9}H_{8.4}$
C <sub>3</sub> N <sub>4</sub> @WO <sub>3</sub> (Schlenk)	11.60	21.38	1.23	34.21 (53.5)	$C_3N_{4.7}H_{3.8}$
C <sub>3</sub> N <sub>4</sub> @WO <sub>3</sub> (Tube)	8.65	18.81	2.17	29.63 (46.4)	C <sub>3</sub> N <sub>5.6</sub> H <sub>9.0</sub>
C <sub>3</sub> N <sub>4</sub> @ WO <sub>3</sub>	14.63	25.93	2.17	42.02 (49.9)	C <sub>3</sub> N <sub>4.6</sub> H <sub>3.6</sub>

Table 4.1. CHN elemental analysis for  $C_3N_4$  metal oxide composite products.

The IR spectra of the  $C_3N_4@SiO_2$  composites are shown in Figure 4.7. Both feature N-H stretching absorptions above 3100 cm<sup>-1</sup>, though there are two absorptions for the Schlenk product as opposed to one broad absorption for the tube products. The C-N and C=N absoptions in the 1200-1700 cm<sup>-1</sup> region partly overlap with the intense Si-O stretch from 1000-1200 cm<sup>-1</sup>. Though this masks the lower energy C-N absorptions, it is clear that the Schlenk product has higher intensity for the absorptions at higher frequency resembling absorption shapes of TCM more than the C<sub>3</sub>N<sub>4</sub> product (see Chapter 2 IR spectra). The ring breathing mode at 800 cm<sup>-1</sup> is present for each, and a sharp absoprtion for Si-O bend appears below 500 cm<sup>-1</sup>.



Figure 4.7 – IR spectra of (a)  $C_3N_4@SiO_2(Schlenk)$  and (b)  $C_3N_4@SiO_2(Tube)$ . The large broad peak centered at 1100 cm<sup>-1</sup> is characteristic of SiO<sub>2</sub>.

The IR spectra of the  $C_3N_4@TiO_2$  composites are shown in Figure 4.8. Both spectra feature a single broad N-H stretching absorption above 3100 cm<sup>-1</sup>. There is some

difference in the relative intensities of the C-N and C=N absorptions in the 1200-1700  $\text{cm}^{-1}$ . The absorptions for the Schlenk product are less resolved and higher intensity at the higher frequency end. The tube product has more distinct absorptions in this region, as well as an extended shoulder absorbance towards 1000 cm<sup>-1</sup>. The ring breathing mode at 800 cm<sup>-1</sup> is visible for the tube product, but is overshadowed by the Ti-O stretching absorption in the Schlenk product's spectrum.



Figure 4.8 – IR spectra of  $C_3N_4$ @TiO<sub>2</sub>(Schlenk) and (b)  $C_3N_4$ @TiO<sub>2</sub>(Tube). The large broad peak centered at 800 cm<sup>-1</sup> is characteristic of TiO<sub>2</sub>.

The IR spectra of the  $C_3N_4@Al_2O_3$  composites are shown in Figure 4.9. The N-H stretching absorptions above 3100 cm<sup>-1</sup> appeared for each product, though the Schlenk has two absorptions. The relative intensities of the C-N and C=N absorptions in the

1200-1700 cm<sup>-1</sup> differ significantly for the two products. The Schlenk product has one intense absorption at 1650 cm<sup>-1</sup> with a broad shoulder feature extending to 1200 cm<sup>-1</sup>. This higher intensity at 1650 cm<sup>-1</sup> could either be the result of more TCM like character or from an O-H bend from the hydroxylated  $Al_2O_3$  surface. The tube product has the evenly intense absorptions in this region consistent with the other high-temperature tube products. The large broad absorption from 500-900 cm<sup>-1</sup> represents Al-O-Al bend and Al-O stretches, and covers the C-N ring mode normally observed at 800 cm<sup>-1</sup>.<sup>121</sup>



Figure 4.9 – IR spectra of (a)  $C_3N_4@$ (Schlenk) and (b)  $C_3N_4@$ (Tube). The broad absorptions centered at 700 cm<sup>-1</sup> are characteristic of  $Al_2O_3$ .

The IR spectra of the different  $C_3N_4@WO_3$  and  $WO_3@C_3N_4$  composites are shown in Figure 4.10. The absorption intensities of  $C_3N_4@WO_3$ (anneal) in Figure 4.10b were low due to the need to reduce the sample concentration in the pellet in order to maintain transmission through the dark product. All products had the broad absorption above 3100 cm<sup>-1</sup> for N-H stretches. The C-N and C=N absorptions from 1200-1700 cm<sup>-1</sup> varied slightly in intensity and shape. The Schlenk product had broader, less resolved absorptions consistent with the other Schlenk C<sub>3</sub>N<sub>4</sub> products. The same set of absorptions in Figure 4.10d are fairly sharp, however, since this started with previously synthesized C<sub>3</sub>N<sub>4</sub>, no significant changes were expected from only heating to 300 °C. The tube C<sub>3</sub>N<sub>4</sub>@WO<sub>3</sub>(Tube) product is similar to the Schlenk, though there is an extended shoulder absorbance towards 1000 cm<sup>-1</sup>. The 800 cm<sup>-1</sup> ring absorption is visible in each, but is partially overlapped with the O-W-O stretching at lower frequency.<sup>118</sup>



Figure 4.10 – IR spectra of (a)  $C_3N_4@WO_3$ (Schlenk) (b)  $C_3N_4@WO_3$ (anneal) (c)  $C_3N_4@WO_3$ (Tube), and (d)  $WO_3@C_3N_4$ .

Powder XRD patterns for the  $C_3N_4@SiO_2(Tube), C_3N_4@TiO_2(Tube)$ , and  $C_3N_4@Al_2O_3(Tube)$  products are shown in Figure 4.11. The  $C_3N_4@TiO_2$  composite shows primarily anatase with some rutile TiO\_2. The first peak for rutile overlaps with where the major  $C_3N_4$  peak would be, however the peak for crystalline TiO\_2 phase is much more intense and the  $C_3N_4$  peak is lost in the background. The  $C_3N_4@SiO_2$  pattern shows the  $C_3N_4$  peak 27.5 degrees on the shoulder of the broad peak for amorphous SiO\_2 near 20 degrees. The  $C_3N_4$  peak is also visible in the  $C_3N_4@Al_2O_3$  composite, while the remaining broad peaks fit to nanoscale  $\gamma$ -Al\_2O\_3.



Figure 4.11 – XRD patterns of (a)  $C_3N_4@TiO_2(Tube)$ , (b)  $C_3N_4@SiO_2(Tube)$ , and (c)  $C_3N_4@Al_2O_3(Tube)$ .

The XRD patterns for the  $C_3N_4@WO_3$  and  $WO_3@C_3N_4$  products are shown in Figure 4.12. The low temperature C<sub>3</sub>N<sub>4</sub>@WO<sub>3</sub>(Schlenk) product (Figure 4.12a) and  $WO_3@C_3N_4$  product (Figure 4.12d) both show monoclinic phase  $WO_3$ . The peak for carbon nitride at 27.5 degrees is visible only in the product from  $WO_3@C_3N_4$ . Since this product started with premade  $C_3N_4$ , the structure was already well formed, having this clearly resolved peak. The  $C_3N_4@WO_3$ (Schlenk) product, starting with TCM to form  $C_3N_4$  around WO<sub>3</sub>, may not have grown as well ordered around the oxide support. The  $C_3N_4$ @WO<sub>3</sub>(anneal) (Figure 4.12b) and  $C_3N_4$ @WO<sub>3</sub>(Tube) (Figure 4.12c) show the formation of a different phase. This phase was found to match hydrogen tungsten bronze  $(H_{0.23}WO_3)$ . The amount of hydrogen can vary in these  $H_xWO_3$  structures (0.1<x<0.6), which have previously been prepared by heating above 400 °C under hydrogen,<sup>122</sup> milling under high pressure hydrogen atmosphere,  $^{123}$  or electrodeposition from Na<sub>2</sub>WO<sub>4</sub> in  $H_2SO_4$ .<sup>124</sup>  $H_xWO_3$  products are described as dark blue or black versus the yellow WO<sub>3</sub>, which is consistent with the dark color of these products. Since it is known that C<sub>3</sub>N<sub>4</sub> transports NH<sub>4</sub>Cl as a by-product when annealed, this hydrogen incorporation into the structure of WO<sub>3</sub> may come from volatile NH<sub>3</sub> or HCl.



Figure 4.12 – XRD patterns of (a)  $C_3N_4@WO_3(Schlenk)$ , (b)  $C_3N_4@WO_3(anneal)$ , (c)  $C_3N_4@WO_3(Tube)$ , and (d)  $WO_3$ - $C_3N_4$ .

# Morphology and surface area

The morphologies of the C<sub>3</sub>N<sub>4</sub> composites were investigated with SEM. The C<sub>3</sub>N<sub>4</sub>@SiO<sub>2</sub>(Tube) composite (Figure 4.13A) shows a cluster of 3-5  $\mu$ m aggregates. Compared to unmodified C<sub>3</sub>N<sub>4</sub>, the composite had a very rough surface with small particle features less than 200 nm. The C<sub>3</sub>N<sub>4</sub>@Al<sub>2</sub>O<sub>3</sub>(Tube) composite (Figure 4.13B) shows a 5  $\mu$ m aggregate particle. This has a similar rough surface as the C<sub>3</sub>N<sub>4</sub>@SiO<sub>2</sub>(Tube) composite, though the smaller 100-300 nm particulate features are more distinct. The C<sub>3</sub>N<sub>4</sub>@TiO<sub>2</sub>(Tube) composite (Figure 4.nC,D) shows a 10  $\mu$ m aggregate. The smaller segments are more irregularly shaped and feature void spaces as small as 100 nm in diameter. The walls of these pieces are smoother, and small individual particles are not distinguishable as in the  $C_3N_4@SiO_2(Tube)$  and  $C_3N_4@Al_2O_3$ . (Tube) composites. The WO<sub>3</sub>@C<sub>3</sub>N<sub>4</sub> composite (Figure 4.13E) was made up of particles varying in size from 400 nm to 2µm. These pieces appear dense, with no pores, and aggregate in 5-10 µm clusters. The  $C_3N_4@WO_3(Tube)$  composite (Figure 4.13F) formed aggregates 10 µm and larger. These had very rough surface features including 150-300 nm particles on extended surfaces and 300-500 nm pores along edges.



Figure 4.13 – SEM images of (A)  $C_3N_4@SiO_2(Tube)$ , (B)  $C_3N_4@Al_2O_3(Tube)$ , (C, D)  $C_3N_4@TiO_2(Tube)$ , (E) WO\_3@C\_3N\_4, and (F)  $C_3N_4@WO_3(Tube)$ .

Physical mixtures of  $C_3N_4$  with TiO<sub>2</sub> and WO<sub>3</sub> were prepared by ball milling. The physically-mixed  $C_3N_4$ -TiO<sub>2</sub> composite (Figure 4.14A,B) featured a large 5-10  $\mu$ m aggregate. This had some small 1-2  $\mu$ m segments and had particles visibly embedded on the surface. Figure 4.14B gives a higher magnification of these 100-200 nm particles embedded on a very contoured wall. The physical  $C_3N_4$ -WO<sub>3</sub> composite (Figure 4.14C,D) showed highly textured 1-4 µm segments on a larger aggregate. Figure 4.14D gives a close view of the disordered thin 100 nm plates of WO<sub>3</sub>. These plates appear to be embedded on a wall made up of  $C_3N_4$  particles.



Figure 4.14 – SEM images of ball milled (A, B)  $C_3N_4$ -TiO<sub>2</sub> and (C, D)  $C_3N_4$ -WO<sub>3</sub>.

The BET surface area for oxide products from tube reactions are summarized in Table 4.2 and compared to the bare metal oxides. While these oxide-supported composites had a higher surface area  $C_3N_4$  material, the growth of  $C_3N_4$  as a coating around the oxide reduced its natural surface area. This reduction in surface area was

roughly by a factor of three for each of the oxides. This loss could be dependent on how thick the  $C_3N_4$  coating on the oxide surface was.

Product	Composite Surface Area (m <sup>2</sup> g <sup>-1</sup> )	Oxide Surface Area (m <sup>2</sup> g <sup>-1</sup> )
$C_3N_4@SiO_2$ (Tube)	135.1	300
C <sub>3</sub> N <sub>4</sub> -Al <sub>2</sub> O <sub>3</sub> (Tube)	33.7	100
C <sub>3</sub> N <sub>4</sub> -TiO <sub>2</sub> (Tube)	16.3	44.8

Table 4.2. BET surface areas of C<sub>3</sub>N<sub>4</sub>-metal oxide composites.

# **Optical properties**

The optical absorption properties and band gaps of the composite materials were explored with UV-vis diffuse reflectance spectroscopy and compared with those of bare  $C_3N_4$ , TiO<sub>2</sub>, and WO<sub>3</sub> (Figure 4.15A and B). The raw reflectance data (Figure 4.16A) showed some level of increased reflectance for wavelengths above 380 nm, except for the  $C_3N_4$ -H<sub>x</sub>WO<sub>3</sub> sample due to its very dark color. The reflectance data was converted into absorbance data using the Kubelka-Munk function and plotted against energy to give the Tauc Plot (Figure 4.16B). The indirect band gaps ( $E_g$ ) can be approximated by extrapolating the linear region of the absorption edge to the energy axis (in eV). The  $E_g$  values for  $C_3N_4$ @SiO<sub>2</sub>,  $C_3N_4$ @Al<sub>2</sub>O<sub>3</sub>,  $C_3N_4$ @TiO<sub>2</sub>, and WO<sub>3</sub>@C<sub>3</sub>N<sub>4</sub> were 2.3, 2.45, 2.7, and 2.25 eV respectively. These values are lower than the component semiconducting metal oxides and comparable or slightly lower than bare  $C_3N_4$ .  $C_3N_4$ @WO<sub>3</sub>(Tube) does not have a clear absorption edge in the range measured. H<sub>x</sub>WO<sub>3</sub> is known to be a

conducting material, and its high visible light absorbance overshadows the contribution of  $C_3N_4$  by this measurement.



Figure 4.15 – UV-vis DRS data for  $C_3N_4$ , TiO<sub>2</sub>, and WO<sub>3</sub> (A) reflectance spectra and (B) Tauc plots.



Figure 4.16 - UV-vis DRS data for  $C_3N_4$ -metal oxide composites (A) reflectance spectra and (B) Tauc plots.
# Photocatalytic dye oxidation

The degradation of methyl orange dye was studied with the tube products of the  $C_3N_4$ -metal oxide composites. Figure 4.17 shows the UV light oxidative degradation in air of methyl orange with each composite. These samples absorbed different amount of the dye during the dark stir, up to 50% for C<sub>3</sub>N<sub>4</sub>@Al<sub>2</sub>O<sub>3</sub> and C<sub>3</sub>N<sub>4</sub>@WO<sub>3</sub>, nearly 30% for  $C_3N_4@SiO_2$ , and less than 20% for  $C_3N_4@TiO_2$  and  $WO_3@C_3N_4$ . Bare SiO<sub>2</sub> and Al<sub>2</sub>O<sub>3</sub> were stirred in the dye solution to test if this high degree of absorption was just a factor of their high surface areas. After allowing these to stir for up to 2 hours, Al<sub>2</sub>O<sub>3</sub> absorbed less than 5% and SiO<sub>2</sub> absorbed 15%. The extra absorption on the composites must be due to the combination of high surface area and the presence of  $C_3N_4$ . This could be from interactions with residual amino functional groups on  $C_3N_4$ . Accounting for the dark absorption, the  $C_3N_4@SiO_2$  and  $C_3N_4@Al_2O_3$  composites still had the highest activity, reaching less than 10% dye remaining by 45 minutes. The  $C_3N_4$ @TiO<sub>2</sub>, WO<sub>3</sub>@C<sub>3</sub>N<sub>4</sub>, and C<sub>3</sub>N<sub>4</sub>@WO<sub>3</sub> reached 40%, 50%, and 30% dye remaining, respectively, after 2 hours of total UV illumination. Comparing the rates (Table 4.3 and Figure 4.17B),  $C_3N_4@Al_2O_3$  is slightly higher than  $C_3N_4@SiO_2$  and 7-8 times more active than the remaining composites.



Figure 4.17 – UV-vis analysis of photodegradation of MO dye (A) UV light data and (B) UV linear rate data.

Visible light degradation using the same catalysts was tested using 400 nm cut-off filters. Dark absorption behavior was similar, though  $C_3N_4@SiO_2$  only absorbed near 30% of the initial dye in this test. Total activity was lower using visible light, with samples ranging from 10-40% decrease in dye concentration after 150 minutes of illumination. In contrast,  $C_3N_4@TiO_2$  maintained a visible rate similar to what was observed under UV light (Figure 4.18B and Table 4.3). The remaining rates decreased from  $C_3N_4@Al_2O_3$ ,  $C_3N_4@SiO_2$ ,  $C_3N_4@WO_3$ , and  $WO_3@C_3N_4$ .

Table 4.3. Rate constants (min<sup>-1</sup>) for UV and visible light degradation of methyl orange dye using  $C_3N_4$ -metal oxide composites.

	C <sub>3</sub> N <sub>4</sub> @SiO <sub>2</sub>	C <sub>3</sub> N <sub>4</sub> @Al <sub>2</sub> O <sub>3</sub>	C <sub>3</sub> N <sub>4</sub> @TiO <sub>2</sub>	WO <sub>3</sub> @C <sub>3</sub> N <sub>4</sub>	C <sub>3</sub> N <sub>4</sub> @WO <sub>3</sub>
UV	0.0399	0.0429	0.0062	0.0052	0.0050
	(R <sup>2</sup> = 0.992)	( $R^2 = 0.994$ )	(R <sup>2</sup> = 0.998)	( $R^2 = 0.970$ )	(R <sup>2</sup> = 0.979)
Visible	0.0033	0.0045	0.0063	0.0004	0.0025
(>400 nm)	(R <sup>2</sup> = 0.980)	(R <sup>2</sup> = 0.999)	(R <sup>2</sup> = 0.982)	(R <sup>2</sup> = 0.43)	(R <sup>2</sup> = 0.86)



Figure 4.18 – UV-vis analysis of photodegradation of MO dye (A) visible light data and (B) visible linear rate data.

The UV and visible light degradation of methyl orange were also tested using ball milled mixtures of  $C_3N_4$  with TiO<sub>2</sub> or WO<sub>3</sub>. The milled composites showed slightly higher dark absorption of the dye, nearing a 15-20% loss compared to 5% for unmodified  $C_3N_4$ . Since the composites were milled, they were finer powders than the  $C_3N_4$  used in this experiment, giving more potential surface sites for the dye to adsorb. The  $C_3N_4$ -TiO<sub>2</sub> composite was most active, reaching less than 10% of dye remaining by 100 minutes of illumination. The  $C_3N_4$ -WO<sub>3</sub> reduced the dye concentration to 20%, and  $C_3N_4$  made it to 40% in the same time. The plots for the linear rate fits and rate constants are shown in Figure 4.19B and Table 4.4. Since TiO<sub>2</sub> benefits from higher activity under UV light due to its band gap (3.2 eV), its having the highest rate is unsurprising.

The visible light experiment is shown in Figure 4.20A. The dark absorption varied slightly from the UV experiment, with each milled composite showing only a 10% loss. After 150 minutes of visible light illumination, each composite reached 70% of dye remaining, while  $C_3N_4$  reached 20%. The rate fits and rate constants are shown in Figure 4.20B and Table 4.4.  $C_3N_4$ -TiO<sub>2</sub> and  $C_3N_4$ -WO<sub>3</sub> had identical visible light rates, which were 6.6% of the UV rate for  $C_3N_4$ -TiO<sub>2</sub> and 14.6% of the rate for  $C_3N_4$ -WO<sub>3</sub>. The rate for  $C_3N_4$ -WO<sub>3</sub>.

Table 4.4. Rate constants (min<sup>-1</sup>) for UV and visible light degradation of methyl orange dye using  $C_3N_4$  and milled  $C_3N_4$ -metal oxide composites.

	C <sub>3</sub> N <sub>4</sub> (Parr)	C <sub>3</sub> N <sub>4</sub> -TiO <sub>2</sub>	C <sub>3</sub> N <sub>4</sub> -WO <sub>3</sub>
UV	0.0094 ( $R^2 = 0.986$ )	$0.0317 \\ (R^2 = 0.988)$	$0.0144 \\ (R^2 = 0.992)$
Visible (>400nm)	0.0016 (R <sup>2</sup> = 0.993)	$0.0021 \\ (R^2 = 0.984)$	0.0021 (R <sup>2</sup> = 0.980)



Figure 4.19 - UV-vis analysis of photodegradation of MO dye using milled  $C_3N_4$ -TiO<sub>2</sub> and  $C_3N_4$ -WO<sub>3</sub> composites (A) UV light data and (B) UV linear rate data.



 $\label{eq:Figure 4.20-UV-vis analysis of photodegradation of MO dye using C_3N_4-TiO_2(milled) \\ and C_3N_4-WO_3(milled) \ composites (A) \ visible \ light \ data \ and \ (B) \ visible \ linear \ rate \ data.$ 

# Photocatalytic hydrogen evolution via water-splitting

Photocatalytic hydrogen evolution tests using  $C_3N_4$ -metal oxide composites contained a target 1 wt% platinum as co-catalyst. The tests used RGA-MS to monitor the amount of hydrogen produced in real time. Figure 4.21 shows the activity for  $C_3N_4@TiO_2(Tube)$  over 4 1 hour illumination cycles. The activity remained stable from cycle to cycle and did not exhibit a first hour activation step like other  $C_3N_4$  products described in the previous chapters. The rate of 190 µmol h<sup>-1</sup> g<sup>-1</sup> for  $C_3N_4@TiO_2(Tube)$ was nearly 3 times that of  $C_3N_4$  without a support, but significantly lower than the activity of unmodified P25 TiO<sub>2</sub> (Table 4.5). Since  $C_3N_4$  was formed on the surface of TiO<sub>2</sub>, it is likely blocking the UV photons from being absorbed by TiO<sub>2</sub>. In this case,  $C_3N_4$  would be the primary light absorber, and any charge transferred to the TiO<sub>2</sub> core would not be accessible to facilitate hydrogen reduction. The increased activity over unsupported  $C_3N_4$  could be from a smaller contribution from TiO<sub>2</sub> or a result of the surface area increase.

Sample	$H_2 Rate \ (\mu mol \ h^{-1} \ g^{-1})$
P25 TiO <sub>2</sub>	20,000 (1,400)
C <sub>3</sub> N <sub>4</sub> (Tube)	68 (8)
C <sub>3</sub> N <sub>4</sub> @TiO <sub>2</sub> (Tube)	190 (11)
C <sub>3</sub> N <sub>4</sub> -TiO <sub>2</sub> (mill)	106 (38)
$C_3N_4@SiO_2(Tube)$	44 (14)
$C_3N_4@Al_2O_3$	0

Table 4.5. Average hydrogen evolution rates for 1 wt% Pt deposited  $TiO_2$ ,  $C_3N_4$ , and  $C_3N_4$ -metal oxide composites over 4 1-hour cycles (standard deviation reported in parentheses).



Figure 4.21 – Photocatalytic hydrogen evolution using  $C_3N_4$ @TiO<sub>2</sub>(Tube) with 1% Pt cocatalyst. The UV lamp was turned on for 1 hour cycles and the reactor was purged with argon for 10 minutes between each cycle.

The ball-milled C<sub>3</sub>N<sub>4</sub>-TiO<sub>2</sub> mixture was less active than the surface-coated

 $C_3N_4@TiO_2$  with a rate of 110 µmol h<sup>-1</sup> g<sup>-1</sup> (Table 4.5). This was still an improvement over unsupported  $C_3N_4$ , but remained much less that the activity of TiO<sub>2</sub>. This physical mixture was prepared with a 2:1 ratio of  $C_3N_4$  to TiO<sub>2</sub>. While it should be an intimate mixture, it is possible that the particles of TiO<sub>2</sub> embedded into  $C_3N_4$ . This could mask the ability of TiO<sub>2</sub> to absorb light as in the case with  $C_3N_4@TiO_2$ .

The  $C_3N_4@SiO_2(Tube)$  and  $C_3N_4@Al_2O_3(Tube)$  product showed a decrease in activity compared to the unsupported  $C_3N_4$ , with a rate of 44 µmol h<sup>-1</sup> g<sup>-1</sup> and no hydrogen production observed respectively (Table 4.5). Unlike the TiO<sub>2</sub> composites, both SiO<sub>2</sub> and Al<sub>2</sub>O<sub>3</sub> are photochemically inert, so any enhancement activity would be solely linked to their increased surface areas. The surface of these products was shown to be much higher than unsupported  $C_3N_4$  in Table 4.2. These products were from the 500 °C tube reactions and, based on IR and elemental analysis, should be fully formed  $C_3N_4$ . Their band gaps were also in the range of some of the products shown in Chapter 2. The presence of these oxides may have interfered with successful surface deposition of platinum. The absence or diminished amount of co-catalyst could account for a drop in activity of these samples.

# 4.4 Conclusions

This chapter has described the synthesis of  $C_3N_4$  on the surface of several metal oxide supports (SiO<sub>2</sub>, Al<sub>2</sub>O<sub>3</sub>, TiO<sub>2</sub>, and WO<sub>3</sub>) and the synthesis of WO<sub>3</sub> on a  $C_3N_4$ support. Low temperature synthesis conditions in the Schlenk reactor set-up yielded poorly-formed products, which had difficulty initiating the decomposition. Heating the coated precursors in the 500 °C tube reactor showed more typical reactions, and gave products with properties similar to unsupported  $C_3N_4$ . Heating  $C_3N_4$  with WO<sub>3</sub> under these conditions led  $H_xWO_3$ , hydrogen tungsten bronze, from the reaction between  $WO_3$ with hydrogen containing volatiles. Because of this, the alternate approach to prepare  $WO_3$  on a  $C_3N_4$  support was used. Photocatalytic activity was evaluated for both the degradation of methyl orange dye and hydrogen evolution. The high surface area  $C_3N_4$ @SiO<sub>2</sub> and  $C_3N_4$ @Al<sub>2</sub>O<sub>3</sub> showed a high degree of surface adsorption for the dye, not matched by either of the individual components. The higher amount of surface adsorption seemed to contribute to the UV light degradation activity for these to composites. By strongly adsorbing to the C<sub>3</sub>N<sub>4</sub> surface, the dye could be more readily oxidized.

## **CHAPTER 5**

# ENHANCED PHOTOCATALYSIS OF TRANSITION-METAL SURFACE-MODIFIED TITANIUM DIOXIDE

# 5.1 Introduction

Titanium dioxide (TiO<sub>2</sub>) is a well-studied semiconducting material for photocatalysis and has promise in solar energy conversion.<sup>37-38, 40, 42</sup> Anatase TiO<sub>2</sub> ( $E_g = 3.2 \text{ eV}$ ) can perform the photolysis of water into hydrogen and oxygen gases in an electrochemical cell.<sup>39</sup> The deposition of platinum metal on the surface of anatase TiO<sub>2</sub> forms a highly-active UV photochemical water-splitting catalyst.<sup>46-47, 125</sup> Platinum metal regions on titania act as a surface catalyst that funnel photoexcited TiO<sub>2</sub> conduction band electrons to protons in solution, that are then reduced to H<sub>2</sub>. In these types of photocatalytic reactions, a sacrificial oxidant is needed to remove photogenerated holes from the semiconductor's valence band and maintain overall charge balance.<sup>126-127</sup> However, even in low ~1 wt% co-catalyst concentrations, platinum has disadvantages related to its high cost and low abundance.

There are extensive efforts to improve the photochemical properties of TiO<sub>2</sub>based materials that do not involve noble metal co-catalysts and also may enable absorption of solar radiation for water-splitting hydrogen production.<sup>128</sup> A wide range of metals (Cu,<sup>129-130</sup> Ni,<sup>131</sup> Co,<sup>132</sup> Cr, Fe, Mo, V, and W<sup>133</sup>) and non-metals (N,<sup>134</sup> F,<sup>135</sup> P,<sup>136</sup> S<sup>137</sup>) have been doped into the bulk structure of TiO<sub>2</sub> by solution or solid-state methods in order to modify the bulk titania band structure and improve visible light absorption or electron-hole charge recombination kinetics. Most prior work has used gas phase H<sub>2</sub> or solution phase NaBH<sub>4</sub> reduction of metal ions mixed with TiO<sub>2</sub> precursors. Previous work by Coleman from our group incorporated ~5-10 at% of several 3d metals into rutile  $TiO_2$  via exothermic solid-state metathesis reactions.<sup>138</sup> These metathesis-derived titanias showed lower photoactivity versus undoped materials, though the 3d metals gave visible color to the products. Modifying  $TiO_2$  with two metals in its bulk structure, such as Co-Ni or Cu-Ce, has been studied for photocatalytic applications in dye degradations and water-splitting.<sup>139-140</sup> CuO coatings or intimate copper oxide mixtures with  $TiO_2$  exhibit partial reduction of the copper oxide during observed enhanced photocatalytic hydrogen evolution.<sup>141-142</sup>

Photodeposition has been utilized for removal of metal ions from aqueous environments with semiconducting WO<sub>3</sub> or TiO<sub>2</sub> as photoactive substrates.<sup>48, 50, 143-146</sup> In addition to environmental remediation, photo-assisted deposition can also function as a potentially facile, low-temperature method to modify photoactive surfaces such as TiO<sub>2</sub> with metal or metal oxide co-catalysts.<sup>47</sup> Copper and nickel have shown promise as photodeposited co-catalysts or *in-situ* catalysts for photocatalytic hydrogen evolution using high surface area titania nanotubes or mesoporous structures.<sup>147-149</sup>

This Chapter describes the UV photodeposition of several 3d metals (Mn, Fe, Co, Ni, Cu) onto the surface of photoactive commercial Degussa P25-TiO<sub>2</sub> (primarily anatase) using a method analogous to noble metal (*e.g.*, Ag, Pt) photoreduction processes, and is an expansion of published work.<sup>88</sup> The bulk titania structure and particle morphologies after 3d metal surface modification are unchanged. The surface and optical properties of the modified TiO<sub>2</sub> were investigated using X-ray photoelectron (XPS) and UV-vis spectroscopies. XPS surface compositions were compared to bulk chemical analysis by inductively-coupled plasma optical emission spectroscopy (ICP-OES) to

better inform the metal deposition process. Real-time reactor headspace gas analysis by portable residual gas analysis mass spectrometry (RGA-MS) was used to quantify the photoactivity of these catalysts for hydrogen production. The 3d metal surface-modified  $TiO_2$  materials show significant improvements in the photoreduction of water to hydrogen gas to levels approaching half that of similarly prepared platinum coated titanias.

# 5.2 Experimental

## Reagents

Reagents used as received were TiO<sub>2</sub> (Degussa P25, ~50 nm size and 45 m<sup>2</sup>/g), MnCl<sub>2</sub> (Aldrich, 98%), FeCl<sub>2</sub>·4H<sub>2</sub>O (Aldrich, 99%), CoCl<sub>2</sub>·6H<sub>2</sub>O (Aldrich, reagent grade), NiCl<sub>2</sub>·6H<sub>2</sub>O (Aldrich, ReagentPlus), CuCl<sub>2</sub>·2H<sub>2</sub>O (Aldrich, 99+%), H<sub>2</sub>PtCl<sub>6</sub>·6H<sub>2</sub>O (99.9%, Strem), methanol (ACS certified, Fisher), and methylene blue (Alfa Aesar, high purity). Deionized water (18 M $\Omega$ ) was prepared using a Photronix MiniQuad system.

# Modification of TiO<sub>2</sub> surface with photodeposited transition metals

<u>Single metal (1%) modified TiO</u><sub>2</sub>. In a 50 mL Erlenmeyer flask, 4 mg (0.017 mmol) of  $MCl_2 \cdot 6H_2O$  (M = Co or Ni) or 3 mg (0.017 mmol) of  $CuCl_2 \cdot 2H_2O$  was dissolved in 30 mL of 50 vol% aqueous methanol. TiO<sub>2</sub> (100 mg, 1.25 mmol) was added to the solution. The flask was covered with a septum and the solution degassed with argon while stirring. The flask was exposed to a water-jacketed UV light (450 W Hg lamp, Ace-Hanovia) for 90 minutes under an argon blanket. Solids were isolated by

centrifugation (3000 rpm), washed with 5-10 mL of methanol, centrifuged again from methanol, and then dried in air. The product from nominally 1 wt% metal relative to  $TiO_2$  was denoted as M(1%):TiO<sub>2</sub>.

<u>Single metal (2%) modified TiO</u><sub>2</sub>. In a 50 mL Erlenmeyer flask, 5 mg (0.039 mmol) of MnCl<sub>2</sub>, 7 mg (0.035 mmol) of FeCl<sub>2</sub>·4H<sub>2</sub>O, 8 mg (0.034 mmol) of MCl<sub>2</sub>·6H<sub>2</sub>O (M = Co or Ni) or 6 mg (0.034 mmol) of CuCl<sub>2</sub>·2H<sub>2</sub>O was dissolved in 30 mL of 50 vol% aqueous methanol. TiO<sub>2</sub> (100 mg, 1.25 mmol) was added to the solution. The flask was covered with a septum and the solution degassed with argon while stirring. The flask was exposed to a water-jacketed UV light (450 W Hg lamp, Ace-Hanovia) for 90 minutes under an argon blanket. Solids were isolated by centrifugation (3000 rpm), washed with 5-10 mL of methanol, centrifuged again from methanol, and then dried in air. The product from nominally 2 wt% metal relative to TiO<sub>2</sub> was denoted as M(2%):TiO<sub>2</sub>.

<u>Mixed metal modified TiO</u><sub>2</sub>. In a 50 mL Erlenmeyer flask, a combination using two of these metals: 6 mg (0.035 mmol) of CuCl<sub>2</sub>·2H<sub>2</sub>O, 8 mg (0.034 mmol) of CoCl<sub>2</sub>·6H<sub>2</sub>O, or 8 mg (0.034 mmol) of NiCl<sub>2</sub>·6H<sub>2</sub>O, was dissolved in 30 mL of 50 vol% aqueous methanol. TiO<sub>2</sub> (200 mg, 2.5 mmol) was added to the solution. The flask was covered with a septum and the solution degassed with argon while stirring. The flask was exposed to UV light for 90 minutes under an argon blanket. Solids were isolated by centrifugation (3000 rpm), washed with 5-10 mL of methanol, centrifuged again from methanol, and then dried in air. These mixed metal products each had nominally 1 wt% metal relative to TiO<sub>2</sub> and were denoted M(1%)/M'(1%):TiO<sub>2</sub>.

<u>Platinum modified TiO<sub>2</sub></u>. In a 50 mL Erlenmeyer flask, 2.4 mg (0.006 mmol) of  $H_2PtCl_6$  was dissolved in 30 mL of 50 vol% aqueous methanol. TiO<sub>2</sub> (100 mg, 1.25

mmol) was added to the solution. The flask was covered with a septum, and the solution was degassed with argon while stirring. The flask was exposed to UV light (450 W Hg) for 90 minutes under an argon blanket. Solids were isolated by centrifugation (3000 rpm), rinsed with methanol, and dried in air. This platinum loading is nominally 1 wt% Pt relative to  $TiO_2$  and is denoted Pt(1%):TiO\_2.

## *Characterization*

A Bruker D8 DaVinci powder X-ray diffractometer was used for analysis of powders uniformly placed on glass slides (Cu K $\alpha$ , 0.050°/s step). X-ray photoelectron spectroscopy (XPS) was performed on a Kratos Axis Ultra XPS (Al K $\alpha$ ) with samples mounted on carbon tape. XPS peak analysis was performed using CasaXPS software (www.casaxps.com). Inductively-coupled plasma optical emission spectroscopy (ICP-OES, Perkin Elmer Optima DV 7000) was used for bulk elemental analysis. Samples were dissolved in a heated 5:1 mixture of concentrated H<sub>2</sub>SO<sub>4</sub>:HNO<sub>3</sub> and diluted with 5% HNO<sub>3</sub>. Calibration standards were prepared from 1000 ppm commercial ICP standards for Ti, Co, Cu (Alfa Aesar), and Ni (Fluka). Diffuse reflectance UV-vis data was collected on a Cary 5000 Series Spectrophotometer (300-800 nm, 10 nm/s). Solid samples were loaded into a round compression sample holder with a 1 cm diameter sampling area. Raw reflectance data was converted to Kubelka-Munk units [(F(R)], and band gaps were calculated using  $[F(R)hv]^{1/2}$  extrapolations of linear regions of curve. Scanning electron microscopy (SEM) images were collected on a Hitachi S-4800 FE-SEM with an accelerating voltage of 1.3 kV. Samples were mounted on aluminum stubs using carbon tape and were not coated prior to analysis. BET surface area measurements

were collected on a Quantachrome Nova 1200 using approximately 50 mg of solid sample.

# Photocatalytic reactions

# Oxidative degradation of organic dyes

The photodegradation of methylene blue ( $C_{16}H_{18}CIN_3S$ ) was studied using the M(2%)TiO<sub>2</sub> and M(1%)M'(1%):TiO<sub>2</sub> samples. A stock 10<sup>-3</sup> M dye solution was prepared by dissolving the solid dye in 18 M $\Omega$  water. The stock solution was diluted to 2·10<sup>-5</sup> M to fit the peak at  $\lambda_{max}$  in the range ( $\lambda_{MB}$ =665 nm, A<1.50) of the UV-vis spectrophotometer (Agilent 8453). To 30 mL beakers, 10 mL of dye solution and 10 mg of catalyst were added. An additional beaker containing only dye was used to evaluate self-degradation in the absence of catalyst. Samples were stirred in the dark for a minimum of 30 minutes to equilibrate dye absorption. Samples were exposed to UV light (Ace-Hanovia, 450 W Hg) in 5 minute intervals. After each irradiation period, the catalysts were separated from the dye solution by centrifugation and the UV-vis spectrum of each solution was collected to quantify the remaining concentration of dye. Experiments were typically carried out for less than 1 hour of total UV light illumination before total degradation of the dye.

## Photocatalytic hydrogen evolution.

Reactions were carried out in a 50 mL Schlenk flask containing 10 mg (0.125 mmol) of the titania catalyst suspended in 10 mL of 50 vol% aqueous methanol solution. The reactor was purged with argon and connected to the capillary of a Stanford Research Systems quadrupole mass spectrometer 300 series Residual Gas Analyzer via the flask side arm. Masses corresponding to N<sub>2</sub> (28), Ar (40), H<sub>2</sub> (2), O<sub>2</sub> (32), CO<sub>2</sub> (44), and H<sub>2</sub>O (18) were monitored as a function of time. The flask remained under a constant blanket of argon for the entire experiment, supplied from a Schlenk line with an oil bubbler. Following a 30 minute period for instrument and background stabilization, the flask was illuminated with a UV lamp in 60 minute cycles with an approximate photon flux of 4.0 x  $10^{17}$  photon s<sup>-1</sup>. Between each cycle, the reactor was opened briefly and its headspace flushed with argon. Pressure signals were converted to molar amounts of evolved H<sub>2</sub> based on a calibration curve created using known concentration H<sub>2</sub>/Ar mixtures. Additional details about calibration and experimental set-up can be found in the appendix.

# In-situ metal photodeposition and hydrogen evolution.

This experiment used the same reactor/analysis setup as described in the previous section. Here, 10 mg (0.125 mmol) of Degussa P25-TiO<sub>2</sub> was suspended in 10 mL of 50 vol% of aqueous methanol containing 0.4 mg (0.0017 mmol) of MCl<sub>2</sub>· 6H<sub>2</sub>O (M = Co, Ni) or 0.3 mg (0.0017 mmol) of CuCl<sub>2</sub>· 2H<sub>2</sub>O. The reactor was purged with argon and connected to the RGA as described above. Following a 30 minute period for instrument and background stabilization, the flask was illuminated with a UV lamp in 60 minute cycles. The reactor headspace was flushed with argon between each cycle. In order to verify the critical nature of the methanol sacrificial oxidant in metal deposition and hydrogen evolution, 10 mg (0.125 mmol) of Degussa P25-TiO<sub>2</sub> was suspended in a 10 mL solution 18 MΩ H<sub>2</sub>O with 3 mg (0.017 mmol) of CuCl<sub>2</sub>· 2H<sub>2</sub>O and was UV irradiated for two cycles, and then 2 mL of methanol was added for the third UV cycle onward.

#### 5.3 Results and Discussion

# *Photochemical M:TiO*<sub>2</sub> synthesis.

Commercial Degussa P25-TiO<sub>2</sub> nanoparticles (~50 nm, 80:20 anatase:rutile) were surface-modified with the 3d metal ions manganese, iron, cobalt, nickel, and copper using solution UV photoreduction in the presence of a sacrificial methanol oxidant. P25-TiO<sub>2</sub> was dispersed in a 50 vol% aqueous methanol solution containing dissolved metal dichlorides. The solution was degassed with argon prior to UV irradiation to favor inert (non-oxidizing) conditions. The initial solutions contained sufficient metal chloride to deposit either 1 wt% or 2 wt% metal onto the suspended titania powder. Following UV exposure, the titania suspension had darkened in color (*e.g.*, grey Ni:TiO<sub>2</sub>, purple Cu:TiO<sub>2</sub>), and the washed and dried M:TiO<sub>2</sub> solids were white (Mn), light yellow (Fe), light grey (Ni), or light blue (Co and Cu). When two different metals were present in the photodeposition solution (Co/Ni, Co/Cu, Ni/Cu), the M:TiO<sub>2</sub> products had a light blue coloration similar to the Cu and Co products. Comparable visible titania colorations was observed for the nominally 1 wt% and 2 wt% metal loadings.

# *M*:*TiO*<sup>2</sup> structure, composition, and surface analysis.

Powder X-ray diffraction (XRD) of the photodeposited M:TiO<sub>2</sub> products showed predominately anatase TiO<sub>2</sub> with a small rutile component in ratios that were unchanged from the starting P25-TiO<sub>2</sub> (Figure 5.1). No additional XRD peaks corresponding to other metals or metal oxide phases were observed, suggesting they are present in amounts below the few percent XRD detection limits, or that they are on the surface in a noncrystalline form.



Figure 5.1 – XRD patterns for select M(2%):TiO<sub>2</sub> products, rutile phase TiO<sub>2</sub> is marked with an asterisk.

The photochemical reactions between P25-TiO<sub>2</sub> and metal halide solutions also did not lead to detectable physical changes in the titania powders. A BET surface area of  $44.8 \text{ m}^2/\text{g}$  was measured for P25-TiO<sub>2</sub>, which is essentially the same as the 44.6 m<sup>2</sup>/g surface area measured for a surface-modified Cu(1%):TiO<sub>2</sub> sample. Scanning electron microscopy (SEM) analysis also showed that the original TiO<sub>2</sub> particle sizes near 50 nm were retained in the Cu(1%):TiO<sub>2</sub> photoproduct (Figure 5.2). Aggregated small particles near 50 nm were similarly observed for Ni and Co deposition products.



Figure 5.2 – SEM images of (A) P25 TiO<sub>2</sub> and (B) Cu(1%):TiO<sub>2</sub> photoreaction product.

Bulk elemental analysis using ICP-OES showed the metals deposited onto TiO<sub>2</sub>, and the M:Ti atomic ratios are summarized in Table 5.1. At the targeted 1 wt% metal loading, the ideal M:Ti atomic ratio would be 1.3 : 98.7 (2.6 : 97.4 for 2 wt%) if all of the metal present in solution was deposited onto the P25-TiO<sub>2</sub> particle surface. Cobalt and nickel were detected at less than 10% of the ideal 1 wt% loading value, while copper approached half of the ideal loading. In the mixed-metal samples, the cobalt and nickel content remained low (<0.07 at% vs. Ti), while the amount of copper was closer to its ideal content based on solution concentrations; thus, the photodeposition of copper onto semiconducting  $TiO_2$  is favored over the other metals. A doubling of the metal chloride solution concentration to 2 wt% did not result in a doubling of metal product content. Similar metal deposition trends have been observed in related photochemical contexts.<sup>149</sup> These surface deposition results are consistent with metal activity series trends and reduction potentials ( $M^{2+} + 2e \rightarrow M$  with ease of reduction Cu>2H>>Ni>Co>Fe>Mn); for example copper is the most easily-reduced of the metal cations and protons.<sup>150</sup> Copper is also the only one of the metal cations that is easier to reduce than  $H^+$  to  $H_2$ . In

the photo-assisted solution reduction process, it is generally accepted that a sacrificial methanol oxidant interacts with the valence band holes in photoexcited TiO<sub>2</sub>, while the photoexcited conduction band electrons in TiO<sub>2</sub> are used for solution H<sup>+</sup> to H<sub>2</sub> reduction and also surface metal reduction (e.g., Cu<sup>2+</sup>  $\rightarrow$  Cu<sup>+</sup>/Cu<sup>0</sup>).<sup>40, 126</sup> This catalytic redox cycle parallels the mechanisms used for platinum-coated titanias produced by UV photoreduction of PtCl<sub>6</sub><sup>2-</sup> in methanol solutions.

Sample	M : Ti (at%) ICP	M : Ti (at%) XPS	M 2p <sub>3/2</sub> (eV) <sup>a</sup>	Ti 2p <sub>3/2</sub> (eV)
Mn(2%):TiO <sub>2</sub>	n/a	0.9 : 99.1	641.5	458.5
Fe(2%):TiO <sub>2</sub>	n/a	1.4 : 98.6	710.2	457.9, 458.9
Co(1%):TiO <sub>2</sub>	0.04 : 99.96	0.3 : 99.7	780.0	458.5
Co(2%):TiO <sub>2</sub>	0.07 : 99.93	1.3 : 98.7	780.8, 786.6	458.6
Ni(1%):TiO <sub>2</sub>	0.08 : 99.92	0.5 : 99.5	855.3, 861.5	458.5
Ni(2%):TiO <sub>2</sub>	0.08 : 99.92	3.4 : 96.6	855.8, 860.9	458.6, 461.3
Cu(1%):TiO <sub>2</sub>	0.44 : 99.56	1.7 : 98.3	932.3	458.4
Cu(2%):TiO <sub>2</sub>	0.64 : 99.36	3.5 : 96.5	932.6	458.4
Co(1%)/Ni(1%):TiO <sub>2</sub>	<u>Co/Ni : Ti</u> 0.05/0.06 : 99.89	<u>Co/Ni : Ti</u> 0.6/0.4 : 99.0	Ni: 855.8 Co: 782.0	458.5
Co(1%)/Cu(1%):TiO <sub>2</sub>	<u>Co/Cu : Ti</u> 0.02/0.97 : 99.01	<u>Co/Cu : Ti</u> 0/5.2 : 94.8	Cu: 932.2	458.4
Ni(1%)/Cu(1%):TiO <sub>2</sub>	<u>Ni/Cu : Ti</u> 0.03/1.02 : 98.95	<u>Ni/Cu : Ti</u> 0/6.0 : 94.0	Cu: 933.0	458.8

Table 5.1. ICP-OES and XPS compositions and peak positions of  $M(x\%)TiO_2$  samples.

a) Literature values for M  $2p_{3/2}$  from NIST XPS database:<sup>151</sup> Mn(0/+2/+4) 639.0/641.9/642.2, Fe(0/+2/+3) 706.9/710.6/711.3, Co(0/+2/+3) 778.1/780.2/781.3,<sup>152-153</sup> Ni(0/+2) 852.8/853-856,<sup>154</sup> Cu(0/+1/+2) 932.4/932.5/933.7.<sup>154</sup>

Surface composition and chemical environment were examined for select  $M:TiO_2$ products using surface-sensitive X-ray photoelectron spectroscopy (XPS). For the nominal 1-2 wt% metal concentrations, signals were observable for Co, Ni, and Cu, though the Co and Ni contents were low and required longer sampling times in order to detect weak signals for these metals above the baseline. Figure 5.3 shows  $2p_{3/2}$  spectra for Ti, Mn, Fe, Co, Ni, Cu peaks in representative M(2%):TiO<sub>2</sub> samples. Table 5.1 summarizes XPS peak positions and relative atomic ratios of the surface metal species. As compared to the bulk ICP composition results, XPS shows higher M:Ti surface atomic ratios. Since the metal species are photodeposited on the surface, rather than being homogenously distributed through the TiO<sub>2</sub> structure, it is not surprising that the XPS surface composition yields a higher M:Ti atomic ratio than bulk analysis. Though the amounts of metal in the Mn(2%) and Fe(2%):TiO<sub>2</sub> samples were not quantified by ICP, the XPS compositions were similar to the other M(2%):TiO<sub>2</sub> samples. For the mixed metal samples, Co(1%)/Ni(1%):TiO<sub>2</sub> have Co and Ni contents similar to that observed for the individual Co(1%) and Ni(1%):TiO<sub>2</sub> samples. In contrast, the mixed-metal samples containing Cu showed no detectable Co or Ni, which is consistent with the bulk ICP results showing decreased photodeposition of Co and Ni when Cu is present in solution. The mixed-metal results provide further support for the ease of copper ion photoreduction onto semiconducting TiO<sub>2</sub> as compared to Co and Ni ions.



Figure 5.3 – Representative XPS spectra for 2p doublets for Ti from Cu(2%):TiO<sub>2</sub>, Mn from Mn(2%):TiO<sub>2</sub>, Fe from Fe(2%):TiO<sub>2</sub>, Co from Co(2%):TiO<sub>2</sub>, Ni from Ni(2%):TiO<sub>2</sub>, and Cu from Cu(2%)TiO<sub>2</sub>.

In addition to surface-sensitive compositions, XPS provides useful information about the nature of the metals deposited on the surface. As shown in the results from Table 5.1, in all M:TiO<sub>2</sub> samples, the Ti  $2p_{3/2}$  peak is between 458.4 and 459.0 eV, agreeing with a  $Ti^{4+}$  environment for TiO<sub>2</sub>. For reference, the starting P25-TiO<sub>2</sub> has a Ti  $2p_{3/2}$  peak centered at 458.2 eV. The O 1s spectra show a major peak at 530.2 eV in all samples, also consistent with the oxide anion for  $TiO_2$ . The other metal peak positions were used to help determine their chemical state on the  $TiO_2$  surface. The  $2p_{3/2}$  peaks positions of Mn, Fe, Co and Ni on P25-TiO<sub>2</sub> are close to their M<sup>2+</sup> reference positions (e.g., MO, M(OH)<sub>2</sub>, or MCl<sub>2</sub>) with a small peak near  $M^{3+}$  reference energies in the Co case. In contrast, the  $2p_{3/2}$  peak in Cu samples indicates the metal has been reduced on the surface to  $Cu^0/Cu^+$  states (e.g.,  $Cu_2O$  or CuCl), which is consistent with related work on copper coated titanias.<sup>141, 154</sup> As noted above, Cu is more readily photodeposited onto P25-TiO<sub>2</sub> than Co or Ni, and XPS compositional analysis supports the idea that the ease of  $Cu^{2+}$  reduction leads to higher surface deposition. The Co(1%)/Ni(1%):TiO<sub>2</sub> sample has a nickel peak position similar to comparable  $Ni^{2+}$  peaks observed in the Ni:TiO<sub>2</sub> samples, but the cobalt peak appeared at slightly higher energy near a  $Co^{3+}$  state. The mixed metal samples show Cu is still in a  $Cu^0/Cu^+$  state, though the Ni(1%)/Cu(1%):TiO<sub>2</sub> sample may have some  $Cu^{2+}$  character. In related studies, chemically-precipitated Cu(OH)<sub>2</sub> or Ni(OH)<sub>2</sub> clusters added to the P25-TiO<sub>2</sub> surface show similar 2p<sub>3/2</sub> shifts from 935 to 933 eV as surface  $Cu^{2+}$  transforms to metallic Cu, and a Ni peak at 856 eV is attributed to surface Ni<sup>2+</sup>.<sup>155-156</sup> Chlorine is also present on the titania surface after photoreaction with metal chloride solutions, and higher surface chlorine is detected for Mn, Co and Ni cations versus partially reduced Cu products (Table 5.2). This is

consistent with metal cations deposited or reacted with oxide surface sites that are accompanied by chloride anions while reduced Cu on the titania surface requires less chloride anion charge balance.

Sample	M (at%)	Cl (at%)
Mn(2%):TiO <sub>2</sub>	50.28	49.72
Fe(2%):TiO <sub>2</sub>	65.36	34.64
Co(1%):TiO <sub>2</sub>	25.5	74.5
Co(2%):TiO <sub>2</sub>	40.59	59.41
Ni(1%):TiO <sub>2</sub>	37.1	62.9
Ni(2%):TiO <sub>2</sub>	56.11	43.89
Cu(1%):TiO <sub>2</sub>	66.8	33.2
Cu(2%):TiO <sub>2</sub>	69.9	30.1
Co(1%)/Ni(1%):TiO <sub>2</sub>	22.13 / 15.86	62.01
Co(1%)/Cu(1%):TiO <sub>2</sub>	0 / 67.92	32.08
Ni(1%)/Cu(1%):TiO <sub>2</sub>	0 / 70.95	29.05

Table 5.2. XPS analysis of surface chlorine relative to deposited metals on M(x%):TiO<sub>2</sub>.

## **Optical properties**

Diffuse reflectance UV-vis spectra for the M:TiO<sub>2</sub> samples relative to P25-TiO<sub>2</sub> show that the 3d metal surface modification leads to visible colors and apparent new absorption events in the red or orange regions as well as a slightly lower energy band gap absorption energy ( $E_g$ ). Figure 5.4 shows the raw reflectance data and Tauc plot for the P25-TiO<sub>2</sub> along with M(2%):TiO<sub>2</sub> products. Each of the M:TiO<sub>2</sub> samples shows a band gap absorption shift to lower  $E_g$  near 2.9 eV from 3.2 eV for P25-TiO<sub>2</sub>. Cu:TiO<sub>2</sub> samples

show a decrease in reflectance at wavelengths above 600 nm, correlating with its lightblue color. For comparison, a blue CuCl<sub>2</sub> solution has an absorbance maximum at 800 nm corresponding to d-d electronic transitions. Similar absorption near 700-800 nm are observed for Cu(OH)<sub>2</sub> deposited onto TiO<sub>2</sub>.<sup>156</sup> The Co:TiO<sub>2</sub> samples show decreased reflectance from 520 to 680 nm, similar to CoCl<sub>2</sub> solutions with an absorbance maximum near 500 nm. The light-grey colored Ni:TiO<sub>2</sub> samples exhibit a broad decrease in total reflectance across the visible spectrum relative to P25-TiO<sub>2</sub>. The spectra for M(1%):TiO<sub>2</sub> and M(1%)/M'(1%):TiO<sub>2</sub> samples show similar absorption events and trends. The increase in visible absorptions and band gap decrease may be due to nanoscale surface plasmon resonance effects,<sup>148, 157</sup> while others have ascribed these optical transitions to charge transfer between TiO<sub>2</sub> bands and the surface deposited metal species.<sup>155-156</sup>



Figure 5.4 – UV-vis DRS data on M(2%):TiO<sub>2</sub> samples (A) reflectance spectra and (B) Tauc plots.

# Photocatalytic dye oxidation

The photodegradation of methylene blue was first tested using the M(2%):TiO<sub>2</sub> samples. Following the dark stir, many of the samples saw in increase in absorbance signal due to scattering from fine particles remaining suspended in the dye solution. The degradation of the dye for each sample was compared with unmodified P25  $TiO_2$  (Figure 5.5). Most of the M(2%):TiO<sub>2</sub> fully decolorized the dye by the end of 30 minutes, with the exception of Cu(2%):TiO<sub>2</sub>. Many of the samples tracked closely with the degradation of MB by P25 at each time point leading up to this, most notably Fe(2%):TiO<sub>2</sub>. The linear rate fits (Figure 5.5) and first-order rate constants (Table 5.3) show that Fe(2%):TiO<sub>2</sub> is nearly as effective as P25. This is followed by Co(2%):TiO<sub>2</sub>, then Mn(2%):TiO<sub>2</sub> and Ni(2\%):TiO<sub>2</sub> with the same rate, and Cu(2\%):TiO<sub>2</sub> with the lowest rate. The degradation of MB dye is an oxidative process, and as described earlier copper is the most reduced of these metals on the surface of  $TiO_2$ . This would make Cu(2%):TiO<sub>2</sub> the least suitable for oxidizing MB, consistent with its low degradation activity. The loss of activity for the other metal-modified  $TiO_2$  products could be due to similar redox processes, but not as notable a loss since these metals maintained higher oxidation states than Cu.

Table 5.3. First-order rate constants (min<sup>-1</sup>) for the degradation of methylene blue dye using M(2%):TiO<sub>2</sub> products.

Sample	Mn(2%): TiO <sub>2</sub>	Fe(2%): TiO <sub>2</sub>	Co(2%): TiO <sub>2</sub>	Ni(2%): TiO <sub>2</sub>	Cu(2%): TiO <sub>2</sub>	<b>P25</b> TiO <sub>2</sub>
UV Rate	0.074	0.123	0.091	0.074	0.015	0.127
$R^2$	0.952	0.986	0.993	0.944	0.973	0.993



Figure 5.5 – UV-vis analysis of the photodegradation of MB dye with M(2%):TiO<sub>2</sub> samples (A) UV light data (B) linear rate data.

The M(1%)M'(1%):TiO<sub>2</sub> products were also tested for the degradation of MB dye. Similar to the M(2%):TiO<sub>2</sub> samples, there was some initial light scattering from suspended particles following the dark stir. Only Co(1%)Ni(1%):TiO<sub>2</sub> approached the same level of dye degradation as P25 TiO<sub>2</sub>. Both of the Cu-containing catalysts performed worse, reaching only 35% and 50% of the initial dye concentration in the 35 minute illumination time. These also had less activity than the individual Co(2%):TiO<sub>2</sub> and Ni(2%):TiO<sub>2</sub> catalysts, further showing the negative effects from the addition of Cu. The linear fits are shown in Figure 5.6, and rate constants are in Table 5.4. The rate for Co(1%)Ni(1%):TiO<sub>2</sub> is 52.6% of that for P25 TiO<sub>2</sub>, while Co(1%)Cu(1%):TiO<sub>2</sub> and Nu(1%)Cu(1%):TiO<sub>2</sub> are 9.7% and 6.2% respectively.

Table 5.4. First-order rate constants (min<sup>-1</sup>) for the degradation of methylene blue dye using M(1%)M'(1%):TiO<sub>2</sub> products.

Sample	Co(1%)Ni(1%): TiO <sub>2</sub>	Co(1%)Cu(1%): TiO <sub>2</sub>	Ni(1%)Cu(1%): TiO <sub>2</sub>	<b>P25</b> TiO <sub>2</sub>
UV rate	0.1216	0.0225	0.0144	0.2311
$R^2$	0.985	0.988	0.999	0.999



Figure 5.6 – UV-vis analysis of the photodegradation of MB dye with M(1%)M'(1%):TiO<sub>2</sub> samples (A) UV light data (B) linear rate data.

# Photocatalytic hydrogen evolution with M:TiO<sub>2</sub>.

Photocatalytic hydrogen gas evolution from aqueous methanol solutions was quantified in real-time using RGA-MS for different M:TiO<sub>2</sub> samples under UV irradiation, with methanol serving as a sacrificial oxidant. Calibration curves for the H<sub>2</sub>/Ar headspace mass analysis data allowed direct conversion to molar quantities of evolved hydrogen (see Appendix). The use of direct mass quantification by RGA-MS allowed rates of H<sub>2</sub> produced to be measured continuously, and it yielded similar evolution rates for Pt(1%):TiO<sub>2</sub> reported by GC analysis.<sup>158</sup> For comparison, photocatalytic hydrogen gas evolution was determined for highly-active platinum coated titania, Pt(1%):TiO<sub>2</sub>, and unmodified P25-TiO<sub>2</sub>, which yielded average H<sub>2</sub> evolution rates of 200 and 5  $\mu$ mol/h, respectively (for 10 mg samples: 20,000  $\mu$ mol h<sup>-1</sup> g<sup>-1</sup> and 500  $\mu$ mol h<sup>-1</sup> g<sup>-1</sup>). Each M:TiO<sub>2</sub> sample showed considerably-higher H<sub>2</sub> evolution rates than the P25-TiO<sub>2</sub> starting material and multiple cycle data for M(1%):TiO<sub>2</sub> samples is shown in Figure 5.7.



Figure 5.7 – Real-time hydrogen gas evolution under UV light illumination for (A) M(1%):TiO<sub>2</sub> and (B) M(1%),M'(1%):TiO<sub>2</sub>. H<sub>2</sub> production was measured during one-hour illumination cycles followed by a 10 minute argon flush. The sequential illumination cycles began at the 30, 100, 170, and 240 minute marks.

Each M:TiO<sub>2</sub> catalyst was tested in several one-hour cycles of UV exposure in the same aqueous methanol solution, and the average hourly hydrogen evolution data is summarized in Table 5.5. The hourly  $H_2$  evolution rates for most samples across the multiple cycles showed small deviation. The samples with the least stability were the Co(1%):TiO<sub>2</sub> and Ni(2\%):TiO<sub>2</sub>, where H<sub>2</sub> evolution dropped by 30-50% after the third cycle. The Cu(1%):TiO<sub>2</sub> sample had the highest hydrogen evolution rate of 85  $\mu$ mol/h  $(8,500 \mu \text{mol h}^{-1} \text{ g}^{-1})$  that is comparable to or greater than prior studies of copper deposition on porous or nanoscale titanias, with H<sub>2</sub> evolution in range from 1,500 - 8,000  $\mu$ mol h<sup>-1</sup> g<sup>-1</sup>, <sup>141, 148-149</sup> or to the structural incorporation of Cu or Ni with 1,100 - 4,300 µmol h<sup>-1</sup> g<sup>-1</sup>.<sup>131, 142</sup> Though Cu(1%):TiO<sub>2</sub> was more active than Cu(2%):TiO<sub>2</sub>, both Cu modified samples had very stable H<sub>2</sub> production for several cycles. The Cu:TiO<sub>2</sub> samples show higher activity than the Co:TiO<sub>2</sub> and Ni:TiO<sub>2</sub> photocatalysts, though this may be due to the higher amount of Cu deposition based on ICP results. Scaling hydrogen evolution activity to the amount of metal cocatalyst indicate that Co(1%):TiO<sub>2</sub> is most active at 8,400  $\mu$ mol h<sup>-1</sup> mg<sup>-1</sup> Co, while Cu(1%):TiO<sub>2</sub> had an activity of 2,400  $\mu$ mol h<sup>-1</sup> mg<sup>-1</sup> Cu. The calculated quantum yields from the broad spectrum UV illumination were generally similar (~1-20% range) to those reported in other Cu or Ni surface modified titanias.<sup>149, 155-156</sup>

Sample	Average H <sub>2</sub> rate (std. dev)	Quantum Yield (%)
P25-TiO <sub>2</sub>	5.3 (1.1)	0.4
Pt(1%):TiO <sub>2</sub>	200 (14)	16.6
Mn(2%):TiO <sub>2</sub>	4.2 (1.2)	0.3
Fe(2%):TiO <sub>2</sub>	5.7 (0.6)	0.5
Co(1%):TiO <sub>2</sub>	24.8 (8.7)	2.1
Co(2%):TiO <sub>2</sub>	14.9 (1.3)	1.2
Ni(1%):TiO <sub>2</sub>	33.9 (2.2)	2.8
Ni(2%):TiO <sub>2</sub>	33.1 (6.4)	2.8
Cu(1%):TiO <sub>2</sub>	84.7 (4.0)	7.0
Cu(2%):TiO <sub>2</sub>	53.0 (2.8)	4.4
Co(1%)/Ni(1%):TiO <sub>2</sub>	15.6 (4.5)	1.3
Co(1%)/Cu(1%):TiO <sub>2</sub>	23.3 (2.5)	1.9
Ni(1%)/Cu(1%):TiO <sub>2</sub>	70.2 (3.1)	5.8

Table 5.5. Hydrogen evolution rates (µmol/h) and quantum yields for M:TiO<sub>2</sub> materials.

There was no synergistic photocatalyst effect observed for the titania surfacemodified with both Ni and Co, as the activity for the mixed metal sample was similar or lower than that of the individual M(1%):TiO<sub>2</sub> products. While the addition of a second photodeposited metal (Co or Ni) species onto TiO<sub>2</sub> does enhance copper photodeposition, this may lead to higher surface metal coatings that inhibit hydrogen evolution activity by favoring redox reactions of the surface metal species (*e.g.*, Cu<sup>+</sup>/Cu) or blocking illumination of the titania core. Based on Table 5.5 results for M(1%)/Cu(1%):TiO<sub>2</sub>, addition of Ni<sup>2+</sup> to the Cu<sup>+</sup>/Cu photodeposited coating has less negative impact on hydrogen production versus the Co<sup>2+</sup> additions. There are many variables that prevent direct comparisons with other related metal coated heterojunction titanias, but Table 5.6 compares several other studies with the current work.

Sample Meas. mol% M <sup>a</sup>	TiO <sub>2</sub> source, surf. area, metal depn.	Conditions	H <sub>2</sub> rate in µmol h <sup>-1</sup> g <sup>-1</sup> (QY)	Ref.
Ni(1%):TiO <sub>2</sub>	P25, 45 m <sup>2</sup> /g	50% MeOH,	3,390	this
0.08 mol% Ni	MCl <sub>2</sub> photorxn	450 W Hg lamp	(2.8%)	work
Ni(OH) <sub>2</sub> on TiO <sub>2</sub>	P25, 45 m <sup>2</sup> /g	25% MeOH, 3 W	3,056	155
0.23 mol% Ni	solution ppt	365 nm UV-LEDs	(12.4%)	
0.5% Ni on TiO <sub>2</sub> ,	P25, 45 m <sup>2</sup> /g	10% EtOH, 100 W	11 600	131
0.67 mol% Ni	H <sub>2</sub> reduction	365 nm UV lamp	11,000	
0.25% Ni on TiO <sub>2</sub> 0.34 mol% Ni (ideal)	mesoporous, 130 m <sup>2</sup> /g <i>in-situ</i> photorxn, NiCl <sub>2</sub>	95% EtOH, 1000 W Hg lamp	700 (20%)	149
Cu(1%):TiO <sub>2</sub>	P25, 45 m <sup>2</sup> /g	50% MeOH,	8,470	this
0.44 mol% Cu	MCl <sub>2</sub> photorxn	450 W Hg lamp	(7%)	work
Cu(OH) <sub>2</sub> on TiO <sub>2</sub> , 0.29 mol% Cu	P25, 43 m <sup>2</sup> /g solution ppt.	0.09M ethylene glycol, 3 W 365 nm UV- LEDs	3,418 (13.9%)	156
$\begin{array}{c} 0.25\% \text{ Cu on TiO}_2\\ 0.31 \text{ mol}\% \text{ Cu}\\ (\text{ideal}) \end{array}$	mesoporous, 130 m <sup>2</sup> /g <i>in-situ</i> photorxn, CuCl <sub>2</sub>	95% EtOH, 1000 W Hg lamp	1,600 (50%)	149
$\begin{array}{c} 10\% \text{ Cu on TiO}_2, \\ 12.4 \text{ mol}\% \text{ Cu} \\ (\text{ideal}) \end{array}$	P25, 45 m <sup>2</sup> /g $H_2$ reduction	10% glycerol, 300 W Xe lamp	5,700	141
$10\% \text{ Cu on TiO}_2,$ $12.4 \text{ mol}\% \text{ Cu}$ (ideal)	P25, <i>in-situ</i> photorxn CuSO <sub>4</sub>	MeOH, 125 W Hg lamp	1,748	148

Table 5.6. Comparison of several Ni and Cu modified titania for hydrogen evolution photocatalysis.

a) Ideal mol% metal noted based on ideal  $M/TiO_2$  solution content, if bulk analysis not reported.
For the results shown in Table 5.6, some cases note the ideal amount of metal that could be deposited onto  $TiO_2$  from solution where bulk chemical analysis was not available. The metal salt solutions vary from using non-oxidizing chloride anions to potentially reducible sulfate and nitrate anions. As P25-TiO<sub>2</sub> alone only yields ~500 µmol h<sup>-1</sup> g<sup>-1</sup> of H<sub>2</sub> in aqueous UV photocatalytic reactions with a methanol sacrificial oxidant, the wide range of surface deposition strategies in Table 5.4 all show enhanced H<sub>2</sub> evolution abilities with various illumination strategies and oxidants.

Recycling tests were performed after recovering M(1%):TiO<sub>2</sub> samples and testing activity in new aqueous methanol solutions. All M:TiO<sub>2</sub> samples retained enhanced hydrogen evolution, though at lower activity than the initial cycling tests. As an example, Cu(1%):TiO<sub>2</sub> had an average rate of 62 µmol/h during a recycle test, roughly 73% of the average activity of the initial cycles. During several catalyst recycle runs, the individual cycles were fairly consistent (Table 5.7).

Sample	Average H <sub>2</sub> rate (std. dev)	% of initial activity	Quantum Yield (%)
Co(1%):TiO <sub>2</sub>	14.3 (2.9)	58	0.9
Ni(1%):TiO <sub>2</sub>	16.5 (2.6)	49	1.4
Cu(1%):TiO <sub>2</sub>	61.6 (6.4)	73	4.8
Co(1%)Cu(1%):TiO <sub>2</sub>	19.3 (4.1)	83	1.1
Ni(1%)Cu(1%):TiO <sub>2</sub>	33.4 (5.8)	48	2.2

Table 5.7. Hydrogen evolution rates ( $\mu$ mol/h) and quantum yields for select recycled M:TiO<sub>2</sub> materials.

Post-reaction XPS analysis was used to check for changes to the catalyst surface composition (Table 5.8). Cu(2%):TiO<sub>2</sub> showed a ~70% decrease in the Cu:Ti atomic

ratio to 1.1 : 98.9, but no change in Cu  $2p_{3/2}$  peak position. The Co(2%) and Ni(2%) samples did not show clearly identifiable surface metal peaks in the post-reaction analysis, though the solids retained some of their initial color. This loss of surface metal may relate to the decrease in activity over the initial cycles as well as the drop in activity during the recycle tests for these catalysts. In related CuO/TiO<sub>2</sub> photocatalytic systems, it was determined that photocorrosion of chemically-deposited copper occurs during UV hydrogen evolution photocatalysis along with possible competing reverse photodeposition of some dissolved copper.<sup>147</sup>

Table 5.8. XPS data on select M:TiO<sub>2</sub> products after use in photocatalytic hydrogen evolution.

Sample	M:Ti:O:Cl (at%)	M 2p <sub>3/2</sub> (eV)	Ti 2p <sub>3/2</sub> (eV)
Co(2%):TiO <sub>2</sub>	0:31.2:68.8:0	n/a	458.34
Ni(2%):TiO <sub>2</sub>	0:29.21:70.79:0	n/a	458.50
Cu(2%):TiO <sub>2</sub>	0.34 : 29.45 : 70.22 : 0	Cu: 932.61	458.31
Ni(1%)/Cu(1%):TiO <sub>2</sub>	0 / 0.45 : 30.73 : 68.64: 0.15	Cu: 932.4	458.50

In the Ti  $2p_{3/2}$  XPS spectrum of Ni(1%)/Cu(1%):TiO<sub>2</sub>, a small peak appeared at 454.9 eV that is consistent with the presence of reduced Ti<sup>3+</sup> on the surface of this sample.<sup>151</sup> The presence of Ti<sup>3+</sup> sites on TiO<sub>2</sub> has been associated with enhanced photoreduction abilities.<sup>159-160</sup> As noted earlier, the copper photodeposition preference may be due to its ease in reduction relative to Co<sup>2+</sup>, Ni<sup>2+</sup>, or H<sup>+</sup>. The Ti<sup>4+</sup>/Ti<sup>3+</sup> reduction is slightly less favored than H<sup>+</sup> reduction and control experiments with P25-TiO<sub>2</sub> alone under UV illumination in methanol solution undergoes no visible change in color or enhanced hydrogen evolution activity (see P25-TiO<sub>2</sub> sample in Figure 5.7).<sup>150</sup> Thus, if

present,  $Ti^{3+}$  is not generated in the absence of other metals under the conditions of this work. Given that chloride ions are present on the surface after photodeposition reactions in metal halide solutions, another control experiment with P25-TiO<sub>2</sub> was performed in the presence of NaCl(*aq*) under the same aqueous methanol/UV conditions. No detectable changes in H<sub>2</sub> evolution were observed for P25-TiO<sub>2</sub>, indicating that the presence of chloride is not a determining feature of enhanced H<sub>2</sub> evolution from the M:TiO<sub>2</sub> materials.

### In-situ photodeposition and hydrogen evolution studies.

Several in-situ photocatalytic hydrogen evolution experiments were run with P25- $TiO_2$  in the presence of MCl<sub>2</sub> in 50% methanol, equivalent to the 1% photodeposition experiments. Cycling data is presented in Figure 5.8 for these single-step deposition/hydrogen evolution experiments. These tests showed that even without the metal previously deposited onto the TiO<sub>2</sub> surface, hydrogen gas is produced very quickly, once some metal surface photodeposition occurs. After the first in-situ cycle for CuCl<sub>2</sub>, the titania catalyst developed its characteristic blue color. The *in-situ* copper chloride photocatalytic test showed an average hydrogen evolution rate of 29 µmol/h (2,900 µmol  $h^{-1}$  g<sup>-1</sup>, 5 times higher than P25-TiO<sub>2</sub> alone) that is higher than observed in other *in-situ* Cu:TiO<sub>2</sub> studies (~1,500  $\mu$ mol h<sup>-1</sup> g<sup>-1</sup>),<sup>149</sup> but about one-third of that observed for the isolated Cu(1%):TiO<sub>2</sub> material described above. Related titania experiments have shown that the addition of copper sulfate to a titania suspension also improves the rate of hydrogen evolution to 2,200  $\mu$ mol h<sup>-1</sup> g<sup>-1</sup>.<sup>148</sup> The initial rates for the *in-situ* Co and Ni tests were comparable to values for the isolated Co(1%) and Ni(1%):TiO<sub>2</sub> materials, but dropped in activity over four cycles. This loss of activity is comparable to that observed

for the isolated Co(1%):TiO<sub>2</sub> sample, but the activity of the isolated Ni(1%):TiO<sub>2</sub> catalyst was more stable under UV photocatalytic cycling than shown by the *in situ* test.



Figure 5.8 – *In-situ* metal deposition and hydrogen evolution data, average rates for all 4 cycles (µmol/h) are Co 12.2, Ni 15.8, and Cu 28.3. H<sub>2</sub> production was measured during 1 hour illumination cycles followed by a 10 minute argon flush. Each illumination cycle began at the 30, 100, 170, and 240 minute marks.

An *in-situ* Cu(1%):TiO<sub>2</sub> experiment was also performed in 18 M $\Omega$  water to verify the importance of methanol as a sacrificial oxidant in this photochemical reaction. Two UV illumination cycles (P25-TiO<sub>2</sub>, 1 wt% CuCl<sub>2</sub>, pure water) were run without methanol, and there was no visible titania color change and no detectable H<sub>2</sub> production by RGA-MS analysis (Figure 5.9). After the second illumination cycle, methanol was spiked into solution and this produced detectable H<sub>2</sub> evolution, and a titania color change was observed during a third UV illumination cycle. This test confirmed the importance of a sacrificial oxidant for both the photo-assisted copper surface deposition/reduction and hydrogen generation processes.



Figure 5.9 – *In-situ* copper UV photodeposition and hydrogen evolution in 18 MΩ water. First UV lamp on cycle began near 30 min and ended at 90 min. Reactor headspace was flushed with argon between UV illumination runs. Second UV lamp on cycle began at 100 min and ended at 160 min. Methanol is spiked into solution during the argon flush prior to the third cycle.

Photocatalytic considerations.

The conduction band energy for anatase  $TiO_2$  is energetically well-suited for the solution reduction of H<sup>+</sup> to H<sub>2</sub>. One of the primary limitations on its activity is the fast charge recombination of photoexcited electrons in the conduction band (CB) with holes in the valence band (VB), before they are utilized in external redox reactions. The addition of metal or metal oxide to the TiO<sub>2</sub> surface provides a pathway to transfer

photogenerated electrons from the TiO<sub>2</sub> CB to the surface-bound metal (or metal ion) cocatalyst that facilitates hydrogen reduction. In the absence of a photocatalyst, the overall reaction of  $CH_3OH$  with water to produce  $CO_2$  and  $H_2$  is a thermochemically uphill process, with a calculated  $\Delta H_{rxn}$  of +53 kJ/mol using standard enthalpy of formation values. One study estimated the endothermic free energy of +16 kJ/mol for this reaction in the absence of a catalyst.<sup>130</sup> Based on the relatively-low energy redox processes for Co, Ni, and Cu ions in solution and metals on titania particles, the enhanced photocatalytic activity of these heterojunction semiconductor composites is usually attributed to the 3d metal engaging in surface redox reactions that lead to  $H^+$  to  $H_2$ reduction. During UV illumination, there is a continuous flow of conduction band electrons from TiO<sub>2</sub> to the surface metal co-catalyst that enables reduction processes, which could transiently involve the metal (e.g., metal oxide to metal particles) as well as the H<sup>+</sup>/H<sub>2</sub> reduction. Given the different reduction potentials for Co and Ni versus Cu, there may be different surface reactions occurring, for example, H<sub>2</sub> from Ni or Co systems via transient production of nanoscale metals that reoxidize during H<sub>2</sub> production versus H<sub>2</sub> production from metallic copper particle surfaces. These general heterojunction photoreactions have been proposed in several related previous systems.<sup>148-</sup> <sup>149, 155-156</sup> The general reaction steps described for these photocatalytic redox reactions are given in Equations 5.1 through 5.4.

$$TiO_2 + hv (> E_g) \rightarrow e^{-}(TiO_2 CB) + h^{+}(TiO_2 VB)$$
(5.1)

$$e^{-}(TiO_2 CB) + surface(M^+/M^{2+}) \rightarrow TiO_2/M(reduced)$$
 (5.2)

$$2e^{-}(TiO_2 CB) + TiO_2/M(reduced) + 2 H^+ \rightarrow H_2 + TiO_2/M(reduced)$$
(5.3)

$$2h^{+}(TiO_2 VB) + CH_3OH \rightarrow CH_2O + 2H^{+}$$
(5.4)

In this work, the lower activity of our photodeposited M:TiO<sub>2</sub> samples for Co or Ni versus Cu can be attributed to the (a) differing amounts of metal species on the TiO<sub>2</sub> surface and (b) differences in metal ion redox activity versus hydrogen reduction. In this photocatalytic metal species deposition, there appears to be an upper limit for metal deposition that competes with hydrogen gas formation. Other studies have also shown that catalytic activity of surface-deposited metals or metal oxides/hydroxides peaks at low metal content (~0.2-0.5 mol% M) and then drops in activity overlap of reactive surface sites, such as with Ni(OH)<sub>2</sub> and Cu(OH)<sub>2</sub> cluster depositions on TiO<sub>2</sub>.<sup>155-156</sup> As demonstrated in the *in-situ* test, the presence of methanol plays a crucial supporting role in the overall photocatalytic reaction. While hydrogen is reduced, methanol is oxidized stepwise to formaldehyde and formic acid by holes in the valence band of TiO<sub>2</sub>, eventually forming CO<sub>2</sub>.<sup>130, 161</sup> This occurs in preference to the oxidation of water, which would limit overall hydrogen production due to back reaction between hydrogen and oxygen.

## 5.4 Conclusions

This chapter has described a facile method of modifying commercial P25-TiO<sub>2</sub> to improve its activity for photocatalytic hydrogen production. The photodeposition of Mn, Fe, Co, Ni, or Cu onto the surface of TiO<sub>2</sub> led to visible optical absorption properties. Compositional analysis showed that low metal contents (~0.04 - 1 at% of bulk material, highest for copper) are photodeposited onto the TiO<sub>2</sub> surface, but this surface modification is sufficient to show beneficial effects on the utilization of titania conduction band electrons for photocatalytic hydrogen reduction using methanol as a sacrificial oxidant. The rate of UV photocatalytic hydrogen production was improved to as high as 8,500  $\mu$ mol h<sup>-1</sup> g<sup>-1</sup> (7% QY) for Cu(1%):TiO<sub>2</sub> versus 500  $\mu$ mol h<sup>-1</sup> g<sup>-1</sup> (0.4% quantum yield) for P25-TiO<sub>2</sub>. Direct atmospheric-pressure headspace gas sampling by portable mass spectrometry was demonstrated as a useful portable tool for real-time monitoring of reactor headspaces and quantification of hydrogen gas production.

### CHAPTER 6

# PRELIMINARY STUDIES ON METAL NITRIDES, SILICA ETCHING, SELECTIVE ORGANIC PHOTO-OXIDATIONS, CONCLUSIONS, AND FUTURE OUTLOOK 6.1 Introduction

Transition metal nitrides have applications in battery materials and a number of catalytic and electrocatalytic reactions, including hydrogen and oxygen evolution reactions.<sup>162-166</sup> Nickel nitride (Ni<sub>3</sub>N), cobalt nitride (Co<sub>4</sub>N), and copper nitride (Cu<sub>3</sub>N) can be prepared from the reaction between the reduced metal, halide, or amide salts with ammonia.<sup>163-164, 167-168</sup> However, these materials are metastable and decompose to the reduced metal and nitrogen gas between temperatures of 300-400 °C.<sup>167</sup>

 $C_3N_4$  is a potential nitrogen reagent for the formation of metal nitrides. Al-Ga-N and Ti-V-N have been formed from their respective chlorides using  $C_3N_4$  as a reactive nitrogen support.<sup>169</sup> Ta<sub>3</sub>N<sub>5</sub> has been prepared using  $C_3N_4$  as a template, but also utilized ammonia during synthesis.<sup>170</sup> However, this work utilized the full decomposition of  $C_3N_4$  at 800 °C to generate a reactive nitrogen containing atmosphere for nitridation. Considering the previously-described transition metal nitrides both formed and decomposed at lower temperatures, it may be possible to use  $C_3N_4$  as the nitrogen source and keep it as an intact support after the synthesis.

Chapter 4 described the preparation of carbon nitride on a silica support  $(C_3N_4@SiO_2)$  to take advantage of its high surface area. The removal of this support should further increase available surface area by generating new pores into the morphology of  $C_3N_4$ , as shown by the salt templating described in Chapter 3. The

removal of the SiO<sub>2</sub> template often utilizes HF as an etchant, giving ordered porous materials with surface areas on the order of 100 m<sup>2</sup> g<sup>-1.81, 107, 171</sup>

The photocatalytic reactions discussed in the previous chapters have either been the oxidative degradation of dyes or hydrogen evolution from water-splitting.  $C_3N_4$  has also been used as a catalyst or photocatalyst in the functionalization of benzene,<sup>98, 172</sup> cyclization of nitriles and alkynes,<sup>99</sup> and different organic coupling and oxidation reactions.<sup>76, 173-175</sup>

#### 6.2 Experimental

## Reagents

Trichloromelamine (C<sub>3</sub>N<sub>3</sub>(NHCl)<sub>3</sub> ,TCM, 98%) was purchased from GFS Chemical and stored in an argon atmosphere glovebox. Other reagents used as received were TiO<sub>2</sub> (Degussa P25), FeCl<sub>2</sub>·4H<sub>2</sub>O (Aldrich, 99%), CoCl<sub>2</sub>·6H<sub>2</sub>O (Aldrich, reagent grade), NiCl<sub>2</sub>·6H<sub>2</sub>O (Aldrich, ReagentPlus), CuCl<sub>2</sub>·2H<sub>2</sub>O (Aldrich, 99+%), H<sub>2</sub>PtCl<sub>6</sub>·6H<sub>2</sub>O (99.9%, Strem), methanol (Fisher, ACS certified), benzene (Fisher, laboratory grade), toluene (Fisher, ACS certified), and hexanol (Aldrich, 98%). Deionized water (18 MΩ) was prepared using a Photronix MiniQuad system. Sodium hydroxide solution (6 M) was prepared from dissolving 48 g of sodium hydroxide pellets (Macron Fine Chemical) in 200 mL of deionized water.

## Metal nitride formation on $C_3N_4$

FeCl<sub>2</sub>·4H<sub>2</sub>O, CoCl<sub>2</sub>·6H<sub>2</sub>O, NiCl<sub>2</sub>·6H<sub>2</sub>O, and CuCl<sub>2</sub>·2H<sub>2</sub>O were used as potential precursors for metal nitride formation. The MCl<sub>2</sub> salt (150 mg) was dissolved in 20 mL

of methanol.  $C_3N_4$  (150 mg, TCM Parr product, see Chapter 2) was added to this solution and the mixture was stirred for 2 hours. The solution was left open in air overnight to allow the methanol to evaporate and deposit the MCl<sub>2</sub> onto  $C_3N_4$ . The drycoated solid was transferred into an alumina boat and was placed in a Pyrex tube closed on one end. The tube was connected to a Schlenk line, evacuated, and refilled with argon. The tube was placed into a clamshell furnace and heated to 500 °C at a rate of 10 °C/min and held for 1 hour. A dark brown solid was recovered (~38-50% by mass).

## Etching of silica from $C_3N_4@SiO_2$

 $C_3N_4@SiO_2$  prepared from the Schlenk reactor method (detailed in Chapter 4) was washed with sodium hydroxide to remove the SiO\_2 support.  $C_3N_4@SiO_2$  (150-300 mg) was put into a 100 mL round bottom flask and the mixture was stirred in 50 mL of 6 M NaOH overnight. Centrifugation (3000 rpm) was used to isolate the solid. The solid was rinsed with additional water over a Büchner funnel, followed by a methanol rinse, and allowed to dry in air. The color of the recovered solid varied, but was usually darker than the initial  $C_3N_4@SiO_2$ . Mass recovery ranged from 5-30%.

#### Photocatalytic oxidation of small organic molecules

 $C_3N_4$ ,  $C_3N_4$ @TiO<sub>2</sub>, and TiO<sub>2</sub> were used as catalysts in the attempted vapor-phase oxidations of different organic molecules. The catalyst (10-20 mg) was either placed into a 1 cm diameter glass cup or deposited onto a microscope slide from an acetone slurry, and then placed inside a gas IR cell. The assembled cell was first purged with nitrogen in order to collect a background. Then oxygen saturated with hexanol, benzene, or toluene vapors was flowed through the cell. A spectrum was collected to observe the initial peak signals of the organic species. The cell was exposed to UV by two different methods. The first used a 25 W UV pen lamp kept in the IR sample chamber with the gas cell. The lamp was turned on and spectra were collected every 15 minutes for up to 3 hours. The second method used a larger UV lamp (Ace-Hanovia, 450 W Hg), which required the gas cell to be moved from the lamp cabinet in order to collect spectra. The lamp was turned on in 30 minute intervals for up to 4 hours of total illumination. An IR spectrum of the catalyst was taken after the reaction to check for adsorbed organics.

#### *Characterization*

FT-IR spectra were collected on a Nicolet Nexus 670. Solid samples were prepared as KBr pellets for data collection. Photocatalytic vapor measurements were collected in a 10 cm long gas IR cell with KBr windows. Elemental analysis was performed on a Perkin Elmer 2400 Series II CHNS/O Analyzer, with 1-2 mg of sample in crimped tin capsules. Powder X-ray diffraction was collected on a Bruker D8 system (source Cu K $\alpha$ , 5-80°, 0.050 °/step) with samples deposited on glass slides. Solid state <sup>13</sup>C and <sup>15</sup>N CP-MAS-NMR spectra were collected on a 500 MHz Bruker spectrometer. Thermogravimetric analysis was performed on a Seiko Exstar 6300 TGA-DTA. Samples were heated up to 1000°C at a rate of 10°C/min under air or argon flow.

## 6.3 Results and Discussion

## Metal nitrides on $C_3N_4$

The inert annealing of different metal chlorides (Fe, Co, Ni, Cu) deposited on the surface of  $C_3N_4$  produced several dark-brown or black products. In addition to the white film of NH<sub>4</sub>Cl observed from the annealing of  $C_3N_4$ , condensation of water droplets was observed on the cool end of the tube from the waters of hydration from the starting metal chloride. XRD of these products only showed the formation of a crystalline nitride phase for the reaction with nickel (Figure 6.1). Ni<sub>3</sub>N is reported to degrade above 440 °C to Ni metal and nitrogen gas.<sup>176</sup> Since this Ni<sub>3</sub>N remained crystalline up to 500 °C, it is possible that the presence of the volatiles from  $C_3N_4$  during the annealing reaction stabilized it at the higher temperature. The remaining metals did not show any crystalline peaks, even for the starting chloride.



Figure 6.1 – XRD pattern of  $Ni_3N$  on  $C_3N_4$  prepared from annealing  $NiCl_2 \cdot 6H_2O$  on  $C_3N_4$  at 500 °C under argon.

Considering the significant color change from annealing, it is likely that the metals underwent some degree of reduction, even if not to the nitride. It is possible that the nitrides may have formed at lower temperatures, but unlike Ni<sub>3</sub>N, decomposed by the annealing temperature at 500 °C. TGA analysis showed residual mass for the samples after heating to 1000 °C in air, and the remaining residues were collected from the TGA pan. XRD analysis of these residues showed formation of the respective metal oxides. These phases and colors of the initial products and oxidation residues are summarized in Table 6.1. The presence of the metal oxide in this residue confirms the metal is on the surface of  $C_3N_4$  in some form after the inert anneal. Since there are no identifiable peaks, it is possible that the metals reduced to a nanocrystalline form. If the crystallite size was small enough, the peaks in the XRD could have broadened enough to be lost in the baseline.

Sample	XRD after argon anneal	Color	XRD after air TGA	Residue color	% Mass remaining in TGA
$C_3N_4 + NiCl_2$	Ni <sub>3</sub> N	Dark brown	NiO	Green or brown	28.7
$C_3N_4 + CuCl_2$	n/a	Dark brown	CuO	Black	35.6
$C_3N_4 + FeCl_2$	n/a	Brown	Fe <sub>2</sub> O <sub>3</sub>	Red-brown	37.8
$C_3N_4 + CoCl_2$	n/a	Dark brown	Co <sub>3</sub> O <sub>4</sub>	Black	19.1

Table 6.1. Summary of XRD phases observed for annealing reactions between  $C_3N_4$  and  $MCl_2$ .

## Etching of SiO<sub>2</sub> from C<sub>3</sub>N<sub>4</sub>@SiO<sub>2</sub>

As described in Chapter 4, the starting  $C_3N_4@SiO_2$  Schlenk products were lightyellow in color and required higher reaction temperatures than TCM in the absence of SiO<sub>2</sub>. These products also had features in the IR spectrum resembling more triazine-like properties (Figure 6.2a). Washing these in NaOH was successful in removing the SiO<sub>2</sub> support, but there were distinct changes in the remaining  $C_3N_4$ . The washed products were notable darker that the original  $C_3N_4@SiO_2$ , and had a mass recovery lower than what was expected based on the estimated  $C_3N_4$  content from elemental analysis and TGA. The IR spectrum of the washed product (Figure 6.2b) confirmed the removal of the SiO<sub>2</sub> due to the absence of the large peak at 1000 cm<sup>-1</sup>. The broad peak above 3200 cm<sup>-1</sup> is representative of N-H stretching. Unlike unmodified  $C_3N_4$ , there was a single broad signal from 1200-1700 cm<sup>-1</sup> for the C-N and C=N stretches. The peak at 1600 cm<sup>-1</sup> is slightly more pronounced than the rest of this broad signal. This lack of sharplydefined features in this region may indicate poorly-defined structural ordering. The peak at 800 cm<sup>-1</sup> for the ring breathing mode remained intact after the wash.



Figure 6.2 – IR spectra of (a)  $C_3N_4@SiO_2$  and (b) the 6M NaOH washed product to etch the SiO<sub>2</sub>.

Elemental analysis of two sets of  $C_3N_4@SiO_2$  and etched products is summarized in Table 6.2. Each washed product showed a similar change of total CHN content after the removal of SiO<sub>2</sub>, increasing from ~27% to above 70%. There was still a large amount of unanalyzed mass, which could be from additional oxygen content.  $C_3N_4$  products have been shown to undergo base hydrolysis at high temperatures.<sup>69</sup> Since  $C_3N_4@SiO_2$  has been shown to have difficulty reaching reaction completion, it is possible this product was more susceptible to hydrolysis at room temperature. In addition to the change in total CHN content, the composition also changed. The washed products contained less nitrogen and hydrogen than the starting products. If the products did undergo hydrolysis, a loss of terminal NH<sub>2</sub> sites could account for the decrease in nitrogen and hydrogen.

Sample	C wt%	N wt%	H wt%	Total wt%	Composition
$C_3N_4@SiO_2$	8.60	17.09	1.93	27.62	$C_{3}N_{5.1}H_{8}$
Etched C <sub>3</sub> N <sub>4</sub>	24.76	45.20	3.42	73.38	$C_3N_{4.7}H_{5.0}$
$C_3N_4@SiO_2$	8.90	17.44	1.23	27.57	$C_3N_{5.0}H_{5.0}$
Etched C <sub>3</sub> N <sub>4</sub>	25.83	42.03	3.54	71.40	$C_3N_{4.2}H_{4.9}$

Table 6.2. CHN elemental analysis of  $C_3N_4@SiO_2$  and etched products.

The morphology of the etched  $C_3N_4$  product was investigated using SEM (Figure 6.3). Similar to other  $C_3N_4$  products, these formed 5-20 µm aggregates. The surface of these particles has a very moss-like appearance. The thin component flakes have dimensions less than 100 nm.



Figure 6.3 – SEM images of  $C_3N_4@SiO_2$  after etching with NaOH.

# Photocatalytic oxidations in the vapor phase

The photocatalytic oxidations of different organic molecules in the vapor phase were attempted. The reagents and targets molecules are summarized in Figure 6.4.



 $\label{eq:Figure 6.4-Targeted photocatalytic oxidations of small organic molecules using C_3N_4-based catalysts.$ 

The reactions were attempted at room temperature using two different intensities of UV light. Very little change was observed for any of the catalyst and starting reagent combinations in the reaction set-up using the 25 W UV lamp. When reactions were illuminated with the 450 W lamp, the absorption for carbon dioxide grew in at 2380 cm<sup>-1</sup>, indicating the formation of the total oxidation product rather than the target intermediates. Figure 6.5 shows the initial and final IR spectra of benzene using  $C_3N_4$ @TiO<sub>2</sub> catalyst with the 450 W lamp. While there was some conversion, the photocatalytic reaction was not selective. In order to achieve better selectivity for the partial oxidation products, a milder oxidant might be more suitable rather than performing the reaction in oxygen atmosphere. Depending on how accessible the surface of  $C_3N_4$  is to adsorption of the analyte, catalysis in the vapor phase might not be efficient.



Figure 6.5 – IR spectra of benzene vapor before and after 4 hour UV light (450 W Hg) illumination in the presence of  $C_3N_4@TiO_2$ .

## 6.4 Overall Thesis Summary and Conclusions

 $C_3N_4$  can be prepared under several different reaction conditions from a trichloromelamine precursor. These products share many properties with those produced from other common literature precursors. There are some clear differences in the band structure, which accounts for some of the observed differences in photocatalytic activity. Other differences include variations in surface area and the quantity of platinum co-catalyst successfully deposited. Increasing amounts of platinum (up to 0.5 wt%) on  $C_3N_4$  from a given synthetic method showed an increase in hydrogen evolution activity. If higher amounts of platinum loading can be achieved, a more optimal loading could be identified. Higher surface area also increased catalytic rates, though even samples with comparable surface areas differed in activity depending on the synthetic route.

The use of salt templates was successful in generating  $C_3N_4$  products with rougher, more porous surface morphologies. Several of the salts completed an exchange reaction with TCM, which resulted in different colored products than using the unmodified precursor. The products formed from TCM mixtures with KI and NH<sub>4</sub>SCN salts showed the greatest improvement for the photocatalytic production of hydrogen over unmodified  $C_3N_4$ , mostly attributed to the increase in surface area.

 $C_3N_4$  composites supported on different metal oxides required higher synthesis temperatures in order to ensure complete reactions. Higher-temperature synthesis using WO<sub>3</sub> as a support resulted in a phase change to H<sub>x</sub>WO<sub>3</sub>. To work around this phase change,  $C_3N_4$  was used as the support for the formation of WO<sub>3</sub> at lower temperatures. The composites with SiO<sub>2</sub> and Al<sub>2</sub>O<sub>3</sub> showed a notable increase in surface area over unmodified  $C_3N_4$ , which contributed to a higher degree of dye adsorption during the dark stir in dye degradation tests. This adsorption contributed to a higher rate of oxidation in the presence of UV light.

The surface modification of  $TiO_2$  provided very-active photocatalysts for hydrogen evolution. This presents an attractive alternative to the use of platinum as a cocatalyst. Modification with copper, even with lower than the targeted quantities, had the greatest effect on hydrogen evolution, which is very promising considering copper's much higher abundance than platinum.

#### 6.5 Future Outlook

 $C_3N_4$  materials present continued opportunities in future photocatalytic reactions. Besides the dye degradation and hydrogen evolution reactions detailed in this thesis, there are many other reactions to explore. The preliminary oxidation reactions discussed in this chapter are in need of optimization. Besides adjustments to the reaction parameters, the other composite and salt templates  $C_3N_4$  products could also be active in these reactions. Investigation of the luminescent properties of  $C_3N_4$  could give insights to the life-time of the photo-excited electrons. A shorter excitation life-time would support the low hydrogen evolution activity in the absence of co-catalyst to aid in charge carrier separation. This could show additional differences between the TCM-salt derived  $C_3N_4$ products.

Work on the formation of transition metal nitrides on the surface of  $C_3N_4$  should be continued. Ni<sub>3</sub>N has been observed from 500 °C inert heating between  $C_3N_4$  and NiCl<sub>2</sub>, but these conditions did not work for Fe, Co, or Cu. Lower-temperatures should be tested to see if the nitride phase can form before higher-temperature decomposition. If the remaining nitrides can be formed, this opens up additional catalytic systems to study. Since the metal nitrides are more conductive than  $C_3N_4$ , they can be tested for electrochemical applications. With  $C_3N_4$  as a photoactive support, photoelectrochemical reactions could be tested, where  $C_3N_4$  can provide photogenerated electrons and the conducting nitride can better transport the charge. Even if the metal nitride cannot be stabilized, the presence of the reduced metal may still have use as a co-catalyst on the surface of  $C_3N_4$  in photocatalytic reactions. The presence of the conducting metal could also be useful in electrochemical applications. The  $C_3N_4@H_xWO_3$  composite could also be useful in these applications.

The modification of  $TiO_2$  by surface photoreduction of first-row transition metals provided highly-active catalysts for hydrogen evolution. This work only modified commercial  $TiO_2$ . Since our group has prepared  $TiO_2$  using botanical and other natural templates, exploring the same surface modification with these unique morphological structures could show further enhancement of catalytic activity.

#### APPENDIX

#### A.1 Photocatalytic Hydrogen Evolution

## Signal calibration

Residual gas analysis mass spectrometry was used as the method of measuring the amount of hydrogen produced during photocatalytic reactions. The raw signal reported from the instrument was pressure (in Torr), while hydrogen quantification is typically reported in µmol or mmol. In order to convert this pressure signal to µmol, a signal calibration method was developed using known mixtures of hydrogen and argon within the reaction flask used for photocatalysis.

A 50 mL Schlenk flask was filled with 10 mL of water in order to account for the liquid volume present during photocatalytic reactions. The flask was capped with a septum and purged with argon. A second flask was filled with hydrogen, and a syringe was used to transfer 0.5, 1, 2, and 5 mL volumes to the argon-filled Schlenk flask. The gas mixture in the flask was sampled with the RGA capillary while scanning on Pressure vs. Time mode (mass 40 and 2). The flask was re-purged with argon before each addition of a different volume of hydrogen.

While a 50 mL Schlenk flask was used for this calibration, the true total volume including the headspace in the neck of the flask was 70 mL. Accounting for the 10 mL of water added to the flask, the total gas volume available was 60 mL. The volume % of hydrogen within the flask was determined for each volume addition used. The ratio of the partial pressures for hydrogen and argon (pH<sub>2</sub> and pAr) were also calculated. These values are presented in Table A.1. Plotting pH<sub>2</sub>/pAr against the vol.% H<sub>2</sub> gave the linear calibration in Figure A.1.

vol H <sub>2</sub>	vol% H <sub>2</sub>	pH <sub>2</sub>	pAr	pH <sub>2</sub> /Ar
0.5	0.83	2.34 x 10 <sup>-7</sup>	1.04 x 10 <sup>-6</sup>	0.225
1	1.67	4.42 x 10 <sup>-7</sup>	1.04 x 10 <sup>-6</sup>	0.426
2	3.33	8.22 x 10 <sup>-7</sup>	1.04 x 10 <sup>-6</sup>	0.791
5	8.33	2.06 x 10 <sup>-6</sup>	9.91 x 10 <sup>-7</sup>	2.08

hydrogen and argon used for calibration.

Table A.1. Volume additions of hydrogen and corresponding pressure signals for



Figure A.1 – Linear calibration for the quantification of hydrogen using RGA-MS.

With this calibration fit, data from each photocatalytic hydrogen evolution experiment can be used to calculate the µmol amount of hydrogen. First the pressure signals in each experiment and the calibration fit can give a value for the volume% of hydrogen. The volume% multiplied by the headspace volume of 60 mL gives the volume of hydrogen detected. The moles of gas can be calculated from the volume using the ideal gas law.

## Photocatalysis set-up

While the main chapters of the thesis have described the methods used for photocatalytic hydrogen evolution, a more detailed description of the reactor design will be given here. Once the methanol or triethanolamine solution and catalyst were placed into the Schlenk flask, the solution was degassed by bubbling argon through a needle for 5-10 minutes. After this the flask was placed into the reactor cabinet and connected via gas inlet adapter to the Schlenk line (Figure A.2). The capillary from the RGA was connected to the side-arm of the flask using a quick connect fitting. During this time, data was collected in broad spectrum mode to observe the stabilization of the argon background and reduction of other atmospheric signals (N<sub>2</sub>, O<sub>2</sub>, H<sub>2</sub>O, CO<sub>2</sub>). When the background stabilized, the instrument was switched to a Pressure vs Time scan, selecting masses 28, 40, 2, 1, 18, 32, and 44, and changing from triggered to continuous scanning. In between illumination cycles, the produced hydrogen was vented from the flask by disconnecting the quick connect between the flask and RGA. It is important to keep the capillary close to the side arm while venting the hydrogen, in order to maintain the argon background and minimize the amount atmospheric gases that could leak in.



Figure A.2 – Photo of Schlenk flask inside the photoreactor cabinet. The hose connected to the gas inlet adapter is connected to an argon Schlenk line in the adjacent hood. The side-arm is connected to the sampling capillary from the RGA-MS. The flask is clamped above the stir plate, positioned 20 cm from the UV lamp.

## A.2 Quantum Yield Calculations

Quantum yield is defined by the ratio between the number of molecules produced or converted per time and the number of incident photons per time (photon flux). Equation 1 gives an example for the quantum yield of photocatalytic hydrogen production.

$$QY(\%) = \frac{2 \times Hydrogen \ molecules \ produced \ s^{-1}}{number \ of \ incident \ photons \ s^{-1}}$$
(A.1)

Photon flux was determined using the manufacturer's reported energy distribution of the radiated mercury lines for a 450 W Ace-Hanovia UV lamp on the exterior surface of the glass cooling jacket. The total power density at the exterior of the water cooling jacket for wavelengths absorbed greater than the  $TiO_2$  bandgap was 0.426 W cm<sup>-2</sup>. Using the energy for mercury emission wavelengths, the power can be converted to number of photons (Equation 2).

$$\Phi(photons \, s^{-1} \, cm^{-2}) = \frac{Power \, density \, (W \, cm^{-2})}{Photon \, energy \, (J)} \tag{A.2}$$

Totaled for the reported mercury emission wavelengths, this gives  $6.81 \times 10^{17}$  photons s<sup>-1</sup> cm<sup>-2</sup>. The reactor has an illumination area of 19.6 cm<sup>2</sup> and was a distance of 20 cm from the lamp (17 cm from the jacket exterior). The light intensity decay over this distance was determined using a digital light meter sensitive to the 550 nm emission line (Extech Instruments, LT300), giving a scaling factor of 0.0301. The total number of photons s<sup>-1</sup> was then calculated using equation 3.

$$6.81 \times 10^{17} \ photons \ s^{-1} \ cm^{-2} \times 19.6 \ cm^{2} \times 0.0301$$

$$= 4.02 \times 10^{17} \ photons \ s^{-1}$$
(A.3)

An example of calculating the QY for Cu(1%):TiO<sub>2</sub> (data from Chapter 5) is given in Equations 4 and 5.

$$84.7\mu mol \ h^{-1} \times \frac{1 \ mol}{10^6 \ \mu mol} \times \frac{6.022 \times 10^{23} \ molecules}{1 \ mol} \times \frac{1 \ h}{3600 \ s}$$
(A.4)  
$$= 1.41 \times 10^{16} \ molecules \ s^{-1}$$
$$QY(\%) = \frac{2 \times 1.41 \times 10^{16} \ molecules \ s^{-1} \ hydrogen}{4.02 \times 10^{17} \ photons \ s^{-1}} \times 100\% = 7.01\%$$
(A.5)

# A.3 Protocol for CHN Elemental Analysis

## Start-up

- 1. Check gas connections and that the three gas supplies are open and flowing at the correct pressures (Nitrogen 60 psi, Helium 20 psi, Oxygen 20 psi).
- 2. Turn on the Perkin Elmer 2400 CHNS/O Analyzer. Enter time (24 hour format), date (dd/mm/yy), and press enter through the rest of the start-up parameters. Press yes when prompted for helium purge and enter 60. Repeat for oxygen. Make sure the exhaust fan is running.
- Under diagnostics, select gas, then leak tests, and enter "1" to run combustion zone leak test. Repeat, entering "2" for mixing zone leak test. If leak tests fail check connections or manually actuate valves.
- Enter Parameter 12 and press "1" to turn on the furnace. Wait 2-3 hours for temperatures to stabilize before continuing.

# Calibration and sampling (Manual 4-58)

- Run a minimum of three blanks, until H, N, and C read reproduce within C±30, H±100, N±16. If the instrument has not been in operation for an extended period of time additional runs will be needed as adsorbed water vapors are removed from the gas lines. If the nitrogen blank does not stabilize and significantly increases in magnitude, the reduction tube needs to be changed. Depending on usage, this may occur before the run counters indicate.
- 2. Weigh samples and standards on microbalance. Use between 1-2 mg for best results.

- a. Tare the balance with empty tin capsules on the weigh pan and the reference pan.
- b. Add powder into the sample capsule, targeting 1-2 mg.
- c. Fold capsule shut using tweezers according to the diagram in manual (4-57).
- Run acetanilide standard, values should fall within tolerances theoretical values (71.09% C, 10.36% N 6.71% H).
  - a. If measured values for the standard do not match, a full calibration sequence must be run. For full calibration after first standard, run blank, standard, blank, and three k-factors. Check manual for detailed steps and tolerance values (4-63).
- 4. Run samples by single run or auto run
  - a. Single Run: Load Sample, Enter sample name (letters entered by .xx corresponding to order in alphabet A= .01, Z= .26), enter sample weight, press start
  - Auto run: Load samples into carousel and enter name and weight for each sample, then press start. If running more than five samples, leave a space after the fifth sample in the carousel, and insert a blank run in the run sequence.
  - c. Optimize combustion: If samples are graphite-like or are largely inorganic with less than 5% total CHN, enter parameter 9 to optimize combustion timing. Enter 1 and change OXYFILL to 3, enter 2 and change COMB to 10, enter 3 and change OXB1 to 1. Additional recommended values are in

the manual (5-34). After analyzing samples with optimized combustion, return OXYFILL, COMB, and OB1 to default value of 1.

## Shutdown

- 1. Enter parameter 12 and press "2" to turn off the furnace.
- Enter parameter 22 and press "1" to turn on Gas Saver. Enter a start time 2-3 hours after when the furnace was shut off. Enter an arbitrarily distant end time.
- 3. If the instrument will not be in immediate use, shut off main power, only if the furnace has reached room temperature. Close all gas valves at the tanks.

# Changing the combustion and reduction tubes

- When the instrument is powered on it will display the remaining run counts for the combustion and reduction tubes. Depending of the parameters of the runs until this point, the reduction tube may need to be replaced before the remaining counts reaches zero.
- Page 4-6 in the manual describes the necessary reagents and filling method for each tube.
- 3. Figures A.3A and A.3B show photos of correctly prepared tubes.
- 4. A full calibration sequence should be run before the next analysis after replacing either the combustion or reduction tubes.



Figure A.3 – Photographs of (A) prepared reduction tube and (B) prepared combustion tube.

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