

Simulating Spectra

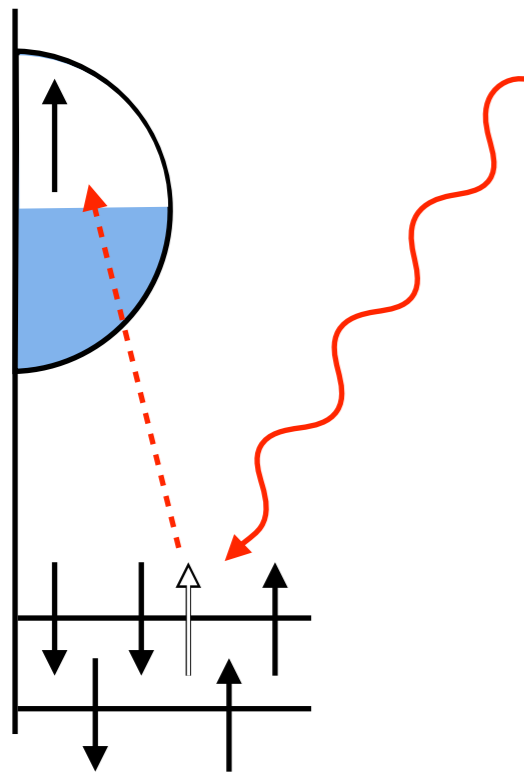
Travis Jones
19 Jan 2018

Introduction

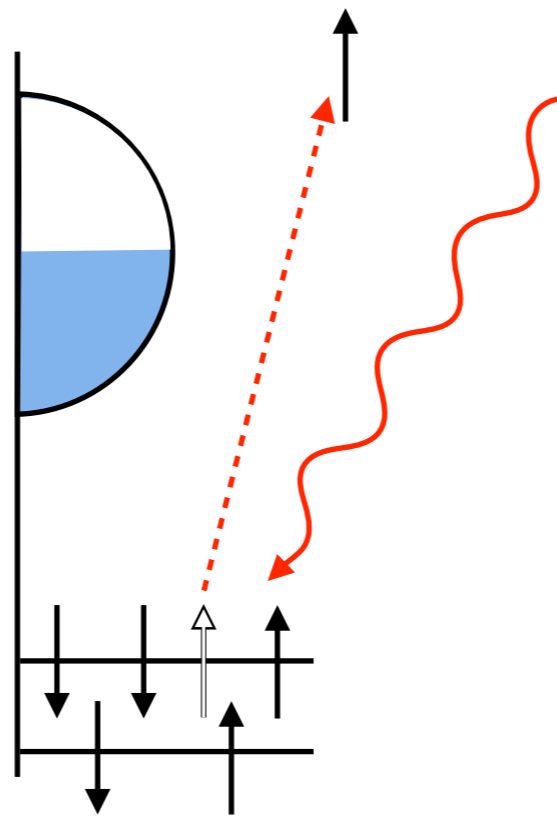
- Why should you care about calculating spectra?
- What kinds of spectra can you compute?
- What types of approaches are there?
- What are the pitfalls?

A few types of X-ray spectroscopy

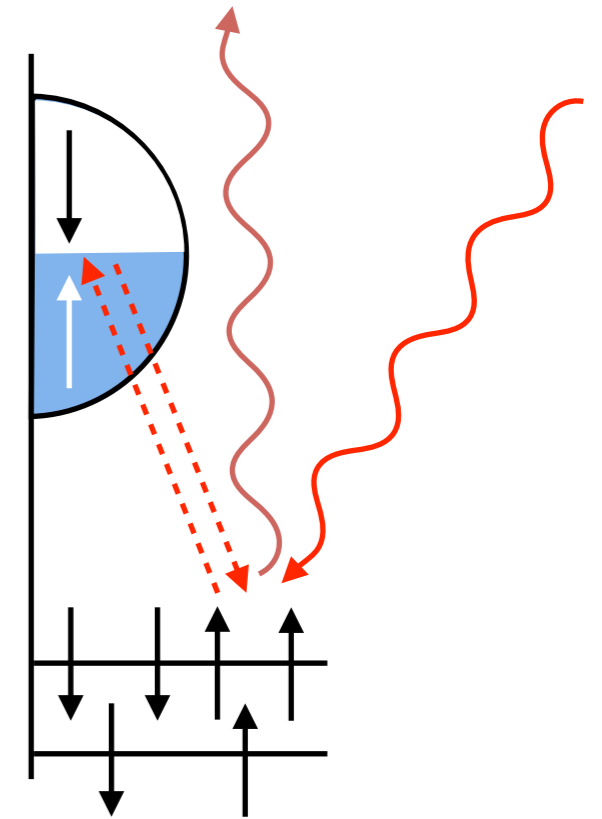
Absorption



Photoemission



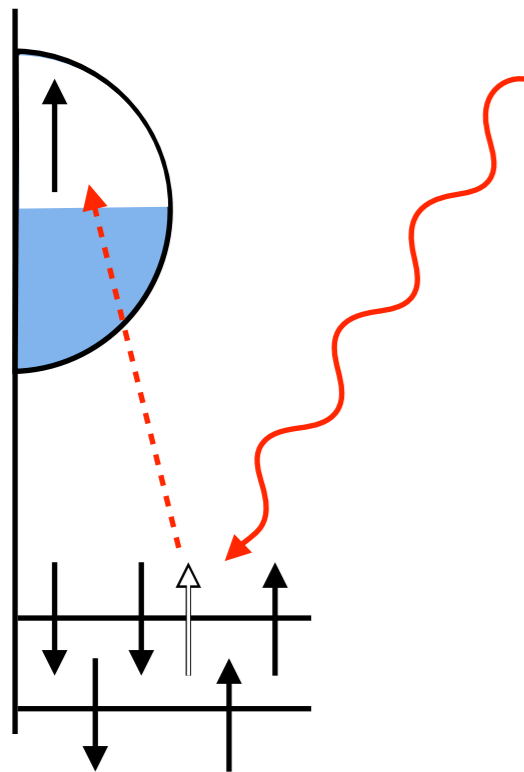
RIXS



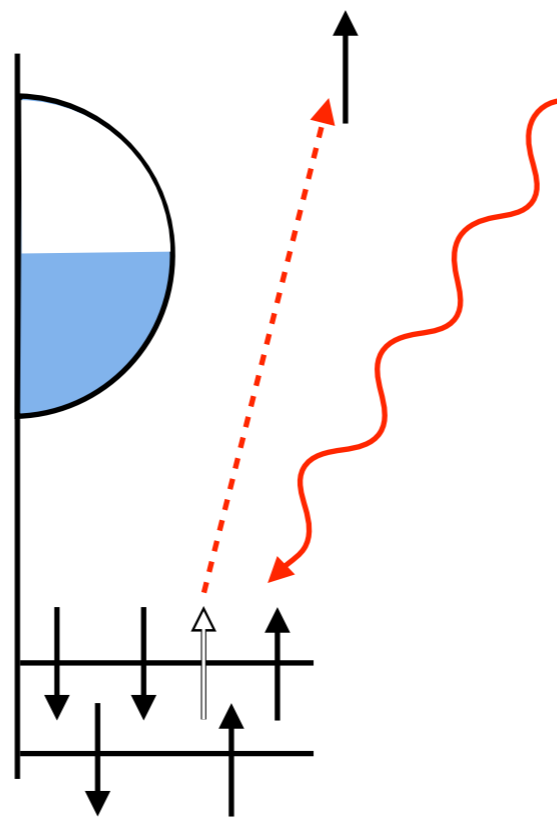
A few types of X-ray spectroscopy

- Give different but complementary information

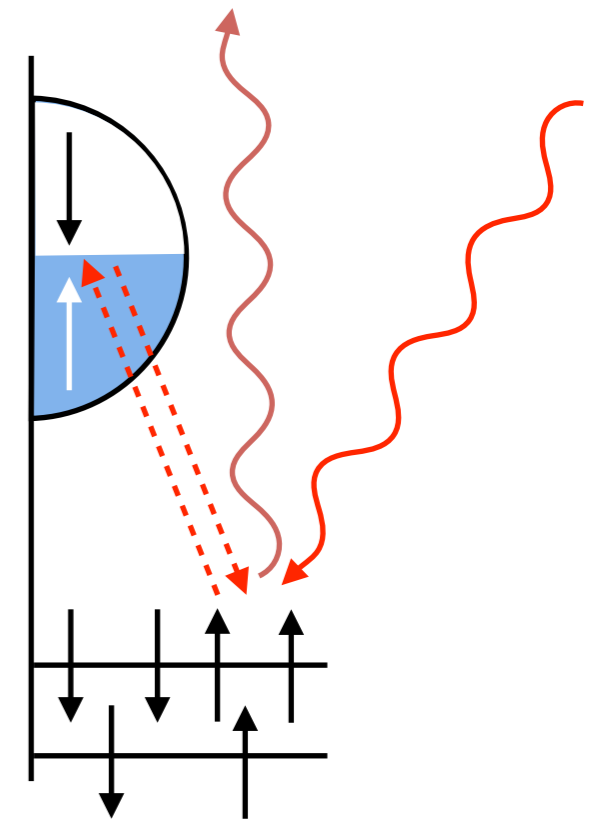
Absorption



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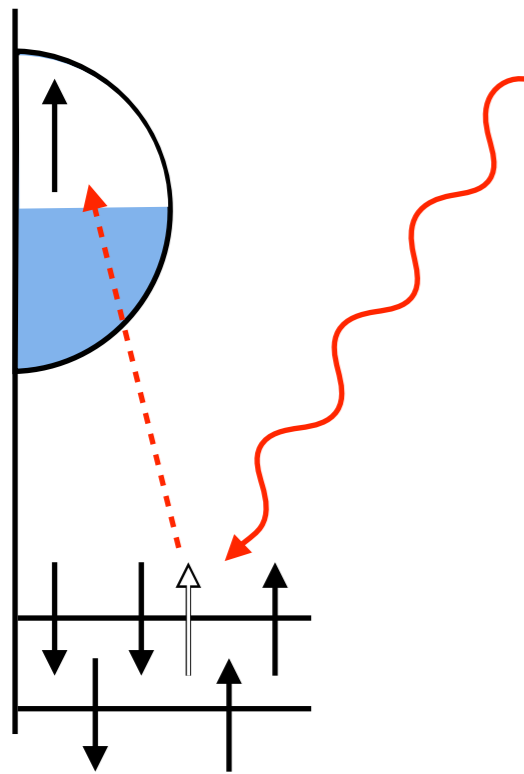
RIXS



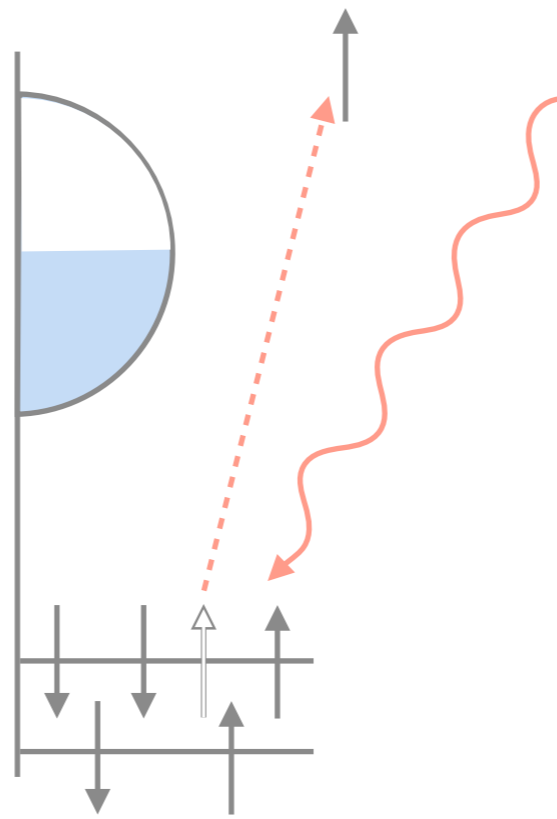
A few types of X-ray spectroscopy

- Give different but complementary information

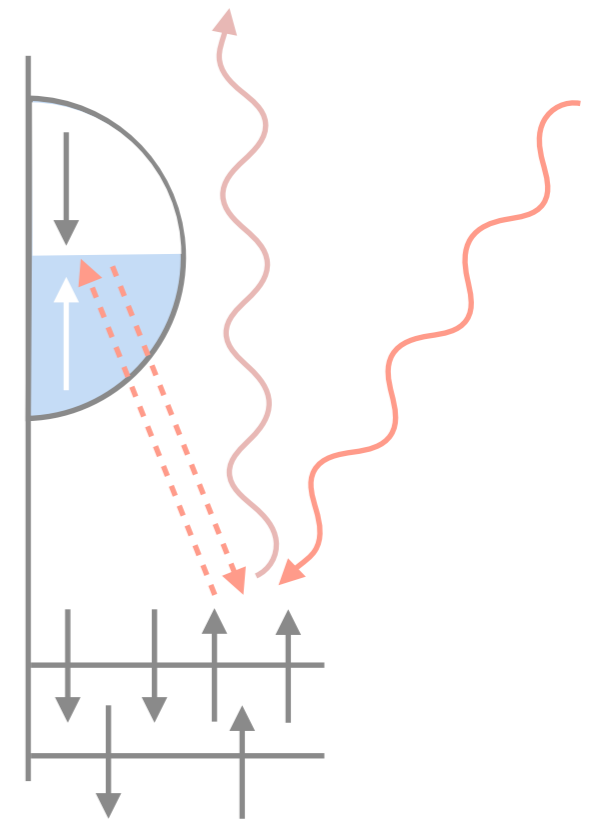
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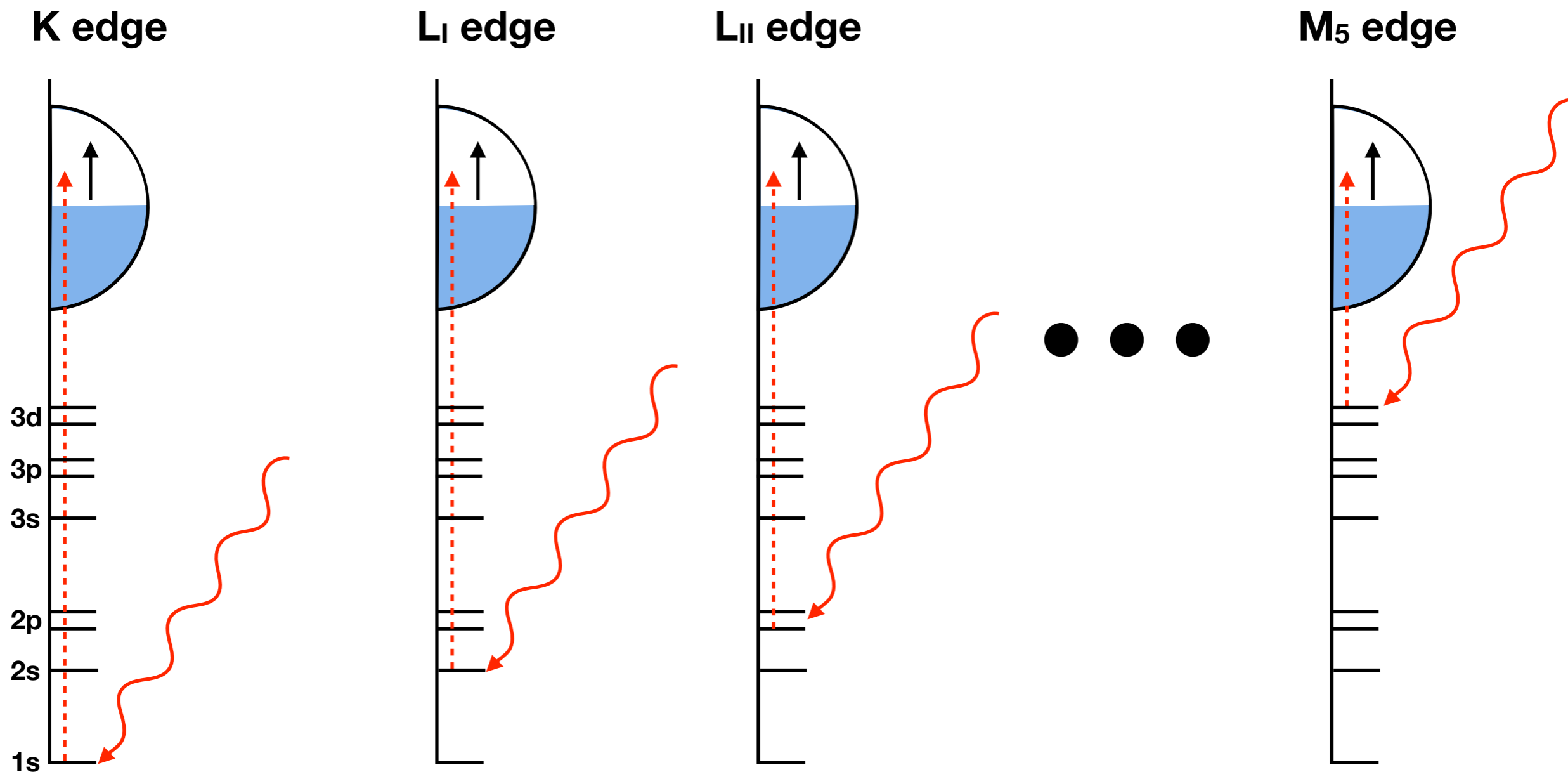


RIXS



Vocabulary

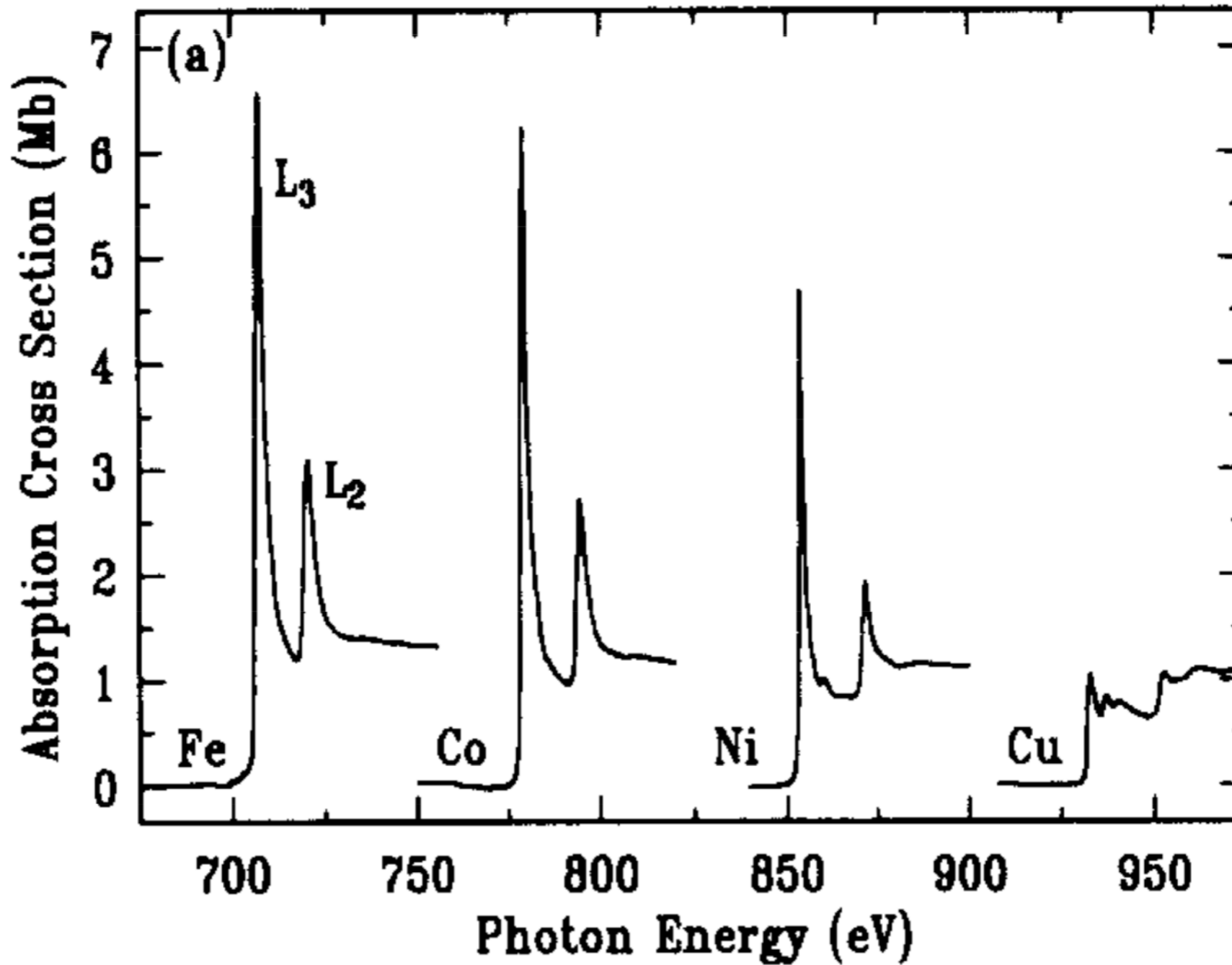
- Name identifies (n, l) quantum number of core electron involved
 - Letter give n starting with K
 - Number gives l starting with I



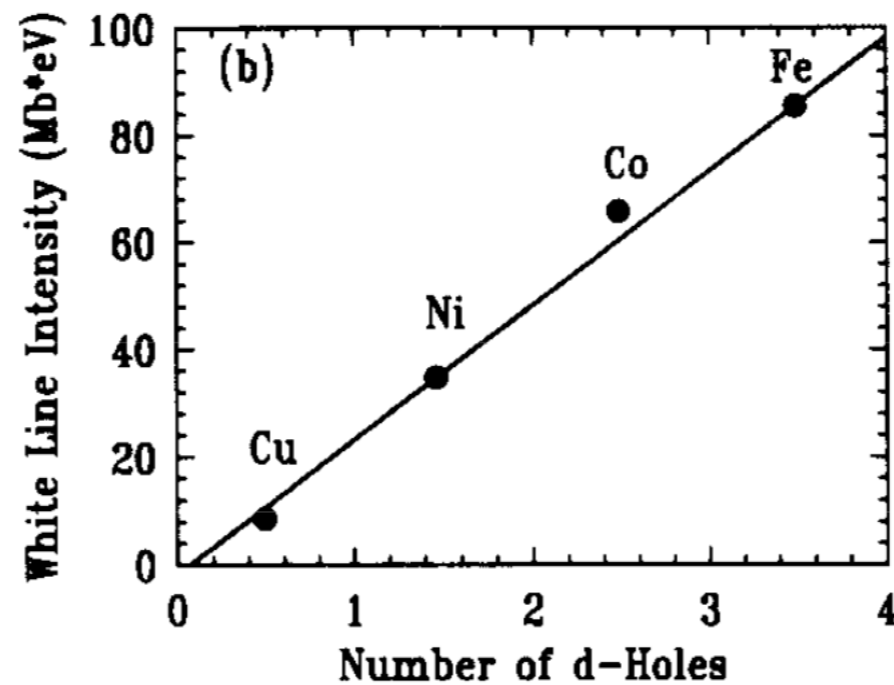
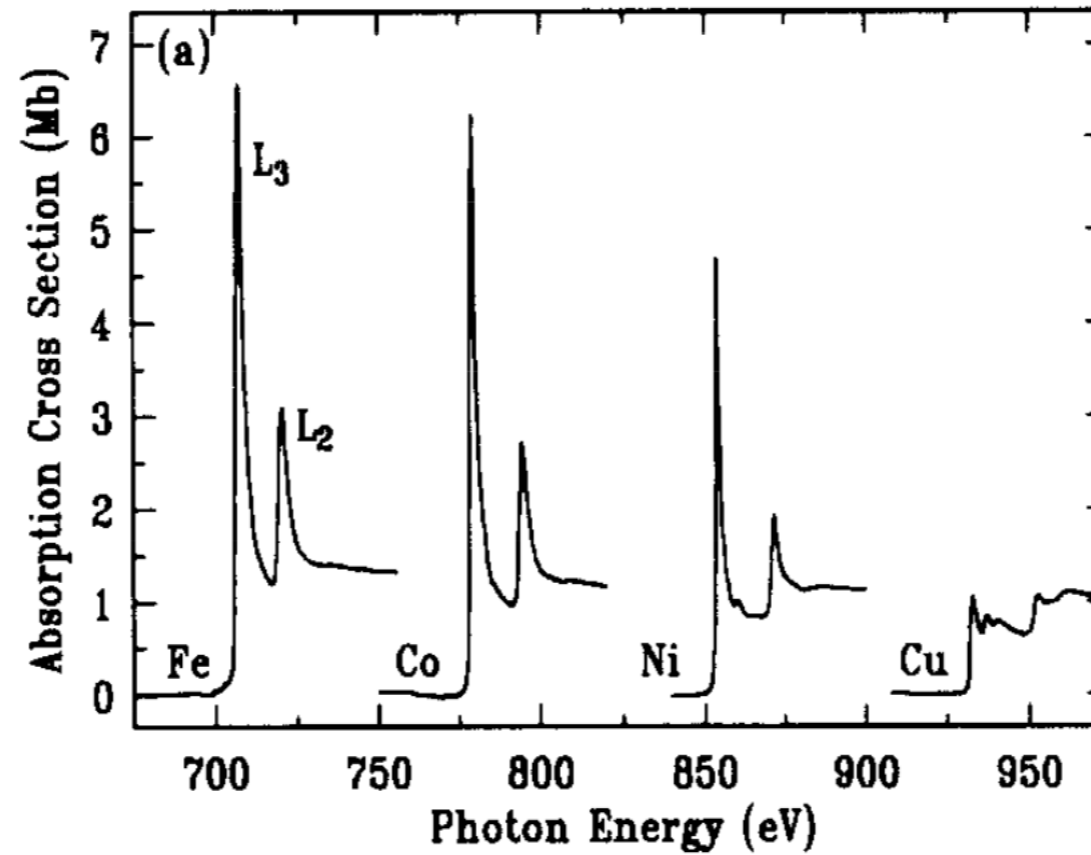
What do we need to understand to describe XAS using theory?

XAS $L_{III,II}$

- Edge onset and hydrogen like atoms



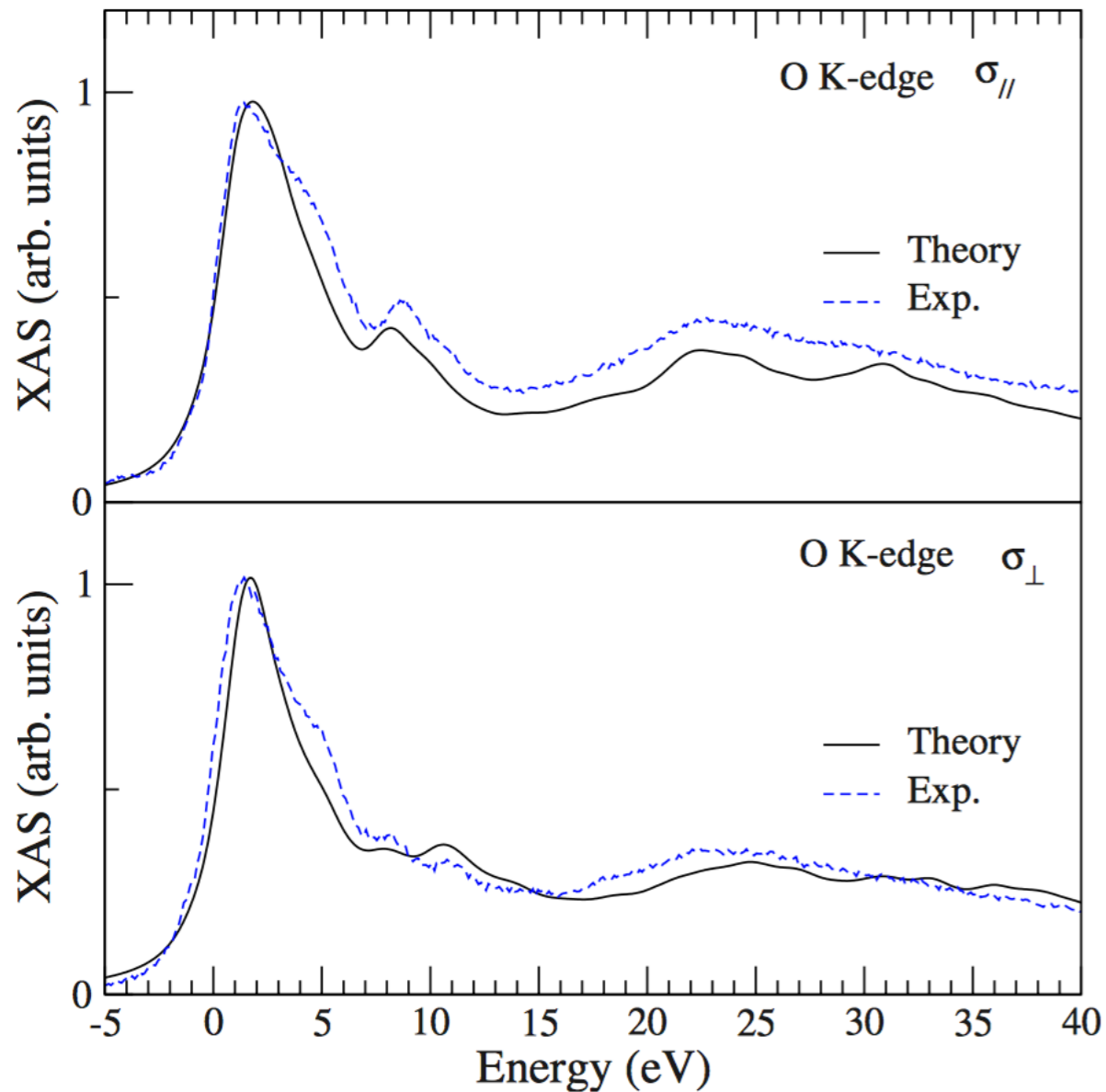
Useful sum-rules



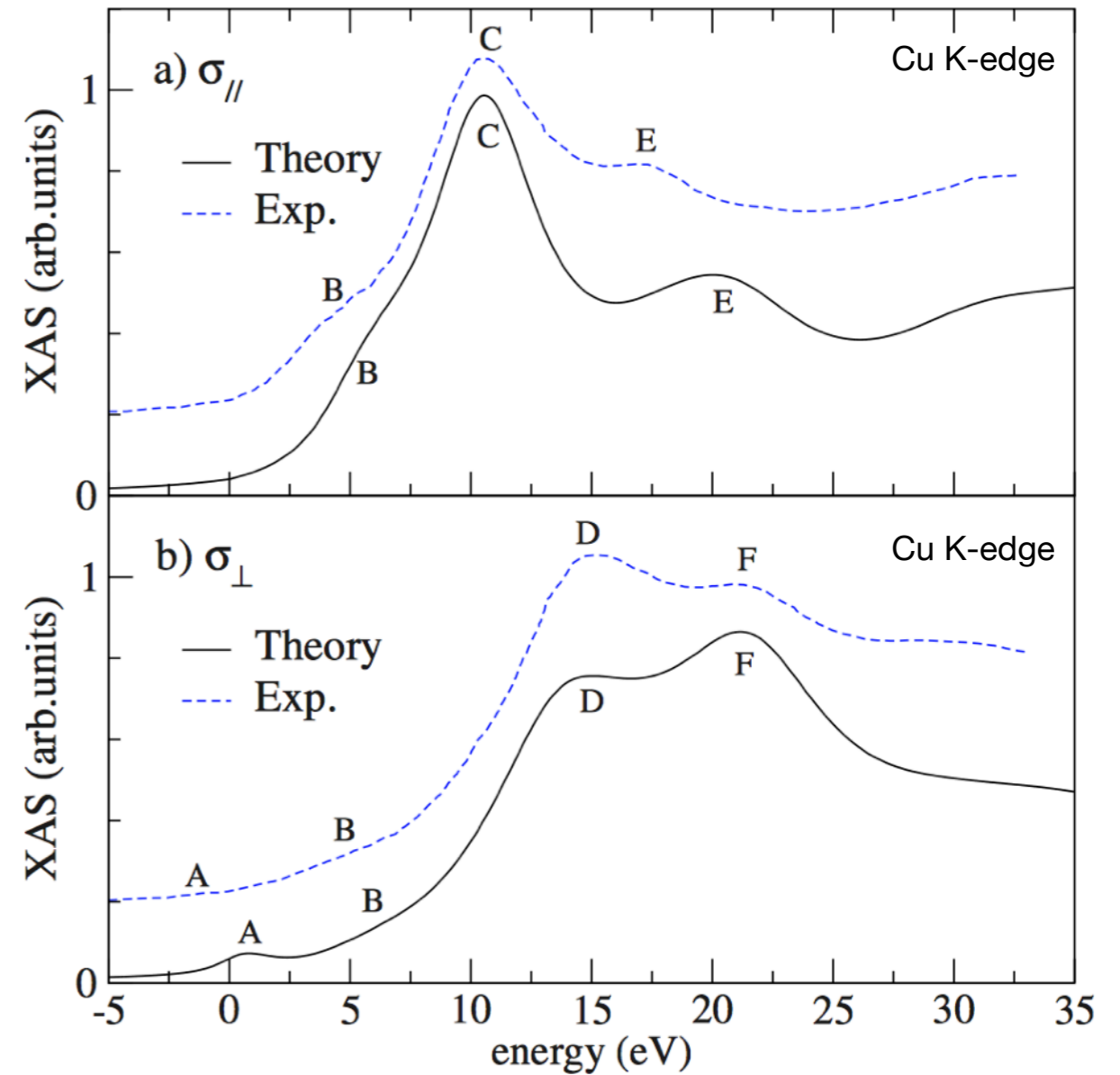
XAS K-edge

- K-edge looks like the empty density of states on absorber

α -quartz SiO_2



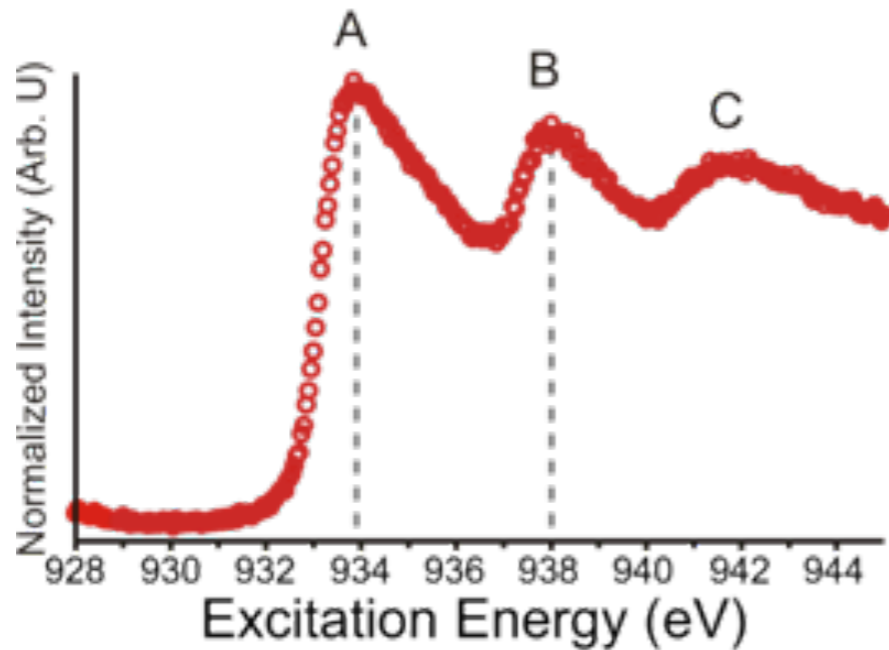
La_2CuO_4



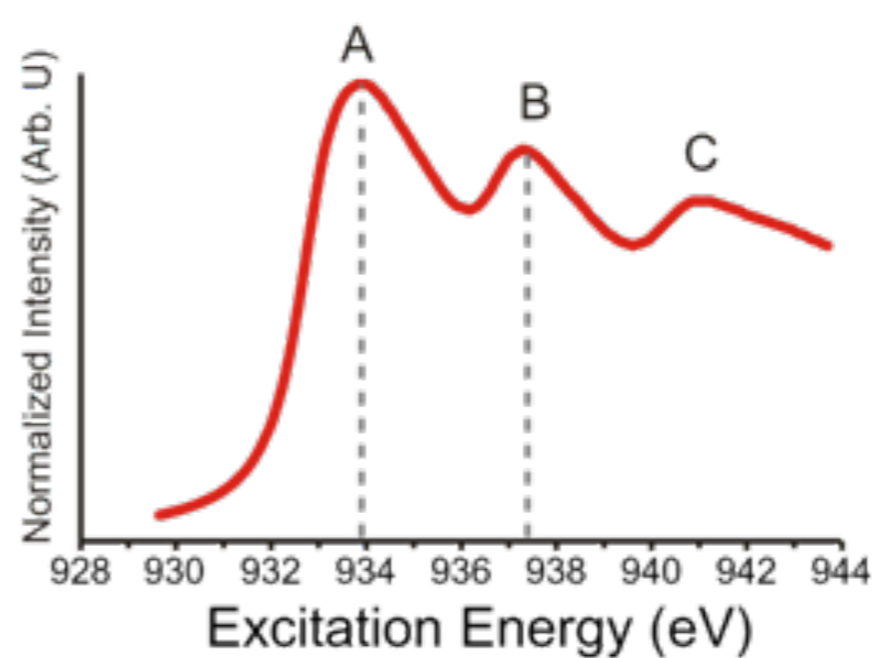
XAS $L_{II,III}$ -edge

- $L_{II,III}$ -edge looks like the empty density of states on absorber

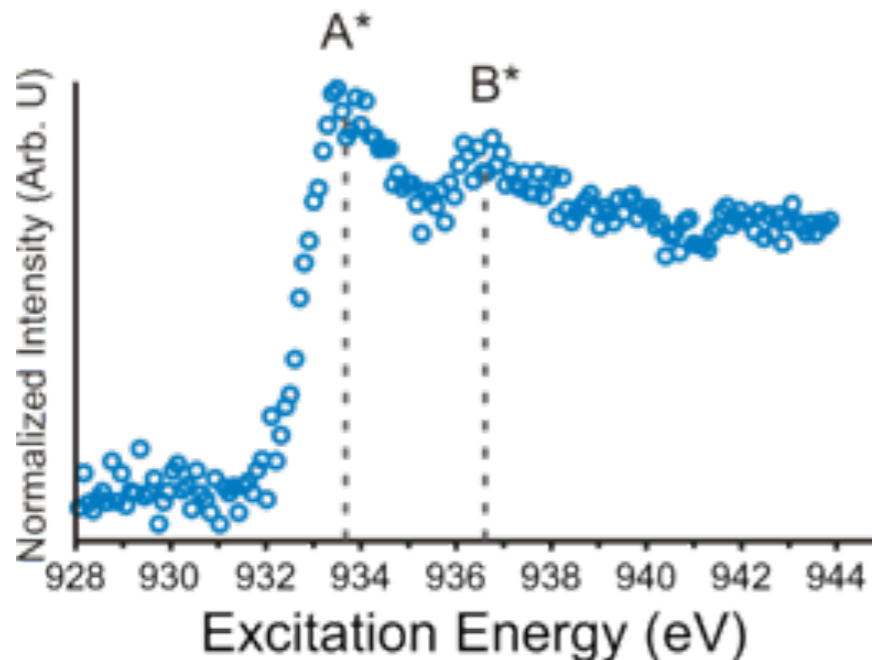
a) Measured Cu L_3 of metallic Cu



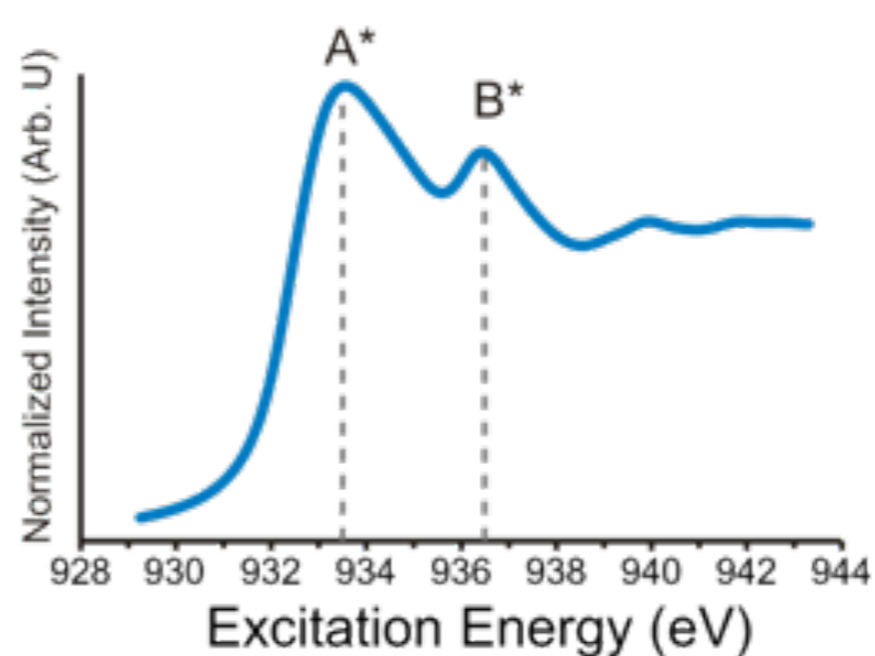
b) Calculated Cu L_3 of metallic Cu



c) Measured Cu L_3 of AgCu (0.3%Cu)

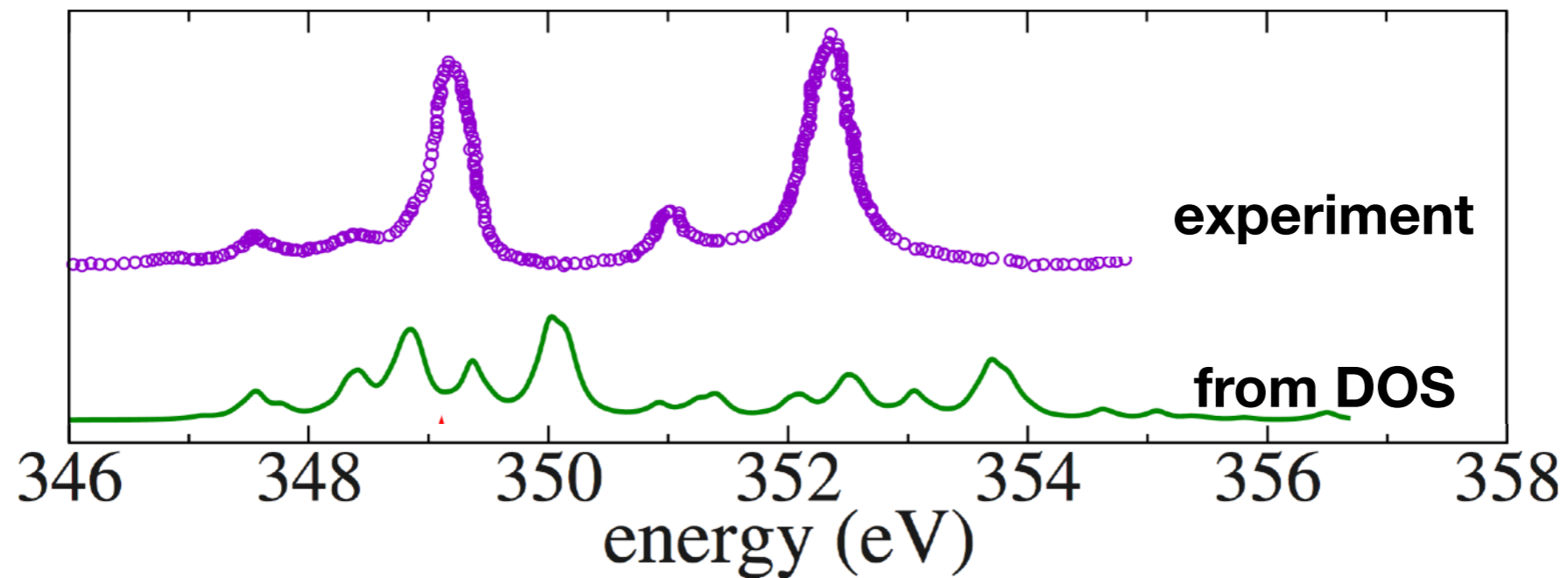


d) Calculated Cu L_3 of $Ag_{31}Cu_1$



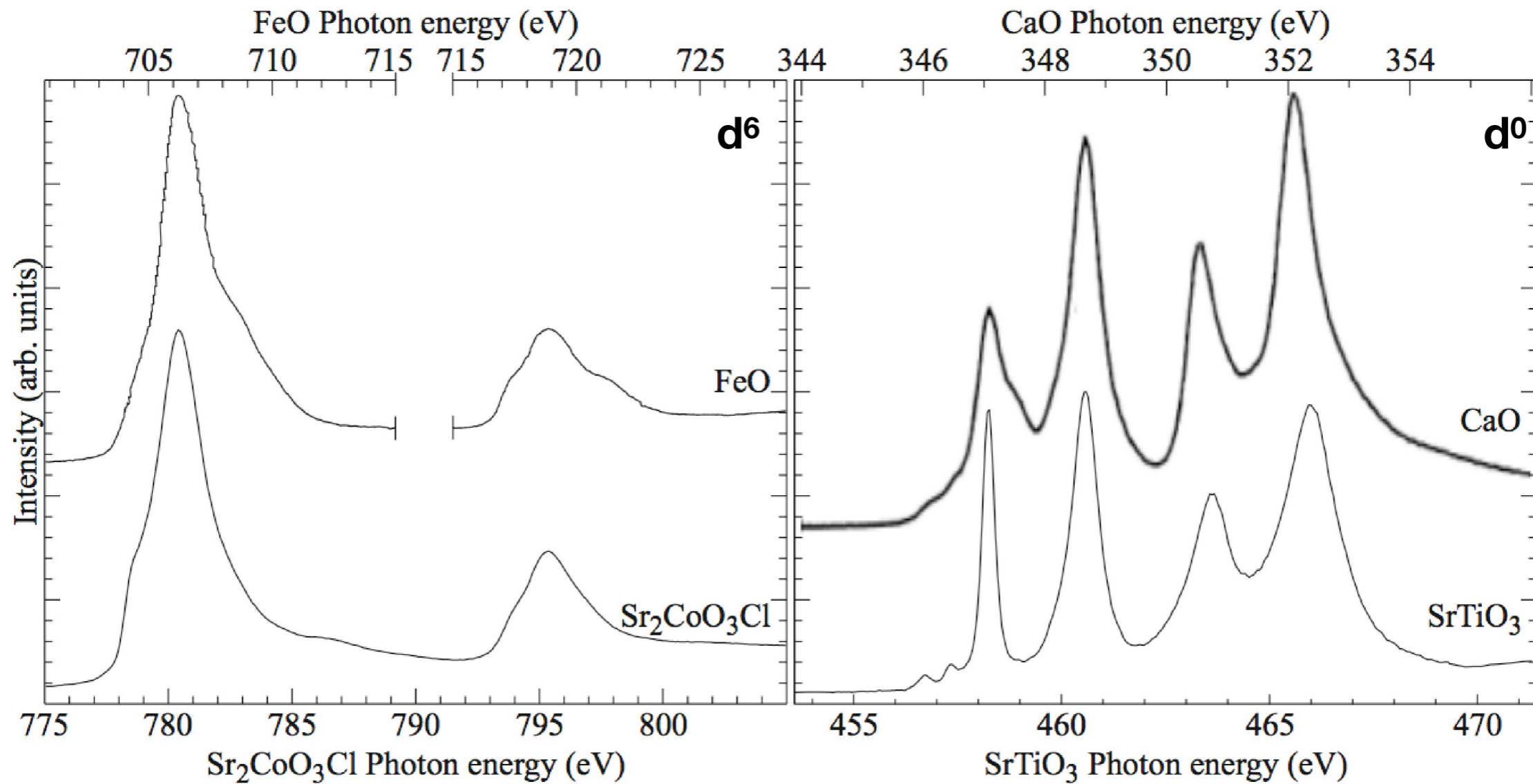
XAS L_{II,III}-edge

- ~~L_{II,III}-edge looks like the empty density of states on absorber~~
- Apparently depends on the material



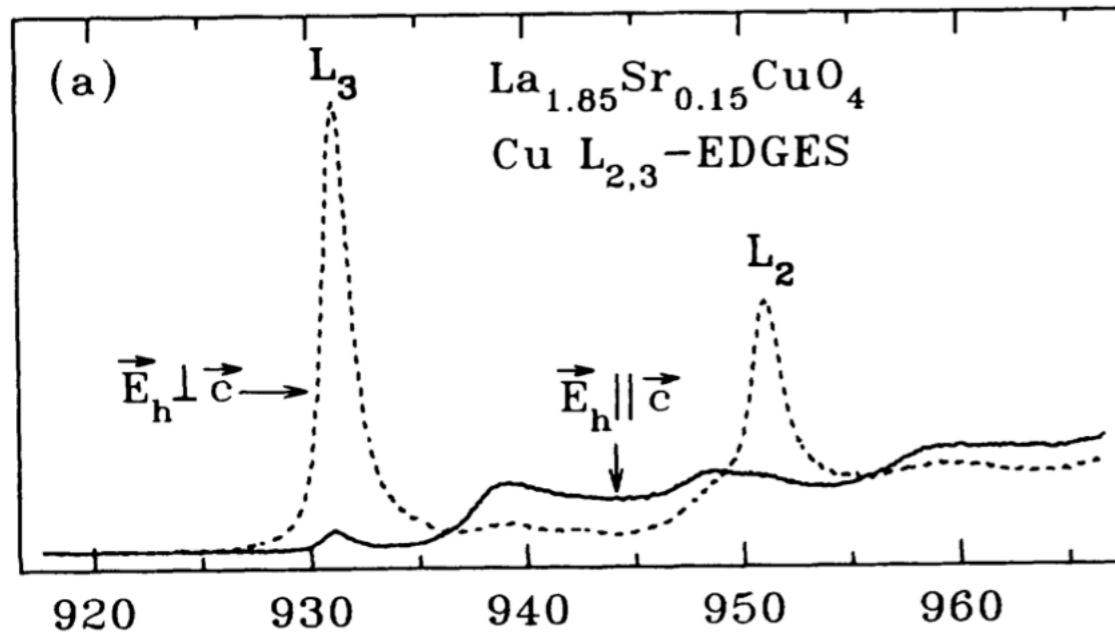
XAS $L_{II,III}$

- In some cases spectrum seems to reflect d electron count

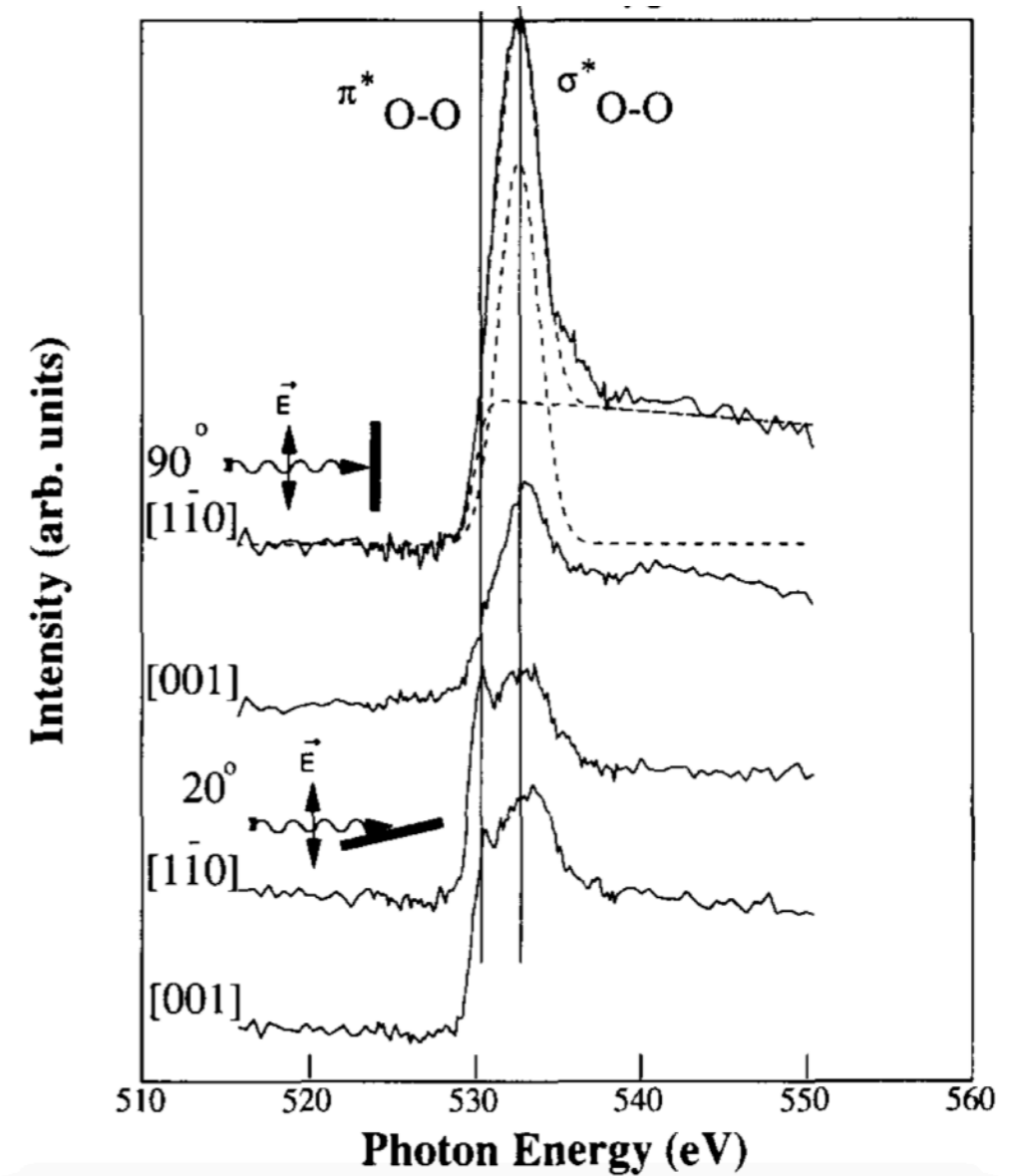


Polarization dependence

- Changing E vector gives different spectra for single crystals



C. T. Chen et al PRL 68, 2543 (1992)



J. Pawel-Crew et al Surf. Sci. 339, 25 (1995)

How can we understand these spectra?

One electron Selection Rules

- Transition described by Fermi's golden rule

$$\sigma(\omega) = \sum_f |\langle f | T | i \rangle|^2 \frac{i/\pi}{\omega - E_f + i\Gamma/2}$$

Diagram illustrating Fermi's Golden Rule for transition cross-section $\sigma(\omega)$. The equation is annotated with arrows pointing to its components:

- final state**: points to $|f\rangle$ in the bra part of the matrix element.
- initial state**: points to $|i\rangle$ in the ket part of the matrix element.
- Transition operator**: points to T .
- Energy Conservation**: points to the denominator $\omega - E_f + i\Gamma/2$.

One electron Selection Rules

- Transition described by Fermi's golden rule

$$\sigma(\omega) = \sum_f |\langle f | T | i \rangle|^2 \frac{i/\pi}{\omega - E_f + i\Gamma/2}$$

final state

initial state

Transition operator for light-matter interaction

Energy Conservation

$$T = \vec{p} \cdot \hat{\epsilon} e^{i\vec{k} \cdot \vec{r}}$$

Multipolar approximation

- We can simplify the transition operator

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Multipolar approximation

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$$T = \vec{p} \cdot \hat{\epsilon} e^{i\vec{k} \cdot \vec{r}}$$

- By expanding the plane wave as

$$e^{i\vec{k} \cdot \vec{r}} = 1 + i \left(\vec{k} \cdot \vec{r} \right) + \frac{1}{2!} i^2 \left(\vec{k} \cdot \vec{r} \right)^2 \dots$$

Multipolar approximation

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- How do we know which terms to keep?

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 - Option 1: Dimensional analysis

Multipolar approximation

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- How do we know which terms to keep?
 - Option 1: Dimensional analysis
 - Option 2: Look at $\left(\vec{k} \cdot \vec{r} \right)$

Multipolar approximation

- Edge energy approximately energy of electron in n shell

$$E = \hbar\omega \approx \frac{Z}{n} \approx \frac{Z\alpha\hbar c}{R_n}$$

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Multipolar approximation

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$$E = \hbar\omega \approx \frac{Z}{n} \approx \frac{Z\alpha\hbar c}{R_n}$$

- Wavenumber is

$$\left| \vec{k} \right| = k = \frac{E}{\hbar c}$$

- Keep 1 term for $Z \ll 137$

$$\left| \vec{k} \cdot \vec{r} \right| \approx Z\alpha \approx \frac{Z}{137}$$

One electron Selection Rules

- Looks like the dipole approximation is pretty good

$$T = \vec{p} \cdot \hat{\epsilon}$$

- What does it do in our Golden rule expression?

$$\sigma(\omega) = \sum_f |\langle f | \vec{p} \cdot \hat{\epsilon} | i \rangle|^2 \frac{i/\pi}{\omega - E_f + i\Gamma/2}$$

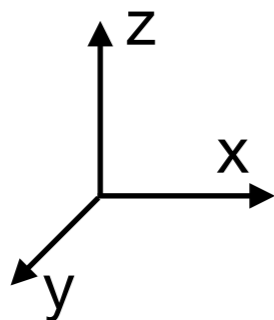
Dipole transitions

$$\sigma(\omega) = \sum_f |\langle f | T | i \rangle|^2 \frac{i/\pi}{\omega - E_f + i\Gamma/2}$$

$$\langle \bullet \circ \mid \begin{array}{c} \bullet \\ \circ \end{array} \mid \bullet \rangle =$$

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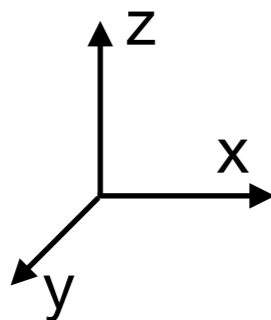
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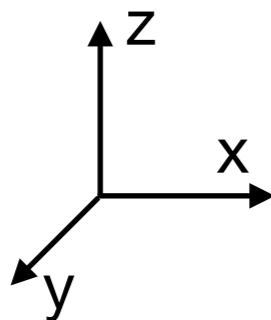
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$$\langle \bullet \circ \mid \bullet \circ \mid \bullet \rangle = 1$$

$$\langle \bullet \circ \mid \begin{array}{c} \circ \\ \bullet \end{array} \mid \bullet \rangle =$$



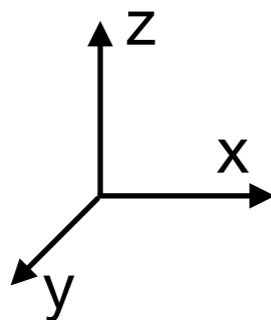
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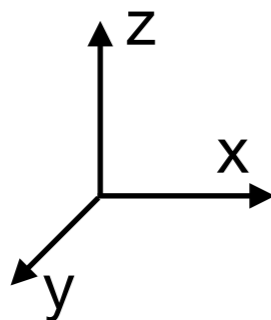
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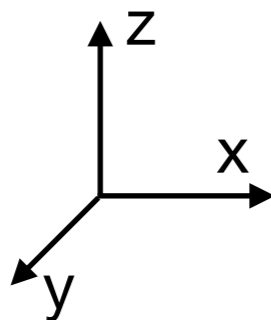
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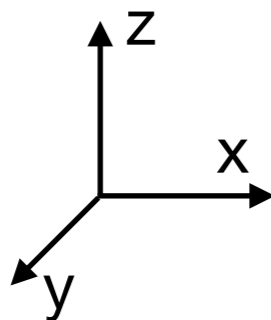
Dipole transitions

$$\sigma(\omega) = \sum_f |\langle f | T | i \rangle|^2 \frac{i/\pi}{\omega - E_f + i\Gamma/2}$$

$$\langle \text{p-orbital} | \text{p-orbital} | \text{d-orbital} \rangle = 0$$

$$\langle \text{p-orbital} | \text{p-orbital} | \text{p-orbital} \rangle = 0$$

$$\langle \text{p-orbital} | \text{d-orbital} | \text{d-orbital} \rangle = 1/2$$



Dipole transitions

$$\sigma(\omega) = \sum_f |\langle f | T | i \rangle|^2 \frac{i/\pi}{\omega - E_f + i\Gamma/2}$$

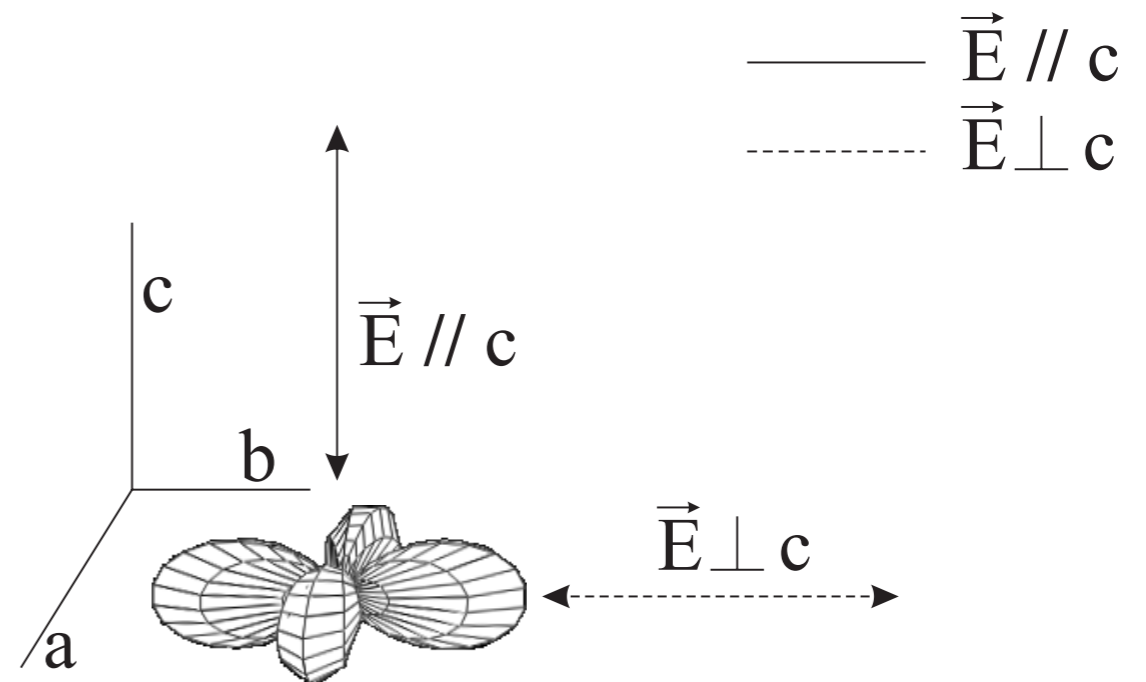
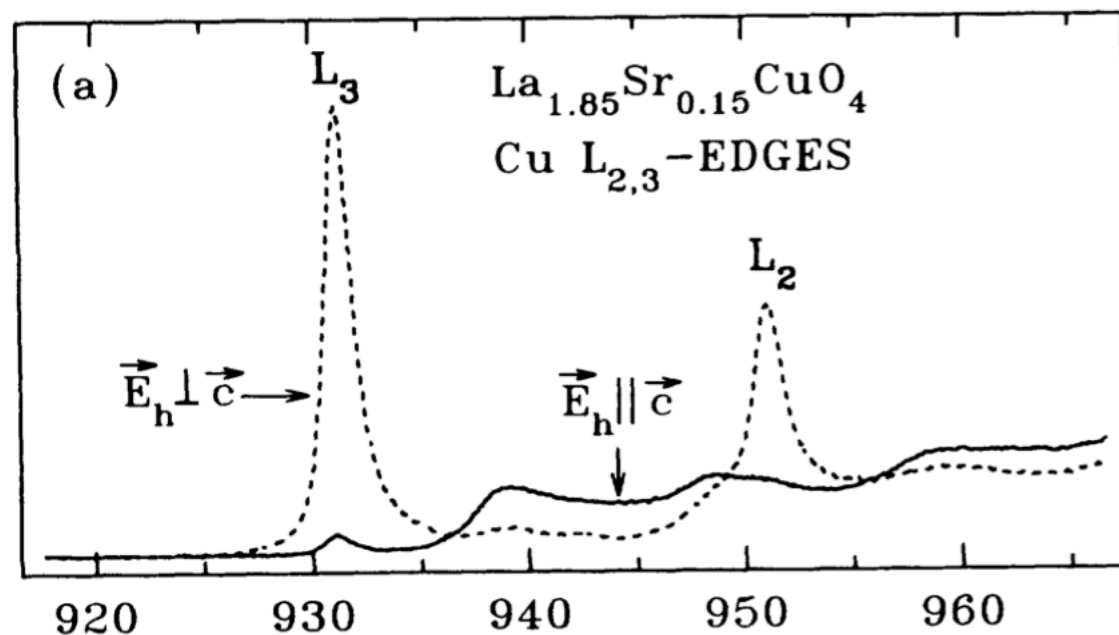
- Dipole transitions are between $\Delta l = \pm 1$
- σ is a tensor that obeys the point group symmetry

$$\begin{pmatrix} \sigma_{xx} & \sigma_{xy} & \sigma_{xz} \\ \sigma_{yx} & \sigma_{yy} & \sigma_{yz} \\ \sigma_{zx} & \sigma_{zy} & \sigma_{zz} \end{pmatrix}$$

- Powders are isotropic—trace of σ

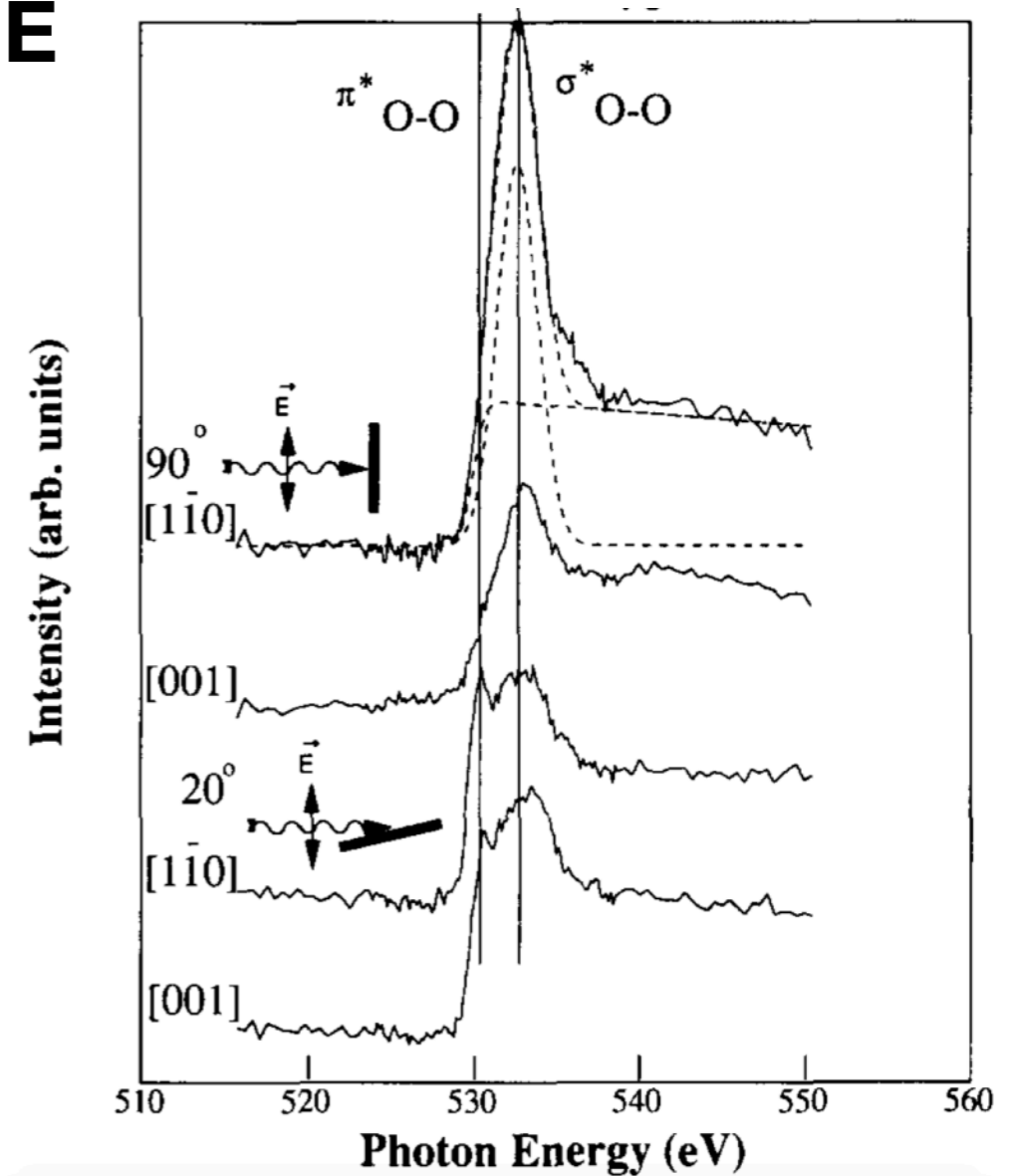
Now we can understand angular dependence

- Changing E vector probes different elements of σ
- Holes in the $d_{x^2-y^2}$



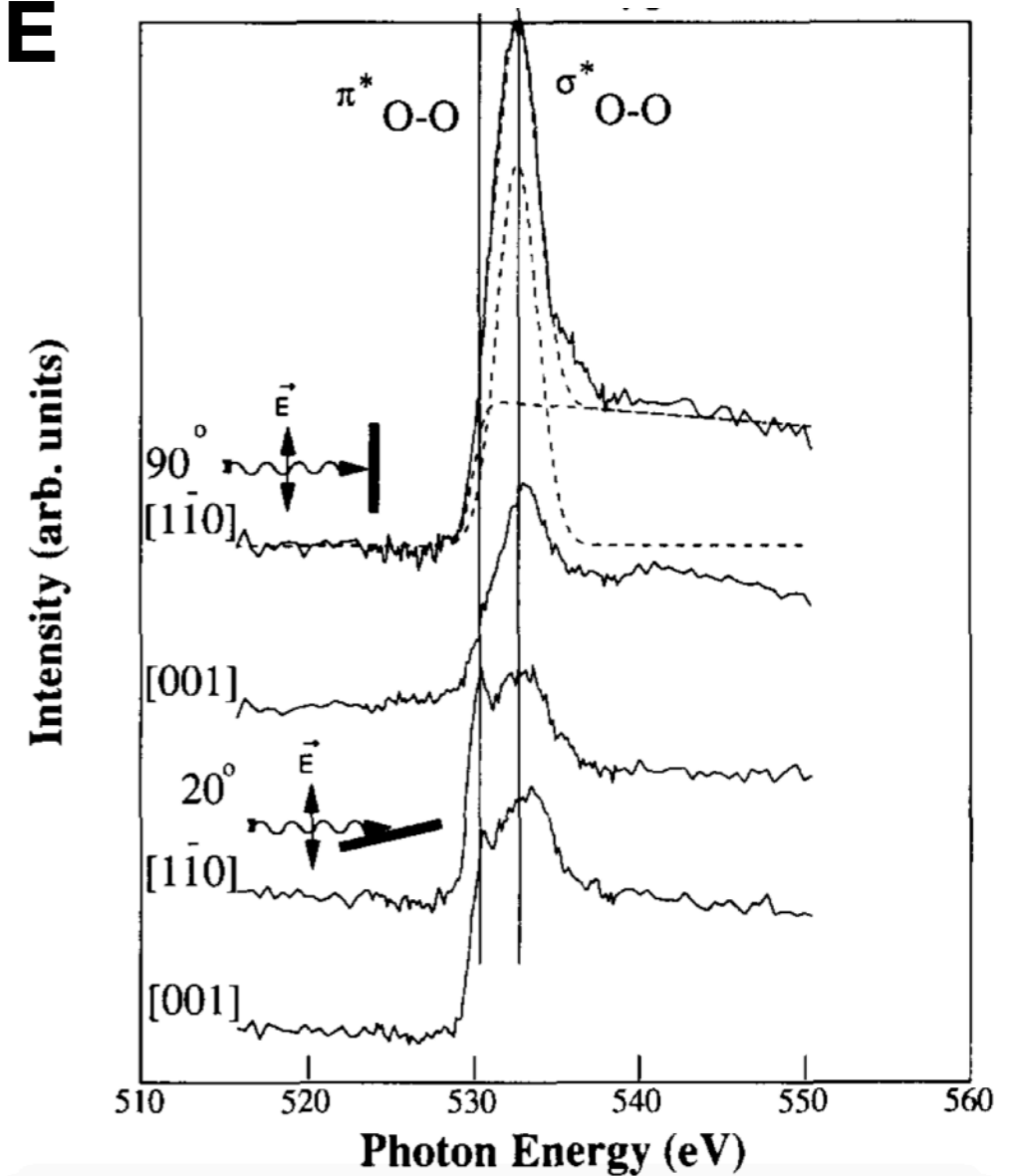
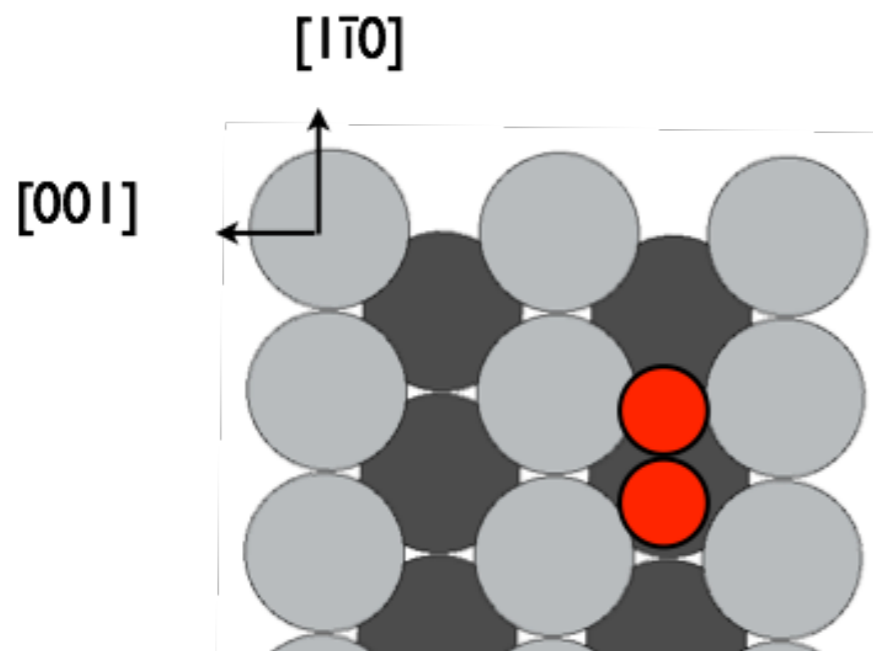
Now we can understand angular dependence

- Changing \mathbf{E} vector probes different elements of σ
- Empty π^* along $[1\bar{1}0]$ with out-of-plane \mathbf{E}
- Empty σ^* along $[1\bar{1}0]$ with in plane \mathbf{E}



Now we can understand angular dependence

- Changing \mathbf{E} vector probes different elements of σ
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How to compute

- Can we describe all behavior with one framework?
- In practice pick methodology depending on the type of material and the measurement
- Let's start with deep K edges for materials that are not strongly correlated because DFT is sufficient for them

Quick DFT

- Most electronic structure calculations done with Kohn-Sham DFT
- Works amazingly well but has serious limits
- Basic idea is to solve KS equation

$$\left(-\frac{\hbar^2}{2m} \nabla^2 + v_{ext}(\vec{r}) + v_H(\vec{r}) + v_{xc}(\vec{r}) \right) \phi(\vec{r}) = \epsilon_i \phi_i(\vec{r})$$

- Formally scales as $O(N^3)$
- Not totally correct because we don't know v_{xc}
- To solve we need some basis functions that we can pick

Choices to make

- We have already settled on DFT
- Now we need to think about the type of system
 - Solid?
 - Molecule?
 - Metal?
 - Insulator?
- The answers to these questions will impact the approach
- Say we have solid
 - For metals and insulators we could then use periodic boundary conditions
 - Planewaves are a good basis set for periodic boundary conditions

Planewave DFT

- Expand wave functions in a plane wave basis set
- Advantages
 - FFTs reduce scaling from $O(N^3)$ to $O(N^2)$
 - Systematically improvable with a convergence parameter
- And drawbacks
 - Vacuum costs the same as atoms
 - Cannot handle rapid oscillations of wave functions—need pseudo potentials

PAW approach

- $|i\rangle$ and $|f\rangle$ are *all electron* states
- We want to use pseudo states

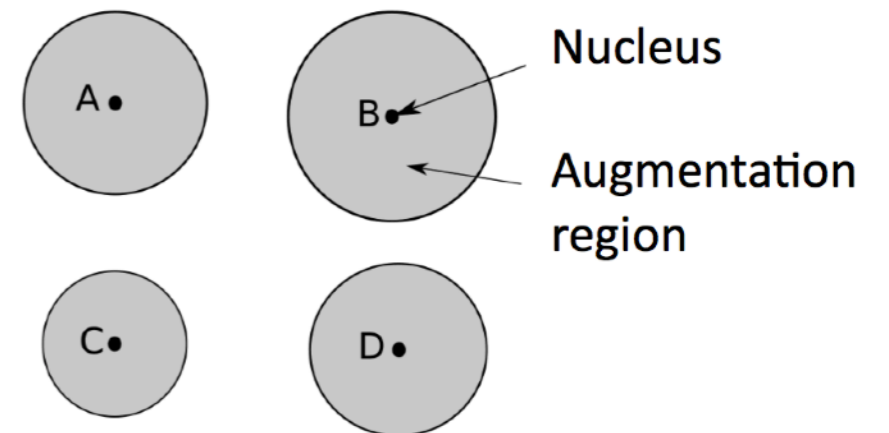
linear mapping between pseudo and all electron wave functions

$$|\Psi\rangle = T |\tilde{\Psi}\rangle$$

$$T = \hat{1} + \sum_{\vec{R},m} \left(|\phi_{\vec{R},n}\rangle - |\tilde{\phi}_{\vec{R},n}\rangle \right) \langle p_{\vec{R},n}| \xrightarrow{\text{projectors (non-zero in augmentation region)}}$$

AE partial waves

pseudo partial waves



Dipole and PAW

- Because the core hole is localized only terms on the absorbing atom survive

$$\sigma(\omega) = \sum_f \left| \langle \tilde{\Psi}_f | \tilde{\Phi}_{\vec{R}_0} \rangle \right|^2 \frac{i/\pi}{\omega - E_f + i\Gamma/2}$$

Sum over projectors

$$\tilde{\Phi}_{\vec{R}_0} = \sum_n \left| p_{\vec{R}_0, n} \right\rangle \left\langle \tilde{\phi}_{\vec{R}_0, n} \left| D \right| \psi_i \right\rangle$$

- To compute the spectrum you need
 - Core wave function on absorber ψ_i
 - AE partial waves centered on absorber
 - PAW projectors

What does it all mean?

XAS is (in this case) measuring a lifetime broadened projected density of states in the presence of a core.

$$\sigma(\omega) = \sum_f \left| \langle \tilde{\Psi}_f | \tilde{\Phi}_{\vec{R}_0} \rangle \right|^2 \frac{i/\pi}{\omega - E_f + i\Gamma/2}$$

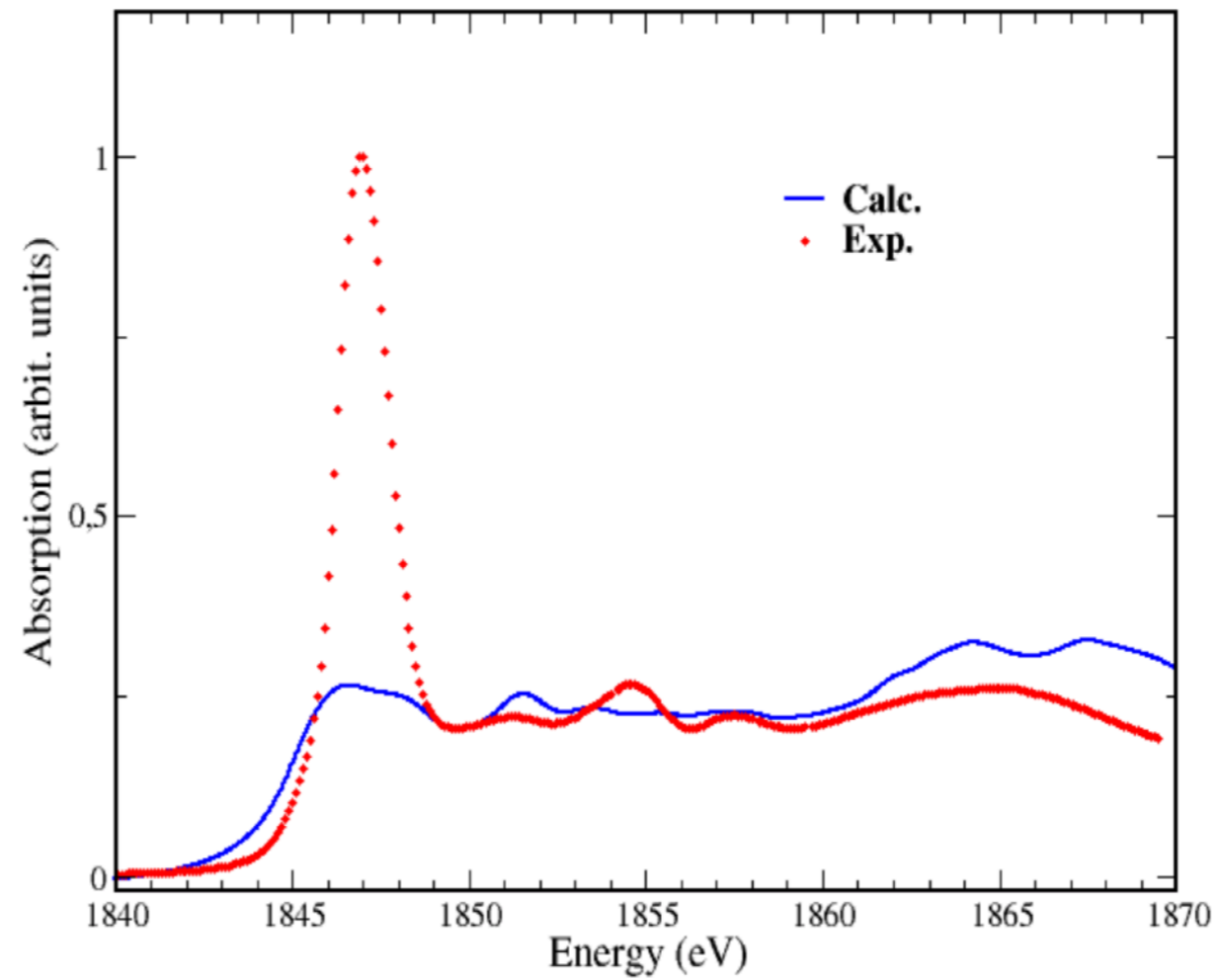
$$\tilde{\Phi}_{\vec{R}_0} = \sum_n \left| p_{\vec{R}_0, n} \right\rangle \left\langle \tilde{\phi}_{\vec{R}_0, n} \left| D \right| \psi_i \right\rangle$$

Oh the core hole

- There is a core hole in the final state
- But we are using an independent particle approach
- Turns out you don't always need a core-hole
- Some common approaches are
 - Full core hole
 - Half core hole aka Slater's transition potential approximation
 - No core hole

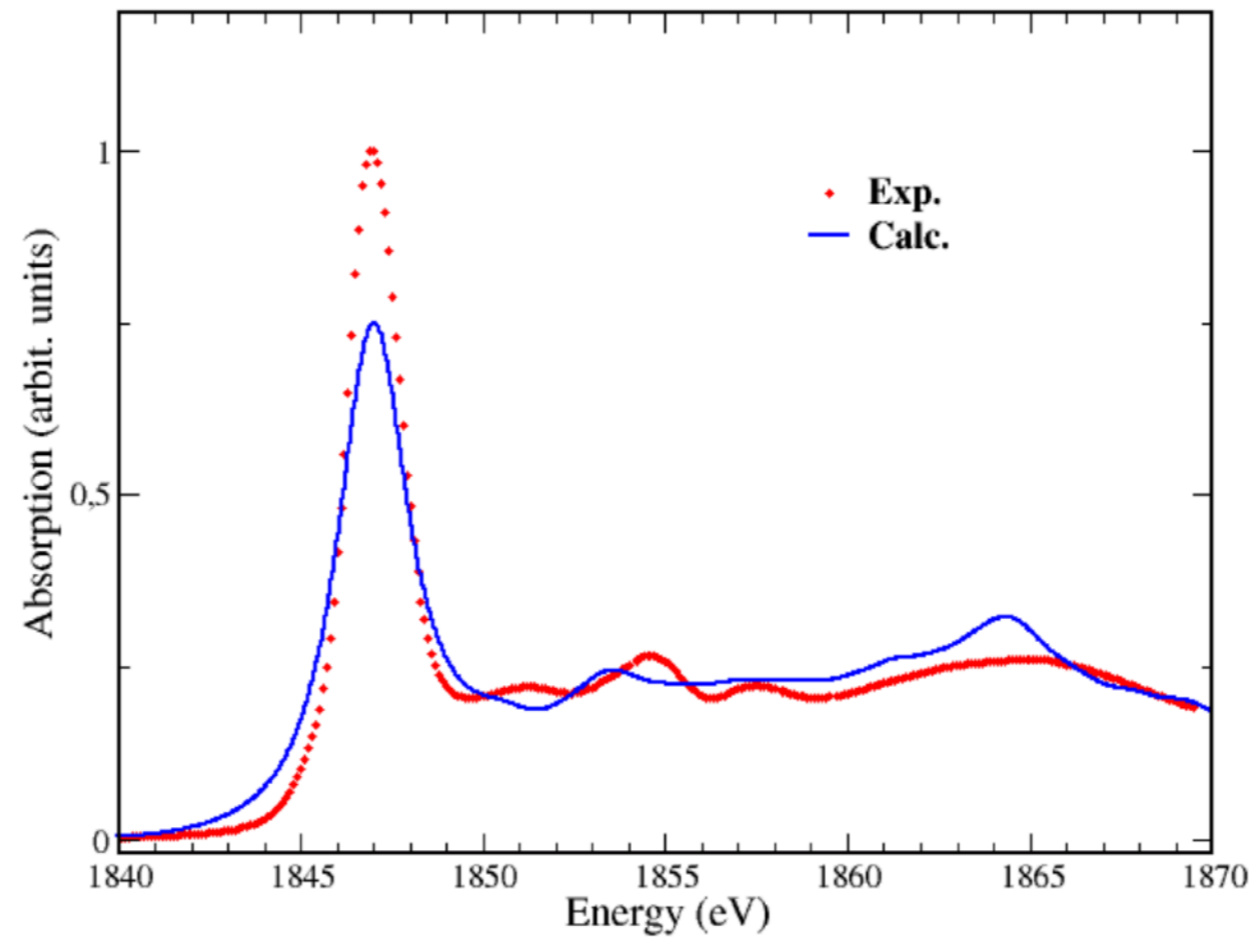
Example α -quartz

- Ignoring the core hole is not a good idea



Example α -quartz

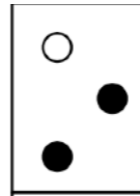
- Adding a core hole does better but there are still problems



Example α -quartz

- Can we do better?

(1×1) cell



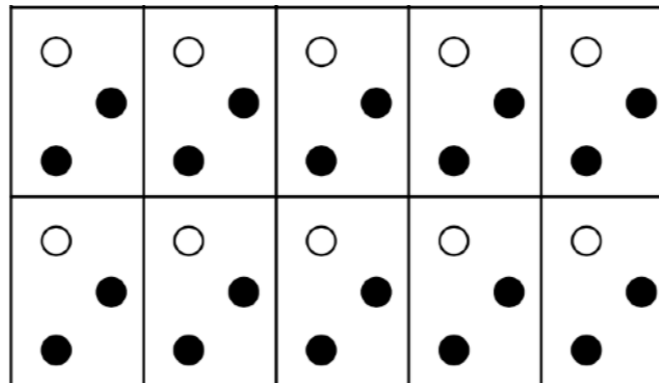
○ **core hole**

● **no core hole**

Example α -quartz

- The core hole sees is periodic image

(1×1) cell with PBC

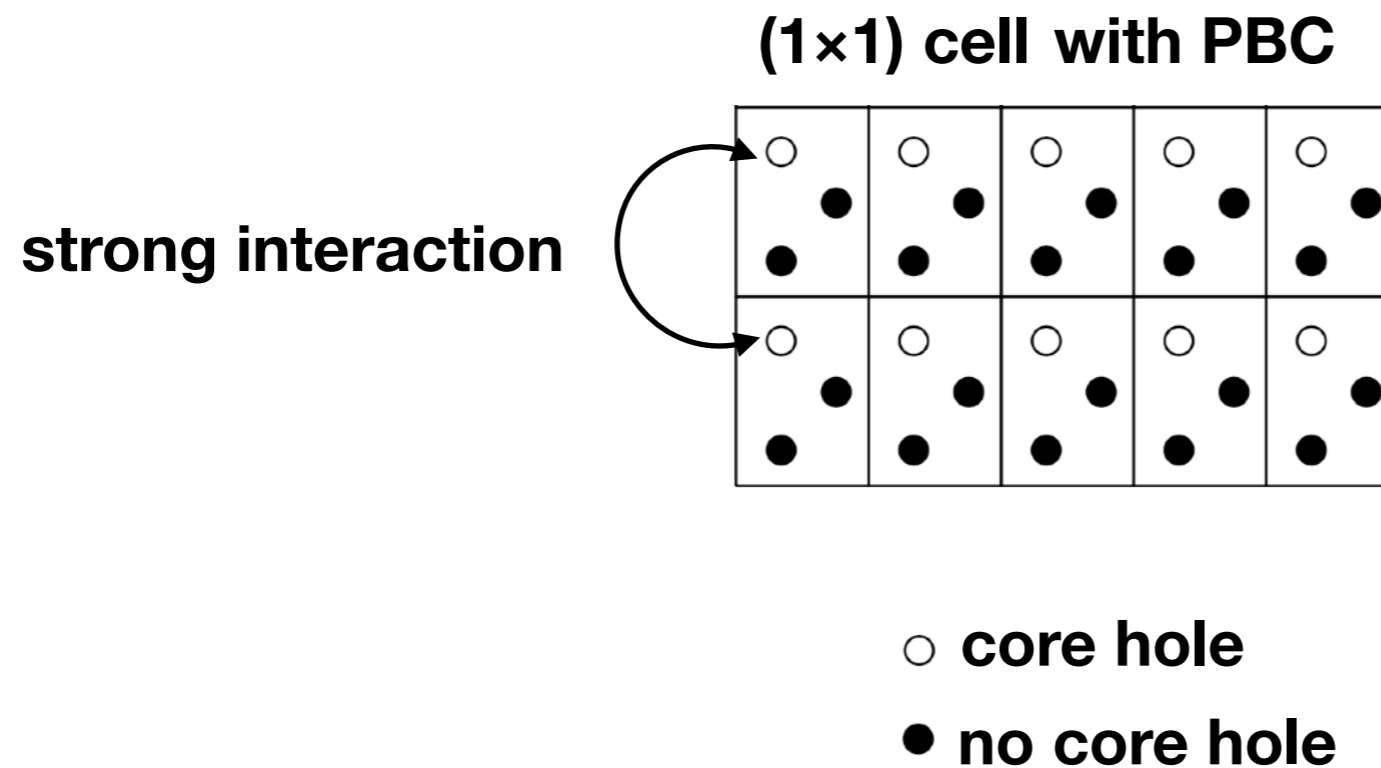


○ **core hole**

● **no core hole**

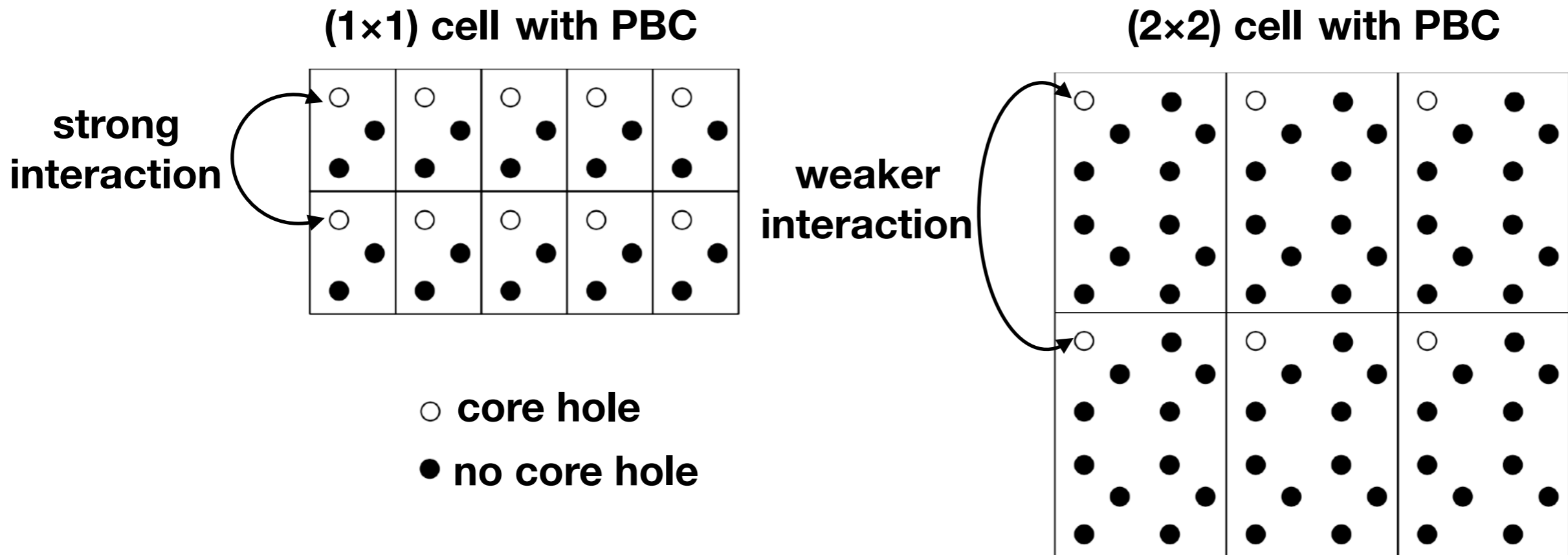
Example α -quartz

- The core hole sees is periodic image



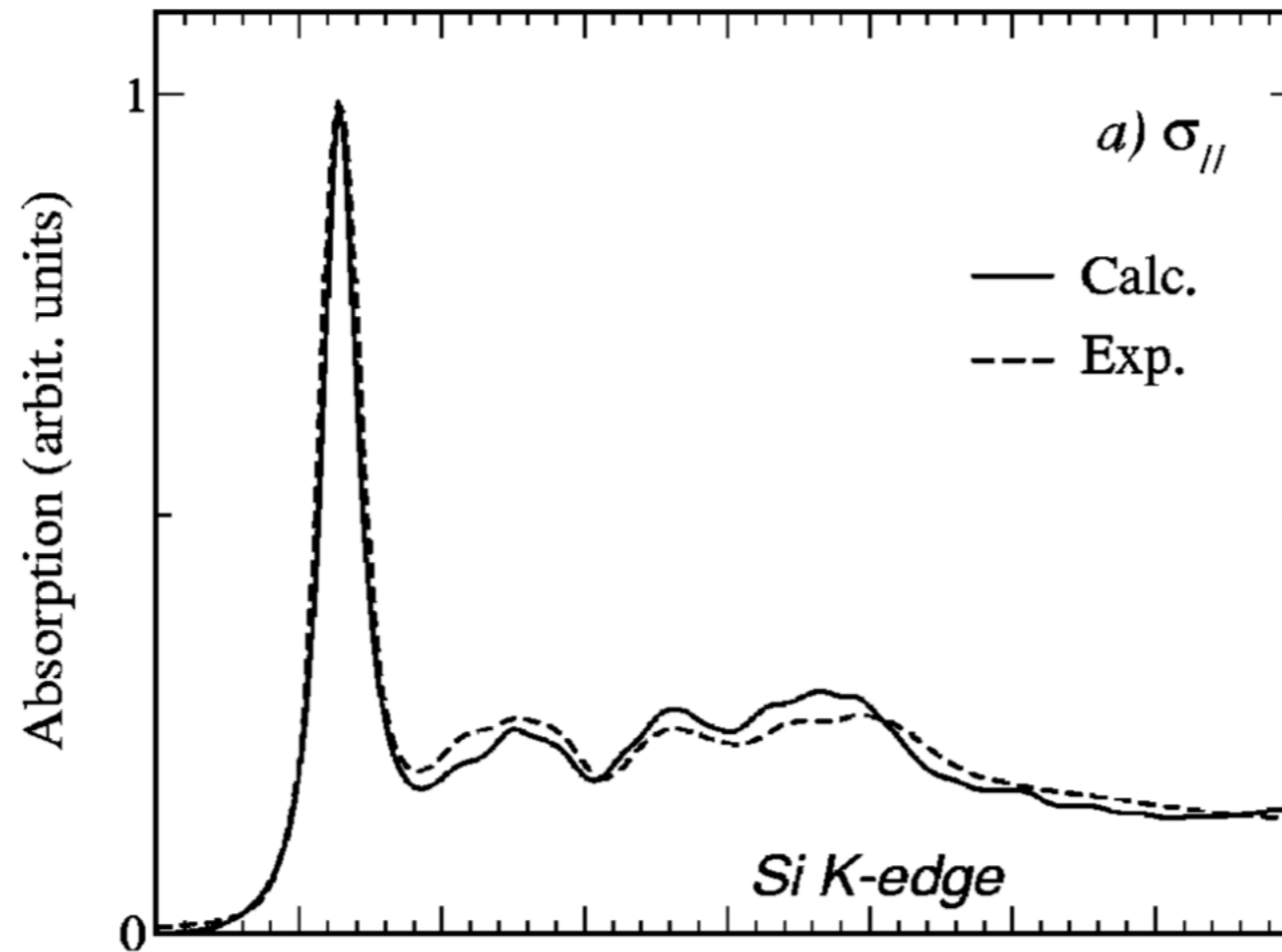
Example α -quartz

- The core hole sees is periodic image
- Double unit cell size to reduce spurious interaction



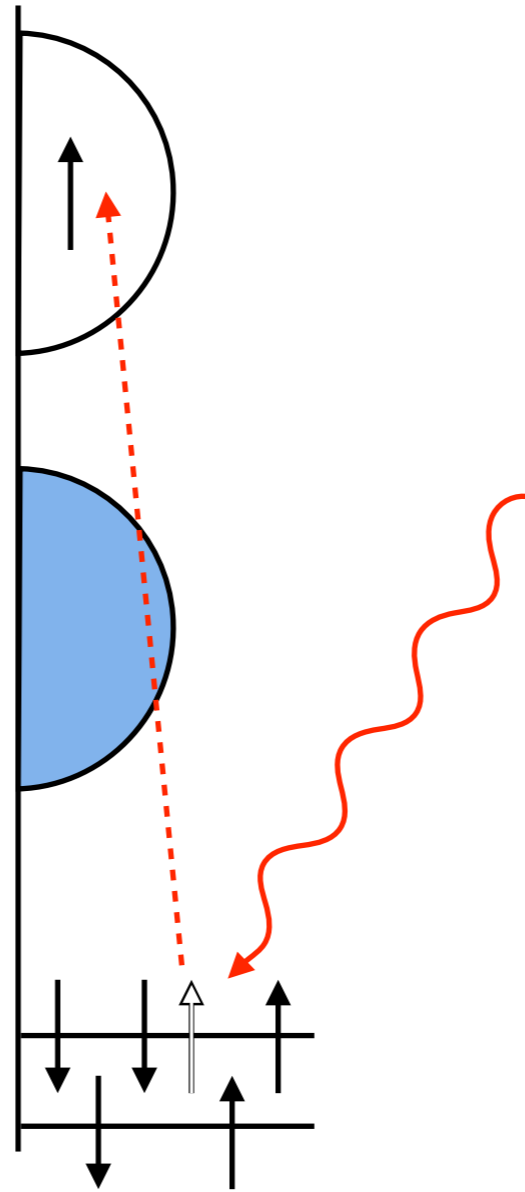
Example α -quartz

- With a (2x2x2) cell the calculation is converged



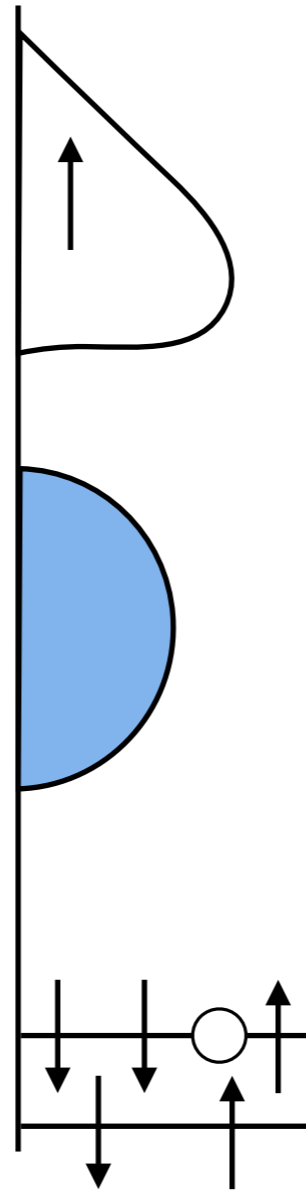
What happened?

- Core-valence interaction pulls states down



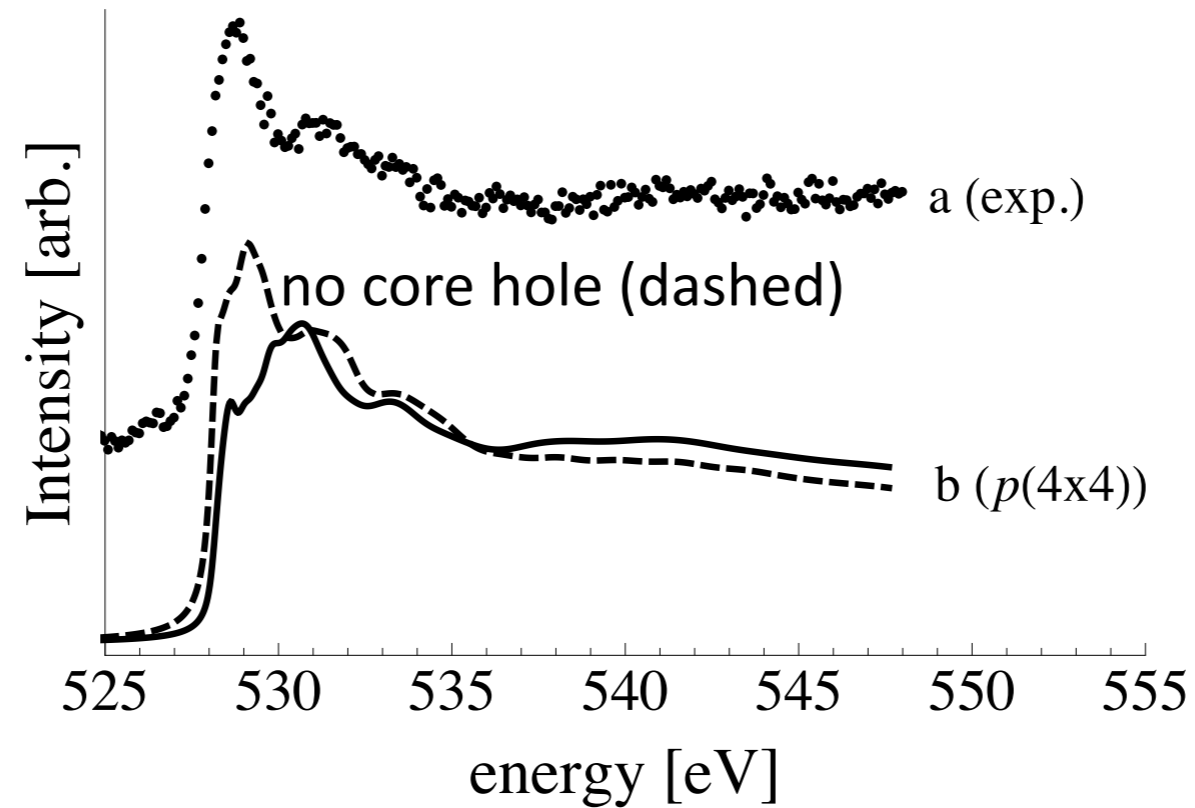
What happened?

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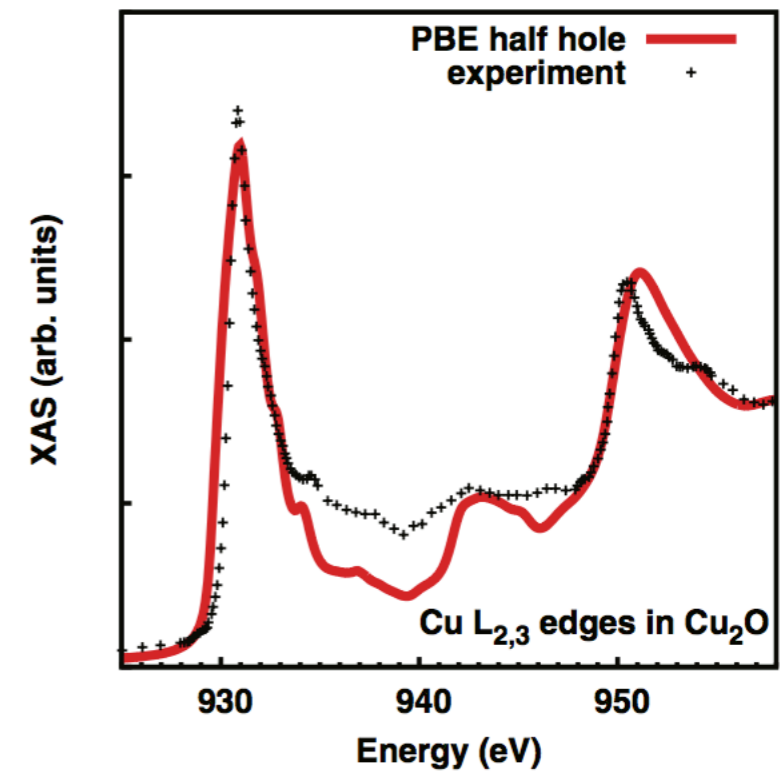
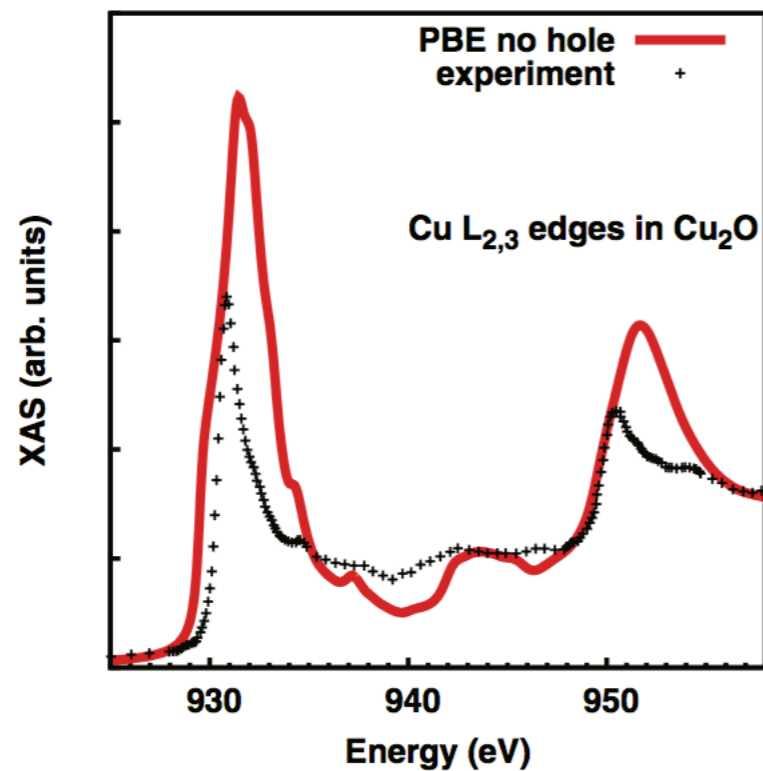
Sometimes no hole

- The core valence interaction can appear to be completely screened



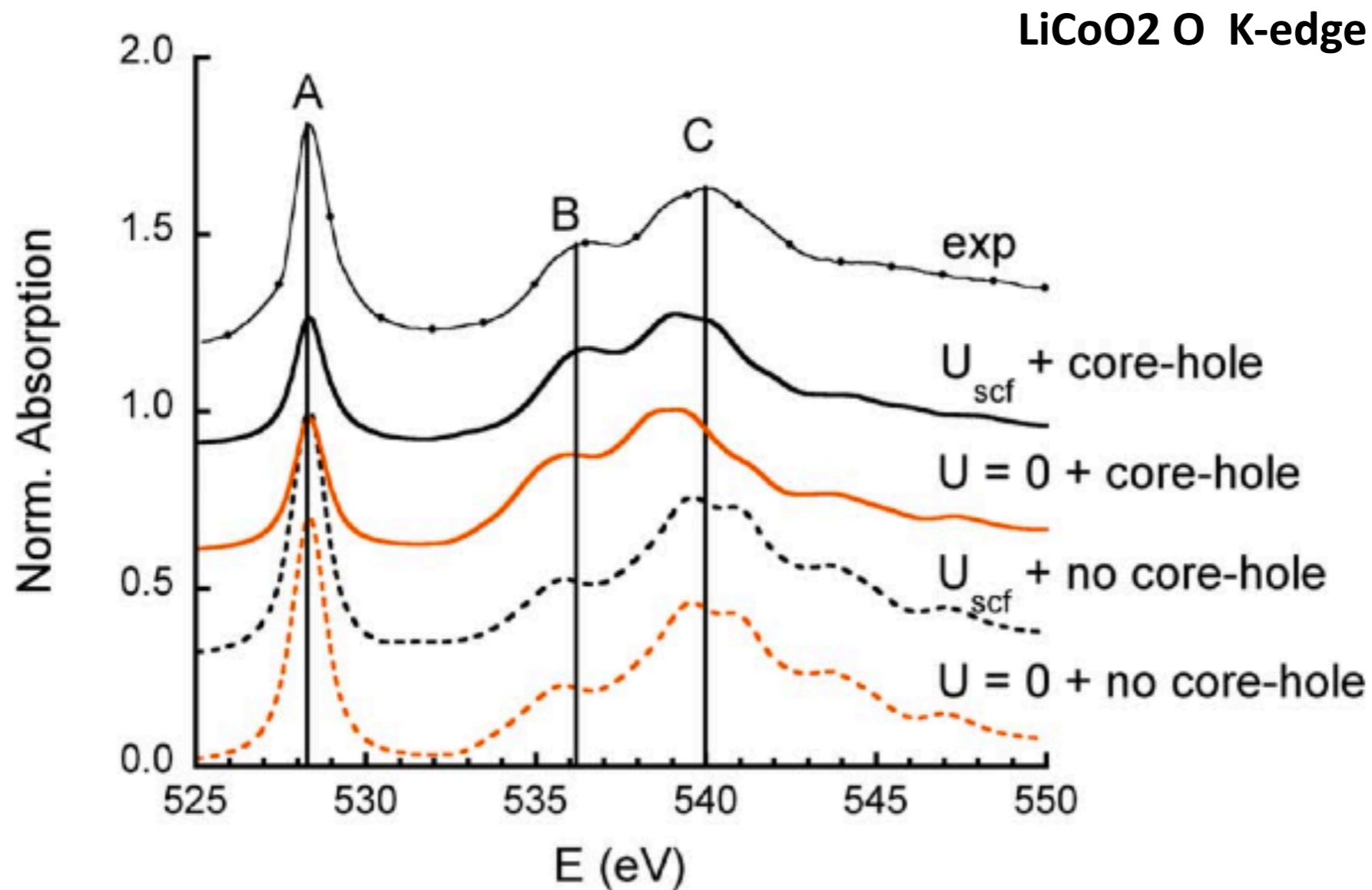
Sometimes half a hole

- The core valence interaction can appear to be partially screened



Sometimes it does not matter

- Or the core hole potential may not do much



The good and bad about core holes

- The core hole problem comes from describing two particle system with independent particle approximation (IPA)
 - Good news is IPA works well for K edges
 - Bad news is we don't know how much core hole potential to include *a priori*
- If we want a better answer we need to turn to a two-particle theory
 - Good news is we have one in the Bethe-Salpeter equation (BSE)
 - Bad news is its computationally expensive

BSE

- Hamiltonian similar to before

$$H_{\text{BSE}} = H_e - H_h - V_D + V_X$$

- Single particle electron and hole Hamiltonians

$$H_e - H_h$$

- Screened Coulomb interaction

$$V_D$$

- Non-local exchange

$$V_X$$

BSE

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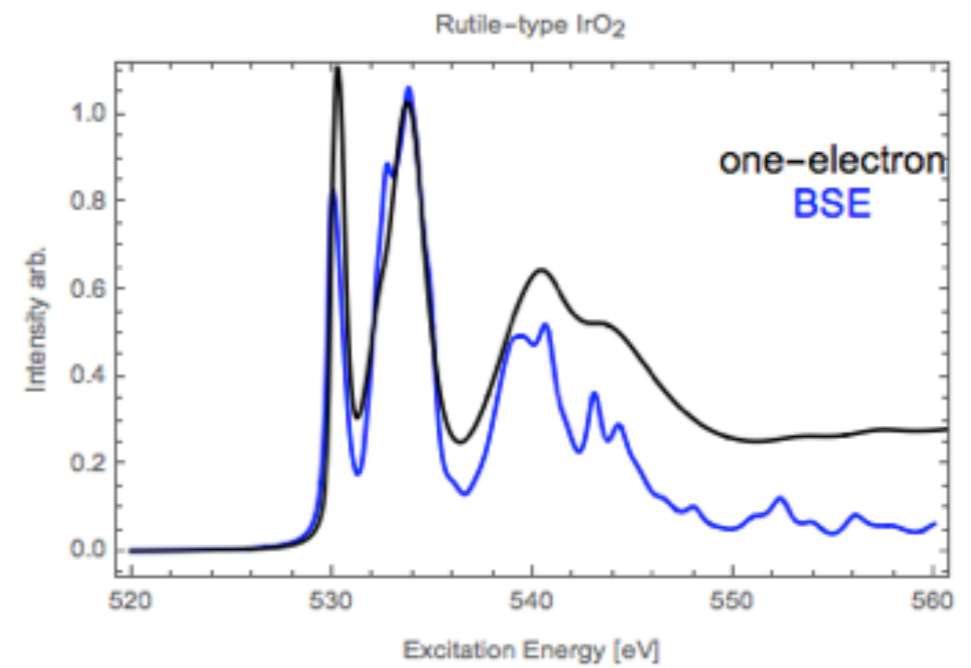
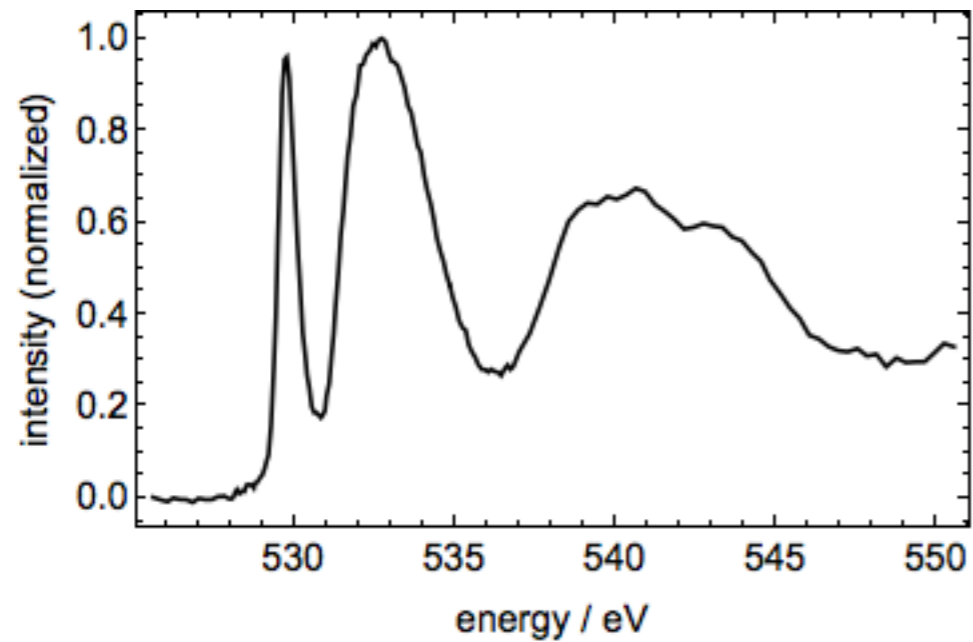
$$V_D$$

- **Non-local exchange**

$$V_X$$

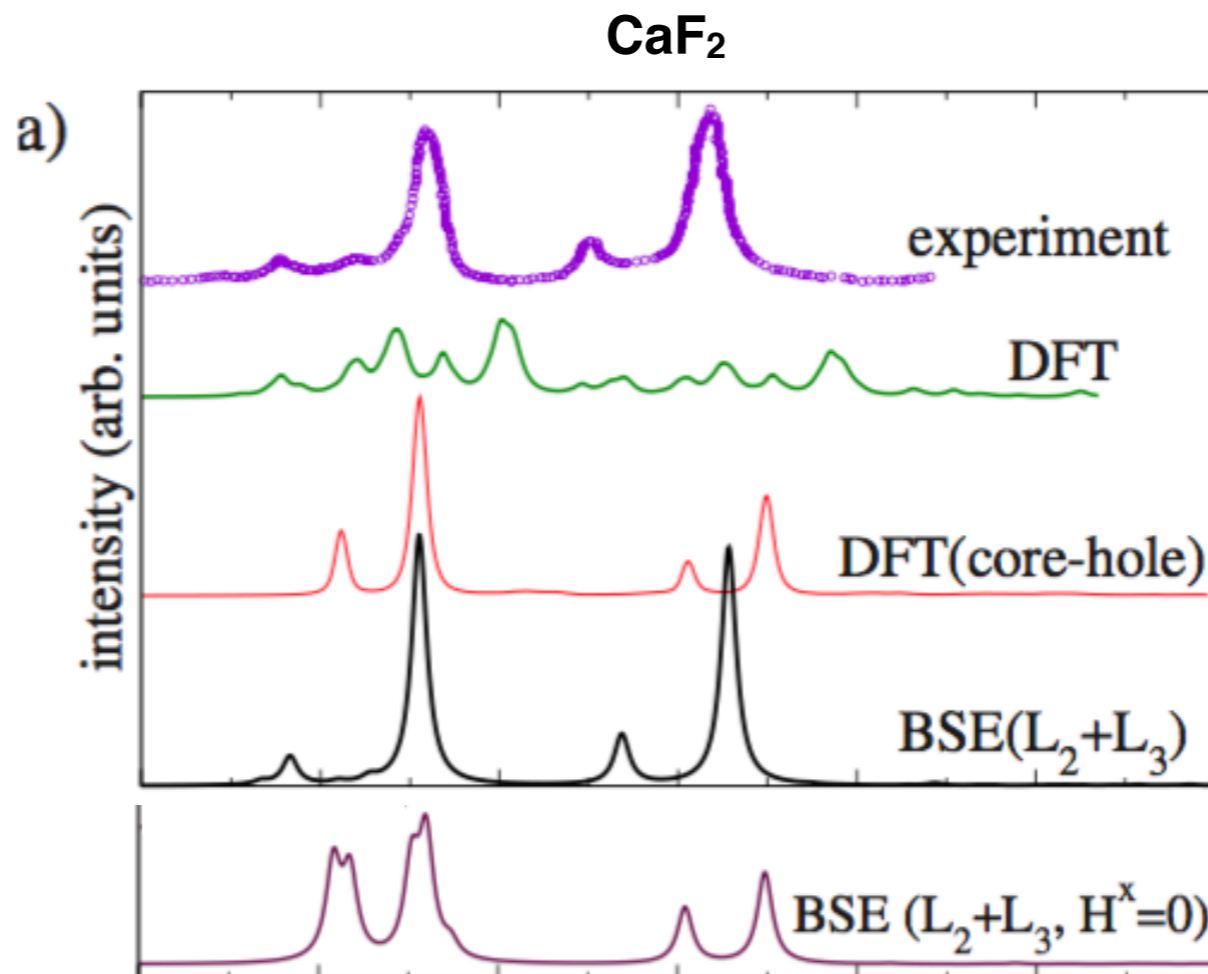
BSE

- BSE and IPA are formally similar
- One difference is screened Coulomb interaction



BSE

- A bigger difference is the non-local exchange
- It mixes L_{II} and L_{III} edges

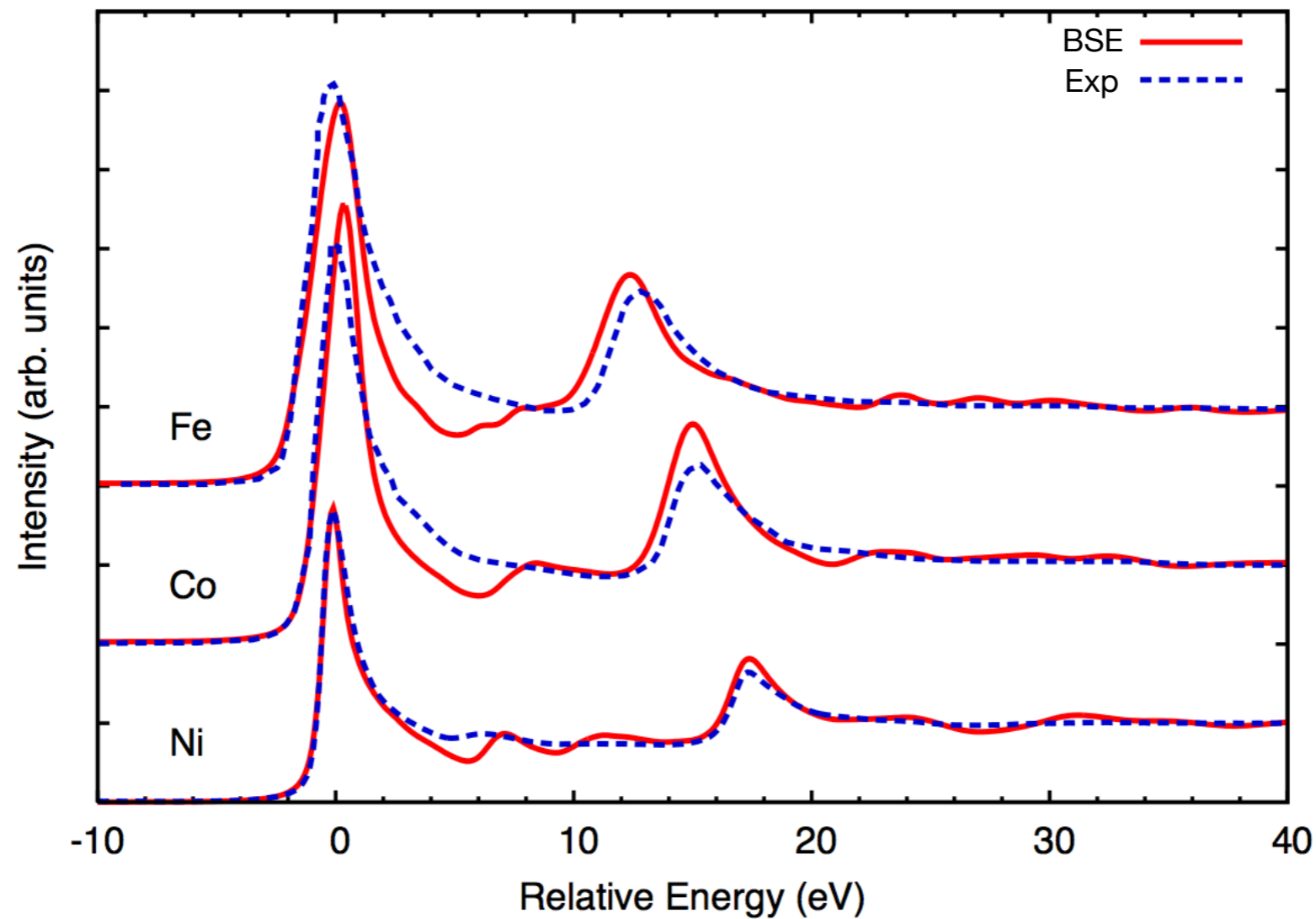


no core hole

no non-local exchange

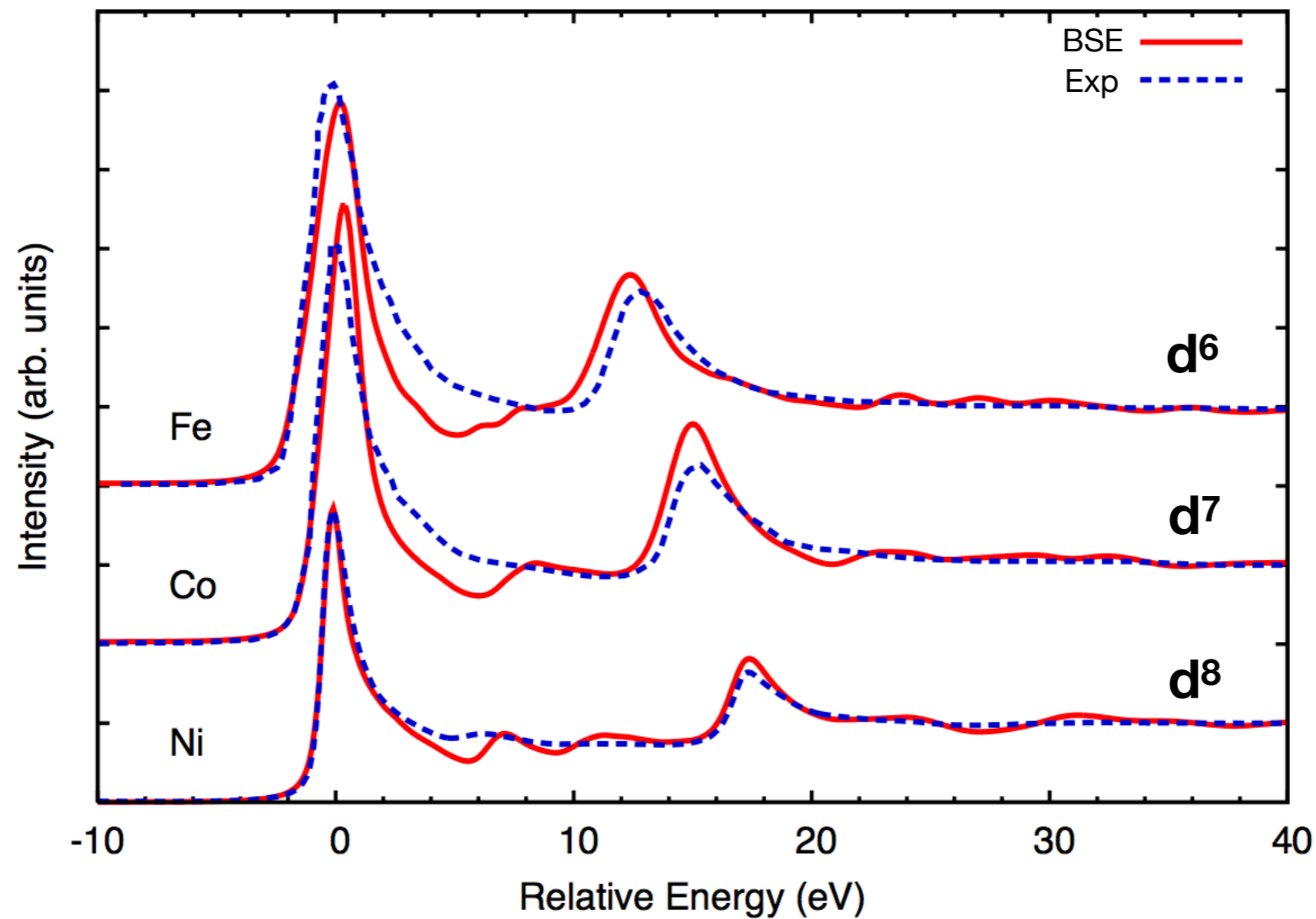
Computing $L_{II,III}$ edges

- BSE can capture non-statistical L_{III}/L_{II} branching



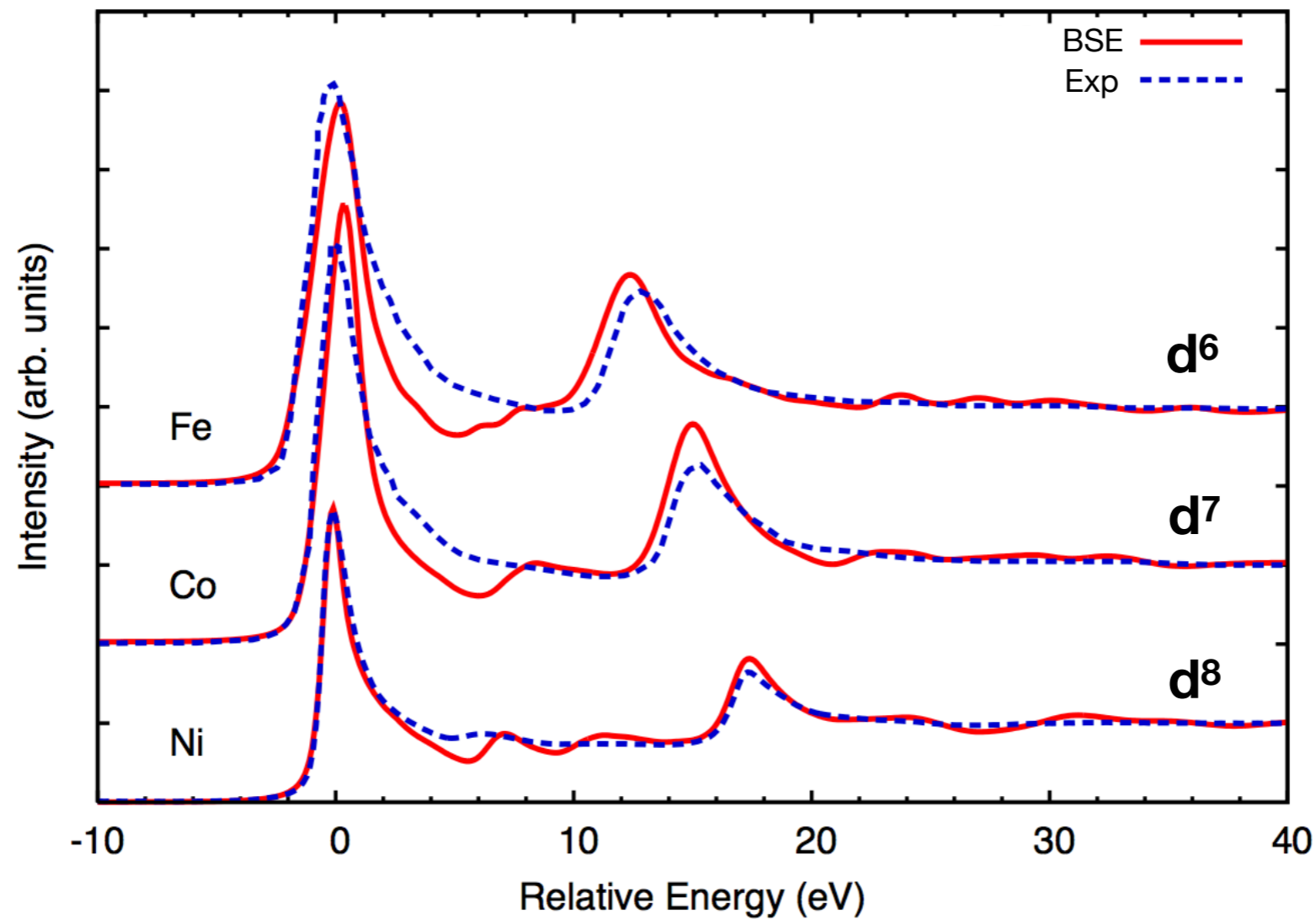
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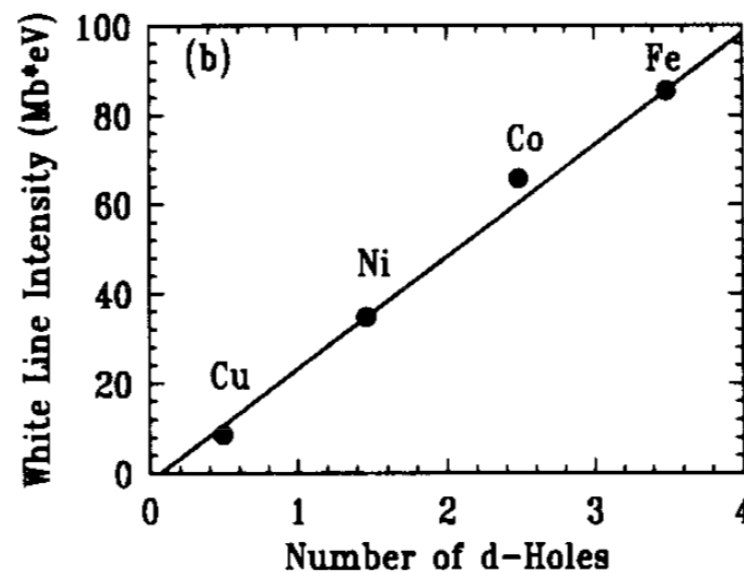
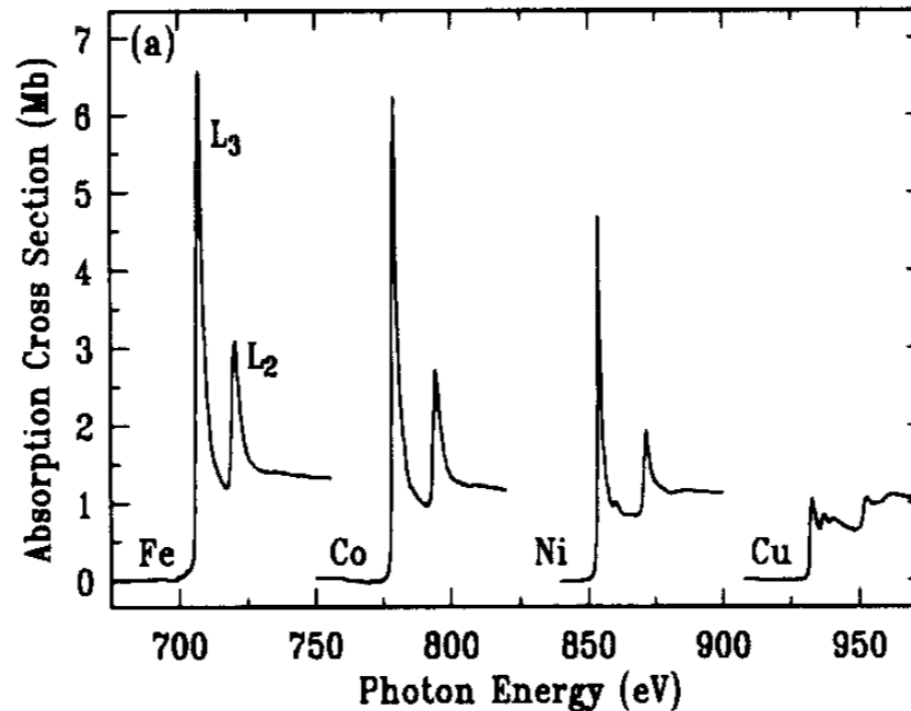
Computing $L_{II,III}$ edges

- Non-local exchange means L_{III} alone does not give d hole count



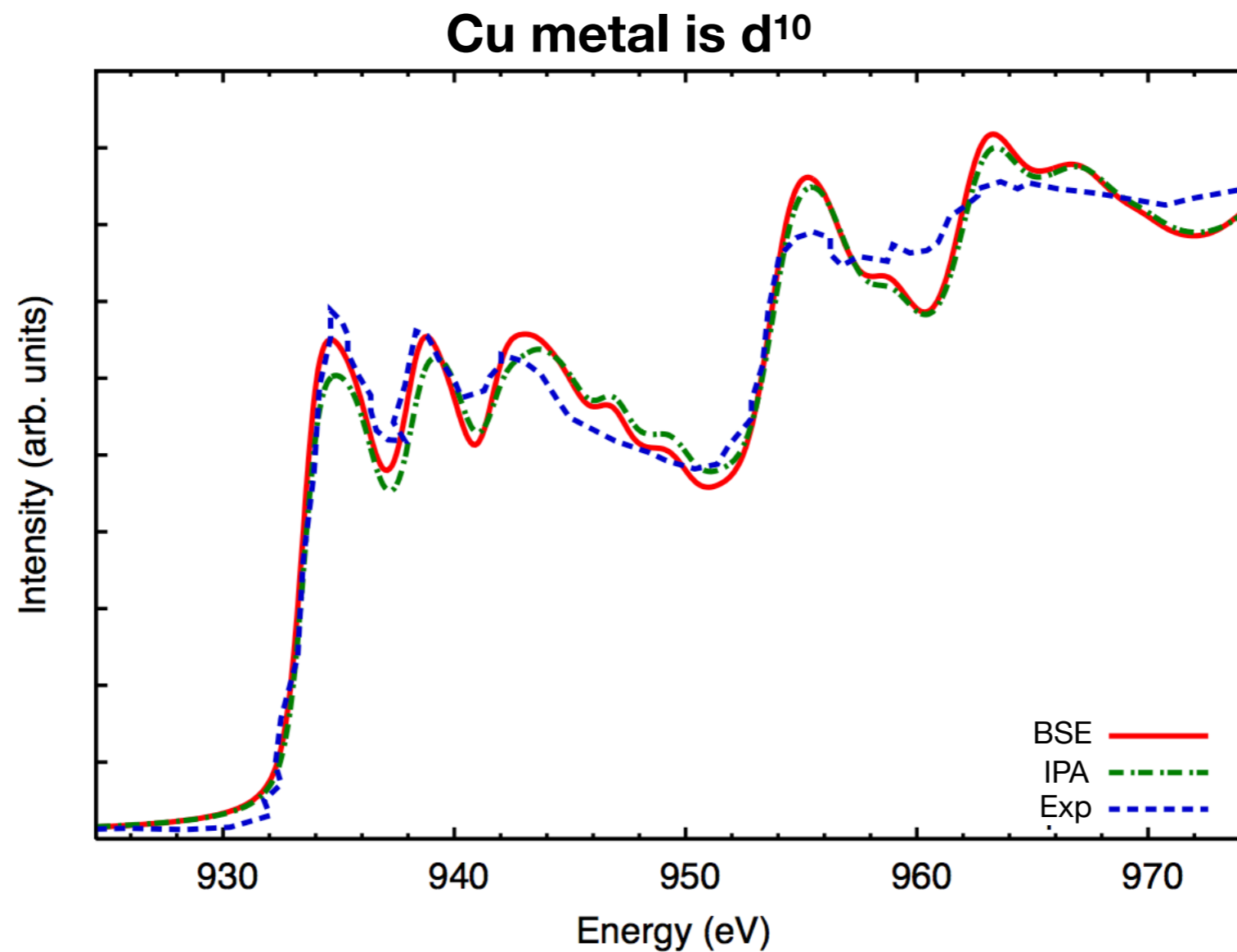
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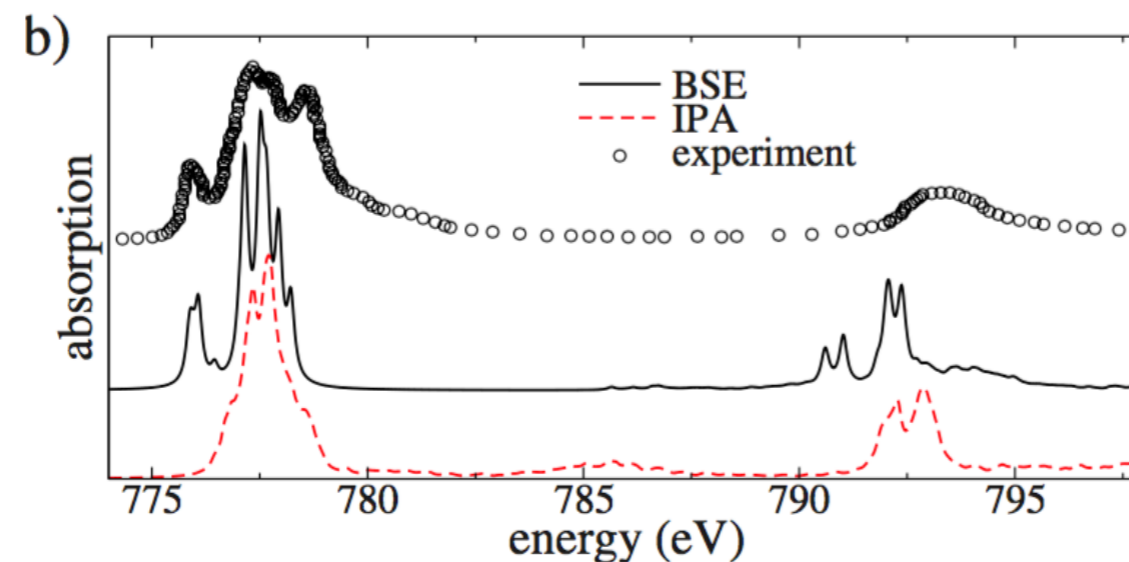
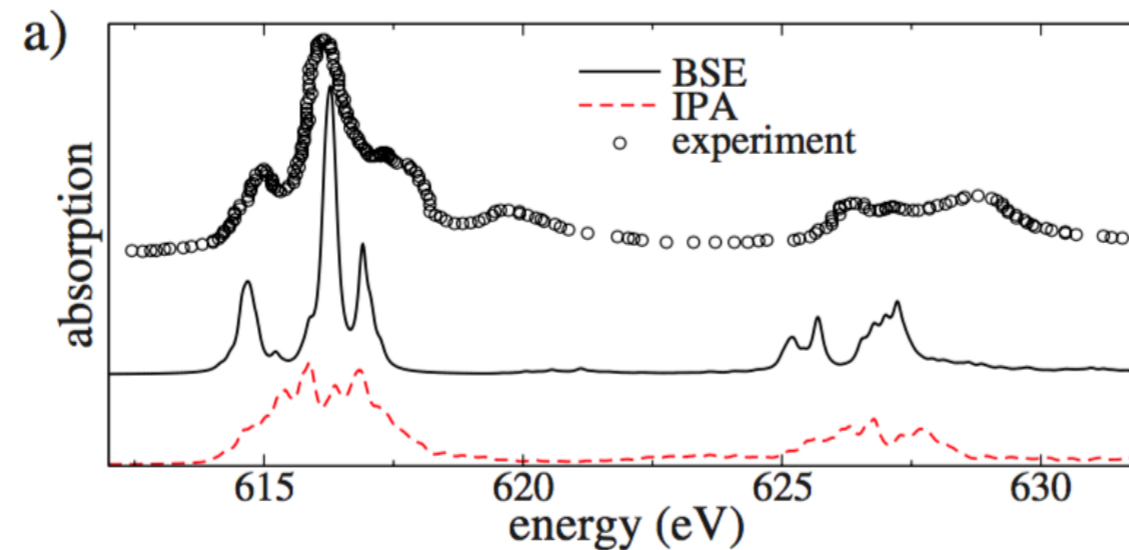
Computing $L_{II,III}$ edges

- But when there is no white-line IPA can work



When does BSE fail?

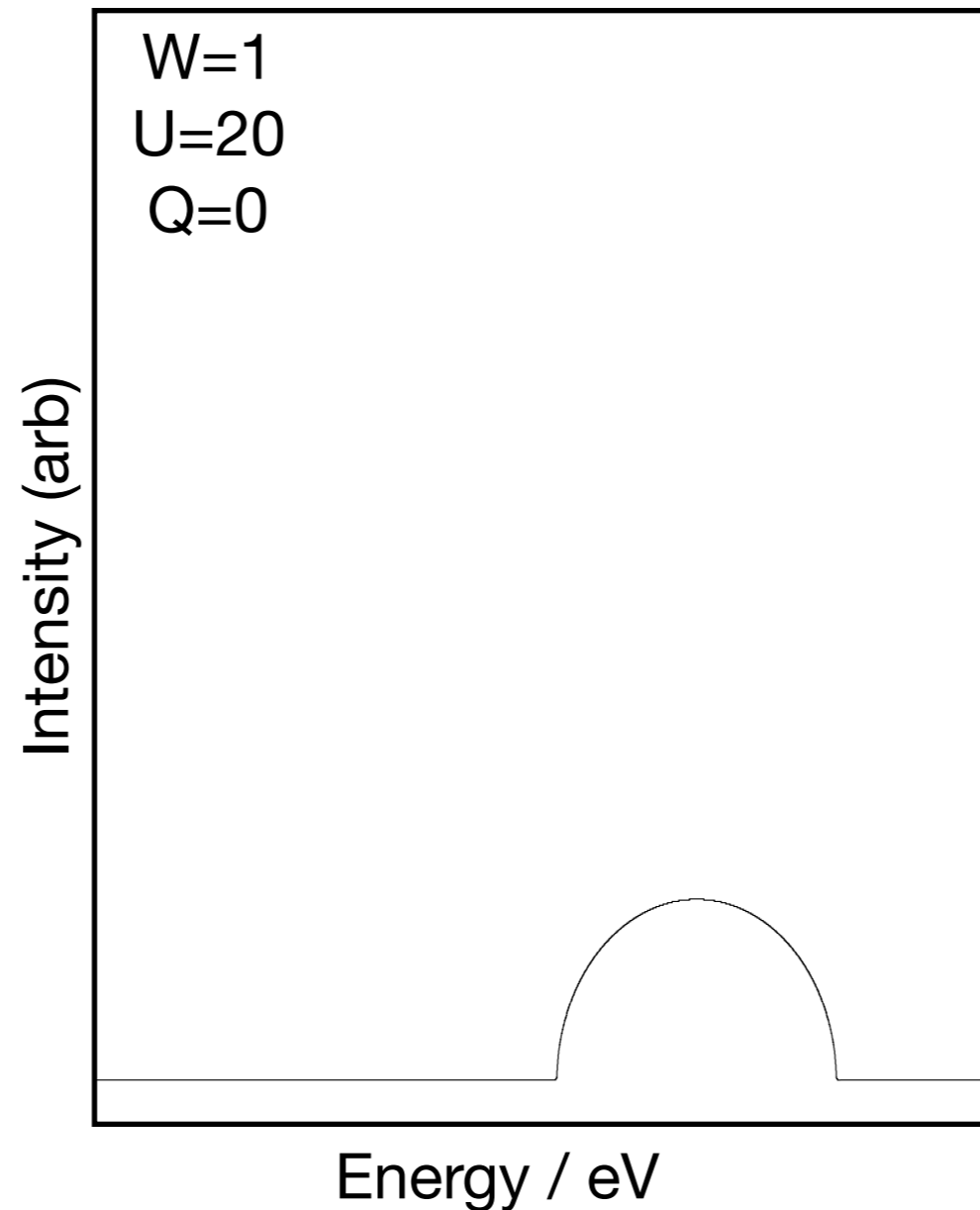
- Formally defined for one-hole and one-electron excitation
- If many particles are present it might not work—especially when correlated



Many-body problem

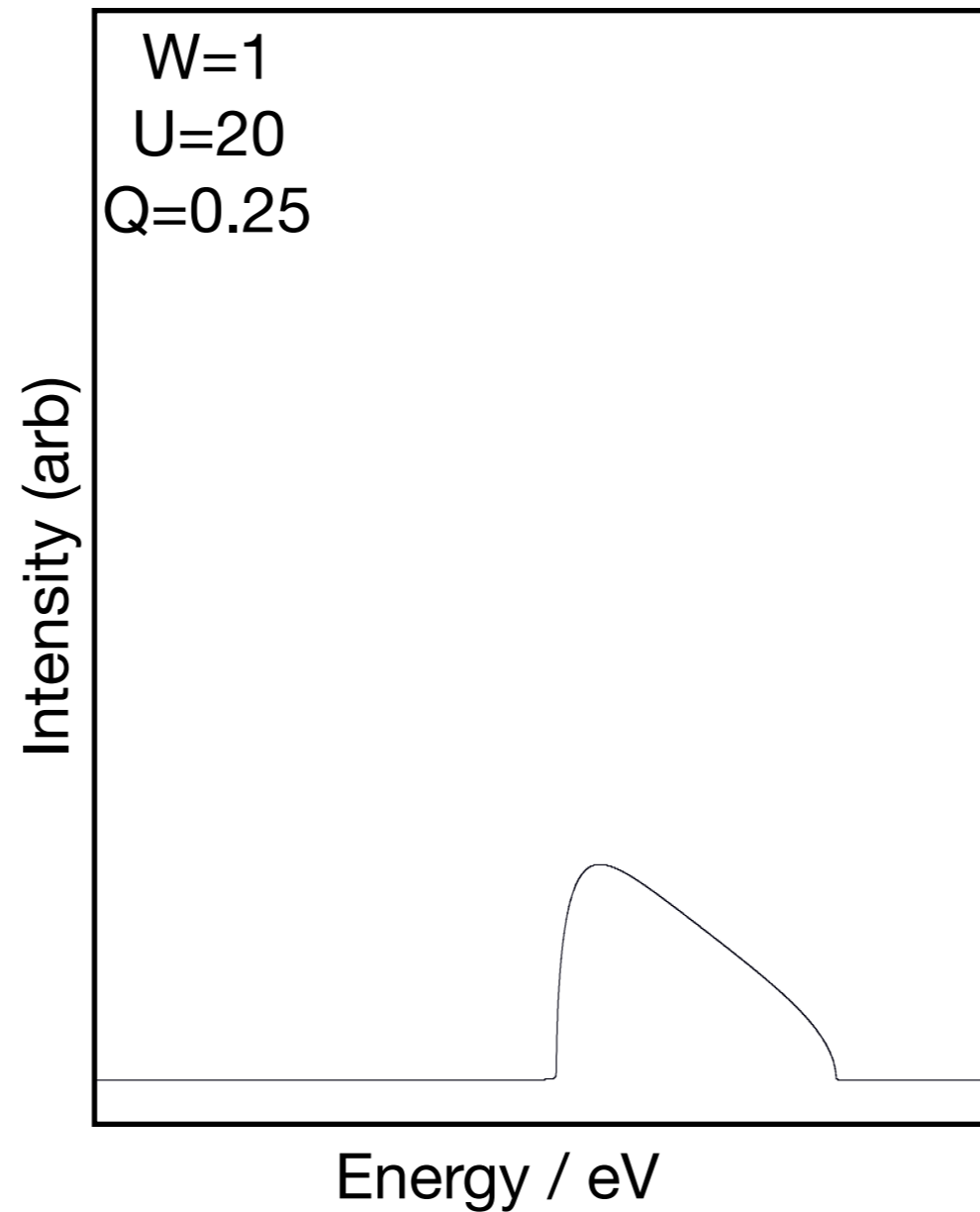
Core-hole potential

- What if we have a strong core hole potential (Q)?



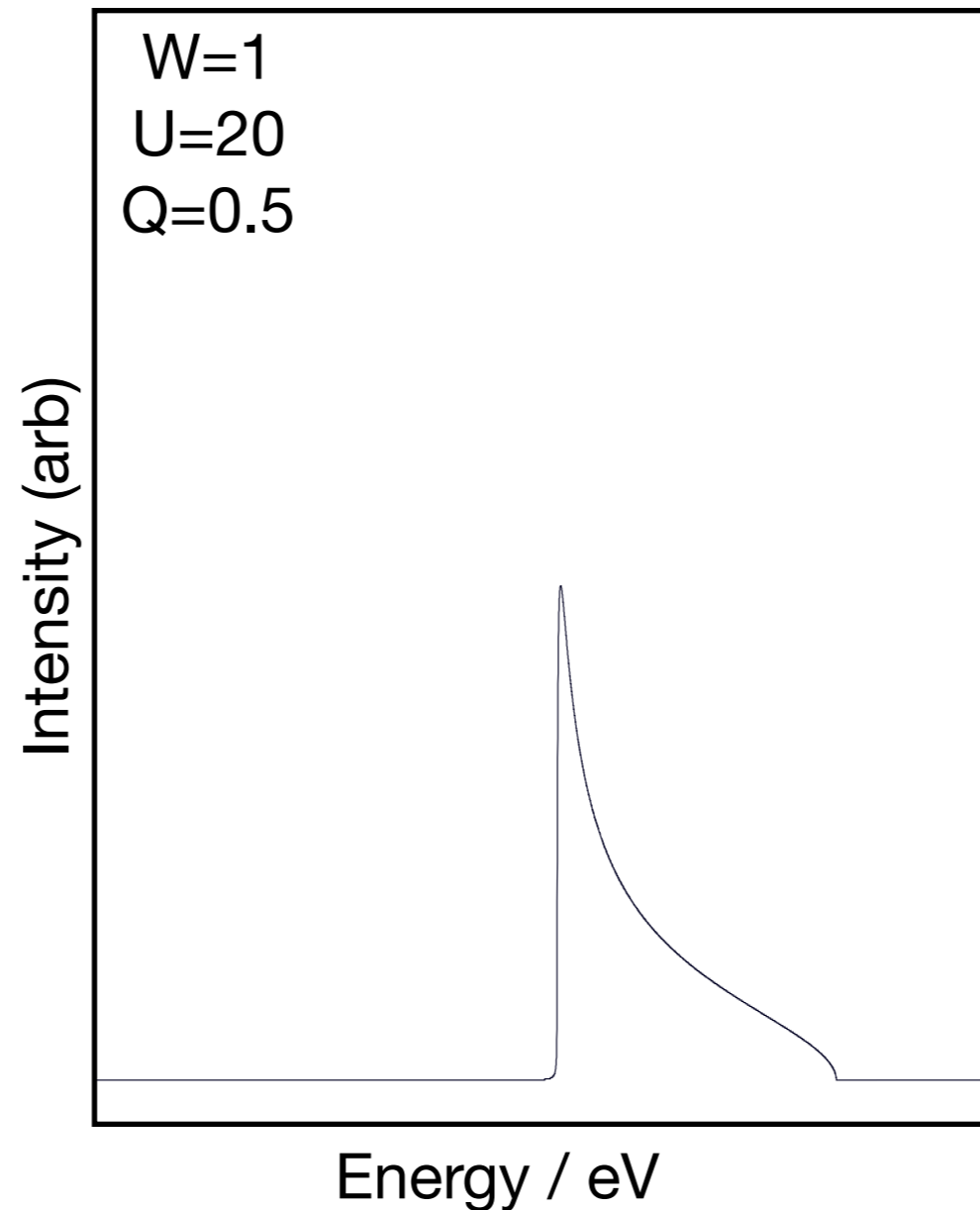
Core-hole potential

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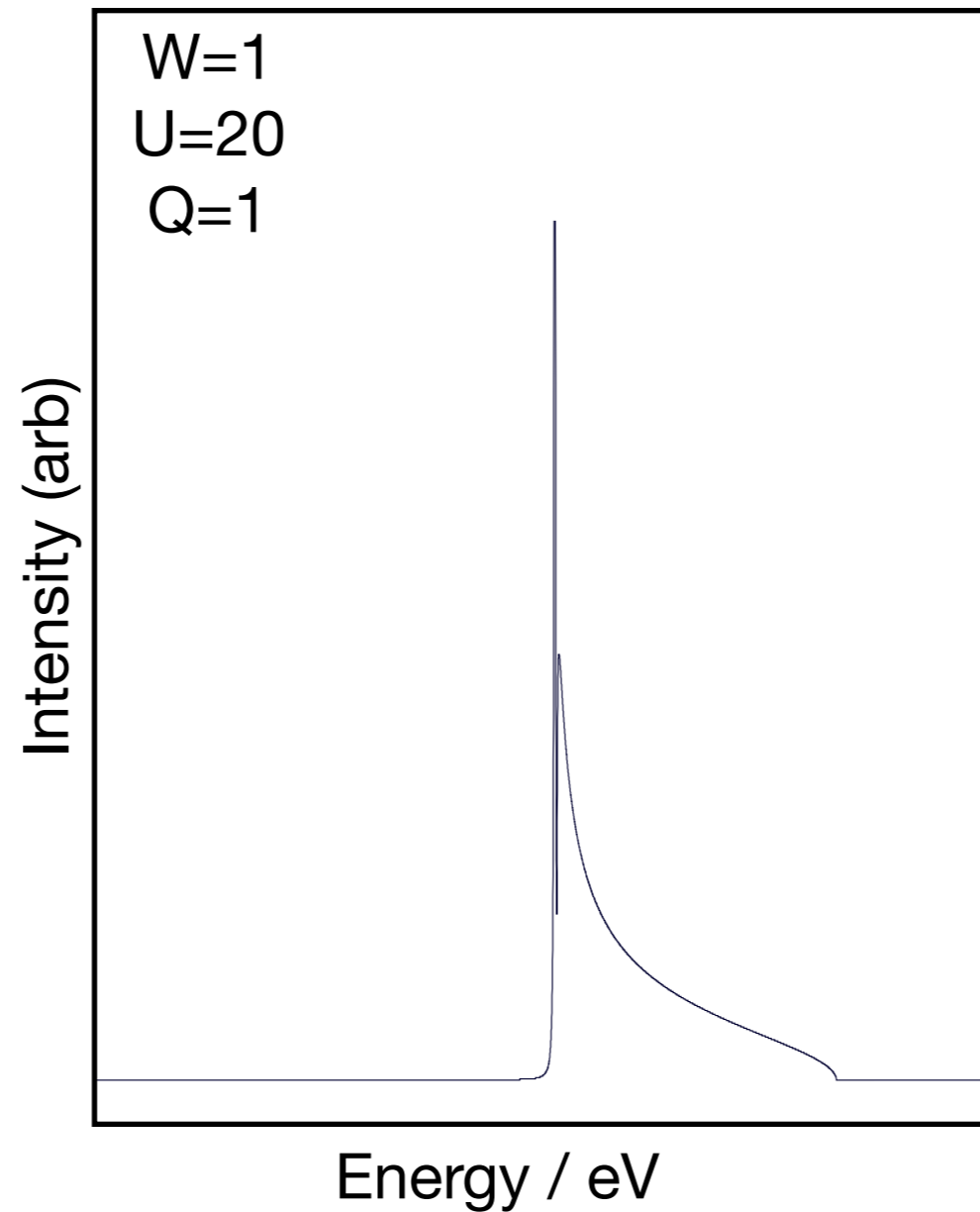
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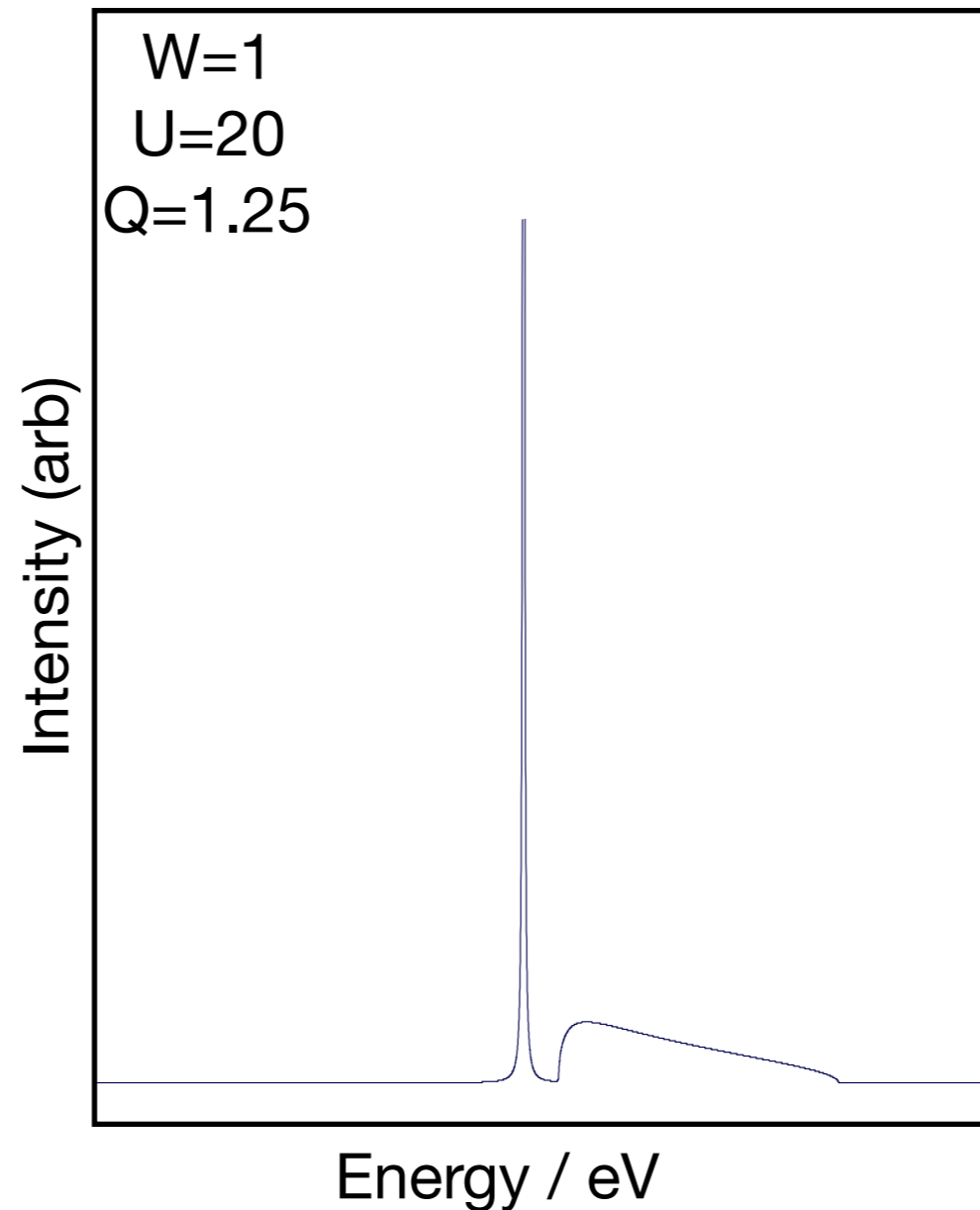
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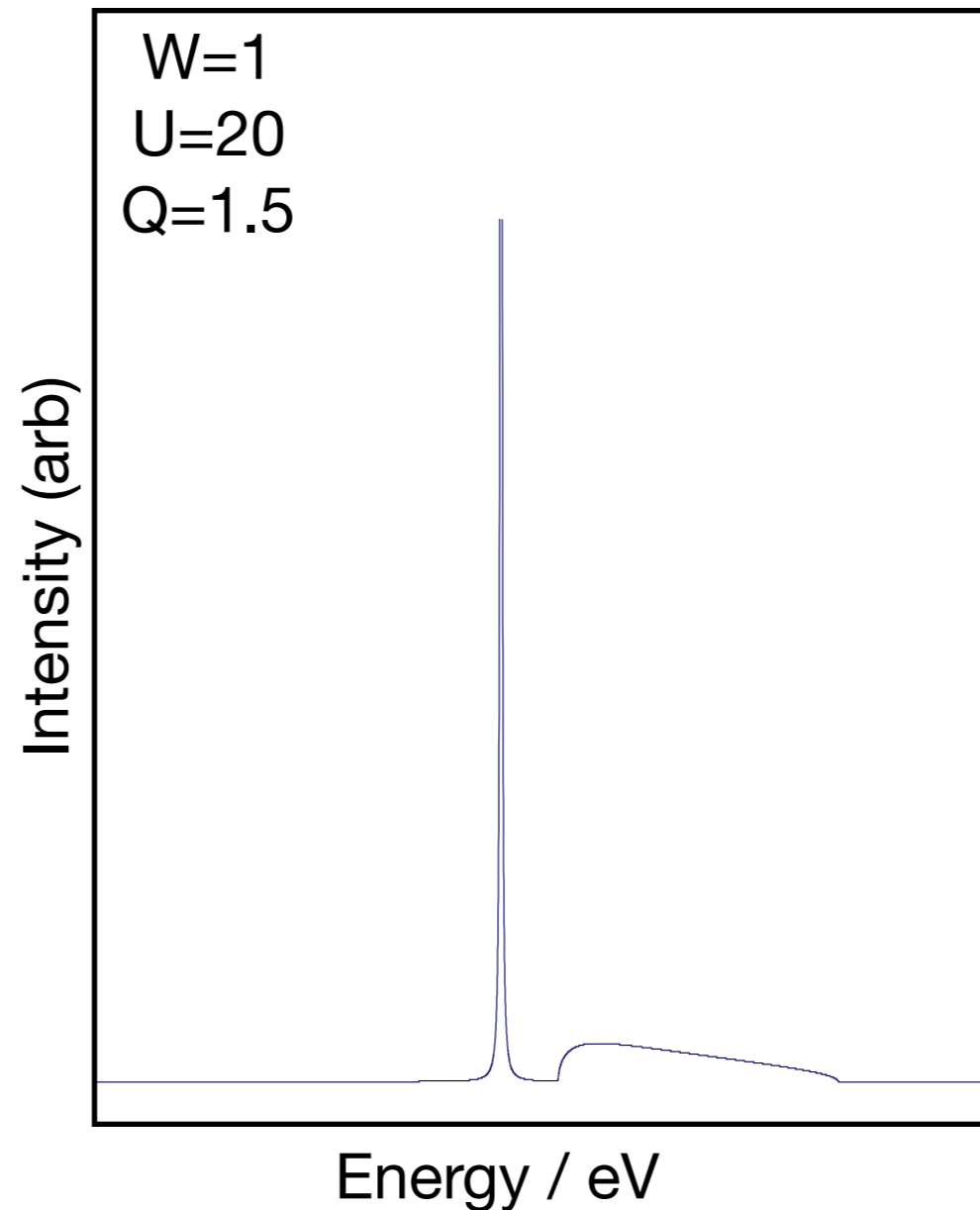
Core-hole potential

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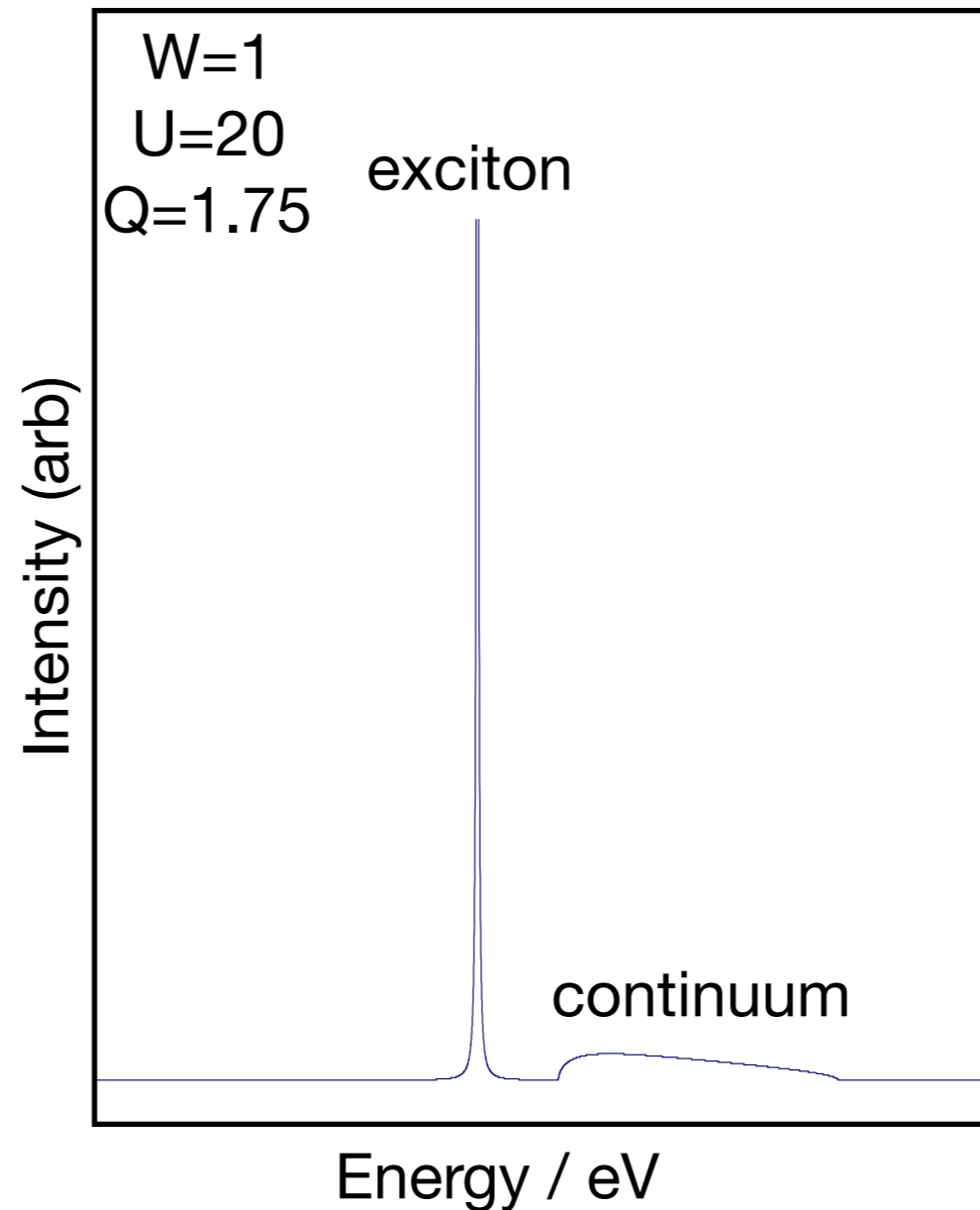
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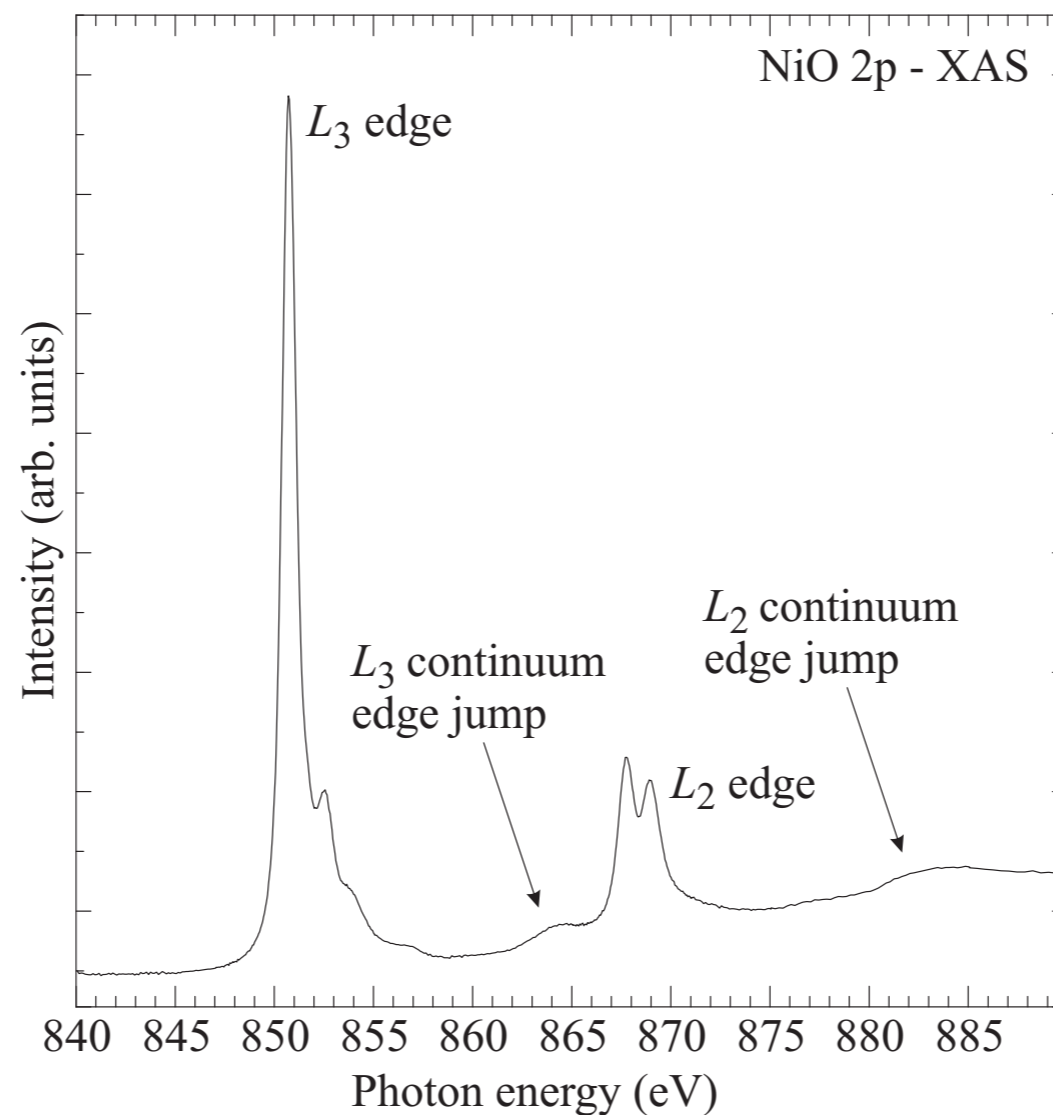
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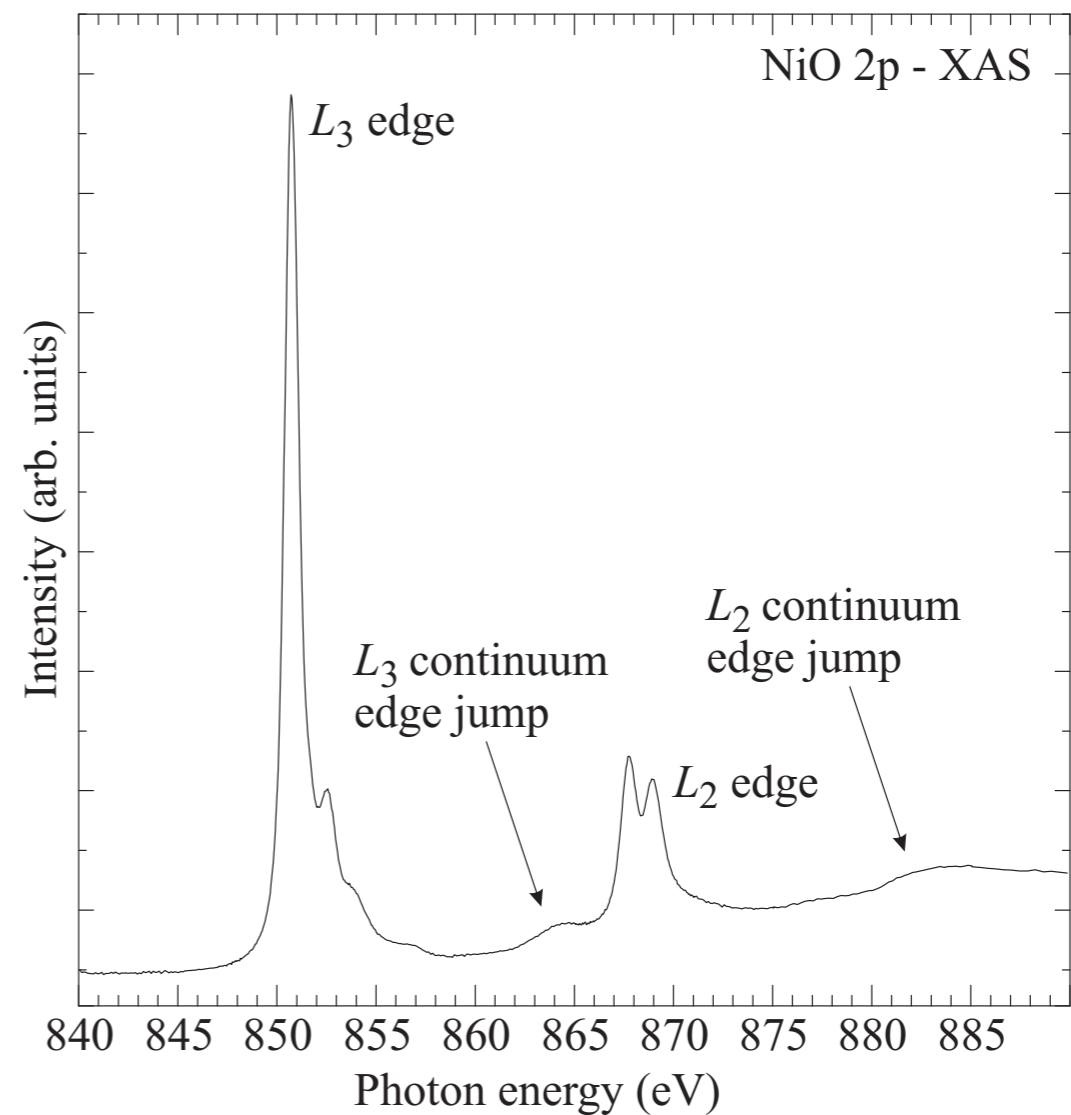
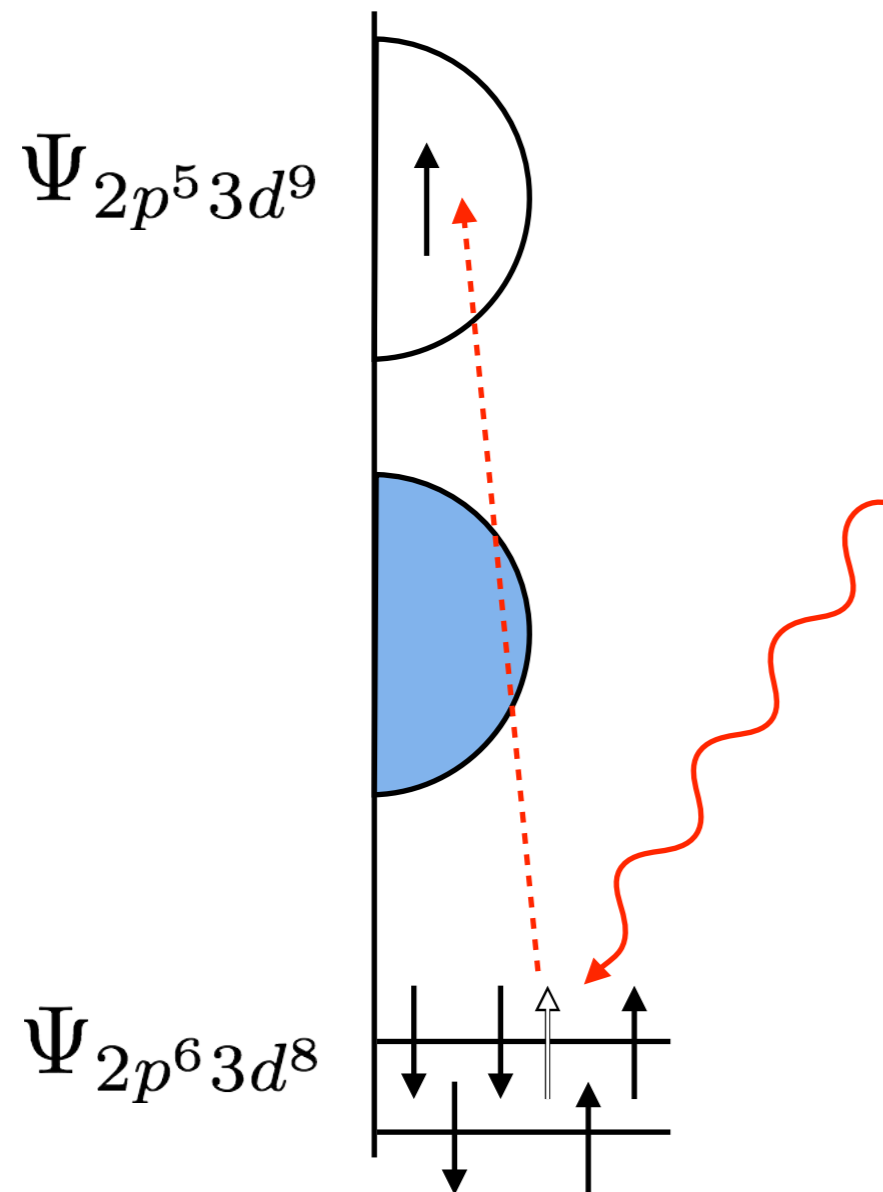
Strong core-valence interaction

- We no longer see a PDOS—exciton instead
- We can use a different approach to capture the physics



Configuration interaction

- Now we need to capture correlations



Configuration interaction

- Expand ground state wave function as

$$\psi_0 = \sum_i c_i |d^{n+i} \underline{L}^i\rangle$$

- Define energies for charge fluctuations

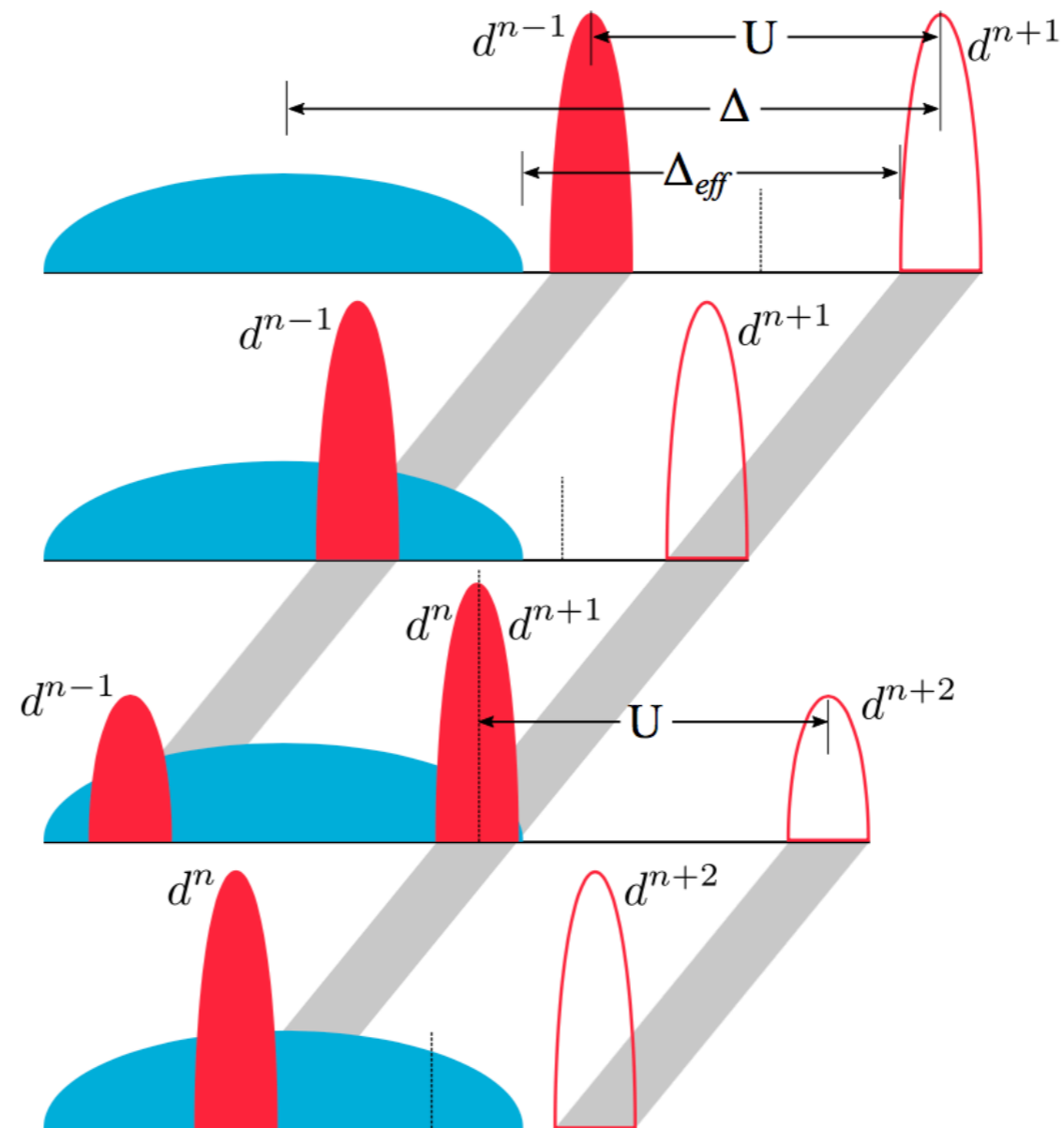
$$U \equiv \left[E(d^{n+1}) + E(d^{n-1}) \right] - 2E(d^n)$$

$$\Delta \equiv E(d^{n+1} \underline{L}^1) - E(d^n)$$

- Now we need a basis

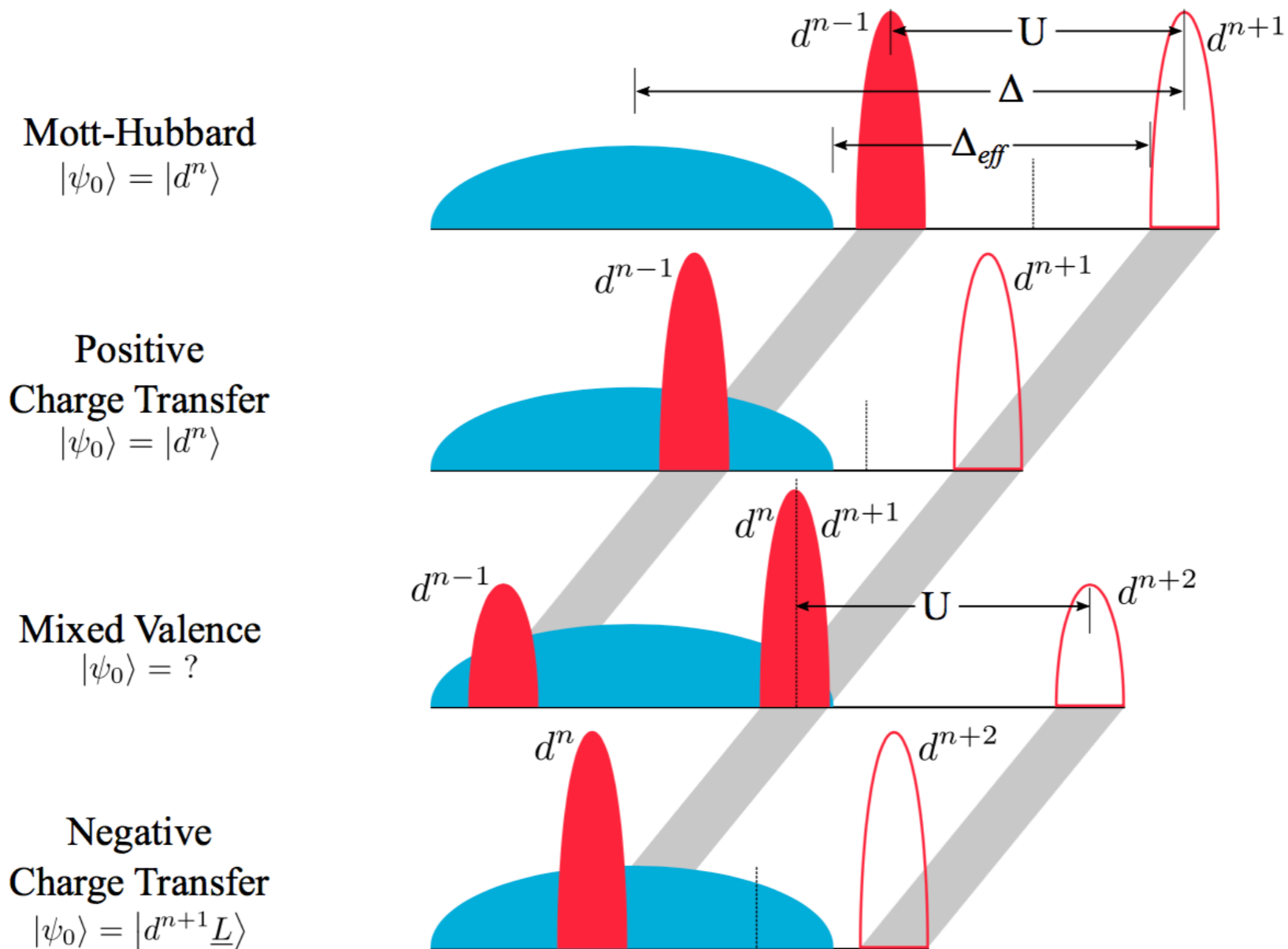
Parameters

- If view the parameters U and Δ graphically...



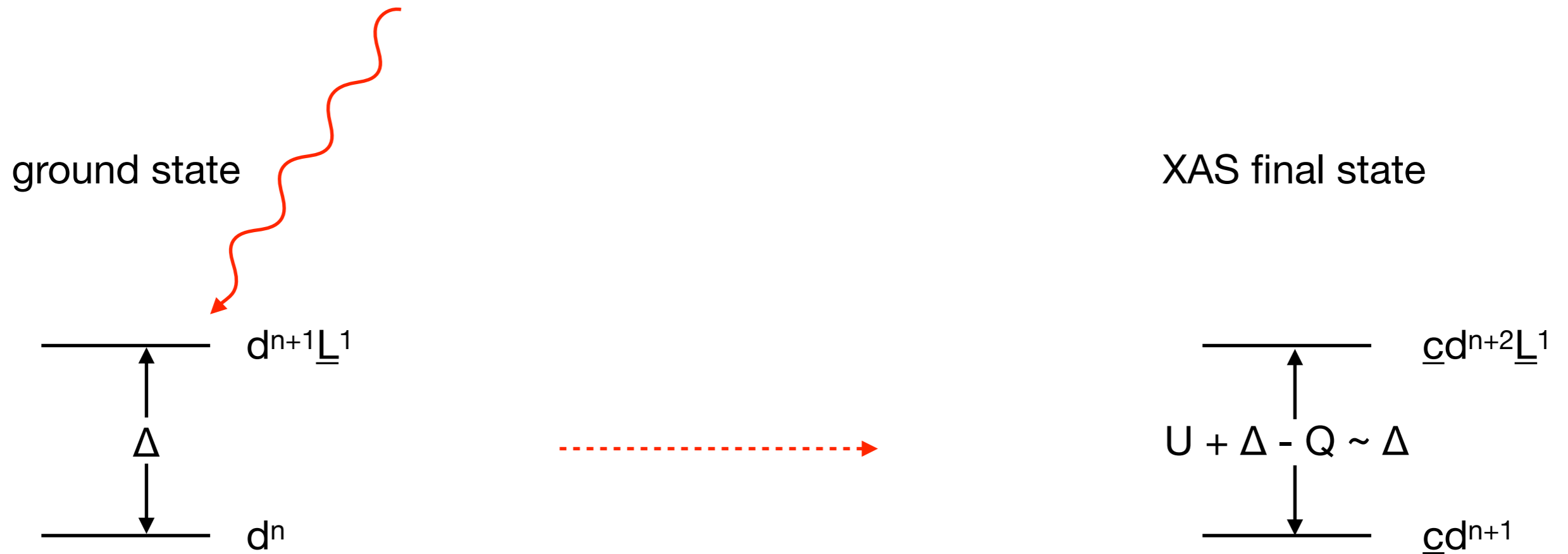
Parameters

- If view the parameters U and Δ graphically...
- They tell us something about the electronic structure



Parameter sensitivity

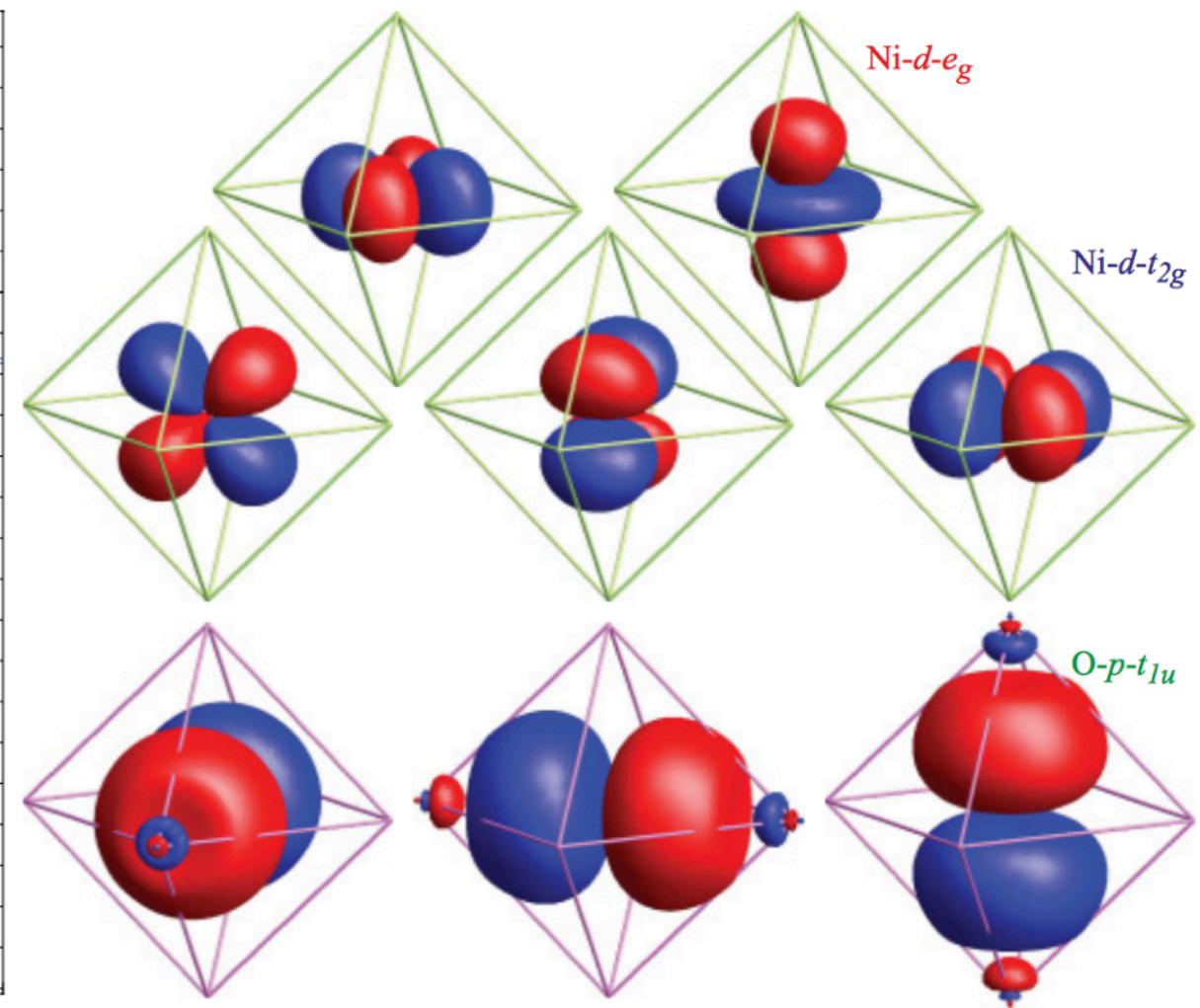
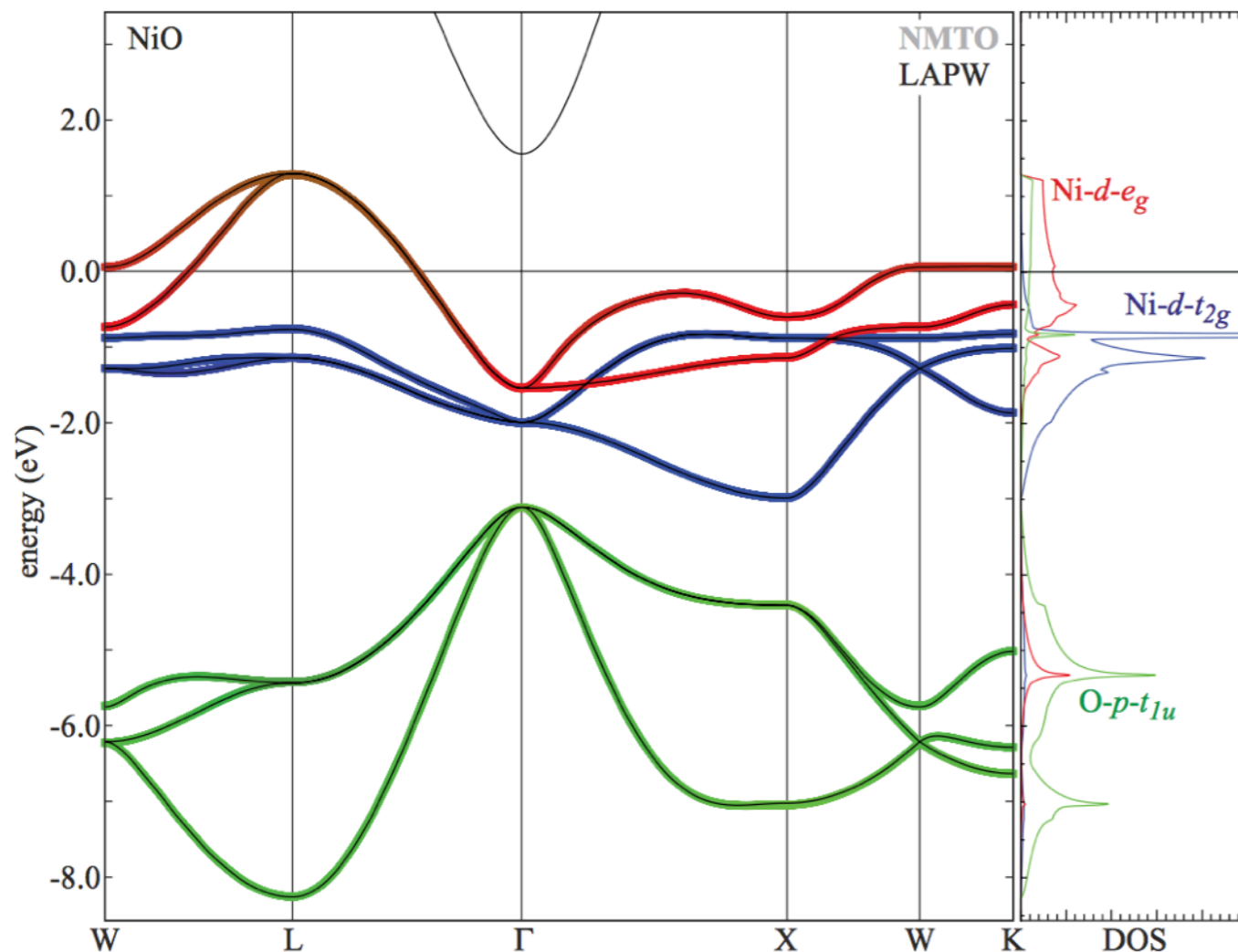
- XAS is not sensitive to U and Δ because excited electron screens



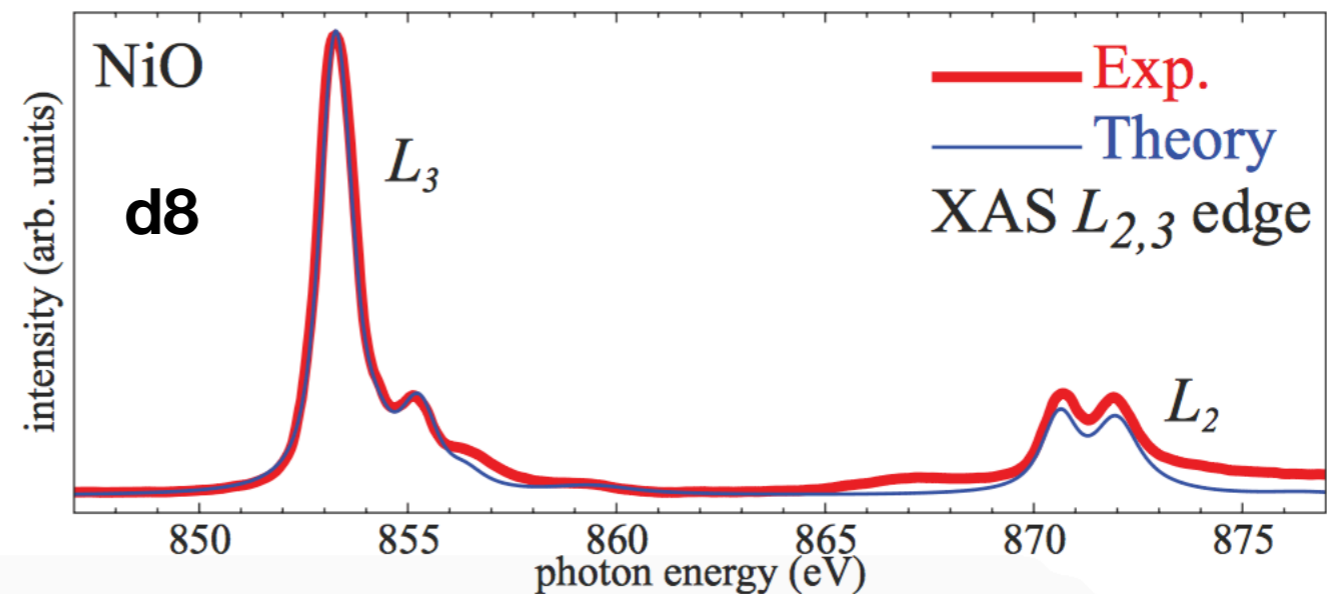
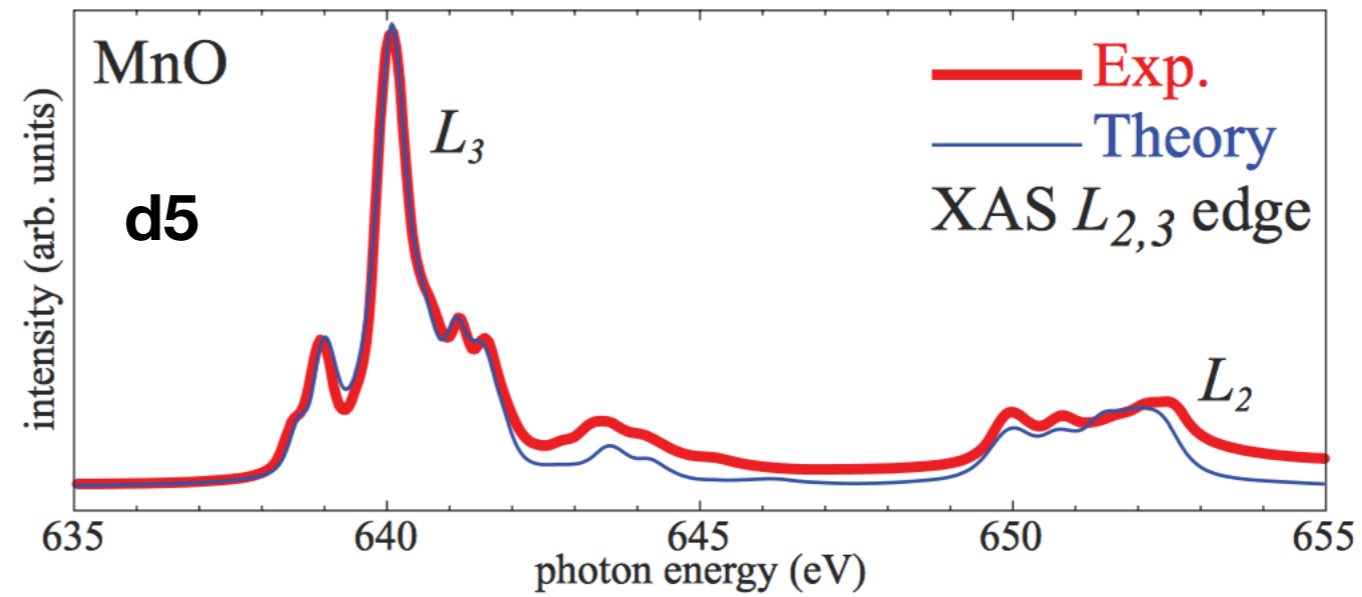
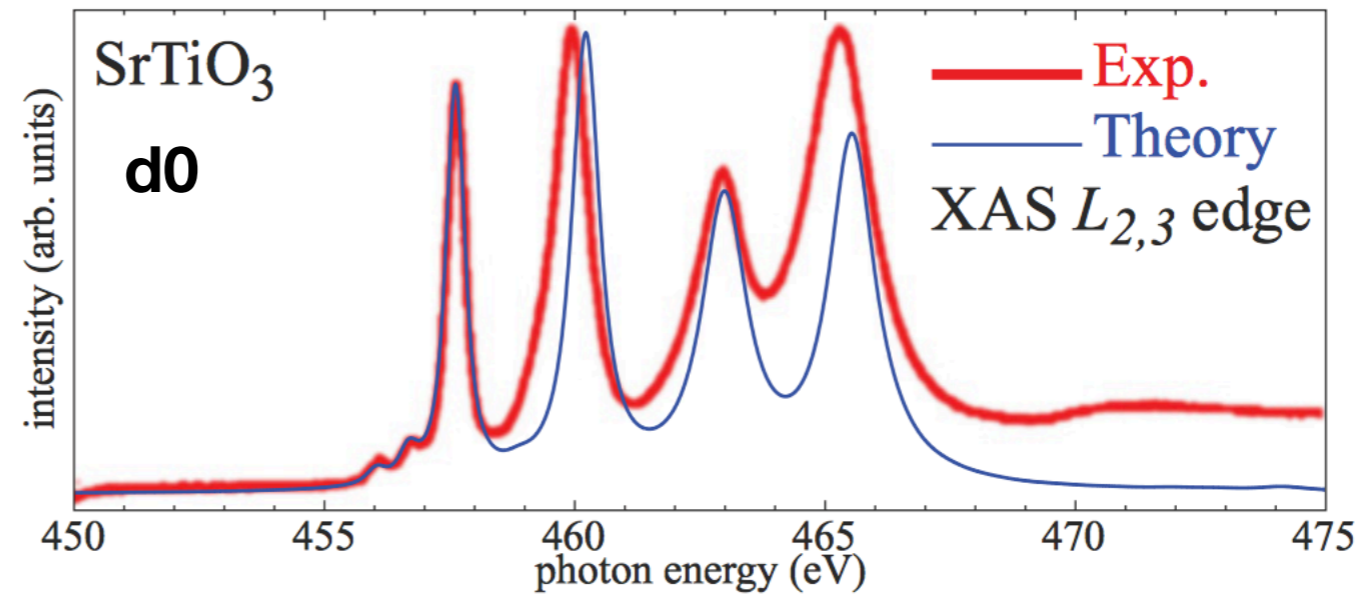
Q is the core hole potential

Local many-body problem

- Start with the LDA (or better) potential
- Create a set of Wannier functions
- Build local Hamiltonian on this basis and potential and include all local many body interactions



Spectra dominated by multiplets

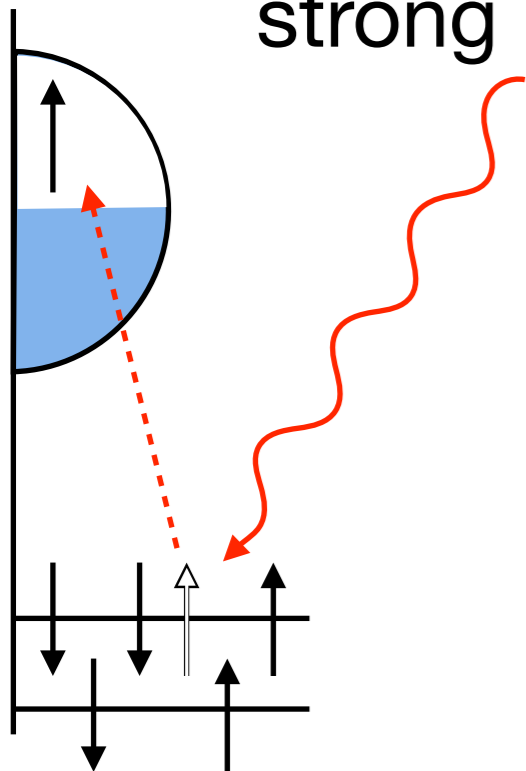


Summary of CI approach

- The bad
 - Based on parameters (U and Δ) that are not easy to compute *a priori*
 - Computationally expensive for large basis
- The good
 - XAS rather insensitive to U and Δ
 - If you can find the values of U and Δ you will know a great deal about the system
 - Small basis often fine for excitons

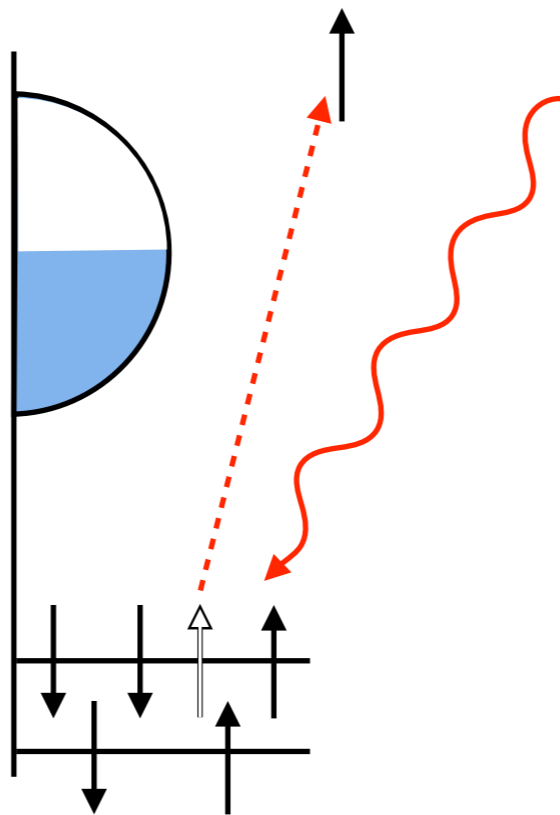
Summary of XAS

- Polarization dependence comes from dipole selection rule
- $L_{II,III}$ needed in sum rule — mixing due to non-local exchange
- Dichotomy of results on different materials/energy scales
 - PDOS when correlations and core hole potential are weak
 - Multiplets when correlations and core hole potential are strong



XPS

- Typically used for chemical shift, stoichiometry, etc...
- To get the zero of XAS you need to know the core level binding energy



Core level binding energy

- Computing a core level binding energy is straightforward with DFT
- You only get relative binding energy
- You can compute the initial state contribution separately from the final state contributions
 - Initial state shift is just the energy of the KS level relative to some reference
 - Final state shift involves relaxations

Final state shift

- Energy difference between final and initial state

$$E = \boxed{\begin{array}{c} \circ \\ \bullet \\ \bullet \end{array}} - \boxed{\begin{array}{c} \bullet \\ \bullet \end{array}} \quad \begin{array}{l} \circ \text{ core hole} \\ \bullet \text{ no core hole} \end{array}$$

- Perform SCF calculation with core hole to get energy of final state

- With PBC will require supercell

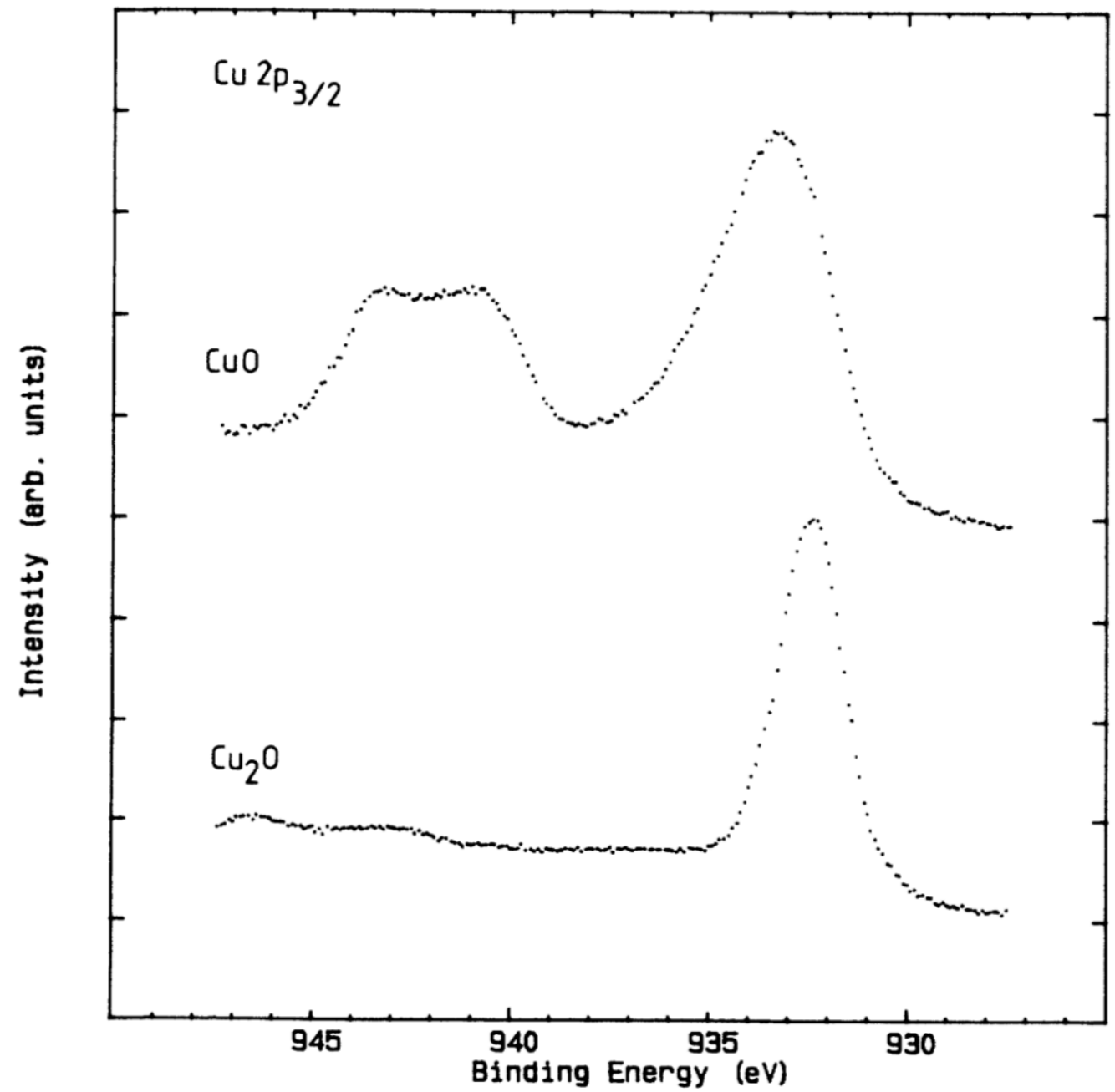
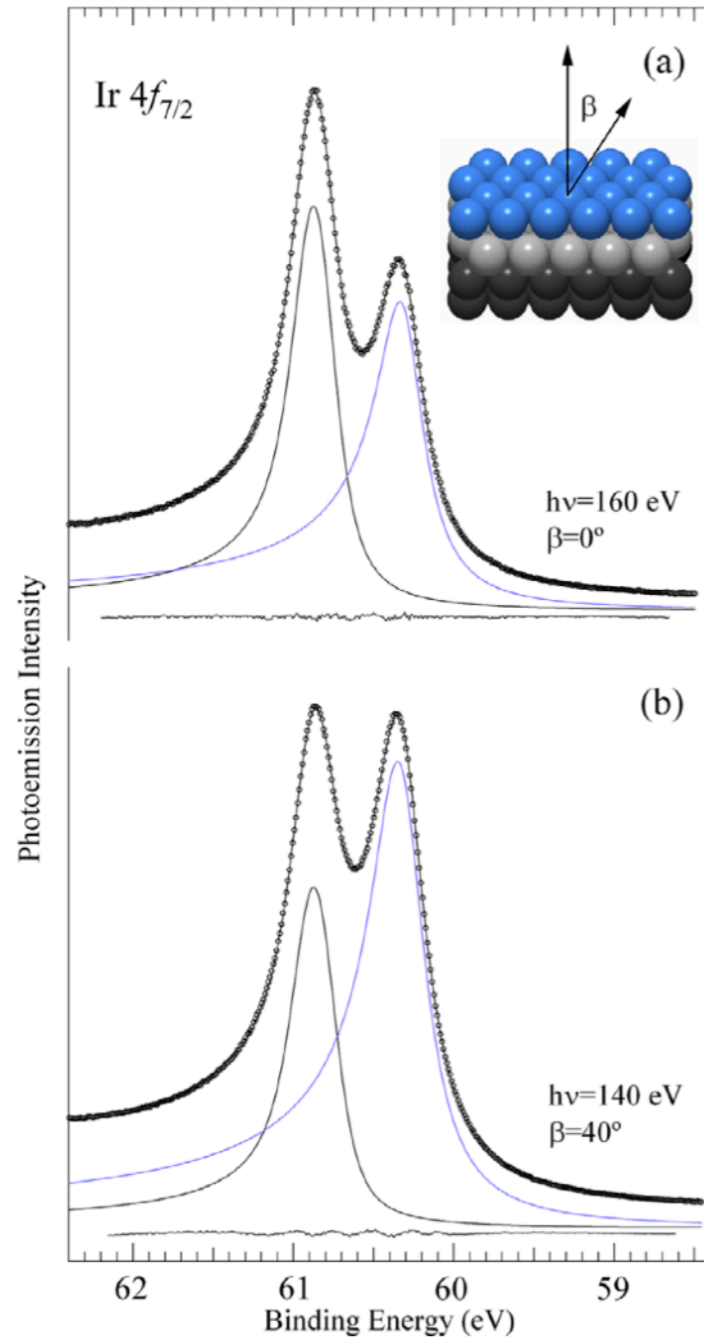
$$E = \boxed{\begin{array}{ccc} \circ & \bullet & \bullet \\ \bullet & & \bullet \\ \bullet & & \bullet \\ \bullet & \bullet & \bullet \end{array}} - 4 \times \boxed{\begin{array}{c} \bullet \\ \bullet \end{array}} \quad \begin{array}{l} \circ \text{ core hole} \\ \bullet \text{ no core hole} \end{array}$$

- Can also use Slater's transition state approximation

XPS

- But the spectra are not delta functions
- Final states are always screened
- How they are screened matters
- Multiple peaks/shape give insight into contributions to ground state

XPS

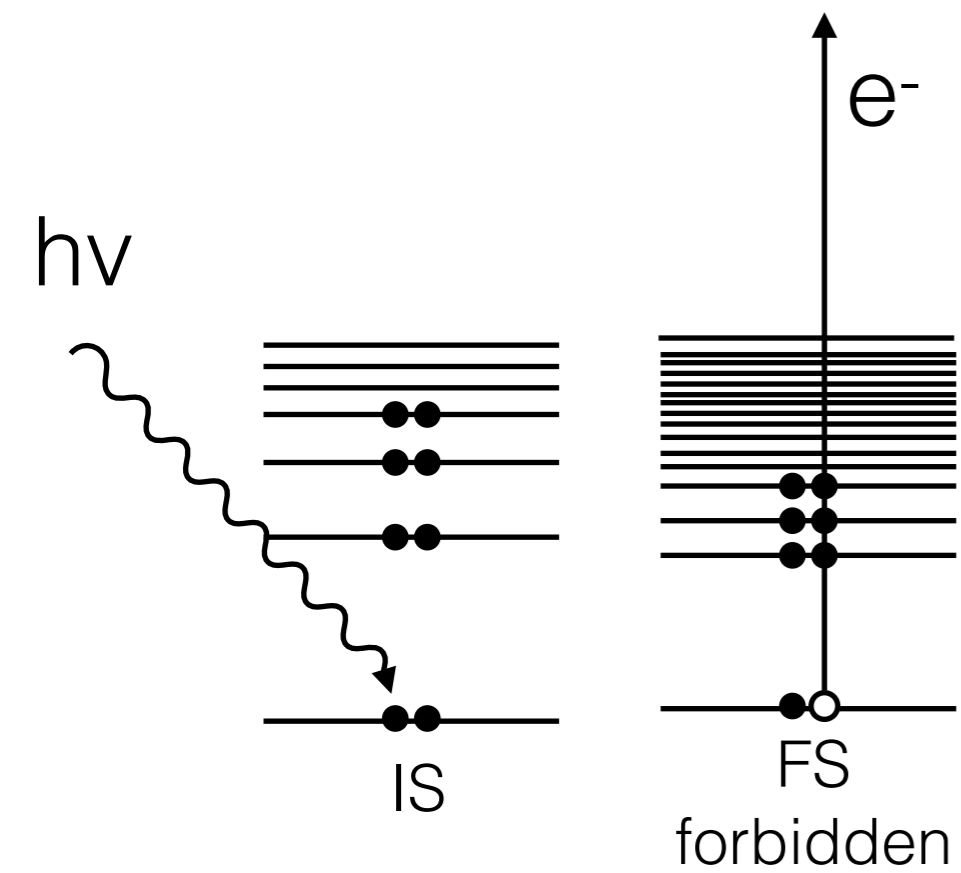


Line shapes for metals

- X-ray edge of a metal is complicated because 10^{23} conduction electrons respond to creation core hole
- Original work on absorption
 - Exciton theory predicts adsorption edges have power law divergence near threshold (Mahan 1967)
 - Orthogonality catastrophe requires electron-hole excitations for non-zero absorption (Anderson 1967)
 - Asymptotically exact solution available near threshold (Nozieres and deDomincis 1969)
- Same many body physics present in XPS (Doniach and Sunjic 1970)

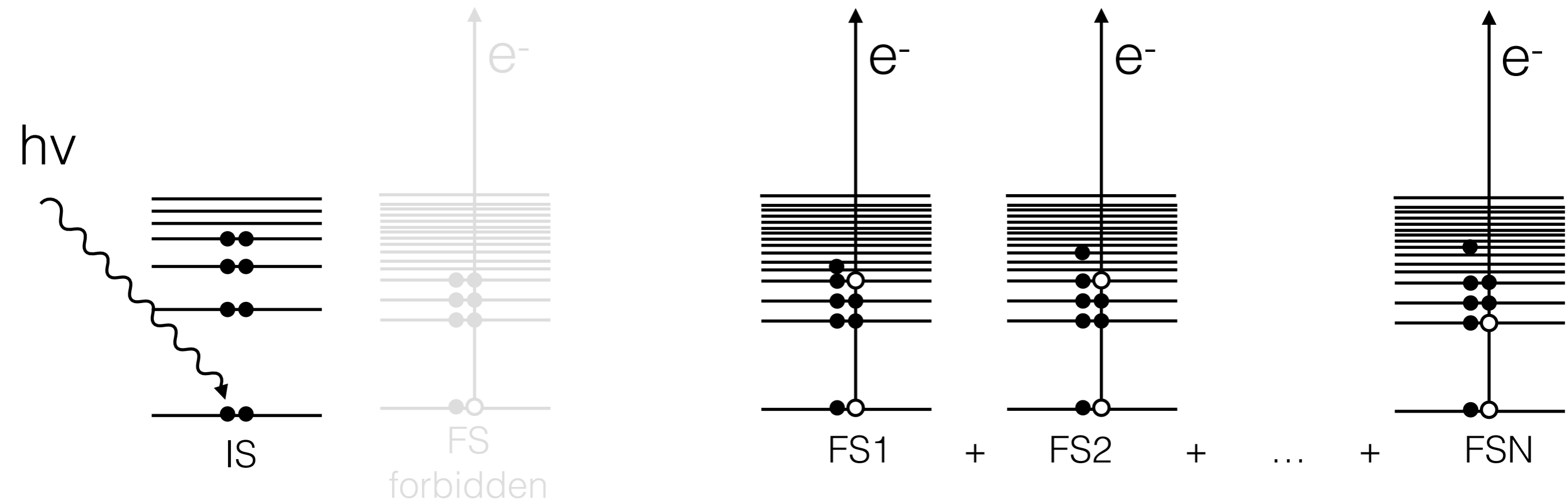
DS

- Orthogonality catastrophe makes main line asymmetric



DS

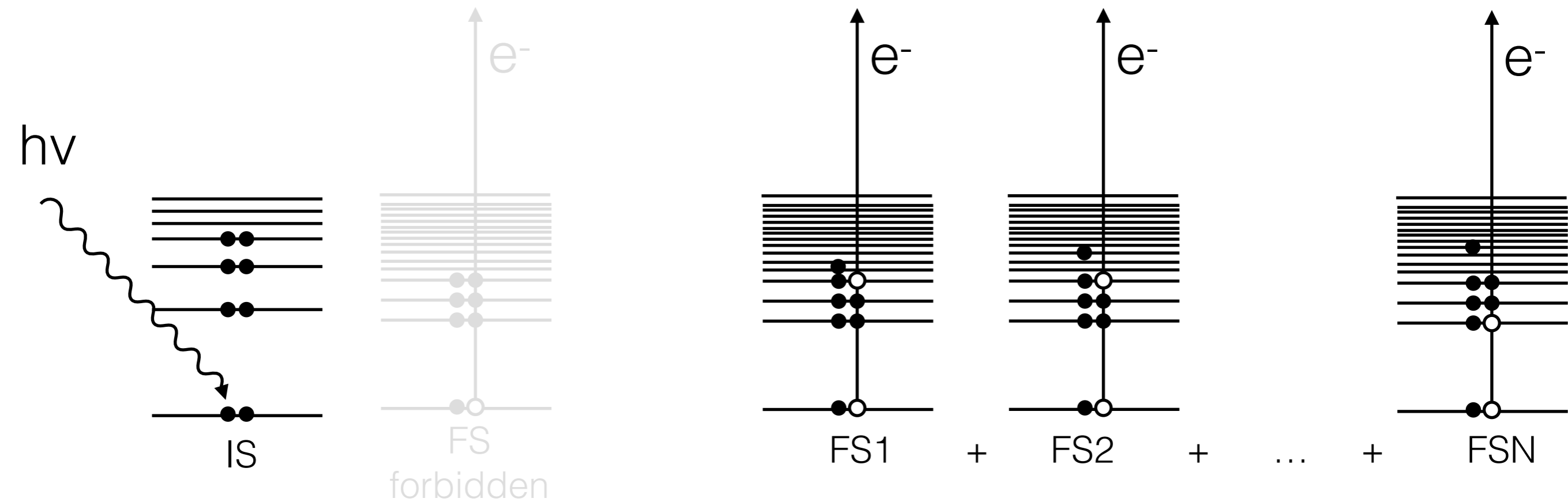
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DS

- Orthogonality catastrophe makes main line asymmetric

States increasing in E



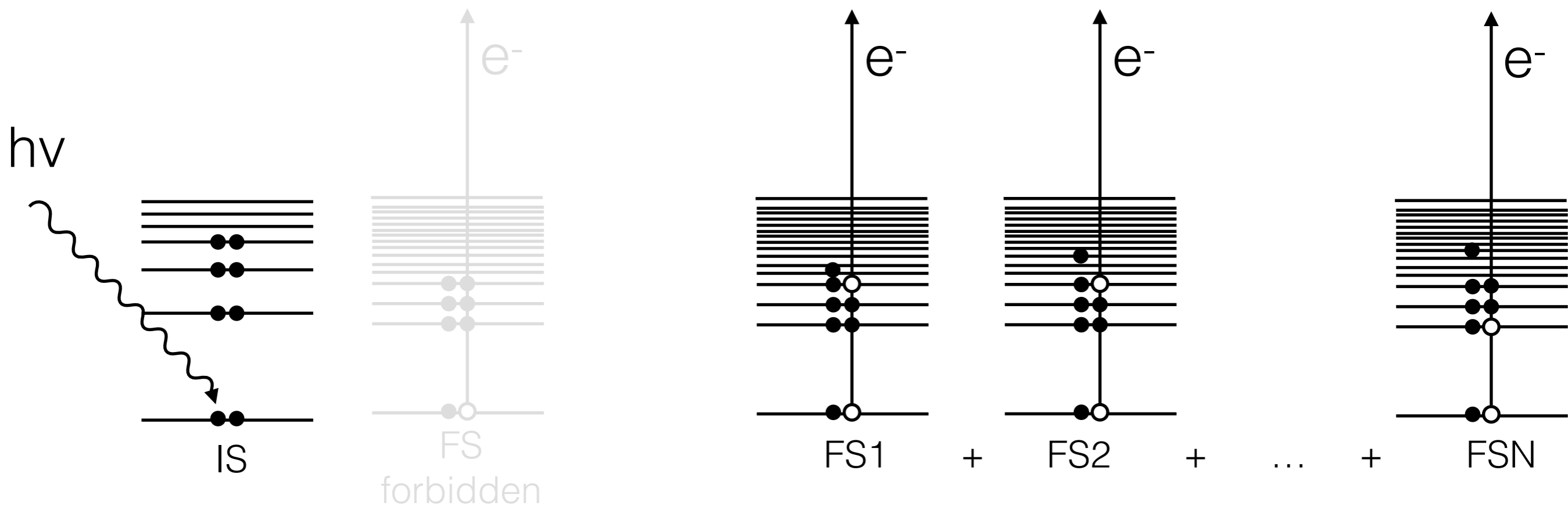
DS

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States increasing in E

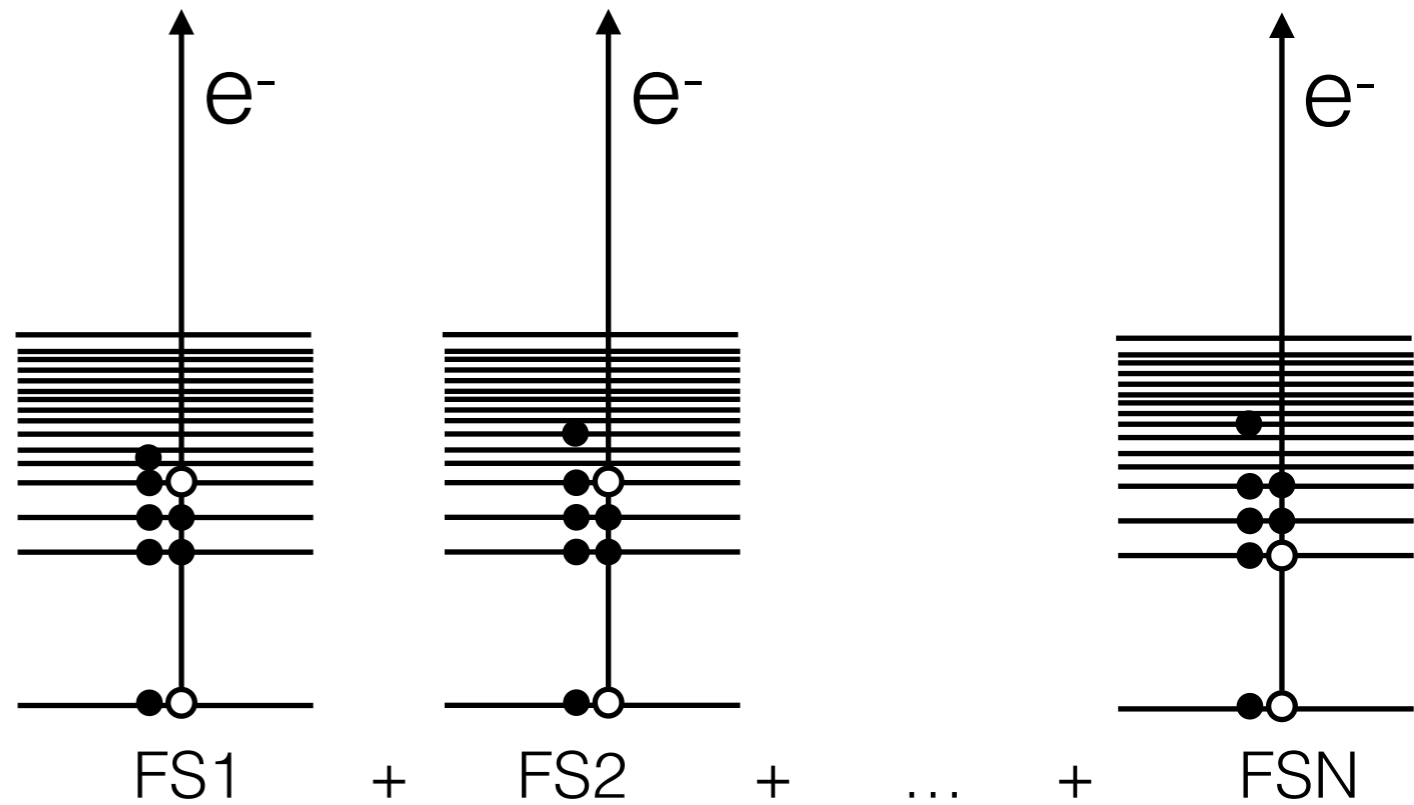
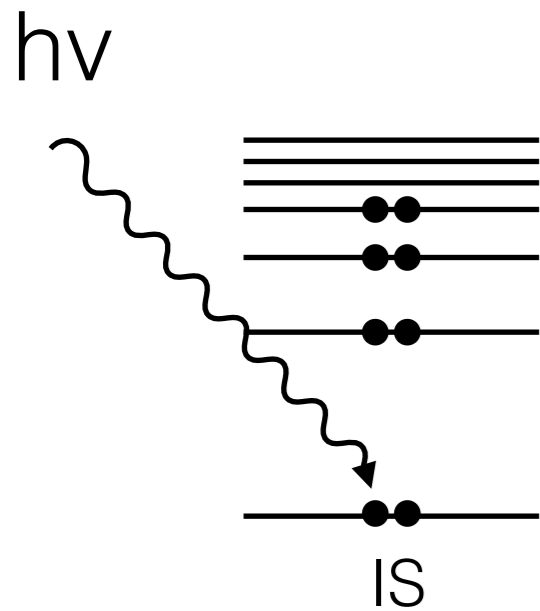


How does each state contribute to $I(\text{XPS})$?



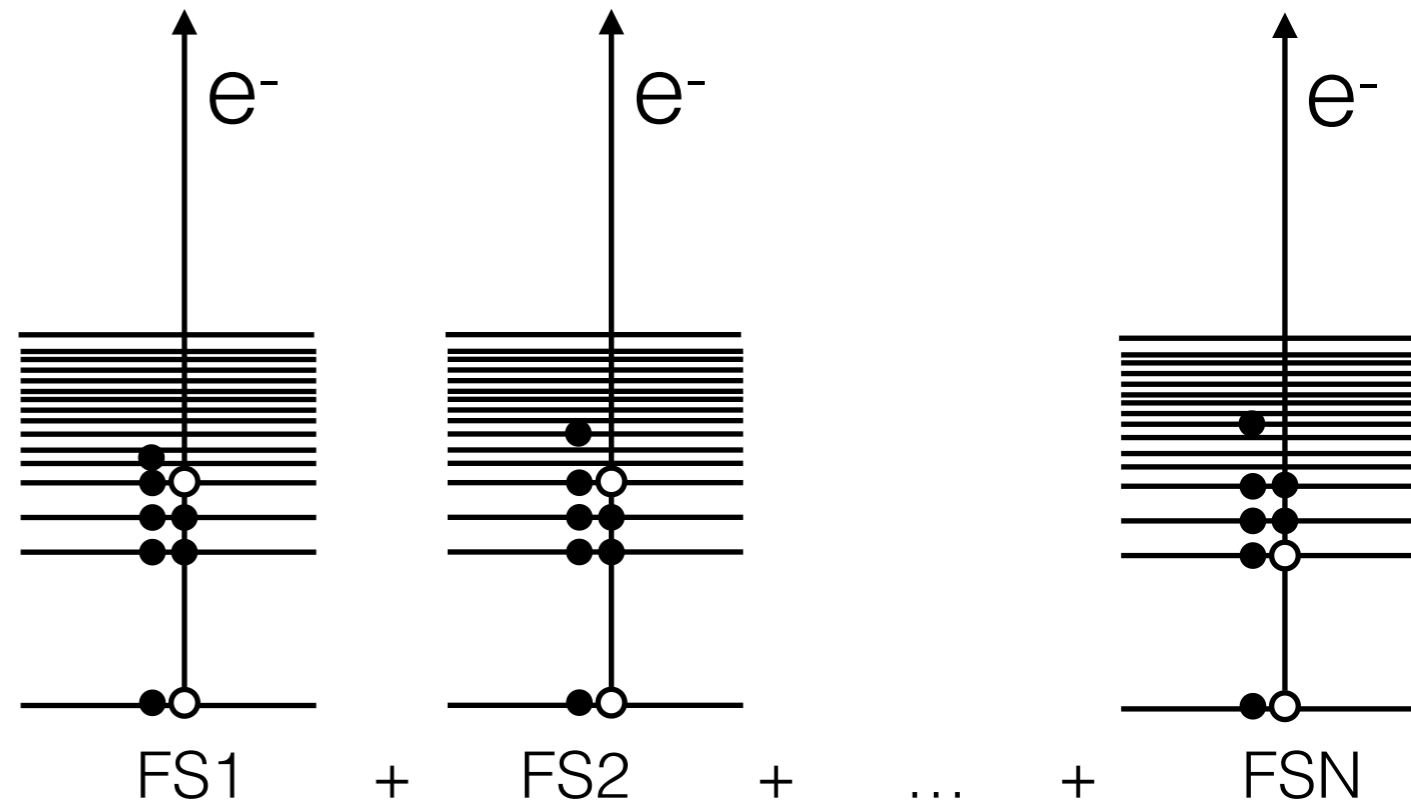
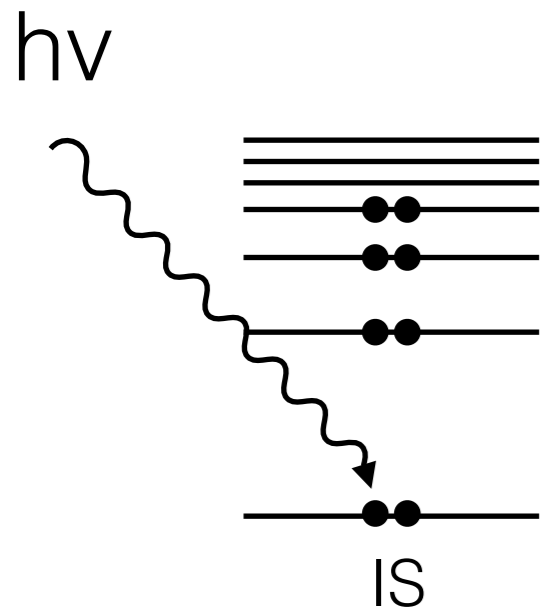
DS

- Orthogonality catastrophe makes main line asymmetric
- Probability for an excitation from i to j is M^2/ϵ_{ij}^2



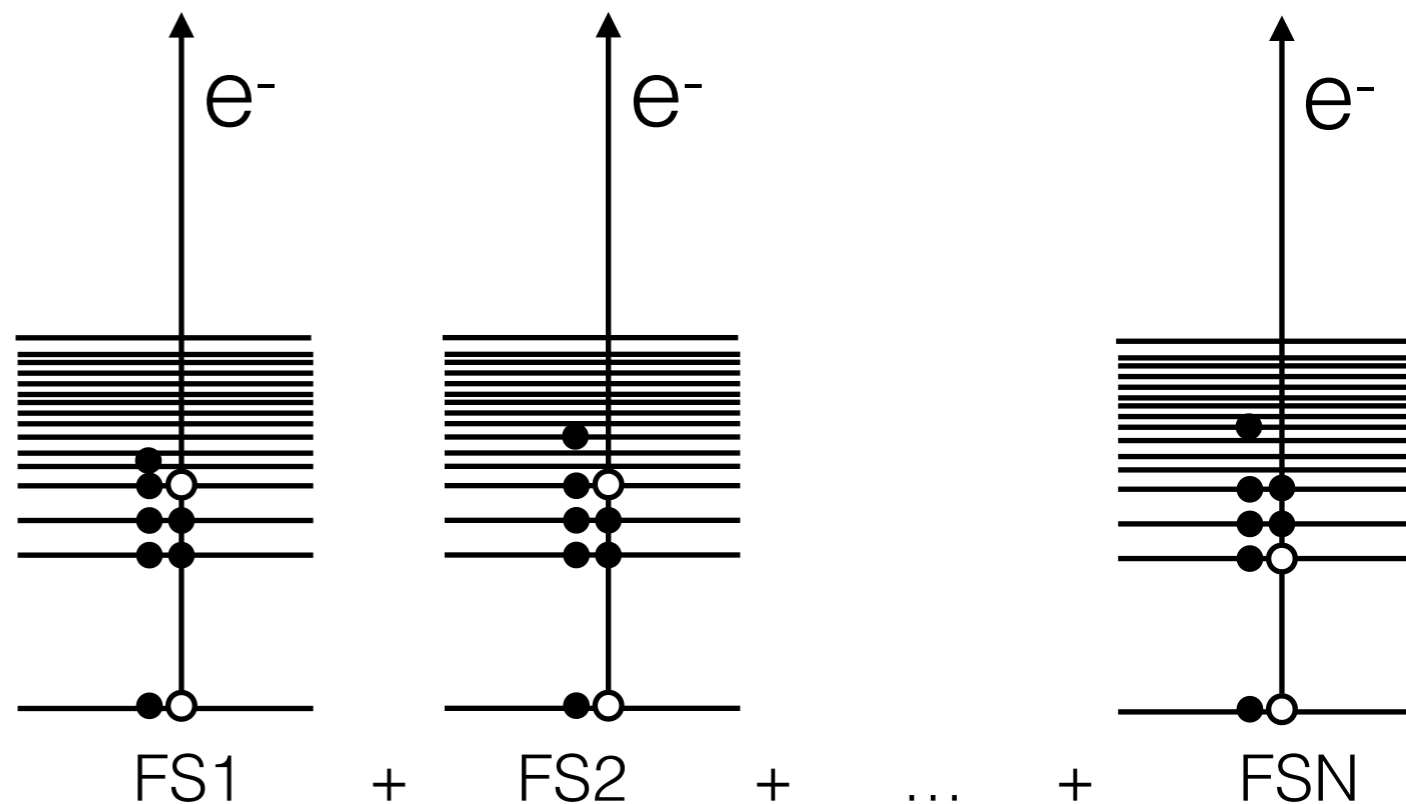
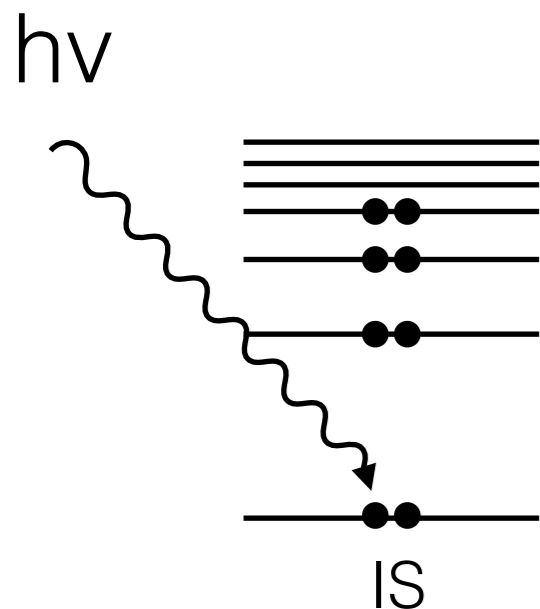
DS

- Orthogonality catastrophe makes main line asymmetric
- Probability for an excitation from i to j is M^2/ϵ_{ij}^2
- Total probability density function is $P(E) \propto \int_{-\infty}^{\infty} \exp(-iEt) \exp\left(\sum_{ij} \frac{M^2 \exp(i\epsilon_{ij}t) - 1}{\epsilon_{ij}^2}\right) dt$



DS

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- Convolve with Lorentzian and Gaussian to get line shape
- Make it solvable by assuming
 - matrix elements small and equal
 - DOS continuous \Rightarrow replace sum with integral over joint DOS
 - DOS constant \Rightarrow JDOS is linear

DS

- Assumptions restrict validity to neighborhood near E_f !

Requirement of constant DOS problematic

- $I(E)$ integrals do not coverage because JDOS is linear

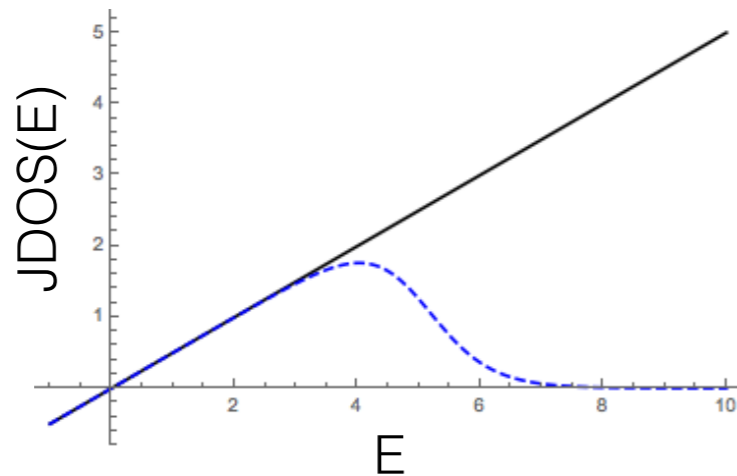
$$I(E) \sim \int_{-\infty}^{\infty} e^{-iEt} e^{-\lambda|t|} e^{-\frac{\sigma^2 t^2}{2}} \exp\left(\int_0^{\infty} \text{JDOS}(E') \frac{e^{iE't-1}}{(E')^2} dE'\right) dt$$

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- JDOS must tend to zero as E goes to infinity

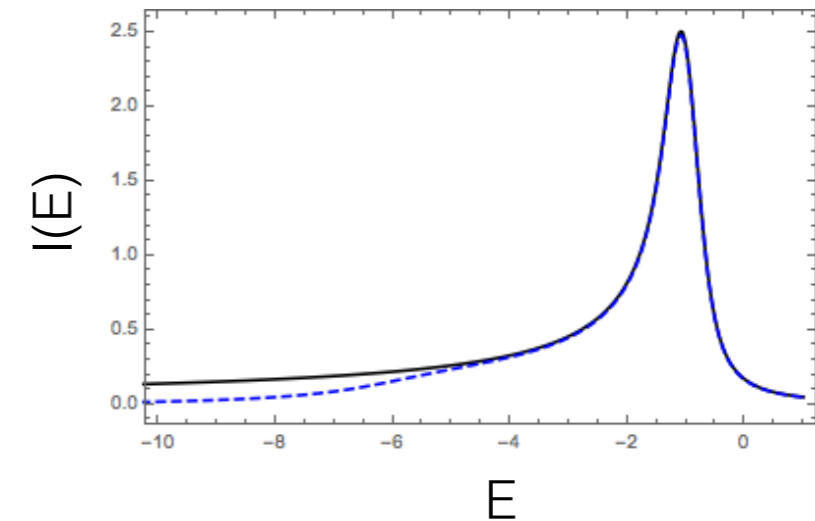
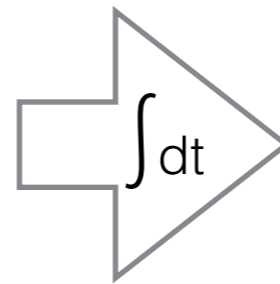
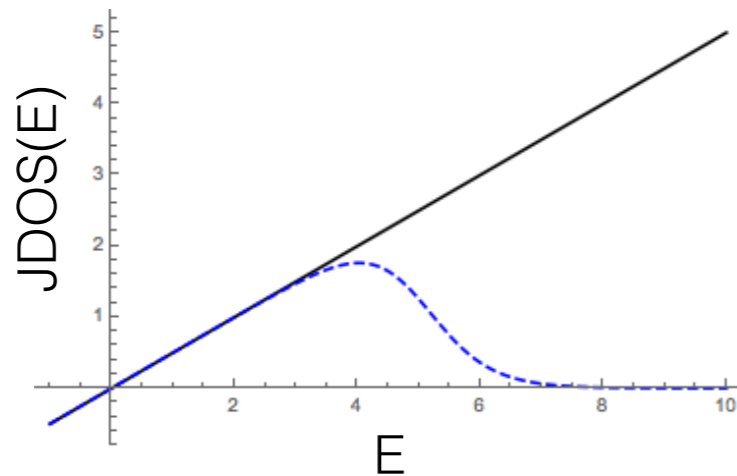


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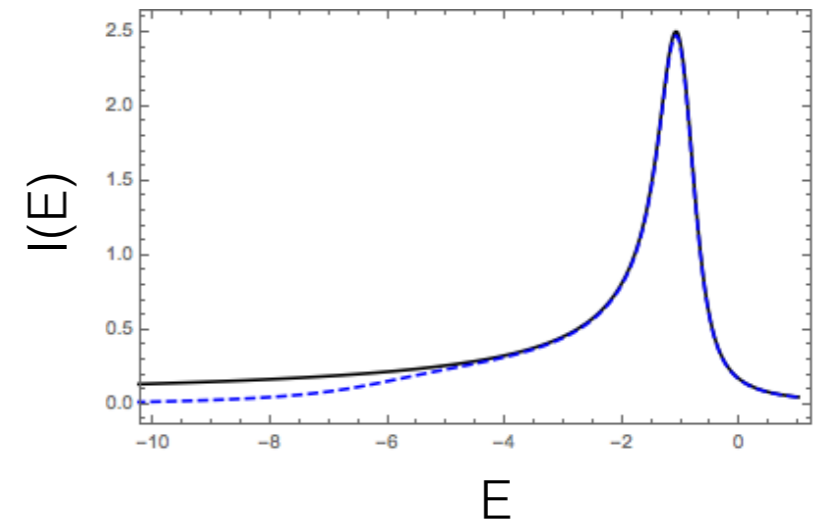
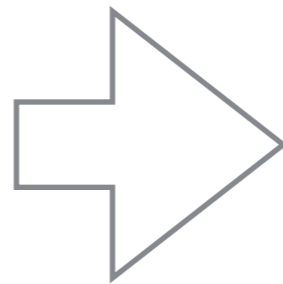
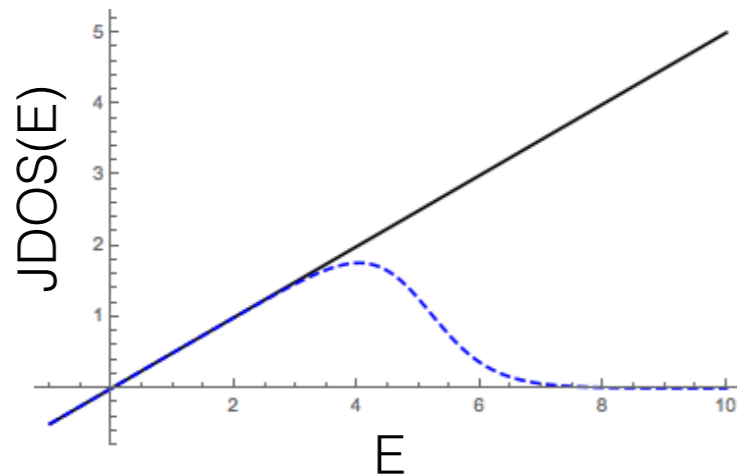


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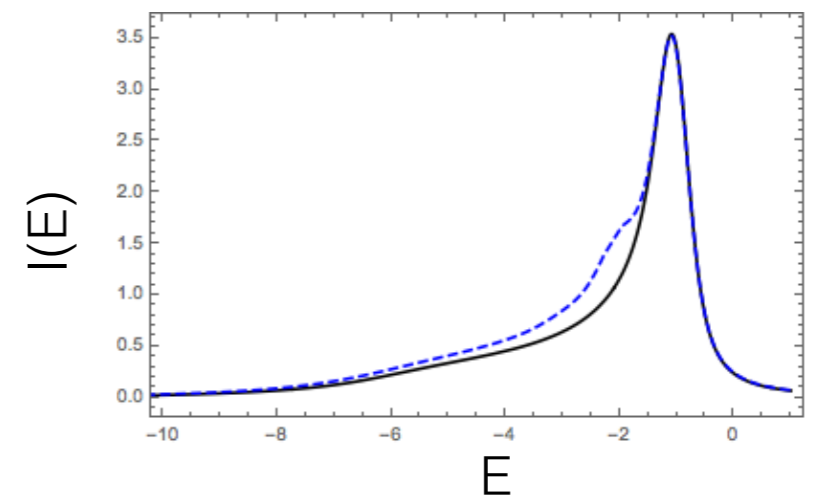
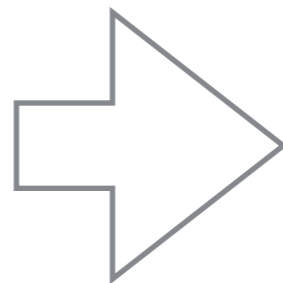
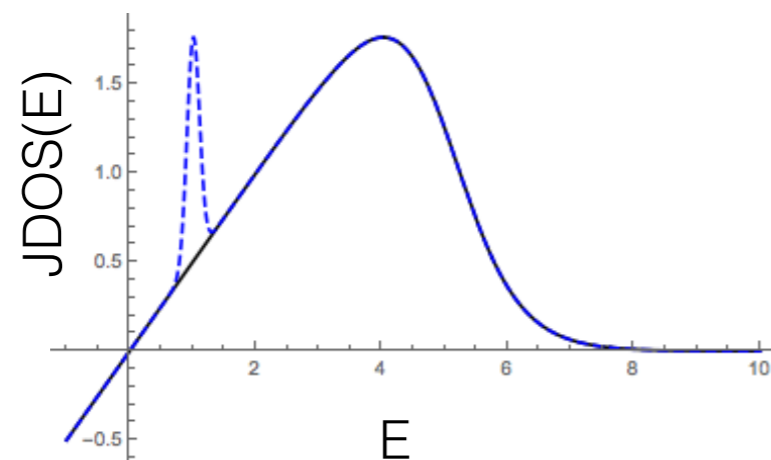
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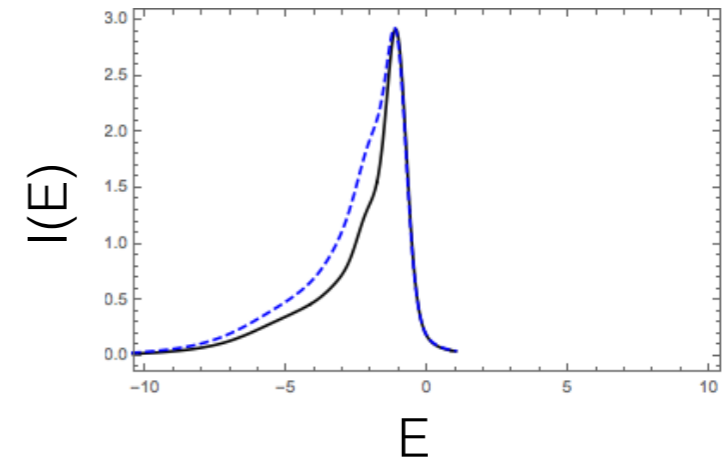
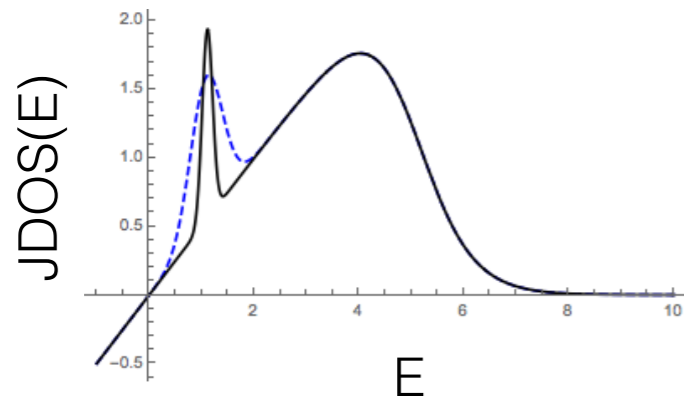


- Features in DOS near E_f will appear in spectra

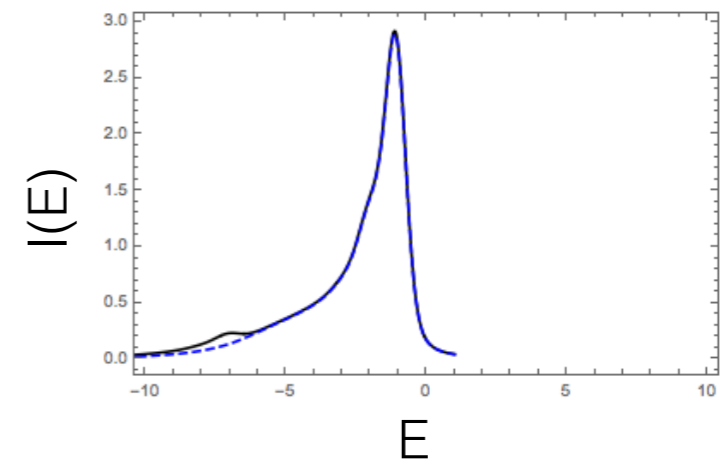
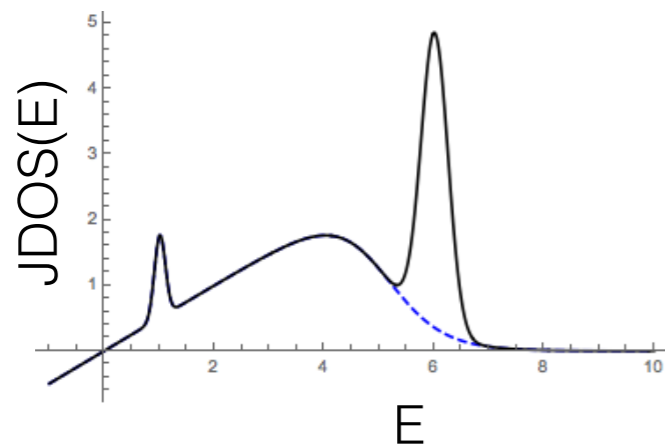


JDOS Features

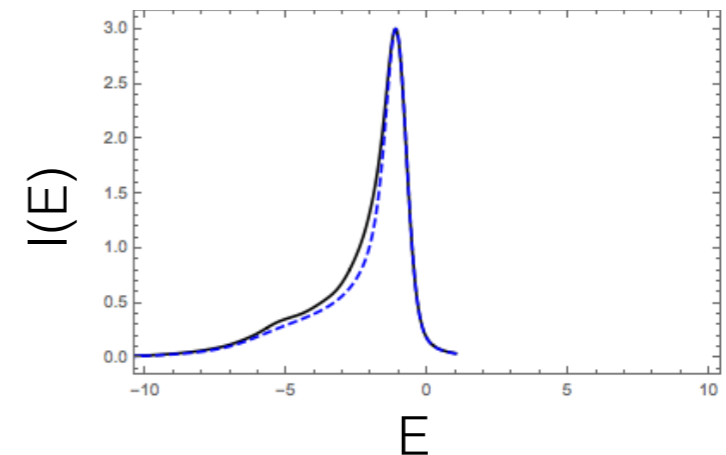
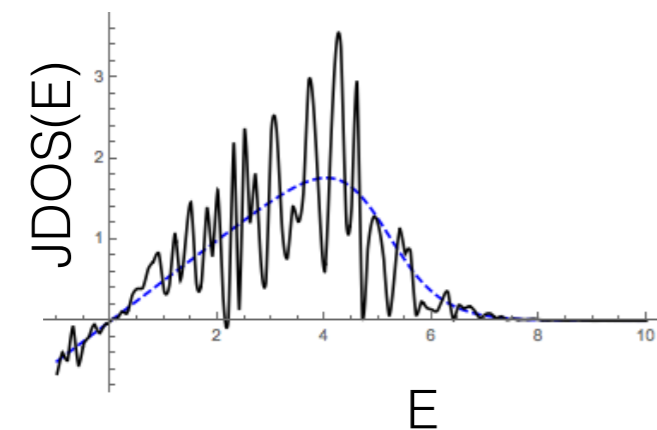
- Size and shape of JDOS features influence peak shape



- High energy features have weaker influence

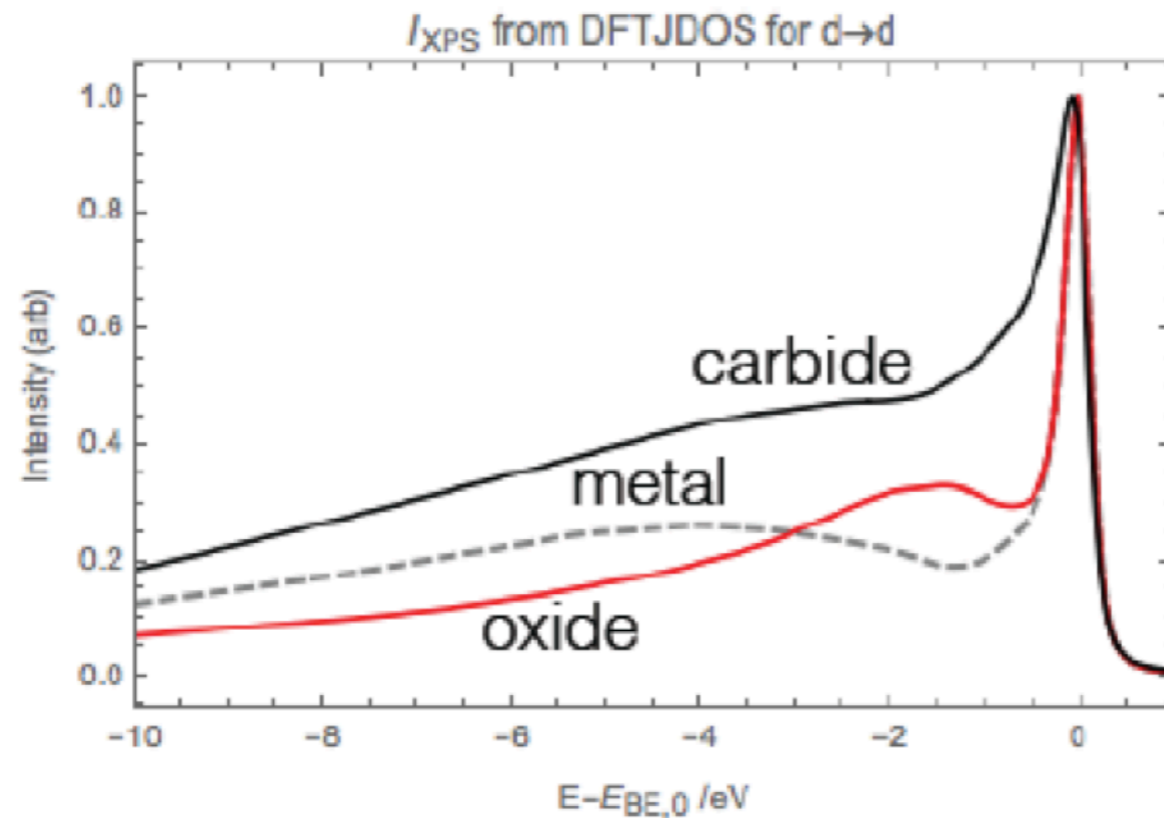
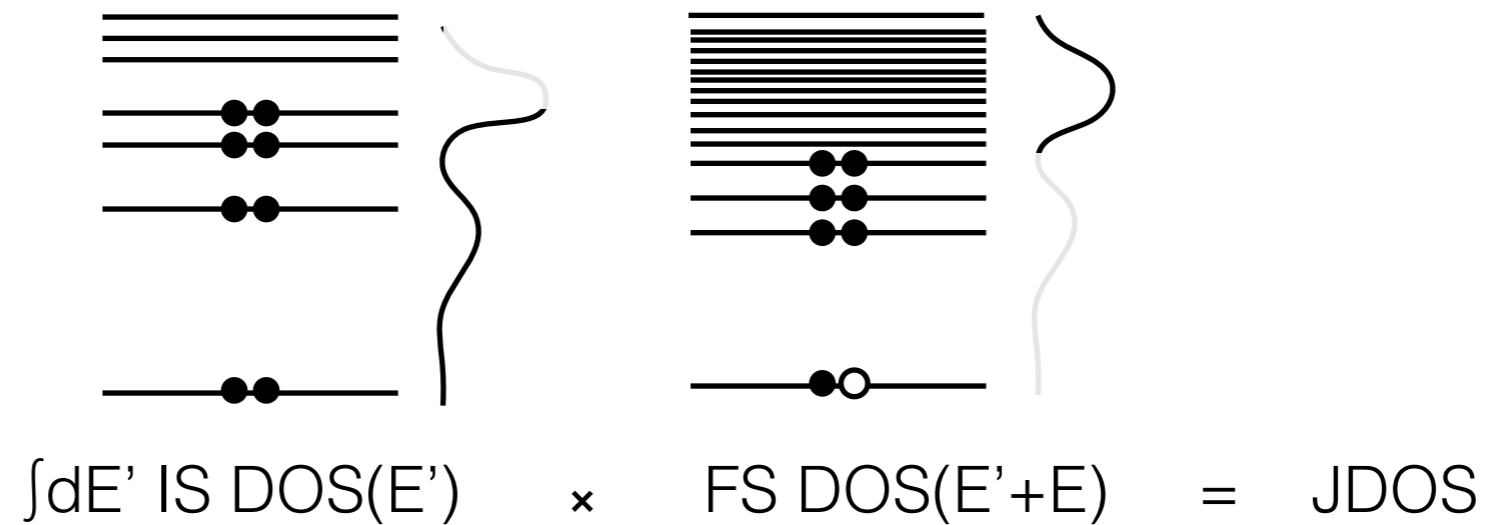


- JDOS features can appear averaged



JDOS from DFT

- Can just use JDOS from DFT calculations (still ignoring matrix elements)
- Use occupied IS DOS and unoccupied FS DOS to generate JDOS



Non-metallic?

- Primary screening is charge transfer from ligands
- Return to CI picture

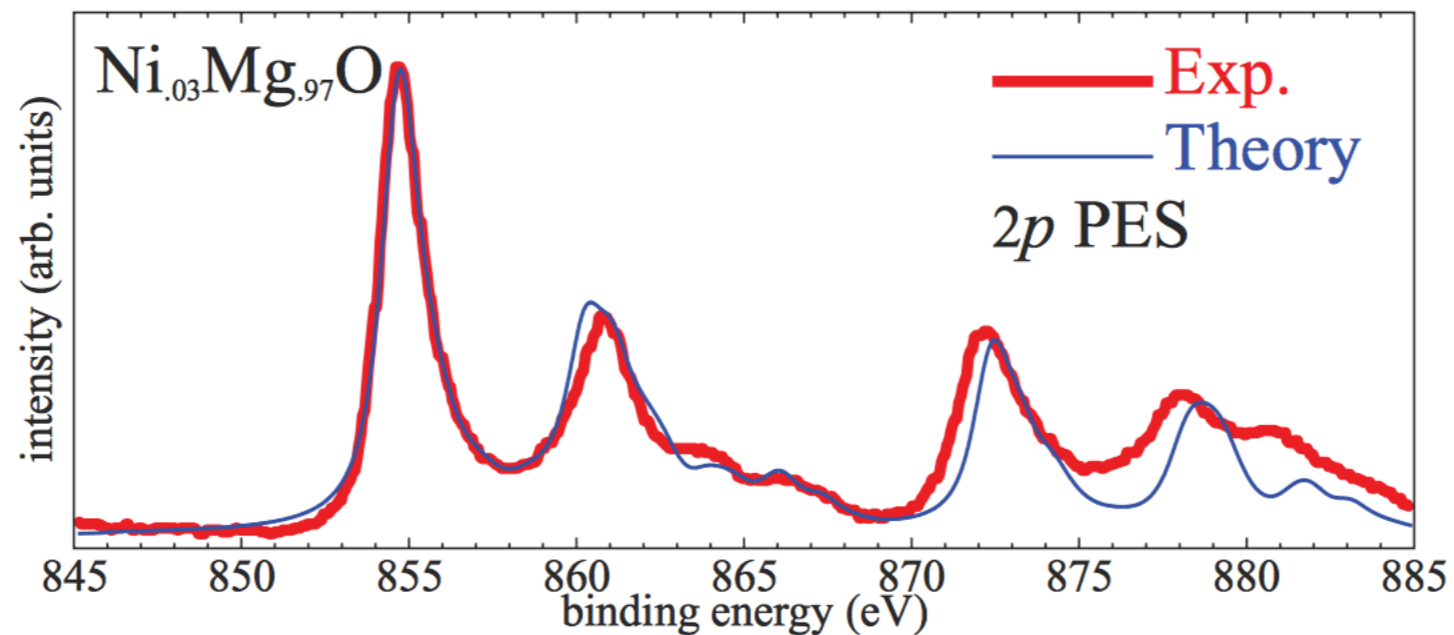
$$\psi_0 = \sum_i c_i |d^{n+i} \underline{L}^i\rangle$$

$$U \equiv \left[E(d^{n+1}) + E(d^{n-1}) \right] - 2E(d^n)$$

$$\Delta \equiv E(d^{n+1} \underline{L}^1) - E(d^n)$$

Local many-body problem

- Use Wannier functions from LDA to build local Hamiltonian and potential
- Include all local many body interactions

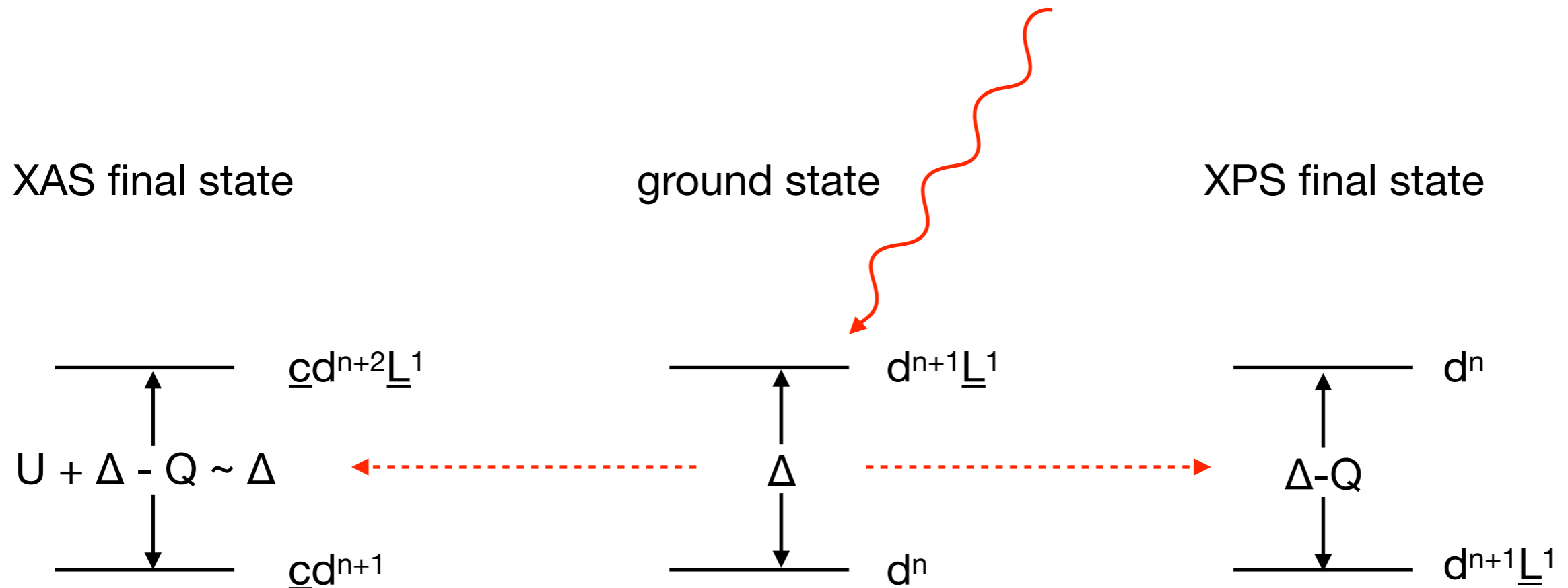


M.W. Haverkort, M. Zwierzycki, O.K. Andersen, PRB **85**, 165113 (2012).

- Nice results ... but sensitive to U and Δ

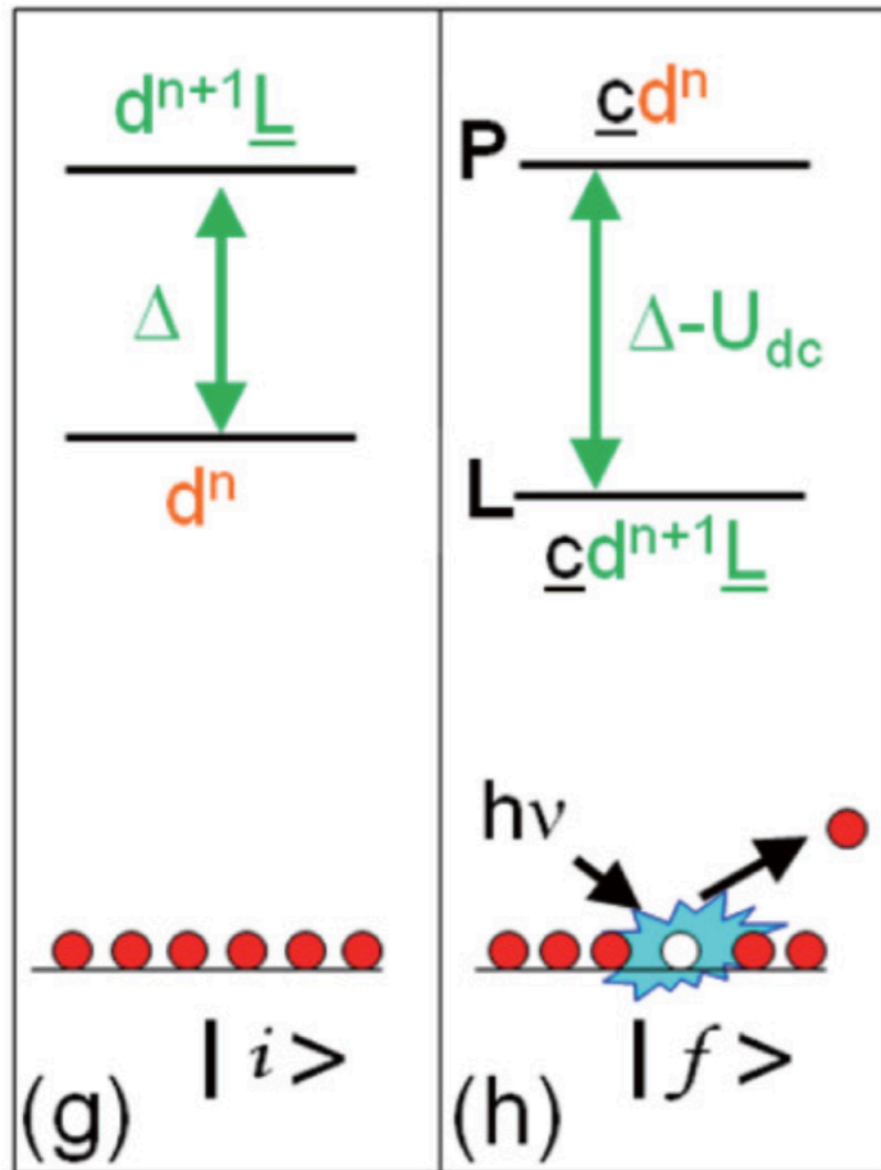
U and Δ

- XAS is not sensitive to U and Δ because it is charge neutral
- XPS is very sensitive to U and Δ because valence electrons screen



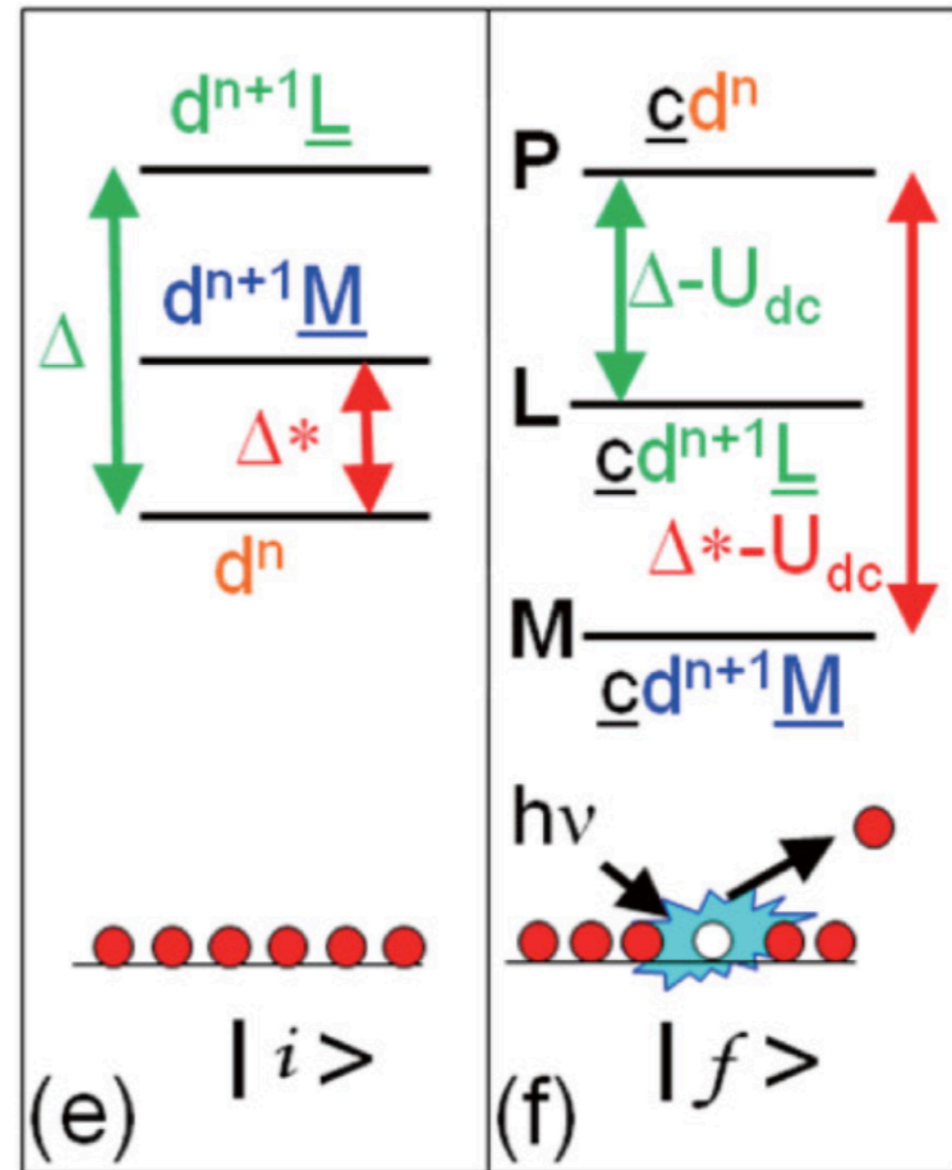
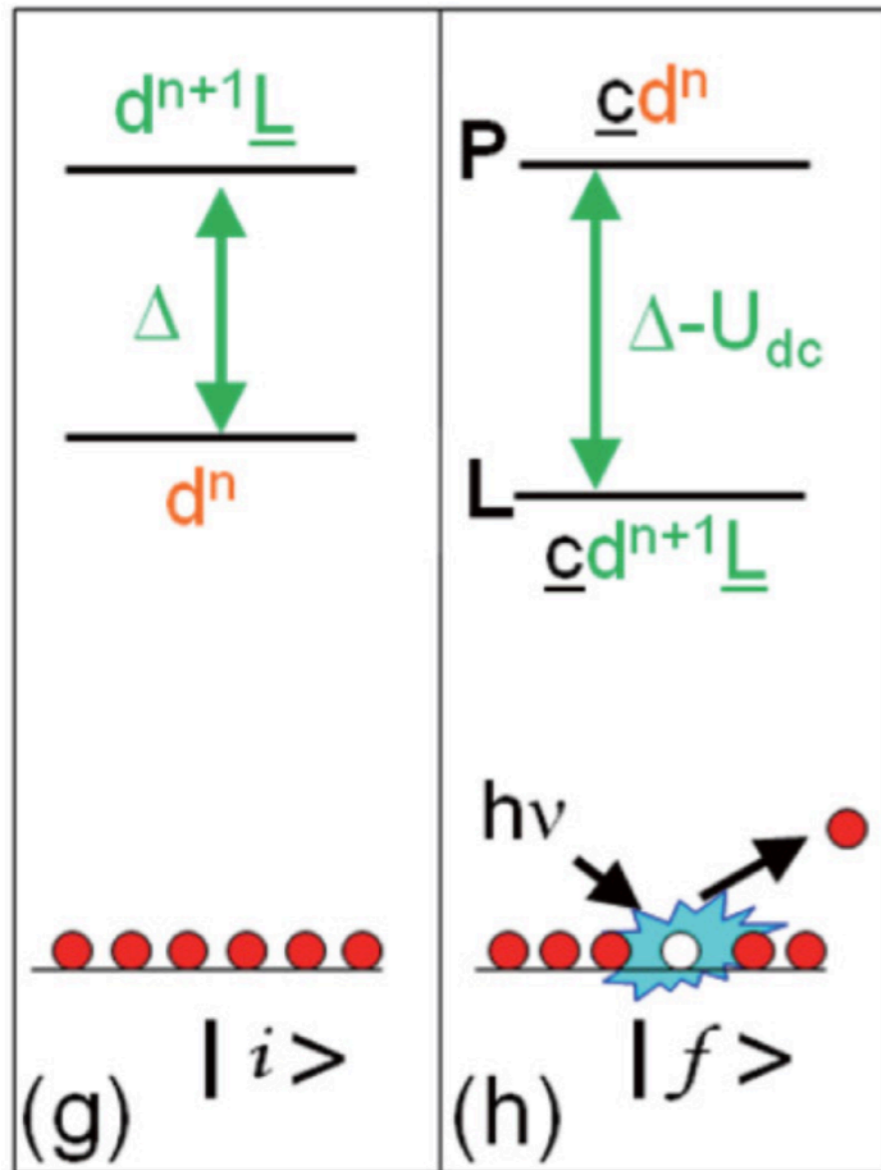
And non-local screening

- Can even add non-local screening to channel to handle metallic and insulating systems within same framework



And non-local screening

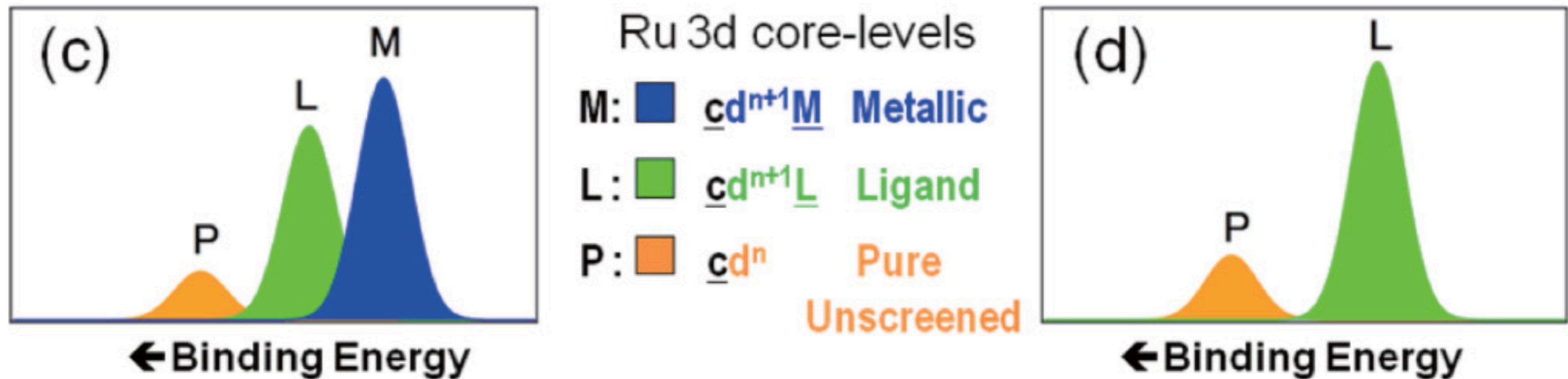
- Can even add non-local screening to channel to handle metallic and insulating systems within same framework



And non-local screening

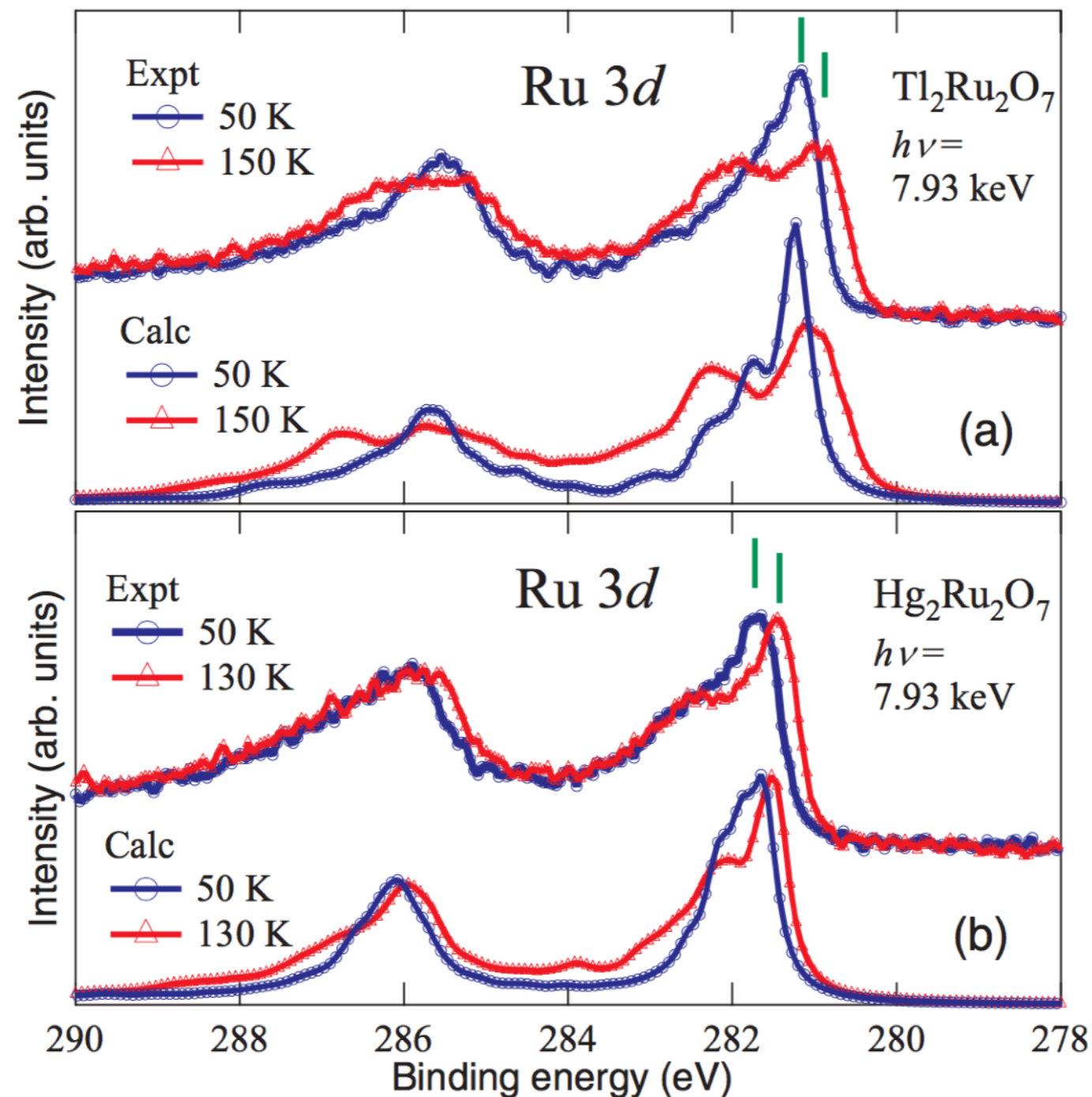
- Can even add non-local screening to channel to handle metallic and insulating systems within same framework

Schematic result for $\text{Hg}_2\text{Ru}_2\text{O}_7$ metal insulator transition



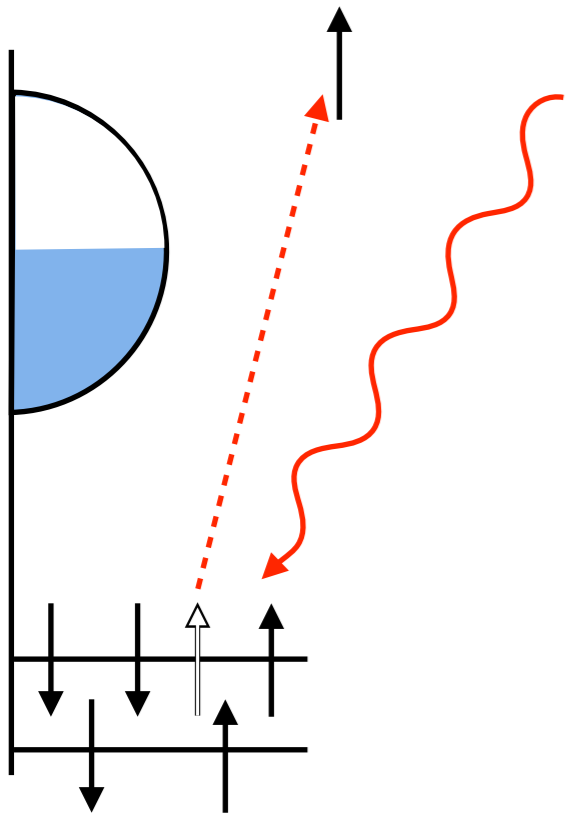
And non-local screening

- Can even add non-local screening to channel to handle metallic and insulating systems within same framework



XPS summary

- Core level binding energies straight from DFT
- IPA gives a delta function
- Many body physics gives useful line shapes
 - Metallic systems can be tackled with DFT
 - Metallic and non-metallic line shapes can be computed with CI-approach



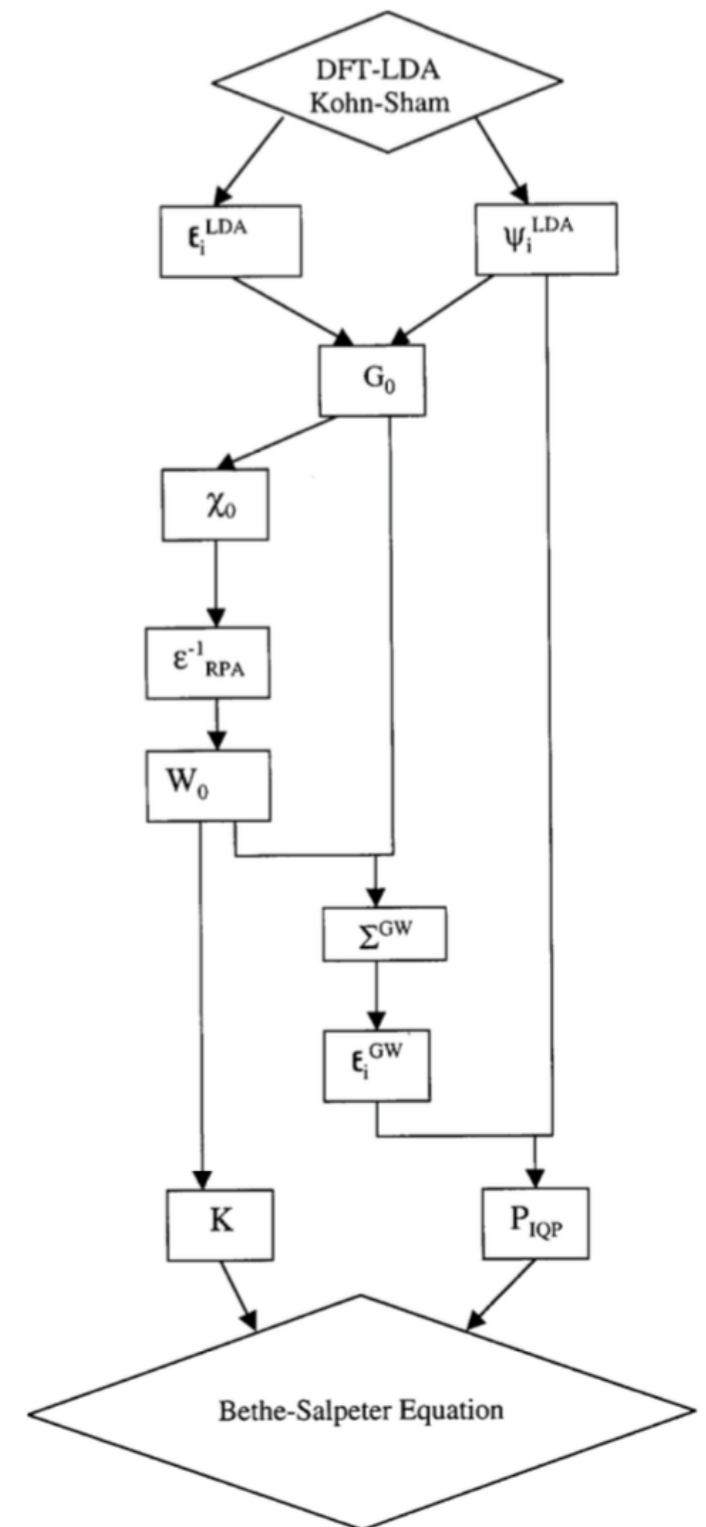
Bringing it all together

- Absorption and emission give complimentary information
 - Screening in XPS due to ligands and/or condition band
 - Screening in XAS from excited electron
 - In insulator XAS edge will not align with XPS BE!
- For non-correlated materials there are lots of options
 - XAS—BSE and IPE (for K edges) ... in principle CI and DMFT
 - XPS—JDOS and CI and DMFT
 - Core level shift—Slater transition state, Δ SCF, initial state
- For correlated materials (DFT might even fail you so be careful) the options are limited
 - XAS—BSE for one-electron one-hole excitations otherwise CI and DMFT
 - XPS—CI and DMFT

Questions

BSE NEXAFS

- Treat core-hole correctly
- Get excitonic effects
- L2,3 branching
- Still only with 10% of peak positions and 20% peak strength
- Using NLPP possible errors with non-locality
- Use DFT wave functions -> missing $-1/r$ tail in LDA may be problematic for clusters/surfaces
- Static electron-hole screening -> fine when exciton binding energy not too large otherwise might need Strinati equation
- RPA electron-hole screening -> kernel W evaluated with RPA can overestimate exciton binding energy



Why does IPA work?

one-electron approach

BSE

$$h' = \frac{p^2}{2m} + V'_C + \Sigma'$$

$$\equiv h + [V'_C - V_C] + [\Sigma' - \Sigma].$$

Why does IPA work?

one-electron approach

$$h' = \frac{p^2}{2m} + V'_C + \Sigma'$$
$$\equiv \underbrace{h}_{\text{non-interacting response}} + [V'_C - V_C] + [\Sigma' - \Sigma].$$

non-interacting response

BSE

Why does IPA work?

one-electron approach

BSE

$$h' = \frac{p^2}{2m} + V'_C + \Sigma'$$
$$\equiv \underbrace{h}_{\text{non-interacting response}} + \underbrace{[V'_C - V_C]}_{\text{screened core hole}} + [\Sigma' - \Sigma].$$

non-interacting response

screened core hole

Why does IPA work?

one-electron approach

BSE

$$h' = \frac{p^2}{2m} + V'_C + \Sigma'$$

$$\equiv \underbrace{h}_{\text{non-interacting response}} + \underbrace{[V'_C - V_C]}_{\text{screened core hole}} + \underbrace{[\Sigma' - \Sigma]}_{\text{dynamically screened exchange}}.$$

non-interacting response

screened core hole

dynamically screened
exchange

Why does IPA work?

- Same quasi-particle hamiltonian

one-electron approach

$$h' = \frac{p^2}{2m} + V'_C + \Sigma'$$
$$\equiv \underbrace{h}_{\text{non-interacting response}} + \underbrace{[V'_C - V_C]}_{\text{screened core hole}} + \underbrace{[\Sigma' - \Sigma]}_{\text{dynamically screened exchange}}.$$

non-interacting response

screened core hole

dynamically screened
exchange

BSE

$$h'_{\text{eff}} = \underbrace{h}_{\text{non-interacting response}} + V_D(i) + V_X(i).$$

non-interacting response

Why does IPA work?

one-electron approach

$$h' = \frac{p^2}{2m} + V'_C + \Sigma'$$
$$\equiv \underbrace{h}_{\text{non-interacting response}} + \underbrace{[V'_C - V_C]}_{\text{screened core hole}} + \underbrace{[\Sigma' - \Sigma]}_{\text{dynamically screened exchange}}$$

non-interacting response

screened core hole

dynamically screened
exchange

BSE

$$h'_{\text{eff}} = \underbrace{h}_{\text{non-interacting response}} + \underbrace{V_D(i)}_{\text{direct screened core hole}} + V_X(i).$$

non-interacting response

direct screened core hole

Why does IPA work?

one-electron approach

$$h' = \frac{p^2}{2m} + V'_C + \Sigma'$$
$$\equiv h + [V'_C - V_C] + [\Sigma' - \Sigma].$$

$$V'_C(\vec{r}) - V_C(\vec{r}) \equiv V_i(\vec{r}) + \Delta V_{\text{scf}}(\vec{r}).$$

bare potential

induced potential

BSE

$$h'_{\text{eff}} = h + V_D(i) + V_X(i).$$

non-interacting response

direct screened core hole

Why does IPA work?

- Direct screened core hole analogous to BSE

one-electron approach

$$h' = \frac{p^2}{2m} + V'_C + \Sigma'$$

$$\equiv h + [V'_C - V_C] + [\Sigma' - \Sigma].$$

$$V'_C(\vec{r}) - V_C(\vec{r}) \equiv V_i(\vec{r}) + \Delta V_{\text{scf}}(\vec{r}).$$

bare potential

induced potential

BSE

$$h'_{\text{eff}} = h + V_D(i) + V_X(i).$$

$$V_D(\vec{r}) = \int d^3r' W(\vec{r}, \vec{r}'; \omega = 0)$$

$$\times \frac{1}{N} \sum_{\vec{R}} |\chi_{i, \vec{R}}^*(\vec{r}')|^2 \exp[-i(\vec{k} - \vec{q}) \cdot \vec{R}]$$

$$\equiv V_i(\vec{r}) + V_{\text{ind}}(\vec{r}),$$

bare potential

induced potential

$$\Delta V_{\text{scf}}(\vec{r}) \approx V_{\text{ind}}(\vec{r}).$$

Why does IPA work?

one-electron approach

$$h' = \frac{p^2}{2m} + V'_C + \Sigma'$$
$$\equiv h + [V'_C - V_C] + [\Sigma' - \Sigma].$$

non-interacting response

screened core hole

dynamically screened
exchange

BSE

$$h'_{\text{eff}} = h + V_D(i) + V_X(i).$$

non-interacting response

direct screened core hole

unscreened direct
exchange

Why does IPA work?

one-electron approach

$$h' = \frac{p^2}{2m} + V'_C + \Sigma'$$

$$\equiv h + [V'_C - V_C] + [\Sigma' - \Sigma].$$

self energy is like dynamically screened exchange

$$\Sigma(\omega) = iG[v + (\epsilon^{-1}(\omega) - 1)v] = V_{ex} + \Sigma_C(\omega)$$

unscreened exchange

dynamic part of self energy

BSE

$$h'_{\text{eff}} = h + V_D(i) + V_X(i).$$

Why does IPA work?

- Exchange differs a bit

one-electron approach

$$h' = \frac{p^2}{2m} + V'_C + \Sigma'$$

$$\equiv h + [V'_C - V_C] + [\Sigma' - \Sigma].$$

self energy is like dynamically screened exchange

$$\Sigma(\omega) = iG[v + (\epsilon^{-1}(\omega) - 1)v] = V_{ex} + \Sigma_C(\omega)$$

unscreened Fock exchange

dynamic part of self energy

$$\Sigma' - \Sigma = V_X(i) + \Delta V_{ex} + \Delta \Sigma_C,$$

change in valence relaxation for exchange

change in dynamic self energy

BSE

$$h'_{\text{eff}} = h + V_D(i) + V_X(i).$$

BSE exchange is non-local unscreened for BSE hamiltonian

$$V_X = \hat{a}_c^\dagger(\mathbf{r}, \sigma) \hat{a}_b(\mathbf{r}', \sigma') \frac{1}{|\mathbf{r} - \mathbf{r}'|} \hat{a}_c(\mathbf{r}', \sigma) \hat{a}_b^\dagger(\mathbf{r}, \sigma')$$

creation annihilation of e/h pairs

Why does IPA work?

one-electron approach

$$h' = \frac{p^2}{2m} + V'_C + \Sigma'$$
$$\equiv h + [V'_C - V_C] + [\Sigma' - \Sigma].$$

BSE

$$h'_{\text{eff}} = h + V_D(i) + V_X(i).$$

$$h = h$$

$$\Delta V_{\text{scf}}(\vec{r}) \approx V_{\text{ind}}(\vec{r}).$$

$$\Sigma' - \Sigma = V_X(i) + \Delta V_{\text{ex}} + \Delta \Sigma_C,$$

Why does IPA work?

one-electron approach

$$h' = \frac{p^2}{2m} + V'_C + \Sigma'$$
$$\equiv h + [V'_C - V_C] + [\Sigma' - \Sigma].$$

BSE

$$h'_{\text{eff}} = h + V_D(i) + V_X(i).$$

$$h = h$$

$$\Delta V_{\text{scf}}(\vec{r}) \approx V_{\text{ind}}(\vec{r}).$$

$$\Sigma' - \Sigma = V_X(i) + \Delta V_{\text{ex}} + \Delta \Sigma_C,$$

terms evaluated locally

Why does IPA work?

one-electron approach

$$h' = \frac{p^2}{2m} + V'_C + \Sigma'$$
$$\equiv h + [V'_C - V_C] + [\Sigma' - \Sigma].$$

BSE

$$h'_{\text{eff}} = h + V_D(i) + V_X(i).$$

$$h = h$$

$$\Delta V_{\text{scf}}(\vec{r}) \approx V_{\text{ind}}(\vec{r}).$$

$$\Sigma' - \Sigma = V_X(i) + \Delta V_{\text{ex}} + \Delta \Sigma_C,$$

terms evaluated locally

For (deep) K edge there is no spin orbit and $\langle \mathbf{r} | \psi_i \rangle$ is localized so this does not matter

Why does IPA work?

one-electron approach

BSE

$$h' \equiv h + [V'_C - V_C] + [\Sigma' - \Sigma].$$

$$h'_{\text{eff}} = h + V_D(i) + V_X(i).$$

(>100 eV) K edge both are equivalent!

$$h = h$$

$$\Delta V_{\text{scf}}(\vec{r}) \approx V_{\text{ind}}(\vec{r}).$$

$$\Sigma' - \Sigma = V_X(i)$$

$L_{\text{II,III}}, M_{\text{II,III}}, \dots$ edges not so lucky

$$h = h$$

$$\Delta V_{\text{scf}}(\vec{r}) \approx V_{\text{ind}}(\vec{r}).$$

$$\Sigma' - \Sigma = V_X(i) + \Delta V_{\text{ex}} + \Delta \Sigma_C$$