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# H-MOR: Density functional investigation for the relative strength of Brønsted acid sites and dynamics simulation of NH<sub>3</sub> protonation–deprotonation

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#### Abstract

The adsorption energies of NH<sub>3</sub> at different positions in acidic mordenite, viz., main channel, side pocket, and double four-membered rings, are investigated using periodic density functional theory method. Furthermore, for the first time, the dynamic behavior of NH<sub>3</sub> interacting with Brønsted acid site in the main channel has been monitored. The results reveal that the adsorption energies of ammonia on Brønsted acid sites in the main channel ( $T_4$ ,  $T_2$ , and  $T_1$ ) are higher than that in the side pocket ( $T_3$ ). Consequently, the strength of Brønsted acid sites follows the same order. Ammonia dynamics results show that the protons are in continuous transfer, where NH<sub>3</sub> acts as a bridge for transferring protons in between ammonium ion and framework oxygen ions.

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Keywords: Mordenite; Brønsted acid sites; Ammonia adsorption; DFT; Molecular dynamics

#### 1. Introduction

In the field of heterogeneous catalysis by molecular sieves, ammonia is extensively used as a basic probe molecule for acidity characterization owing to its accessibility for all acid sites locating in pores, channels, or windows  $\geq 4$  Å. Some of the encountered problems on using NH<sub>3</sub>, in experimental studies, are: (i) strong basicity, which levels out the clear distinction of different acid sites, (ii) the inherent complexity of the vibrational spectra of NH<sub>4</sub><sup>+</sup> and coordinated NH<sub>3</sub>, and (iii) its dissociation on some oxide surfaces. These observations led to the use of other probes viz., CO, pyridine,

alkyl amines, and alkylnitriles [1-4]. However, many valuable quantitative analyses were carried out by fostering the above-mentioned problems on using  $NH_3$  as a probe [5–8]. Recently, we have shown that the adsorption energy of NH<sub>3</sub> can be effectively used to determine the relative strength of Brønsted acid sites, in isomorphously substituted chabazite, and aluminophosphate-34 [9], and Lewis acid sites in microporous molecular sieves [10]. Lamberov et al. [11] showed that NH<sub>3</sub> is more appropriate than CO as a probe for different Lewis acid sites in molecular sieves. Hence, ammonia adsorption is important to scale the strength of different acid sites in microporous molecular sieves. It is well accepted that the interaction between NH3 and Brønsted acid sites, in zeolites, is strong enough to form  $NH_4^+$  [12]. Earlier studies indicated the possible interaction of ammonia with the lattice oxygen ions around aluminum site via two, three, or four hydrogen bonds [13-15].

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The important zeolite structure, mordenite (MOR), is known by its pronounced acidity among many other zeolite structures [6,7]. This has led to the use of mordenite for several applications e.g., cracking, isomerization of hydrocarbons, dewaxing of heavy petroleum fractions [16,17]. Therefore, detailed characterization of various acid sites in mordenite is a significant issue for manipulating and understanding its catalytic applications. Moreover, investigating the dynamic behavior of an adsorbate interacting with its host framework will enrich our understanding on how adsorption and desorption processes are taking place at a specific acid site. The two most popular types of computer simulations are Monte Carlo (MC) and molecular dynamics (MD) techniques. MD is more expensive than MC. However, MD simulation is usually more revealing because it provides information on how molecules move. Thus, with MD one can study time dependent phenomena, fluid flow, and other transport phenomena, which cannot be done with MC [18]. Dynamics of several adsorbates as, H<sub>2</sub>O [19,20] and CH<sub>3</sub>OH [21,22] in zeolites have been studied. Furthermore, the spontaneous proton transfer in between framework oxygen ions of mordenite has been observed recently [23]. Temperature-programmed desorption (TPD) and microcalorimetry techniques (MC) are commonly used for acidity characterization. However, the assignment of its responses to specific acid sites is still unclear. Further, the results depend directly on the experimental conditions [24]. To the best of our knowledge measuring the strength order of Brønsted sites in mordenite could not be tackled in the previous experimental or theoritical studies [4,15, and 25]. This may be due to the complexity of the spectra of probe molecules inside zeolite lattices. Therefore, the aim of this study is to find out the relative strength of Brønsted acid sites in mordenite using NH<sub>3</sub> as a basic probe and to monitor the interaction of NH3 with Brønsted acid site in the main channel using a novel quantum chemical molecular dynamics (QCMD) code [26].

### 2. Theory

### 2.1. Models

Mordnite has an orthorhombic unit cell. In order to create a Brønsted acid site, in silicalite structure, one silicon is replaced by aluminum and proton (H Al Si47 O96). The proton is attached to the bridging oxygen by covalent bond. The main channel ( $6.7 \text{ Å} \times 7.0 \text{ Å}$ ) in [001] direction consists of 12-membered ring (MR) connected with 8-MR side pocket in the [010] direction as shown in Fig. 1a and b [27]. For labeling the atoms, we followed Alberti et al. notation [28]. It should be mentioned that there are four distinct tetrahedral (T) sites per unit cell: three lies in the main channel  $T_4$ ,  $T_2$ ,  $T_1$  and one in the side pocket  $T_3$  (Fig. 1b). Replacement of Si by (Al, H) has been done to produce Brønsted sites. The protons were attached to  $O_{10}$ ,  $O_2$ ,  $O_7$ , which are accessible to the interaction with adsorbates in main channel

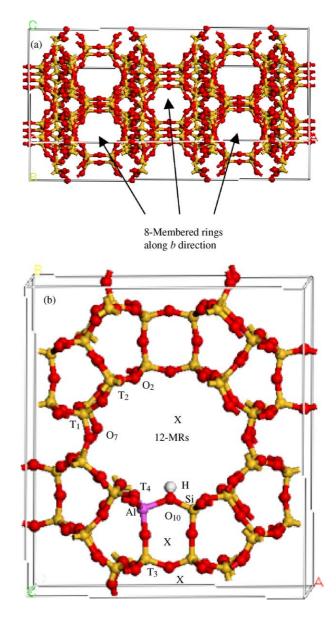


Fig. 1. The optimized structure of acidic mordenite (H-MOR) showing the eight-membered rings along b direction (a), and the investigated adsorption sites marked by x (b).

and O<sub>9</sub> in side pocket. There are 14 possibilities for placing the proton in the unit cell. Brändle and Sauer [29] and Demuth et al. [30] have found that T<sub>4</sub> is the most preferable location for sitting Al. Hence, we have chosen this site for investigating the dynamics of NH<sub>3</sub> where the proton (H<sub>1</sub>) is attached to O<sub>10</sub>. Here we are interested in high Si/Al ratio, which permits studying the isolated Brønsted acid sites in MOR and could be extended to mesostructures or amorphous materials.

#### 2.2. Computational details

The initial structures of NH<sub>3</sub>, H-MOR, and NH<sub>3</sub>–H-MOR were optimized by density functional (DF) method at constant pressure employing Dmol<sup>3</sup> program package [31–33],

with Vosko–Wilk–Nusair [34] local density approximation (LDA) and the energies were evaluated at the generalized gradient approximation (GGA) level with Perdew–Wang (PW91) exchange and correlation functionals [35]. Thus, our choice allows for reasonable timing and sufficient accuracy especially when a comparative analysis is the target. Double numerical with polarization (DNP) basis set was used taking into account all core electrons. This basis set is comparable to 6-31G\*\* set, however, the numerical basis set is much more accurate than Gaussian basis set of the same size [31]. The real space cut-off value was set to 5.5 Å, while the convergence criteria were set as follows: energy =  $2 \times 10^{-4}$  Ha; force =  $4 \times 10^{-3}$  Ha/Å; displacement = 0.05 Å. The adsorption energy  $E_{ads}$  was calculated as per the following relation.

## $E_{\text{ads}} = E_{\text{NH}_4-\text{MOR}} - E_{\text{H-MOR}} - E_{\text{NH}_3}$

where  $E_{\text{NH}_4-\text{MOR}}$ ,  $E_{\text{H-MOR}}$  and  $E_{\text{NH}_3}$  refer to the total energies for each term. The thermal energy correction to the total energy was not considered here since the frequency calculation for such a large system is still a big computational challenge. Molecular dynamics simulation for NH<sub>3</sub> adsorption on Brønsted acid in the main channel of H-MOR was carried out using "Colors" code, based on the tight binding approximation, for 2000 steps with time intervals of 0.1 fs at ambient temperature and pressure. Parameterization for "Colors" was carried out on the basis of DFT calculations of valence state ionization potential, Slater exponents for the atomic orbitals, and the best fitting of the diatomic potential energy curves in the system. Further details about the code

and its parameterization have been given in earlier papers [26,36–38]. Bond population between two atoms *X* and *Y* is given by the relation  $M_{XY} = \sum_{r}^{\text{on } X} \sum_{s}^{\text{on } Y} N_{rs}$ , where  $N_{rs}$  is the atomic orbital bond population.

#### 3. Results and discussion

# 3.1. Adsorption of NH<sub>3</sub> in the main channel, side-pocket, and D<sub>4</sub>-MRs of H-MOR

Table 1 summarizes the optimized geometrical parameters for H-MOR and NH<sub>4</sub>-MOR structures as shown in Fig. 1b and Fig. 2, respectively, as well as the corresponding experimental geometries for natural mordenite with the formula  $Na_{1.5}K_{2.8}Ca_{2.1}Al_9Si_{39}O_{96}$  [28]. First, it can be seen from this table that there is a considerable local distortion in the framework upon the substitution  $Si^{4+} \rightarrow Al^{3+} + H^+$ . This is reflected from the increase of Al– $O_{10}$  bond length (1.897 Å) and the decrease of Al–O<sub>10</sub>–Si bond angle (138.1°) with respect to  $T_4$ – $O_{10}$  (1.637 Å) and  $T_4$ – $O_{10}$ – $T_4$  (146.8°) in natural MOR [28]. In H-MOR, the other Al-O bond lengths viz., Al–O<sub>2</sub> (1.712 Å), Al–O'<sub>2</sub> (1.711 Å) and Al–O4 (1.696 Å) are affected slightly by this substitution. After NH<sub>3</sub> adsorption, the Al– $O_{10}$  (1.771 Å) and Al– $O_{10}$ –Si (138.7°) are changed to a small degree with respect to H-MOR due to the formation of NH<sub>4</sub><sup>+</sup> ion. Thus, the local distortion, in the case of H-MOR, is larger than that of NH<sub>4</sub>-MOR or natural MOR. A similar observation has been noticed when the optimized geometries of pure silica mordenite, Na-MOR, and H-MOR structures

Table 1

Structural parameters, total energies (au), and Hirshfeld charges for H-MOR and NH<sub>4</sub>-MOR at T<sub>4</sub> site

Structural parameters and charges	H-MOR	Experimental [28]	H-MOR [30]		NH <sub>4</sub> -MOR
			LDA	GGA	
Si-O <sub>10</sub>	1.686		1.669	1.682	1.603
Al-O <sub>10</sub>	1.897	1.637	1.850	1.874	1.771
Al-O <sub>2</sub>	1.712	1.614	1.700	1.716	1.771
Al-O <sub>2</sub>	1.711	1.614	1.700	1.716	1.739
Al-O <sub>4</sub>	1.696	1.620	1.685	1.700	1.701
O <sub>10</sub> —H <sub>1</sub>	0.982	-	0.995	0.993	1.638
O <sub>2</sub> -H <sub>2</sub>					1.674
O <sub>2</sub> '-H <sub>3</sub>					2.618
$\tilde{N-H_1}, N-H_2$					1.077, 1.037
N-H <sub>3</sub> , N-H <sub>4</sub>					1.030, 1.027
∠Al—O <sub>10</sub> —Si	133.2	146.8	136.2	133.3	138.7
$\angle O_4$ -Al- $O_{10}$	104.2	112.4	107.5	107.7	114.8
$\angle O_2$ -Al- $O_{10}$	98.9	105.5	98.5	95.6	101.2
$\angle O'_2$ -Al- $O_{10}$	102.7	105.5	98.5	95.6	103.9
$\angle O_2 - A1 - O_{21}$	114.7	109.7	113.1	116.6	105.9
$\angle O_2$ -A1-O <sub>4</sub>	117.8	111.6	117.5	115.8	114.6
qH <sub>1</sub> , qH <sub>2</sub>	0.220				0.126, 0.123
qH <sub>3</sub> , qH <sub>4</sub>					0.174, 0.188
$qO_{10}, qO_2$	-0.163, -0.345				-0.309, -0.343
$qO_4, qO'_2$	-0.352, -0.352				-0.352, -0.316
qSi, qAl, qN	0.588, 0.516				0.56, 0.495, -0.040
Total energy	-21082.62955				-21139.24742

Interatomic distances are given in (Å) and bond angles in (°).

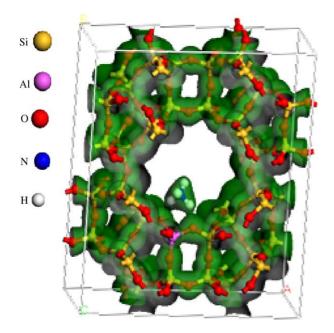


Fig. 2. The optimized structure of NH<sub>4</sub>-MOR, the total electron density distribution showing the interaction between  $NH_4^+$  and the lattice oxygen ions.

were compared [30]. The calculated structural parameters by Demuth et al. [30] using DF with plane-wave basis set and LDA or GGA approximations are given in Table 1 for making a comparison with our results. An average difference of  $\sim 0.014$  Å in bond lengths and  $\sim 0.8^{\circ}$  in bond angles are noticed between our results and Demuth et al. results. This can be attributed to the different bases sets used in the calculations. They showed that the structural parameters obtained by LDA and GGA were in good agreement with the experimental data for natural MOR. However, in the case of H-MOR, the local distortion was larger by using GGA than by using LDA. A recent neutron diffraction study for three deuterated mordenite (D-MOR) samples with Si/Al=5.5, 5.6, and 10 suggested the presence of four types of Brønsted acid sites at O<sub>10</sub>, O<sub>9</sub>, O<sub>6</sub> and O<sub>5</sub> [39]. However, the relation between the structural parameters and the strength of these sites was unclear. For examples, it was mentioned that the strength of Brønsted acid site decreases with the increase of Al content. Nonetheless, the O<sub>10</sub>-D bond distances (1.01, 0.99, and 1.00 Å) and  $T_4-O_{10}-T_4$  angles were 136, 156, and 152° for D-MOR with Si/Al = 5.5, 5.6, and 10, respectively. Furthermore, although Al–O is longer than Si–O, the measured  $T_4-O_{10}$  bond lengths were 1.63, 1.62, and 1.62 Å for the same samples order. The difficulty to designate the geometrical distortions predicted by computational calculations may be due to the deficiency to determine the exact location of Al or Si in the framework.

The structural parameters during the molecular dynamics simulation are not stable, especially around the adsorption site, due to the thermal energy of the atoms at 298.15 K.

Fig. 2 shows the electron density distribution of NH<sub>3</sub> adsorbed on the Brønsted acid site  $(T_4-O_{10}H)$  in the main channel of H-MOR. It is clear that H<sub>1</sub> is transferred to NH<sub>3</sub> molecule to form NH<sub>4</sub>-MOR. This is supported by the change of N-H<sub>1</sub> distance (1.077 Å), which becomes comparable to the other N-H bonds in NH<sub>4</sub>-MOR, viz., N-H<sub>2</sub> = 1.037 Å, N-H<sub>3</sub> = 1.030 Å, and N-H<sub>4</sub> = 1.027 Å. At the same time, O<sub>10</sub>-H<sub>1</sub> distance (1.638 Å) is significantly elongated compared to that of H-MOR (see Table 1). Moreover, the negative charge on O<sub>10</sub> is significantly increased with respect to that in H-MOR. Also, the charge on nitrogen is changed from (-0.3), in free ammonia molecule ( $E_{NH_3} = -56.56152$  a.u.; N-H 1.022 Å), to (-0.04) in NH<sub>4</sub>-MOR. All these observations indicate the formation of NH<sub>4</sub><sup>+</sup> ion.

Other adsorption sites viz., main channel  $(T_2-O_2,$  $T_1-O_7$ ), side pocket ( $T_3-O_9$ ), and  $D_4$ -MRs have been investigated. The calculated geometries and adsorption energies at these sites are summarized in Table 2. In the main channel and D<sub>4</sub>-MRs Al occupies T<sub>4</sub>, T<sub>2</sub>, and T<sub>1</sub> tetrahedral sites while in the side pocket Al occupies T<sub>3</sub> site. Local distortions in the framework and ammonium ion formation on these sites have been observed similarly to the above-discussed T<sub>4</sub> site. The calculated adsorption energies of NH<sub>3</sub> in the main channel at  $T_4$ ,  $T_2$ , and  $T_1$  are -147.9, -142.2, and -146.1 kJ/mol, respectively. These values are in very good agreement with the experimental results obtained by Katada et al. [6,7]. The higher adsorption energy in the main channel than that in the side pocket (-111.0 kJ/mol) indicates that Brønsted acid sites in the main channel are stronger than the Brønsted acid site in the side pocket. This result agrees with an earlier FTIR study using CO and pyridine as probe molecules [1]. In order to see whether the lower adsorption

Table 2

Structural parameters and adsorption energies of NH<sub>3</sub> on H-MOR at main channel, side pocket, and double four-membered rings

Adsorption site	The optimized geometries for H-MOR	Adsorption energy (kJ/mol)
Main channel		
$T_4 - O_{10}$	Si—O, 1.686; Al—O, 1.897; ∠A1—O—Si, 133.2; O—H, 0.982	-147.9
$T_2 - O_2$	Si—O, 1.697; Al—O, 1.895; ∠A1—O—Si, 131.6; O—H, 0.980	-142.2
T1-O7	Si−O, 1.676; Al−O, 1.889; ∠A1−O−Si, 131.9; O−H, 1.028	-146.1
Side pocket		
$T_3 - O_9$	Si—O, 1.686; Al—O, 1.898; ∠Al—O—Si, 136.8; O—H, 0.894	-111.0
Double 4-MR	Si—O, 1.573; Al—O, 1.741; ∠Al—O—Si, 159.4	-80.4

Interatomic distances are given in (Å) and bond angles in (°).



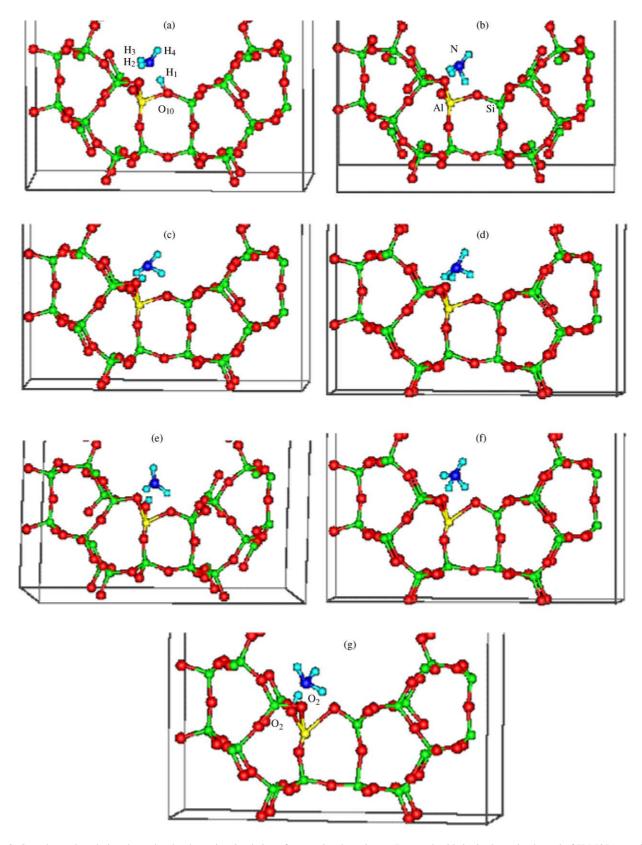


Fig. 3. Snapshots taken during the molecular dynamics simulation of ammonia adsorption on Brønsted acid site in the main channel of H-MOR at various steps at ambient temperature and pressure: (a) 50, (b) 280, (c) 490, (d) 720, (e) 880, (f) 1320 and (g) 2000.

energy of ammonia in the side pocket originates from weak hydrogen bonds to the framework or from the intrinsic acidity, we have calculated the deprotonation energy (DPE) for H-MOR at  $T_4$  and  $T_3$  sites. DPE is defined as the energy required to detach a proton from the framework of H-MOR. The calculated DPE at  $T_4$  (1388.8 kJ/mol) is lower than that at T<sub>3</sub> (1396.5 kJ/mol). Clearly, this indicates that the weaker Brønsted acid site at T<sub>3</sub> is mainly responsible for the lower adsorption energy of ammonia in the side pocket compared to that at T<sub>4</sub> site in the main channel. It should be mentioned here that the calculated DPE value at T<sub>4</sub> site is very close to the calculated value for O<sub>2</sub>'H (1388 kJ/mol) by Brändle and Sauer using the hybrid QM-Pot method [29]. The lower adsorption energy in D<sub>4</sub>-MR (-80.4 kJ/mol) compared to that in main channel reveals that NH<sub>3</sub> adsorbs preferably in the main channel rather than to be in the D<sub>4</sub>-MR. Hence, there are mainly two adsorption sites of NH<sub>3</sub> viz., main channels and side pockets. In the D<sub>4</sub>-MR, NH<sub>3</sub> is tricoordinated to the neighboring lattice oxygen ions. Tetra-coordinated ammonium ion has not been observed at the investigated sites in this study.

It is important to note that the calculated adsorption energies of ammonia here are slightly higher than the previously reported values using both bare-cluster models (-114 kJ/mol) [13] and embedded-cluster models (-133 kJ/mol) [29]. This may be due to the enhanced interaction between ammonium ion and the lattice oxygen ion upon using periodic models or to the different functionals used or the two factors together.

Interestingly Demuth et al. [30] have found that the polarity of the chemical bond  $AIO_{10}$ –H is influenced by applying GGA, resulting in a stronger  $AIO_{10}$ –H bond (see Table 1;  $O_{10}$ –H = 0.995 Å by LDA versus 0.993 Å by GGA). Therefore, higher adsorption energy values are expected for structures optimized by LDA. This can explain the relatively higher adsorption energies of ammonia obtained in this study compared to those calculated in previous papers [13,29].

# 3.2. Dynamics of $NH_3$ adsorption on the Brønsted acid at $T_4$ site in main channel of H-MOR

Fig. 3 shows snapshots taken during the molecular dynamics simulation of  $NH_4^+$  interacting with the lattice oxygen ions of MOR structure at different steps. At 50 step the proton  $H_1$  starts to protonate  $NH_3$  molecule. By the 280 step,  $NH_4^+$  formation can be observed (Fig. 3b). Because of  $NH_4^+$ dynamics, another proton viz,  $H_3$  becomes closer to  $O_2$  ion in the framework. Consequently, at 490 step  $O_2$ – $H_3$  bond is formed leaving  $NH_3$  back as free molecule. Next, at 720 step this proton ( $H_3$ ) moves back to  $NH_3$  to form  $NH_4^+$  again. This scenario is repeated with another proton ( $H_1$ ) when it becomes close to  $O'_2$  it forms  $O'_2$ – $H_1$  bond at 880 step. Thereafter,  $NH_4^+$  is noticed by the 1320 step. By the end of the simulation at 2000 step (Fig. 3g), it can be seen that  $H_2$  and  $O'_2$  form a bond indicating the continuous proton

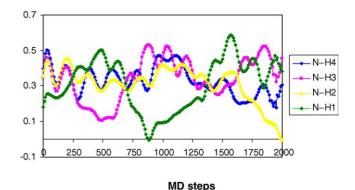


Fig. 4. N—H<sub>x</sub> bond population analysis during the molecular dynamics simulation of NH<sub>3</sub> adsorption.

transfer. Another way to follow the dynamics of ammonia protonation-deprotonation during the adsorption process is to plot the N–H bond population viz, N–H<sub>1</sub>, N–H<sub>2</sub>, N–H<sub>3</sub>, and N-H<sub>4</sub> during the dynamics simulations as shown in Fig. 4. This figure shows clearly when NH<sub>4</sub><sup>+</sup> is formed during the dynamics simulation. Also it reflects when and which proton is attached to the framework oxygen ion. In more details, the bond population values for all N–H bonds are relatively high when  $NH_4^+$  is formed. However if a bond population value is minimum for a specific  $N-H_x$  bond, it means that this proton  $H_x$  is attached to the framework oxygen. Accordingly, Fig. 4 shows that three different protons viz., H<sub>3</sub>, H<sub>2</sub>, and H1 go to framework oxygen ions while NH<sub>4</sub><sup>+</sup> is observed in the meanwhile. Therefore, the dynamics study reveals that there are three oxygen ions in the framework, around the aluminum ion, involved in the interaction with  $NH_4^+$  ion. Furthermore, there is a continuous and consecutive transfer of the protons between the  $NH_4^+$  and the neighboring lattice oxygen ions around Al. From the simulation, it seems that NH3 molecule is acting as a bridge facilitating the protons transfer to different oxygen ions. This point of view may interpret the reason for the weak shoulder observed at  $3375 \text{ cm}^{-1}$  [15]. It should be mentioned here that although the quantum molecular dynamics simulation was carried out for short period 0.2 ps, it was sufficient to monitor the protonation-deprotonation event of NH<sub>3</sub> on the Brønsted acid site in the main channel of H-MOR.

#### 4. Conclusions

The relative strength of the four Brønsted acid sites in H-MOR are investigated using DF method. It has been found that Brønsted acid sites at  $(T_1, T_2, \text{ and } T_4)$  in the main channel are stronger than Brønsted acid site in the side pocket at  $T_3$  site. Moreover, molecular dynamics simulation of NH<sub>3</sub> adsorption on Brønsted acid site in the main channel could monitor the protonation–deprotonation of ammonia resulting from the continuous protons transfer in between ammonium ion and oxygen ions in the framework.

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